TERNARY ION EXCHANGE IN FIXED BEDS
- EQUILIBRIUM AND DYNAMICS -

by
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Dissertation submitted to the Faculty of the Graduate School
of the New Jersey Institute of Technology in partial fulfillment
of the requirements for the degree of
Doctor of Engineering Science
1985
Title of Thesis: Ternary Ion Exchange in Fixed Beds - Equilibrium and Dynamics -

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ABSTRACT

Title of Thesis: Ternary Ion Exchange in Fixed Beds - Equilibrium and Dynamics -
David W. H. Roth, Jr., Doctor of Engineering Science, 1985

Thesis directed by: Professor Ching-Rong Huang

A model was developed to simulate ion exchange within fixed beds for ternary systems. Models of the fluid phase material balance, phase equilibria, and diffusion of ions through the film and within the resin phase incorporated the latest advances in ion exchange theory. The separate model elements were combined after testing into an overall general model. Non-linear regression support programs were developed to estimate equilibrium parameters and resin phase diffusion coefficients.

A computer program was developed to estimate axial dispersion coefficients from experimental data. General correlations derived from literature sources were tested, and axial dispersion terms were included in the electrolyte phase material balance equations.

The rational thermodynamic equilibrium constant, utilizing resin phase activity coefficients based on the 3 suffix Redlich-Kister equation and the Bromley equation for electrolyte phase activity coefficients, was selected. The Wilson and NRTL equations were tested but were not as good. This model was used to correlate published data on 13 binary systems and to predict ternary compositions for comparison to published data on 4 ternary systems. Average root mean square of % normalized difference was about 3% on the binary systems and 4-12% on data predicted for the ternary systems.

The Nernst-Planck equation was used to model resin phase diffusion. An integrated form of the Nernst-Hartley equation, based on the Bromley...
equation, was developed and tested to predict the effect of concentration on electrolyte phase diffusion coefficients. These coefficients were used in a pseudo electric field model which was developed and tested to approximate the electric field effect on diffusion of ions in the film.

The overall ternary system model resulted in four coupled non-linear second order parabolic partial differential equations, with appropriate boundary conditions. The equations were reduced to a set of algebraic equations by finite difference approximations and solved by the implicit Crank-Nicholson method. Non-linear terms were quasilinearized. The resulting five diagonal coefficient matrix describing the fluid phase, coupled with the 7 diagonal coefficient matrices describing the resin phase, were inverted with algorithms developed in this work. An iterative procedure resolved all non-linear terms at each time step. Comparison of concentration histories generated by the model with experimental results obtained by previous researchers showed that the ternary model could be used in practice to optimize process design applications with a bed in a condition of partial presaturation, and for favorable or unfavorable ion exchange.

Resin phase activity coefficients developed in correlation of the equilibrium data were used to test chemical potential as a driving force in the systems simulated. Indication that use of chemical potential would obviate the need for ion pair specific diffusion coefficients in the Nernst-Planck model, or the use of ion pair corrector coefficients (Stefan-Maxwell), is shown by comparison of results on seven binary systems. Implications for industrial application and directions for further research are discussed.
ACKNOWLEDGEMENTS

The author gratefully acknowledges the support and encouragement afforded him by the Department of Chemical Engineering and Chemistry, New Jersey Institute of Technology. In particular, the author wishes to express his gratitude to his advisor, Professor C. R. Huang, for his guidance and counsel during the course of this research. During Dr. Huang's sabbatical leave in 1984-1985, Professor E. C. Roche, Jr., not only provided outstanding advice and direction to the author during the final review and editing of this dissertation, but provided invaluable counsel regarding the completion and fulfillment of all other degree requirements during that period. Special thanks to Dr. Roche.

The author also wishes to express his gratitude to the Allied Corporation for providing all necessary support for completion of this research and the author's graduate studies. In particular, the author is deeply grateful to Dr. L. J. Colby, Jr., Allied Senior Vice President-Technology, for his personal support, patience, and encouragement during the seven years over which this doctoral program extended. At the Allied Corporation, Dr. A. K. S. Murthy provided invaluable technical advice to the author during the course of this research, particularly in the area of numerical solution methods, solution of non-linear equations, and non-linear regression analysis techniques. Indeed, his ingenious and very powerful non-linear regression method is built into two of the parameter fitting programs developed and used in this research. Special thanks to my friend and colleague, Dr. Murthy.
The author also deeply appreciates the continuing advice, encouragement, and support he received from his Office of Science and Technology colleagues during the sometimes hectic last year and a half. In particular, thanks go to Dr. Eve Menger for reviewing and critiquing my dissertation draft and for her continuing moral support, and to Mr. B. J. Samuels who, during the course of this research, provided his usual excellent advice and judgement on a variety of technical issues, as well his moral support.

The author takes this opportunity to express his deep gratitude to his secretary, Ms. Dorothy Simonutti, for handling all of the details related to the typing and preparation of this dissertation. Her patience with my penchant for detail and preciseness is especially appreciated and, in this regard, I extend my appreciation to Mrs. Pat McShane who assisted Ms. Simonutti in this effort.

Finally, and above all, my very deepest appreciation is reserved for my wife, Barbara, whose help, support, sacrifice, and encouragement during the last seven years made the completion of this work possible.

April 1985
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A. Introduction

"But now it should be entirely in the realm of possibility to develop computer programs to handle general (ion exchange) entry conditions...and that the time is not far off when ion exchange adsorption separations can be designed in the same way as distillation separations are today." These words, printed in CEP of October 1977, by Dr. Friedrich Helfferich, who is a world leader in the science of ionic separations, clearly summarize the objectives of this work. The computer based simulation model developed in this work has moved Dr. Helfferich's predictions in that direction, since the purpose of this research was to advance the science of the engineering of ion exchange systems.

Specifically, the objective was to develop a simulation model for ternary systems which would embody most of the currently proven factors applying to prediction of phase equilibria, and to the prediction of the dynamics of ion exchange in fixed beds. The goal was to utilize as many parameters as could be derived from reliable published correlations and to minimize the amount of experimentation required to supply the requisite data for unknown parameter estimation. The simulation objective, therefore, could not be complete until computer based regression analysis support programs, specific to estimation of typical ion exchange parameters, were included as a codevelopment of the dynamic model.
1. General Overview

The use of ion exchange as a unit process in chemical operations dates back more than half a century and, for a large number of commercial ion exchange installations made during that period, the major utility was to produce deionized water for process or boiler feed water makeup. The most important class of ion exchange resins are the gel type beads formed by polymerization of styrene and crosslinked to varying degrees by addition of divinyl benzene. The cation exchange types commonly in use have (1) strongly acidic functionality resulting from sulfonic acid groups attached to the polymer, or (2) weakly acidic functionality due to carboxylic acid groups being part of the polymer structure. Bifunctionality, which may yield a particular cation selectivity relationship, may be obtained by combining in one resin both functionalities.

The anion exchange types of resins in common use involve (1) quaterary ammonium functional groups to yield strongly basic types, and (2) polyamine functional groups to yield weakly basic anion characteristics.

Resin types can be created having both anionic and cationic functionality and, for example, are used in the production of high quality deionized water. Macroreticular (macroporous) resins have been developed with the same functional groups described above. These resins have large discrete pores for use in recovery of high molecular weight
species, or in nonaqueous applications, and where resin
toughness and resistance to breakdown is required.

In recent years, interest in recovery of heavy metals in
hydrometallurgical applications or in waste water cleanup has
led to the development of resin with high selectivity towards
these ionic species. These are resins containing functional
groups, such as iminodiacetic acid, which have the ability to
chelate the desired species and can be designed to be highly
selective for say Cu\(^{++}\) in Cu\(^{++}/Fe^{+++}\) mixtures (leach
dump solutions) at a given pH. Elution with strong acid at
low pH is then possible in order to recover the copper for
subsequent electro-winning.

The number of commercial applications for ion exchange
operations has grown rapidly since the 1950s, and the
following tabulation represents just a sampling of
applications other than water demineralization:

a) Isolation and recovery of metals (hydrometallurgy or
waste treatment.

b) Isolation and recovery of high molecular weight organic
acids or bases.

c) Separation and purification of amino acids.

d) Separation of rare earths.

e) Purification of sugars.

f) Uranium recovery.

g) Antibiotic recovery.

h) Catalysis such as inversion of sugar.
i) Recovery of actinides.

j) Catalysis of organic reactions in non-aqueous solutions.

2. Objectives

2.1 General Objectives

The objective of this work, as stated, was to create a simulation model for the exchange of three counterions between an aqueous electrolyte solution and a monofunctional ion exchange resin performed within a fixed bed. The type of ion exchange resin upon which the model was premised was the gel type having either cationic or anionic functional groups. Most investigators seeking to develop mechanisms and models for ion exchange equilibria and dynamics have standardized on the cation exchange resin Dowex 50W X8 for their experiments. The extensive data available on Dowex 50W X8 was used in this work for parameter estimation and model verification. Dowex 50W X8 is a strongly acidic cation exchange resin produced by Dow Chemical Company with about 8% divinyl benzene cross linkages.

Models dealing with the exchange of two ionic species in such resins are available based upon varying degrees of simplifying assumptions. However, in many of the above-listed applications the selective exchange of two ionic species is performed in the presence of a third ionic species, and in many instances this ion is
H⁺. General models for fixed bed ion exchange involving ternary systems do not exist, except for cases involving major simplifying assumptions (22, 78).

The most definitive publication to date dealing with all aspects of ion exchange (including ionic separations by membranes) is the book, Ion Exchange, by Friedrich Helfferich, McGraw Hill, New York, 1962. The book is out of print but an authorized facsimile was produced in 1978 by University Microfilms International, Ann Arbor, Michigan, USA. Dr. Helfferich continues to be a world leader in the science of ionic separation and, in a complete quotation from the CEP issue of October 1977 stated, "I have so often seen it in practice that, when it comes to sizing or designing a (ion exchange) column, just about anything that goes beyond the equilibrium capacity is left to a strange mixture of experience, intuition, folklore, and black magic. But I believe a change here is imminent, and my confidence is based on two factors. One is the advent of multicomponent theory, which inherently is much nearer to many of the practical problems, more ions being involved than just two. The second is the availability of faster and cheaper computers. What the man in practice usually needs is to calculate a process where an ion exchange bed is run to breakthrough and then is incompletely regenerated. The solutions
tabulated and easily looked up in Perry's Handbook are largely or exclusively for calculation of behavior of a bed of uniform initial composition and receiving a constant feed, and they are only of limited use. But now it should be entirely in the realm of possibility to develop computer programs to handle general entry conditions. I believe that this will happen shortly and that such programs will be used to a much greater extent than we have been accustomed to, and that the time is not far off when ion exchange adsorption separations can be designed in the same way as distillation separations are today."

2.2 Approach

The "generalized approach" used in this research project to develop a fixed bed ion exchange model for binary and ternary systems is shown in Figure 1-1. In this instance, the goal is to optimize, scale up, and design controllable ion exchange processes. In many instances, industrial applications of ion exchange can prove to be more economic than other separation process. The model development methodology depicted, including experimentation to generate equilibrium data and dynamic information plus model verification techniques, are general to any process modeling effort. A more detailed "road map" of modeling effort specifically related to ion exchange processes is
Figure 1-1

ION EXCHANGE PROCESS OPTIMIZATION
BINARY AND TERNARY SYSTEMS

Installation and profitable operation

Ion Exchange Process Application

assumptions
scientific principles

Mathematical Model with unknown parameters

scale-up and control

model validation

Computer based Numerical Approximations
test model

Analytical Solutions simplified cases
test for stability and convergence

Experimental Data ion exchange equilibria

Thermodynamic Model with unknown parameters

Parameter Estimation through regression analysis

and/or map areas for further experiments

Parameter Estimation through trial & error curve fitting w/model

model validation

Thermodynamic parameter input

and/or map areas for further experiments

model validation

Computer based Numerical Model - simulation of complex cases

model support for regression analysis

model utilization in simulations

Plant Design and Optimization

model validation

and/or map areas for further experiments
outlined in section C of Chapter VIII, Summary, in Figure 8-1. The reader is invited to read section C of Chapter VIII following Chapter I, since this dissertation is prepared so that each subsequent chapter deals with a major element of the overall model on a stand-alone basis. The model development approach shown in Figure 8-1 provides the unifying framework for all of the elements as they are combined in an ion exchange process modeling and design optimization procedure.

2.3 Specific objective

The objective of this research work was to advance the engineering of ion exchange systems. Specifically, the goal was to create a general computer based simulation model which would describe ion exchange in fixed beds for ternary systems. The model was to be based on the best currently held theories related to phase equilibria, ionic transport through the film phase, and ionic transport within gel type ion exchange resins. Development of support programs which would perform non-linear regression analytic estimation of phase equilibria parameters and resin phase diffusion coefficients from experimental data was a necessary companion research objective. All other information required to design and optimize fixed bed ion exchange processes was to be derived from reliable published correlations. These include (1) film coefficients in
packed beds, (2) ionic species diffusion coefficients in aqueous solutions, (3) ionic species activity coefficients in aqueous solutions, and (4) axial dispersion coefficients in packed beds.

2.4 Experimental Data Utilized

In order to test the model components, and ultimately the overall simulation model, suitable experimental data had to be found which described both ion exchange equilibria and fixed bed dynamics in binary and ternary systems. The following three major sources of data were utilized in this work:


System NH₃ - Ethylene Diamine in H₂O.

Column operation using Dowex 50w X8 resin at 25°C and 1.0 normal Cl⁻ solutions.

Conditions

Two favorable (Ethylene Diamine (ETDA) for NH₃) exchanges at two flow rates, 1.85 cm/sec and 0.94 cm/sec superficial velocity. Reynolds No. 20 and 40.

One unfavorable (NH₃ for ETDA) exchange at 1.85 cm/sec superficial velocity. Reynolds No. 40.
Equilibrium Data at 25° C for 1.0 normal and 0.1 normal Cl⁻ solutions. (15 and 12 data points respectively.)


System Ag⁺, Na⁺, H⁺ in H₂O.

Column operation using Dowex 50w X8 resin at 25°C and NO₃⁻ solutions of 0.05 and 1.5 normality.

Conditions

Binary runs for favorable exchange of all ion pairs at 2 normalities (6 runs).

- 0.05 normal, superficial velocity - 0.18-0.2 cm/sec; Reynolds No. 4.-4.4.
- 1.5 normal, superficial velocity - 0.08 cm/sec; Reynolds No. 1.9.

Ternary runs for favorable exchange

1. Resin in H⁺ form.
   Feed comp. X_Ag=0.25, X_Na=0.24, X_H=0.51 (equivalent fraction) at 0.0533 normality of NO₃⁻ ion.
   Superficial velocity = 0.097 cm/sec, Reynolds No. 1.94.

2. Resin in H⁺ form.
   Feed comp. X_Ag=0.35, X_Na=0.24, X_H=0.41
(equivalent fraction) at 1.51 normality of NO$_3^-$ ion.

Superficial velocity = 0.0346 cm/sec,
Reynolds No. 0.70.

**Ternary run for unfavorable exchange**

1. Resin in Na$^+$ form.

Feed comp. $X_{Ag} = 0.187$, $X_{Na} = 0.133$, $X_{H} = 0.68$
(equivalent fraction) at 1.5 normality
of NO$_3^-$ ion.

Superficial velocity = 0.046 cm/sec,
Reynolds No. 0.92.

**Equilibrium Data at 25°C for all binary pairs** at NO$_3^-$ solutions of 0.1 & 1.5
normality (16 total points for all 6 sets).

**Ternary Equilibrium data for 0.1 normal & 1.5 normal at 25°C** (2 points + 3 material
balance points generated in this research project from the three rate runs).


Systems Sr$^{++}$, Mn$^{++}$, Cs$^{+}$ in H$_2$O & Mn$^{++}$, Cs$^{+}$, Na$^{+}$ in aqueous solutions of Cl$^{-}$ anion.

Single particle "batch operation" using Dowex 50W X8 resin at 23°C in 0.1 normal Cl$^{-}$ solutions.

Binary runs; concentration vs. time curves were generated for the systems SrMn, MnSr, MnCs, SrCs, MnNa, NaMn, & NaCs, where the bar refers to the species in the resin.

Superficial velocity - 25 cm/sec, Reynolds No. 700.

Ternary runs;

22 runs of various resin and electrolyte compositions. Conditions were similar to the binary runs. Since only one species in three could be radioactively tagged, only one concentration vs. time curve could be generated; whereas for ternary modeling, two curves or concentration traces are desirable.

Equilibrium Data

Experiments were performed using tagged ionic species for all binary pairs, and various ternary compositions for the ternary systems Sr$^{++}$, Mn$^{++}$, Cs$^{+}$ & Mn$^{++}$, Cs$^{+}$, Na$^{+}$. 
(16 points for the first ternary system and 15 points for the second ternary system.)

The quality of the experimental procedures and attention to detail was judged to be very good for the Erickson column experiments on NH$_3$-EDTA, as was the work of Bajpai, Gupta, & Rao on single particle techniques. There have been questions raised on the general extendability of single particle experiments to full column operation, but the investigators carried out the experimentation carefully, and reproducibility on the equilibrium experiments was ± 2%. Omatete's experimental work appeared to be less carefully controlled, and certainly did not yield a sufficient number of data points to satisfy the equilibrium correlation requirements.

B. Background

1. Equilibria

1.1 Equilibrium Relationships

A multitude of conventions are used to relate the concentration of an ionic species in the electrolyte phase with its concentration in the resin phase. The concentration units used in the various conventions can involve molar, molal, mole fraction, or equivalent fraction for both phases, or different units for each phase in all combinations. Helfferich (36) discusses these conventions in his outstanding book on ion
exchange. The exchanging ions have an electric charge opposite in sign to the charge of the fixed functional groups in the resin and are called counterions. The ionic species in the electrolyte having an electric charge of similar sign to the resin phase charge are called coions. A favorable exchange involves replacement of ions in the resin phase with ions from the electrolyte phase which are preferred by the ion exchange resin. The opposite is an unfavorable exchange.

The simplest equilibrium convention utilized is the distribution coefficient which implies a linear relationship:

\[ \lambda_i = \frac{x_i}{x^*_i} \]  

(1-1)

where: \( x_i \) = equivalent fraction of species i in the resin
\( x^*_i \) = equivalent fraction of species i in the electrolyte in equilibrium with the resin.

Other concentration units could be used. Rao and his coworkers (93) used this relationship but corrected for the non-linearity actually experienced in most ion exchange equilibrium relationships by using the expressions:

\[ \lambda_i = a + b \cdot x_i + c \cdot x_i^2 \]  

(1-2)
Ternary systems were correlated by Rao and his coworkers based on pseudo equivalent fractions, i.e., excluding the concentration of the ion not involved in the binary distribution coefficient being calculated.

A more frequently used convention is the separation factor involving any of the concentration units described above, and is given in equivalent fraction units as:

\[ \alpha = \frac{\frac{x_1^*}{x_{2\text{int}}}}{\frac{x_2^*}{x_{1\text{int}}}} \]  

where ion 1 is the favored species. This is similar to the \( K \) factor used in distillation and contains no provisions for non-ideality in either phase. Omatete (78) used this representation for ternary systems with the convention:

\[ x_j^*_{\text{int}} = \left\{ \frac{\alpha_{j\text{ref}}}{\sum_{k=1}^{n} \alpha_k^* \bar{x}_k} \bar{x}_j \right\} \text{ for } j=1, 2, 3 \]  

The various mass action forms appear as is shown for equivalent fraction units:

\[ K_2^1 = \left( \frac{x_2}{x_1} \right)^{\frac{z_2}{x_1^*_{\text{int}}}} \left( \frac{\bar{x}_2}{\bar{x}_1} \right)^{\frac{z_1}{z_2}} \]  

\[ \lambda_2 = \frac{(1-x_1)}{(1-x_{1\text{int}})} \]  

(1-3)
where $z_i$ = charge on ionic species $i$.

These forms can be altered by substitution of activity, $a_i^*$ for the electrolyte phase and finally, with the substitution of activity $\bar{a}_i$ for the resin phase, appear as thermodynamic equilibrium constants:

$$tK^L_2 = \left( \frac{\bar{a}_1}{a_1^*} \right)^{z_2} \cdot \left( \frac{a_2^*}{\bar{a}_2} \right)^{z_1}$$

(1-7)

The molal form appears as

$$t\bar{K}^L_2 = \left( \frac{m_1 \cdot \bar{y}_1}{m_1^* \cdot \bar{y}_1^*} \right)^{z_2} \cdot \left( \frac{m_2^* \cdot \bar{y}_2^*}{m_2 \cdot \bar{y}_2} \right)^{z_1}$$

(1-8)

If the standard and reference states are chosen so they are the same in the electrolyte and resin pore solution, then $tK^L_2 = 1$, and selectivity shown by the relationship is vested in the activity coefficients. The rational form,

$$n_{K_2} = \left( \frac{\bar{y}_1 \cdot \bar{y}_1^*}{m_1 \cdot \bar{y}_1} \right)^{z_2} \cdot \left( \frac{m_2^* \cdot \bar{y}_2^*}{\bar{y}_2} \right)^{z_1}$$

(1-9)

is often used and in this case the standard and reference states of the resin phase and the electrolyte phase are quite different. The selectivity shown by the relationship is vested in the equilibrium constant, $n_{K_2}$, with the activity coefficients providing "trim
up" to correct extreme non-linearity conditions.

Both thermodynamic forms are based on the simplification assumptions that electrolyte and solvent sorption and desorption are negligibly small. They are normally used to describe binary systems, since multicomponent models for activity coefficients in either phase have not been extensively explored.

1.2 Electrolyte Phase Activity Coefficients

Electrolyte phase activity coefficients of individual ions in aqueous solutions of mixed salts are required in the thermodynamic equilibrium relationships. The familiar Debye-Huckel relationship which predicts activity coefficients for single ions and mean activity coefficients for pure salts dissolved in water is based on the long range electrostatic interaction of charged particles. It holds only for dilute solutions in which short range (van der Waals) interactions are minimal. At higher concentrations these forces become significant, and second or third terms which are functions of the solution ionic strength, I, are added to the Debye-Huckel equation to account for ion-ion short range interactions. These equations constitute the extended forms of the Debye-Huckel equation, one of which is given by Robinson and Stokes (103) for single ions as follows:
in which the $a_i$ and $b_i$ parameters are given by Truesdell and Jones (128).

The relationship of activity coefficients of salts in mixed electrolytes has been studied by Guggenheim and Turgeon (31), Meissner and Kusik (67), and Pitzer (86), Edwards, Newman and Prausnitz (24) and others. The treatment introduced by Guggenheim involved the normal Debye-Hückel long range interaction effect plus a summation term which related the short range interactions of specific cation-anion pairs. An ion pair parameter was used in combination with the molal concentration of the specific ion (cation or anion) to represent the magnitude of the interaction. Bromley (9) and Meissner and Kusik (67, 68) extended the predictive ability of such models to concentrations as high as 6 molal by introduction of additional terms, but yet requiring only one pair corrector constant for each salt present. Bromley (9), with later evaluation and testing by Zemaitis (139), correlated a large number of extended coefficients for various salts. Based on a variation of the Guggenheim model, Bromley (9) derived an activity coefficient model for single ions in mixed electrolytes with applicability up to the 6 molal range.
Measurement of the activity of a single ionic species is not possible, so the criteria of predictive capability of a model is limited to the mean activity coefficient of a given salt in mixed electrolytes. Several of these methods for prediction of activity coefficients for salts in multicomponent aqueous solutions were evaluated by Sangster and Lenzi (111). The overall efficacy of prediction of single ion activity coefficients from tabulated Debye-Hückel extension parameters in mixed salts up to 6 molal concentrations can best be judged by the results of simulations in which these parameters are important. The success of the use of the Bromley type relationship has been aptly demonstrated by the predictive capability of the systems developed and tested by Zemaitis (139) while at Chem Systems, Inc. to predict complex ionic phase equilibria. Some of this work (25) was done under sponsorship of the AIChE Design Institute for Physical Property Data (DIPPR).

1.3 Resin Phase Activity Coefficients

Models for resin phase activity coefficients have not been studied extensively for multicomponent systems. For binary systems, Rao and David (88) used the observation that the activity coefficients of species in the resin phase are lower but proportional to the activity coefficients of the species in electrolyte solutions having the same ionic strength as the solution
in the resin phase. This correlation was successful in predicting the binary data with a ± 10% maximum error; however, it is noted that ionic strength in the resin may exceed 13, well beyond the electrolyte phase activity coefficient correlation capability. The method has not had wide acceptance. Although several predictive binary models have been proposed by various investigators (40), these models as yet only give qualitative information. The resin phase activity coefficients must be correlated from experimental data with thermodynamically consistent representations. One such early model was developed by Kielland (53) and put on a more sound theoretical footing by Barrer (5). The model is in fact a one suffix Redlich-Kister type equation, and led to trial of multisuffix Redlich-Kister equations in this work. Salmon (110) has investigated a statistical thermodynamic approach for correlation of multivalent anion exchange equilibria in binary systems with some success.

The most significant work on correlation of resin phase activity coefficients in multicomponent systems was performed by Smith and Woodburn (116) on the system $\text{NO}_3^−-\text{Cl}^−-\text{SO}_4^{2−}$ on Dowex 50W X8/Na⁺. The Wilson equation (98) is used in correlation of activity coefficients in liquid systems based on assumptions of non-randomness of local compositions to express the excess Gibbs free
energy relationships. The equation has two adjustable parameters per binary system. Smith and Woodburn showed that the Wilson equation, utilizing fitted binary system coefficients, could predict ternary system equilibrium concentrations for the system investigated with a standard deviation of % normalized difference of 4.3%. This model, along with the 3 suffix Redlich-Kister and NRTL equations, were evaluated in this work.

2. Film Mass Transfer Coefficients

2.1 Film Coefficient Correlations for Packed Beds

Many investigators have studied the mass transfer from fluids to particles in fixed beds. Carberry (11) generated a boundary layer model of the form

\[ j_{b,l.} = \frac{k_i \cdot \varepsilon}{v_a} \cdot Sc^{2} = b \cdot Re^{-\frac{1}{3}} \]  

(1-11)

where:

- \( b = 1.15 \)
- \( v_a = \) superficial velocity
- \( \varepsilon = \) bed void fraction
- \( Re = \) Reynolds No. based on \( \frac{v_a}{\varepsilon} \)
- \( Sc = \) Schmidt No.

Data from numerous experimental studies for gases and liquids was correlated well by this relationship down to Reynolds No. of 1.0. Wilson and Geankoplis (138) showed for transfer from benzoic acid spheres to water or propylene glycol that \( Re' \) raised to the \(-2/3\) power...
correlated the data well ($Re'$ based on superficial velocity). Work by Snowdon and Turner (131) on correlation of film coefficients for mass transfer to ion exchange resins in fluidized beds found a Carberry-type relationship fit the experimental data well. Comparison of the two equations showed:

**Carberry correlation for fixed beds**

$$Sh = \frac{k_L \cdot D}{d} = 1.15 \cdot \left(\frac{Re'}{\varepsilon}\right)^{\frac{1}{2}} \cdot Sc^{\frac{1}{4}}$$  \hspace{1cm} (1-12)

where $Re'$ is based on superficial velocity

**Snowdon & Turner correlation for fluid beds**

$$Sh = \frac{k_L \cdot D}{d} = 0.81 \cdot \frac{Re'}{\varepsilon} \cdot Sc^{\frac{1}{4}}$$  \hspace{1cm} (1-13)

where for fluid beds the coefficient $b = 0.81$ and is raised to the first power.

P. N. Rowe (109) also evaluated particle to liquid mass transfer in fluidized beds based on work done by Nelson and Galloway (73) at low Reynolds No., $\gamma$1-10. Koloini and his coworkers (54, 55) studied mass transfer in liquid fluidized beds at low Reynolds No. ($\gamma$1) and found the following correlation,

$$Sh = \frac{k_L \cdot D}{d} = 1.10 \cdot \frac{Re'}{\varepsilon} \cdot Sc^{\frac{1}{4}}$$  \hspace{1cm} (1-14)

represented data for ion exchange in dilute solutions of HCl and benzoic acid, as contrasted to the Snowdon and Turner correlation which fit the data well in the region of $5 < Re < 100$. 
2.2 Effects of Electric Field in the Film

Helfferich (43) describes the effect of exchange of ions having differing mobilities as they pass through the film surrounding an ion exchange resin. Since the conditions of electroneutrality and no electric current must exist in the film, the coion distributes itself in the film in such a manner as to counterbalance the electric field gradient in the film. The interdiffusivity relationship is described by the Nernst-Planck equation, which is given as:

\[ J_i = - D_i \cdot (\nabla c_i + z_i \cdot c_i \cdot \frac{F}{RT} \cdot \nabla \phi) \]  

(1-15)

where:

- \( F \) = Faraday's constant
- \( \phi \) = electric potential
- \( c_i \) = concentration
- \( D_i \) = diffusion coefficient
- \( z_i \) = charge on particle
- \( J_i \) = molar flux
- \( T \) = temperature
- \( R \) = gas constant

The usual assumption is to assume a quasi-stationary state for the coion in the film which presumes any coion dynamics have occurred prior to the quasi-stationary state condition. In addition to Helfferich, Snowdon and Turner (131) have evaluated the effect for exchange of
Na\(^+\) and H\(^+\) on Zeo-Karb 225 in dilute Cl\(^-\) solutions. Their measurements agree with the Nernst-Planck model. They assumed a stagnant film thickness based on the Reynolds No. and an effective diffusion coefficient determined by the Nernst-Planck model interdiffusivity constraints. Van Brocklin and David (132) extended the Nernst-Planck concept to a calculable ratio, \(R_i\), of effective film coefficient with electric field effects to the film coefficient calculated from packed bed correlations, such as Carberry's. They investigated three models, (1) a Film Model similar to Turner and Snowdon, (2) a Penetration Theory Model, and (3) a Boundary Layer Model. Tables of \(R_i\) values were calculated for various diffusivity ratios of ions being exchanged and for a variety of electrolyte concentrations at the film boundaries. No experimental verification was offered. Van Brocklin and David (133), in a later work, used data developed by other investigators (single particle studies with radioactively tagged ionic species) to test the various models. The data were insufficiently accurate to select the best mass transfer model, although the direction and magnitude of correction given by the models gave better agreement with the data than did uncorrected film coefficients. Pan and David (83) computed the effect of \(R_i\) model on the design of a moving packed bed ion
exchange process. No experimental data was available for comparison but, in unfavorable exchange, more than 50% underdesign could be experienced if the electric field effect was neglected. Kataoka and Yoshida (50) developed rigorous equations to describe homovalent exchange of ions through a film under the influence of an electric field. A factor, $\beta$, similar to the $R_1$ factor, related the film coefficient corrected for electric field effects to the film coefficient without the effects. Experimental effluent concentration histories (ECH) were obtained for the favorable and unfavorable exchange of $\text{Na}^+$$-\text{H}^+$ on DIAION SK1B resin in fixed beds. Chloride solution normality of 0.005 was maintained in order to operate in a film diffusion controlling regime. In order to simplify the mathematics, resistance to mass transport in the resin was assumed to be negligible. The electric field model gave the best representation of the ECH curves, particularly during unfavorable exchange when the least favored ion has the highest mobility. For favorable exchange, particularly when the most favored ion has the lowest mobility, use of the uncorrected diffusion coefficient for the slower ion can be used to approximate column behavior within an error of 30%.

3. Diffusion in the Resin Phase and Various Representations in Fixed Bed Operations

Early representations of rate laws governing diffusion
in the resin phase had more to do with the ability of the resulting equations to be solved analytically then with actual physical process phenomena. Thomas (127) in 1944 represented ion exchange rates in the resin as two opposing second order reactions. Other "rate law" formats were solved by Amundson (3) in 1949 and Vermeulen and Hiester (134) in 1952. Vermeulen and Hiester (136) also treated the dynamics of ion exchange involving uniform partial presaturation of a column utilizing dominant film theory. Dominant film theory is based on the assumption that resistance to ionic transport from the electrolyte phase to the resin phase is controlled by diffusion through a stagnant film around the particle in external diffusion control, or through a hypothetical thickness of resin constituting a "film" in internal diffusion control.

Rosen (107) in 1951 made the first analytical solution of a fixed bed ion exchange process using Fick's Law for diffusion in the resin phase, coupled with ionic diffusion through an adjacent electrolyte film. The major simplifying assumptions were that axial diffusion in the fluid phase was negligible and that the equilibrium isotherm was linear. The analytical solution contained an infinite integral which converged slowly during computation. This was a milestone effort, however, since the solutions represented benchmarks against which later investigators could test numerical solutions. In 1954 Rosen (108) derived a general solution
for solid diffusion in fixed beds by use of a partial integro-differential equation.

Dranoff and Lapidus (22) performed the solution of ternary ion exchange in a fixed bed in 1957. They assumed (1) mass action type equilibria for monovalent systems, (2) Thomas type reaction rate laws, (3) no axial diffusion in the fluid phase, and (4) that an asymptotic solution method could be applied based on achievement of a stable concentration configuration in the column (i.e., infinite column). The method of characteristics was used to transform the partial differential equations into ordinary differential equations, and solutions were obtained by numerical approximation. Typical results for column operation were generated for the system Ag⁺-Na⁺-H⁺ on Dowex 50 using literature values for parameters. No experimental comparison was made, since the parameters had been derived in batch experiments. Tien and Thodos (126) in 1959, assuming Fick's Law for diffusion in the resin phase, essentially added the effects of a non-linear Freundlich type equilibria isotherm relationship to the case treated analytically by Rosen. Numerical methods were used to solve the resulting integro-differential equation. Masamune and Smith (63, 64) in 1965 investigated the case of adsorption into a porous media from a gas stream using the following model features: (1) film resistance, (2) Fick's Law diffusion in the pores, and (3) a reversible first order adsorption rate equation with linear equilibria.
Additionally, asymptotic conditions were assumed with no axial diffusion term in the gas phase. The resulting analytical solution contained an infinite integral which was evaluated numerically.

In 1965 Cooney and Lightfoot (18) proved the existence of asymptotic solutions to constant pattern operation of fixed bed separations, including the case involving non-linear equilibria with axial dispersion and finite mass transfer resistance. Following this, Hall, Eagleton, Acrivos, and Vermeulen (34) in 1966 studied constant pattern behavior in fixed beds based on both pore and solid diffusion mechanisms within the porous particle. The axial dispersion term was assumed to be negligible, non-linear equilibrium of the Langmuir type was assumed, and numerical solutions were obtained for a wide range of dimensionless variables. These solutions merged smoothly into the analytic solutions for linear equilibrium cases. Rhee and Amundson (101) in 1971 extended the asymptotic type of nonequilibrium exchange to include an axial dispersion term in the fluid phase by applying shock layer theory. The mass transfer to the solid phase was based on ionic diffusion through a fictitious "solid film" utilizing an effective film coefficient; however non-linear equilibria of the Langmuir type was assumed in one example. Transient behavior of the column operation was studied by numerical solution of the basic partial differential equations and the profiles compared to those
given by asymptotic shock theory solution. The concentration profile shape and location in the column coincided with that predicted by shock theory as steady state conditions were approached in the "infinite" column. This approach led ultimately to the more general case developed by Bradley and Sweed (8) for multicomponent systems in 1975. All of these asymptotic approaches are, in effect, equilibrium cases since transient behavior cannot be portrayed. However, in systems where constant concentration patterns are attained, they give good representation of profile shape and location in the column of each specie's concentration plateau with minimum calculation effort.

4. Nernst-Planck Representation of Diffusion in the Resin Phase

4.1 Concentration Driving Force

Helfferich and Plesset in 1958 proposed application of the Nernst-Planck equation (42) to describe the interdiffusivity relationship between counterions in binary exchange. By assuming (1) no coion intrusion into the resin, (2) a condition of electroneutrality at all points in the resin, and (3) that no electric currents are present, the following equation results:

\[
\bar{D}_{12}^{\text{eff}} = \left( \bar{D}_1 \cdot \bar{D}_2 \cdot (\bar{x}_1 \cdot |z_1| + \bar{x}_2 \cdot |z_2|) \right) \left( \frac{2}{\sum_{i=1}^{2} \bar{x}_i \cdot |z_i|} \bar{D}_i \right)
\]

(1-16)

Hering and Bliss (46) measured the diffusion rates of various ions in Dowex 50W resins having varying degrees
of crosslinking. The batch type experiments were run under conditions which would minimize film resistance effects (high stirrer speed and high (0.2 - 1.0 normal) solution strength). The Nernst-Planck model was compared to Fick's Law for its ability to represent the experimental data. Hering and Bliss state that there was no observable better fit of the data by the Nernst-Planck model; however, the Nernst-Planck model was theoretically superior in that the same ionic species "self diffusion" coefficients fit both favorable and unfavorable exchange of a given ion pair. Utilizing Fick's Law, different coefficient values are required to fit each exchange direction. The Nernst-Planck "self diffusion" coefficient for a given ionic species was different, however, depending on the nature of the counterion species for which it was being exchanged.

Rao and David (87) investigated the efficacy of the Nernst-Planck model vs. Fick's Law in single particle studies on the system Cu\(^{++}\)-Na\(^+\) on Dowex 50W X8 resin. Their model included the effect of film resistance (without electric field), and the Nernst-Planck model fit the data reasonably well in the regions where the film diffusion resistance was clearly not a factor; whereas the Fick's Law model did not.

Kuo and David (56) in 1963 showed with single particle studies on the system Ba\(^{++}\)-Na\(^+\) on
Dowex 50W X8 that a numerical model, which included coupled film resistance with Nernst-Planck representation of resin phase diffusion, simulated the experimental curves well.

Morig and Rao (89) in 1965, using a similar procedure for the system Sr⁺⁺-Na⁺ on Dowex 50W X8 resin, found that single "self diffusion" coefficients for each ionic species would not fit both the favorable and unfavorable exchange curves satisfactorily. However, they had used the Helfferich-Plesset (42) model which was based on isotopic exchange (i.e., no selectivity) and did not include film resistance, so the discrepancy is not surprising.

Turner, Church, Johnson and Snowdon (129) in 1965 measured the interdiffusion coefficients for the exchange of H⁺ and Na⁺ ions on Zeo-Karb 225-8%, using batch experiments. Progress of the exchange was monitored by solution conductivity. They found good agreement of the Nernst-Planck equation, based on the Helfferich-Plesset model (42) with the experimental data. More recently, as indicated in section A, part 2 of this chapter, the experimental work of Bajpai, Gupta and Rao was cited. Their batch model included a film resistance to mass transport as well as the Nernst-Planck model for ion exchange in binary and ternary systems. They derived the flux equations for
ternary ion exchange based on the Nernst-Planck equations, and these are given in Chapter V of this work. In a later paper, Rao and his coworkers (94) derived the ternary system flux equations from principles of irreversible thermodynamics. The resulting final equations are identical since they assumed conditions of ideal solution in the resin phase, i.e., $\bar{\mu}_i = RT\ln(\bar{c}_i)$. In general, the results for the binary simulations showed that the Nernst-Planck model was effective in representing the concentration histories for a variety of ions tested. The agreement with the ternary system data was just fair, so only qualitative conclusions could be drawn. The Fick's Law model was not used in the comparisons, since the investigators felt that the superiority of the Nernst-Planck model had been adequately demonstrated by others. However, in a later study Gupta and Rao (95) used isotopic exchange techniques, in which tagged Mn$^{++}$ ion was exchanged in the presence of Ba$^{++}$, Sr$^{++}$, Cs$^+$, and Na$^+$ ions, and found that the effective diffusivity of Mn$^{++}$ changed as a function of the concentration of Mn$^{++}$ ion in the resin. Their explanation was that as the water takeup of the resin increased as a function of ionic composition, the effective diffusivity of the exchanging isotopes increased as well. As will be developed in
this work, use of activity driving force versus concentration driving force should minimize the dependency of effective ionic diffusivity on the resin water content, since the equilibrium resin water content is a function of system excess free energy.

Kataoka, Yoshida, and Ozasa (52) in 1977 conducted experiments in shallow beds on the system \( \text{Na}^+/\text{H}^+/\text{OH}^- \) and \( \text{Na}^+/\text{Ce}^{3+}/\text{NO}_3^- \) on various resins with differing degrees of crosslinking. Because of the neutralization reaction, the R-H/NaOH system has an infinite equilibrium constant. Their Nernst-Planck based model did not consider axial diffusion in the electrolyte phase, nor was film mass transport resistance considered. The agreement of the simulated results with the experimental curves was excellent.

The work of Erickson (26) is outlined in part section A, part 2 of this chapter, as is the work of Omatete (78). Since their data was utilized along with that of Rao and his coworkers, the detailed analysis of their work is contained in the body of this dissertation.

4.2 Chemical Potential Driving Force

Helfferich (45) derived the Nernst-Planck relationship in which chemical potential, \( \mu_i^c \), is used as the driving force versus concentration. Barrer and Rees (6) derived the effective diffusion coefficient for exchange of \( \text{Ca}^{++} \) and \( \text{Sr}^{++} \) ions in chabazite type ion
exchangers using irreversible thermodynamics. They showed that when the cross-phenomenological coefficients are set equal to zero, the equation reduces to the Helfferich form for diffusion in membranes. The additional terms to be added to the Nernst-Planck equation for a binary system are:

$$
\frac{\partial n}{\partial x_1} \left( \sum_{i=1}^{2} \frac{\gamma_i |z_i|}{D_i} \right) \text{ molar}
$$

The derivation for this term is given in Chapter V, along with the derivation in this work of the Nernst-Planck ternary system model in terms of chemical potential utilizing the activity coefficients of ionic species 1, 2, and 3. In order to use this extended Nernst-Planck model, the molar activity coefficients must be known as a function of resin phase composition throughout the binary or ternary system phase diagram. No literature was found dealing with the use of this model to simulate the dynamics of ion exchange.

5. Other Model Considerations Not Included in this Work

5.1 Resin phase mass transfer in ion exchange accompanied by a substantial change in resin volume was studied by Kataoka and Yoshida (49). Four factors come into consideration: (1) the diffusion length changes, (2) specific ion diffusion coefficient values change, (3) the fixed ionic groups in the resin move, and (4) the
activity gradient is altered. The investigators ignored the fourth effect since they did not have an activity driving force model. The expansion or shrinkage of the particular resin must be measured as a function of the concentration of one of the ions, $\bar{x}_i$, and the volume ratio fitted to a polynomial in $\bar{x}_i$. The equations are given which represent the standard Nernst-Planck continuity equation plus a term for resin volume change as a function of composition. Systems exhibiting up to 40% volume change such as $\text{Ba}^{++}-\text{Na}^+$ and $\text{Ba}^{++}-\text{H}^+$ exchange on DIAION SK1B X8 resin were measured for exchange kinetics using single particle tracer techniques. The revised Nernst-Planck model fit the experimental concentration history curves significantly better than the Nernst-Planck model without volume correction.

5.2 Effects of Sorbent Shrinking and Swelling on Fixed-Bed Sorption Operations

Marra and Cooney (17) studied this effect and found that due to effects related to resin shrinkage or swelling as discussed above, the physical bed volume may change by up to 40%. Experimental relationships must be developed for resin diameter change and bed void volume change as a function of composition in the resin. Using the simplifying assumptions of no axial dispersion in the fluid phase, and a "film model" for mass transport
in the resin, the fluid phase continuity equation was developed and solved numerically. The experimentally measured column behavior for the physical adsorption of NaCl into a solute free bed is predicted accurately by the dynamic model.

5.3 Prediction of Effects of Coion Intrusion and Formation of a Stable Complex with the Displaced Counterion

Chmutov, Kalinachev, and Semenovskaya (15) have developed a model in which the Nernst-Planck includes coion diffusion into the resin under conditions of electric neutrality, and the displaced ion, say Ni\(^{++}\) for Na\(^{+}\), forms a stable NiCl\(_2\) complex. This leads to the movement of a sharp Ni\(^{++}\)-Na\(^{+}\) boundary which moves toward the resin core, in a shrinking core model manner. The model fit the experimental data and, since coion intrusion is a factor, the exchange rate increased as solution concentration was increased.

5.4 Intraparticle Ion Exchange Accompanied by Instantaneous Irreversible Reaction

Kataoka, Yoshida, and Ozasa (51) have investigated ion exchange within a resin particle in which the ion exchanged reacts immediately and irreversibly with electrolyte phase coion which has intruded into the ion exchange resin. In this case the diffusion of the coion must be considered, along with the usual considerations of electroneutrality and no electric current. In
experiments involving $\text{H}^+$ form of resin exchanging with $\text{Na}^+$ in the presence of $\text{OH}^-$, water forms to give a stable reaction product. Similarly, $\text{CH}_3\text{COO}^-$ forms of resin exchanging with $\text{HCl}$ formed a stable reaction product in acetic acid. This model fit the experimental data better than the Nernst-Planck model which assumed no coion intrusion. Of course, this somewhat special circumstance of condition limits the models widespread application, but builds on the validity of the Nernst-Planck model and conditions of utility.

5.5 Prediction of Resin Phase Diffusion When Accompanied by Irreversible Exchange and Finite Rates of Reaction

Rao, Gupta, Williams, and Aguwa (96) found that chelating resins coordinate metals irreversibly at a given electrolyte pH, and that, in many cases, the rate of reaction is a significant factor in the exchange process. Using Levenspiel's (60) shrinking core model they successfully simulated exchange of $\text{Pb}$ and $\text{Cd}$ on Dowex A-1.

5.6 Use of Multi-Suffix Diffusion Coefficients in the Nernst-Planck Model

Graham and Dranoff (30) have applied the Stefan-Maxwell equations to ion exchange involving binary systems. The resulting equations are given in Chapter V. By use of these equations with the Nernst-Planck model the authors satisfactorily fit the
experimental concentration history curves for Na\(^+\) displacing Cs\(^+\) and H\(^+\) displacing Na\(^+\) on Dowex 50W X8 resin. The "effective" Nernst-Planck diffusion coefficients are each determined as functions of three parameters, as shown in Chapter V.

C. Other Relevant Literature

1. Literature Search Methodology

The author was assisted in compiling an extensive bibliography on ion exchange and related phenomena by technical information specialists at Allied Corporation. Working with these specialists, a search strategy based on key word and reference-to-reference traceback techniques was worked out to obtain the most pertinent information. Allied searches on-line via the DIALOG INFORMATION SERVICES on the following data bases:

- Chemical Abstracts 1967-date
- Compendex 1970-date (the machine-readable version of the Engineering Index)

Abstracts of relevant article were provided to the investigator based on key word search. Circled abstracts were then obtained from the Allied collection or via outside resources provided by the Allied library. References from the full article many times lead to other relevant articles, also provided by the library.
A. General Background

1. Ion Exchange Equilibrium Conventions

The simulation of fixed bed ion exchange operations requires knowledge of the equilibrium relationship between ionic species contained in the electrolyte phase and those contained in the ion exchange resin. Helfferich (36) has reviewed extensively the many ion exchange equilibria representations, primarily for binary systems. The equilibrium conventions given by Helfferich are as follows:

for the following general binary exchange,

\[ z_i^A A + z_j^B B \rightleftharpoons z_i^A + z_j^B \]

in which:
- \( z_i \) = absolute charge of \( i \)th species
- \( A \) = favored counterion; \( \bar{\cdot} \) represents the resin phase

a) separation factor

\[ \alpha_b^a = \frac{m_a m_b}{m_b m_a} = \frac{C_a C_b}{C_b C_a} = \frac{x_b x_a}{x_a x_b} \]

(1) Although charge on an ionic species is not always the valence of the species (ex., \( \text{NH}_4^+ \)), henceforth in this treatise the commonly used and misapplied term valence will be used as "shorthand" for charge on the defined species.
b) mass action forms

\[ K_{\text{molal}}^a = \frac{z_a}{z_b} \cdot \frac{\gamma_a}{\gamma_b} \quad \text{molal selectivity coefficient} \quad (2-3) \]

\[ K_{\text{molar}}^a = \frac{z_a}{z_b} \cdot \frac{\gamma_a}{\gamma_b} \quad \text{molar selectivity coefficient} \quad (2-4) \]

\[ nK_{\text{equiv}}^a = \frac{z_a}{z_b} \cdot \frac{\gamma_a}{\gamma_b} \quad \text{equivalent ionic fraction or rational selectivity coefficient} \quad (2-5) \]

\[ nK_{\text{equiv}}^a = \frac{\gamma_a}{\gamma_b} \quad \text{or rational selectivity coefficient} \quad (2-5) \]

c) corrected mass action forms

The example given is for the molal selectivity coefficient, but the format applies to all three of the above coefficients:

\[ K_{\text{equiv}}^a = \frac{\gamma_a}{\gamma_b} \quad \text{or rational selectivity coefficient} \quad (2-6) \]

where \( \gamma_i \) = molal activity coefficient of species \( i \) in the electrolyte phase

d) thermodynamic equilibrium constant

molal form

\[ tK_{\text{molal}}^a = \frac{z_a}{z_b} \cdot \frac{\gamma_a}{\gamma_b} = \left( \frac{m_a}{m_b} \cdot \gamma_a \right) \cdot \left( \frac{m_b}{m_a} \cdot \gamma_b \right) \quad (2-7) \]
and the molal scale is used for both the resin and electrolyte phases.

Referring to the diagram below:

\[ \Delta G^0 = -RT \ln \left( \frac{t_k^a}{b} \right) \]  

(2-8)

where

\[ f_i^0 \text{ or } \bar{f}_i^0 \]

Illustration 2.1

Infinite Dilution Standard State Relationships

\[ f_i \text{ or } \bar{f}_i \]

\[ f_i^\infty \text{ or } \bar{f}_i^\infty \]

\[ f_i \]

\[ f_i^0 \]

\[ f_i^\infty \]

\[ m_i \]

\[ m_i^0 \]

\[ \bar{m}_i \]

\[ \bar{m}_i^0 \]

\[ \gamma_i = \frac{f_i}{f_i^0} = \frac{f_i}{m_i f_i^0} \]  

(2-9)

\[ \hat{\gamma}_i = \frac{\bar{f}_i}{f_i} = \frac{\bar{f}_i}{\bar{m}_i \bar{f}_i^0} \]  

(2-10)

\[ \gamma_i \]
So as \( m_i \) or \( \bar{m}_i \rightarrow 0 \) concentration
\[
\bar{f}_i^\infty = \bar{f}_i^\infty \quad \text{and} \quad f_i^\infty = f_i^\infty
\]
resulting in \( \gamma_i = 1 \) and \( \bar{\gamma}_i = 1 \)
The activities, \( a_i \) and \( \bar{a}_i \) are zero for values \( m_i \) and \( \bar{m}_i \) equal to zero concentrations, since \( f_i^\infty \) and \( \bar{f}_i^\infty \) are also zero. If the standard and reference states for the electrolyte and the resin phase are chosen so that they are the same, then:
\[
\begin{align*}
\tau_{ik} &= \left( \frac{f_a}{f_a} \right) \cdot \left( \frac{f_b}{f_b} \right) = 1 \\
\text{since} \quad f_a &= f_a \quad \text{and} \quad f_b = f_b
\end{align*}
\]
Therefore with this convention, the ion exchange selectivity is vested in the activity coefficients \( \bar{\gamma}_i \) and \( \gamma_i \).

Rational form
\[
\begin{align*}
\xi_{ik} &= \left( \frac{n_{-a}}{n_{-a}} \right) \cdot \left( \frac{a_{-b}}{a_{-b}} \right) \\
&= \left( \frac{m_a \gamma_a}{x_a \gamma_a} \right) \cdot \left( \frac{m_b \gamma_b}{x_b \gamma_b} \right) \\
&= \left( \frac{z_{-b}}{z_{-b}} \right) \cdot \left( \frac{z_a}{z_a} \right)
\end{align*}
\]
where \( x_i \) is the ionic equivalent fraction of species \( i \) in the exchanger phase, and the resin is treated as a solid solution of the swollen resinates \( \text{AR} \) and \( \text{BR} \). Both the standard and reference states of \( \text{AR} \) and \( \text{BR} \) are taken to be
the respective monoionic forms of the ion exchanger in
equilibrium with water. Referring to the diagram which
follows:

Illustration 2.2

Standard State Relationships for Resin Phase

\[
\eta_{Y_i} = \frac{n_{F_i}^f}{n_{F_i}^0} = \frac{n_{F_i}^f}{x_i n_{F_i}^0} \quad (2-13)
\]

and

\[
\eta_{a_i} = x_i \eta_{Y_i} = \frac{n_{F_i}^f}{n_{F_i}^0} \quad (2-14)
\]

Values of activity coefficients at the limits are:

- \(\eta_{Y_i}\) at \(x_i = 1.0\), \(n_{F_i}^f = n_{F_i}^0\), so \(\eta_{Y_i} = 1.0\) and \(\eta_{a_i} = 1.0\)

- \(\eta_{Y_i}\) at \(x_i \to 0\), slope \(K = \frac{(n_{F_i}^f)_{i+0}}{(x_i)_{i+0}} = \frac{n_{F_i}^f}{x_i_{i+0}} \quad (2-15)\)

and

\[
\eta_{Y_i} = \frac{n_{F_i}^0}{n_{F_i}^0 x_i_{i+0}} \quad (2-16)
\]

so

\[
\frac{\eta_{Y_i}}{n_{F_i}^0} = \frac{\eta_{Y_i}}{n_{F_i}^0} \frac{k_{x_i}^{+0}}{n_{F_i}^0 x_i_{i+0}} = \frac{K}{n_{F_i}^0} \quad (2-16)
\]
The rational form of the thermodynamic equilibrium constant is then given as:

\[ n_{Y_i}^{\infty} = \frac{K}{n_{F_i}^0} \quad \text{and} \quad n_{A_i}^{\infty} = 0 \quad \text{as} \quad x_i \to 0 \]

The rational form of the thermodynamic equilibrium constant is then given as:

\[ n_{K^a}^{\infty} = \left( \frac{n_{F_i}^0}{F^0_a} \right)^{\frac{z_b}{2}} \cdot \left( \frac{F^0_b}{n_{F_i}^0} \right)^{\frac{z_a}{2}} \]  

\[ (2-17) \]

Since the standard and reference states of the electrolyte and resin phases are different, the ion exchanger selectivity is vested in the equilibrium constant. The activity coefficients mainly trim up compositional variations.

Complete definition of the Thermodynamic Equilibrium Relationship

The above expressions for the thermodynamic equilibrium constant were based on the assumption that electrolyte and solvent sorption and desorption were small and could be neglected.

The full expression (39) is:

\[ tK^a_b = \left( \frac{m_{Y_a}^b}{m_{Y_b}^a} \right)^{\frac{z_b}{2}} \cdot \left( \frac{m_{Y_b}^b}{m_{Y_a}^a} \right)^{\frac{z_a}{2}} \cdot \left( \frac{a_{w}}{a_{w}} \right)^h \left( \frac{a_{w}}{a_{w}} \right)^g \left( \frac{a_{by}}{a_{by}} \right)^f \]  

\[ (2-18) \]

where:

- \( w \) is solvent, \( ay \) is salt \( AY \), and \( by \) is salt \( BY \), with \( h, g, \) and \( f \) representing \( \Delta \) moles of each species between the solvent and exchanger phase.
Chain Rule Relationship

The thermodynamic equilibrium constants, both the molal and the rational form, satisfy the Gibbs free energy equilibrium constraints in that the activity relationship of each species, in both phases (resin and electrolyte), are related by a constant at a given temperature. This leads to the chain rule convention in which:

\[ K_{ij} = \frac{K_{ik}}{K_{jk}} \left( \frac{|z_j|}{|z_k|} \right) \]

for species i, j, and k (2-19)

in which the equilibrium constants \( K_{ij} \) can be all molal forms, or all rational forms, and \( z_i \) represents valence of species i.

2. Ion Exchange Equilibrium Models

2.1 Pressure and Temperature Considerations

In terms of the swelling pressure of the resin and the partial molar volume of each species, the electric double layer or Donnan potential created by electrolyte sorption is defined by Helfferich (35) as:

\[ E_{DON} = \frac{1}{z_i F} \left\{ RT \ln \left( \frac{a_i}{\bar{a}_i} \right) - \pi \nu_i \right\} \]

(2-20)

where:

\( F \) = Faraday's constant

\( \pi \) = swelling pressure

\( \nu_i \) = partial molar volume of species i
For binary exchange this leads to (38):

\[
\ln k_D^a = \frac{\pi}{RT} \left\{ z_a \left( \nu_b - \nu_a \right) - h_{w} w - g_{w} w + f_{w} w \right\} \tag{2-22}
\]

which shows the effect of swelling pressure and partial molar volume on the thermodynamic molal equilibrium constant. Barrer and Rees (6) point out that in a binary exchange of ions in zeolites, the Gibbs-Duhem relationship is:

\[
\bar{n}_a \Delta \bar{n}_a + \bar{n}_b \Delta \bar{n}_b + \bar{n}_w \Delta \bar{n}_w + \bar{n}_{\text{sites}} \Delta \bar{n}_{\text{sites}} = - \frac{\partial (\pi \nu)}{RT} \tag{2-23}
\]

where

- \( n_i \) = moles of species i
- subscripts w referring to water, sites referring to charges in the resin
- \( \nu \) = resin volume, cc
- \( \pi \) = osmotic or swelling pressure, bars

Barrer and Rees (6) did not include coion sorption since in zeolites this tendency is very small.

Barrer and Rees point out that in zeolites, \( \bar{n}_{\text{sites}} \) is constant and water change during ion exchange is small and can be said to be constant. For univalent exchange,
\[ \frac{dn_a}{dn_a} = -dn_b, \text{ so:} \]

\[ \frac{R}{n_b} \frac{dln_{\alpha}}{dn_a} - \frac{R}{n_b} \frac{dln_{\alpha}}{dn_b} = - \left( \frac{R}{n_w} \frac{dln_{\alpha}}{dn_a} \right) \]

\[ + \frac{R}{n_{\text{sites}}} \frac{dln_{\alpha}}{dn_a} + \frac{T}{RT} \frac{dln_{\alpha}}{dn_a} \]

In order for

\[ \frac{dln_{\alpha}}{dn_a} = \frac{dln_{\alpha}}{dn_b} \] (2-25)

the right side of equation (2-23) must be zero. This assumption is implicit in most ion exchange models, even for gel exchangers where osmotic pressure and volume changes may be significant during the exchange, and indeed, equation (2-22) shows that if the thermodynamic equilibrium constant is to remain constant throughout a complete exchange of one species for another, then the volume and pressure changes must be relatively small.

Helfferich (37) points out that in the free energy expression:

\[ \Delta G = \Delta H - T\Delta S \] (2-26)

that the enthalpy term, \( \Delta H \), is usually small. Most of the free energy changes in ion exchange are due to entropic effects, primarily mixing effects but to some extent due to configurational changes in the resin matrix and the adding or losing of solvation shells by the ionic species. The temperature dependence of ion exchange is similar to that...
describing chemical reactions, so:

\[
0 \left( \frac{d\ln n_k^a}{dT} \right)_p = \frac{\Delta H^0}{RT^2}
\]  

(2-27)

where \(\Delta H^0\) = standard enthalpy of reaction

As Helfferich (37) points out, normal \(\Delta H^0\) values for ion exchange are only on the order of 2 kcal/mole, so ion exchange equilibria is not affected greatly by temperature changes.

A number of models for ion exchange equilibrium are discussed by Helfferich (41).

2.2 Resin Phase Activity Coefficient Models

The development of a model which can represent the activity coefficients of ionic species in an ion exchange resin depends on the assumptions made in developing the Gibbs-Duhem equation. As Soldatov (119) points out, the Gibbs-Duhem equation for the resin phase has the general form:

\[
\sum_{i=1}^{k} \tilde{n}_i \tilde{d\tilde{u}}_i = 0
\]

(2-28)

This of course assumes, as discussed above, that the pressure volume effects are negligible, and all references found in this work have included that assumption as inherent in their various model derivations for equilibrium in ion exchanges.

Soldatov (119) made one very good observation
pertaining to ion exchange thermodynamics, and that has to
do with the use of mole fractions or equivalent fractions
as surrogates for molal or molar concentrations in the
Gibbs-Duhem equation. In systems where the total numbers
of moles is a constant, i.e., physical systems with no
reactions, the Gibbs-Duhem for a binary system would be:

\[ \bar{c}_1 \, d\ln \bar{n}_1 + \bar{c}_2 \, d\ln \bar{n}_2 = 0 \quad (2-29) \]

or

\[ \bar{c}_1 (d\ln \gamma_1 + d\ln \bar{c}_1) + \bar{c}_2 (d\ln \gamma_2 + d\ln \bar{c}_2) = 0 \quad (2-30) \]

then

\[ \bar{c}_1 \, d\ln \gamma_1 + \bar{c}_2 \, d\ln \gamma_2 = 0 \quad (2-31) \]

since \( \bar{c}_1 + \bar{c}_2 = \bar{c}_{\text{total}} \), \( d(\bar{c}_1 + \bar{c}_2) = 0 \)  

Soldatov's equation for ion exchange was given as:

\[ \sum_{i=1}^{n} \bar{c}_i \, d\ln \gamma_i + d\bar{c}_{\text{total}} = 0 \quad (2-33) \]

where in heterovalent binary exchange, \( d\bar{c}_{\text{total}} \neq 0 \)

but is equal to \(- \left( \frac{|z_1|}{|z_2|} \right) \cdot d\bar{c}_1 \), since:

\[ \bar{c}_{\text{total}} = \left( \frac{Q_r}{|z_2|} \right) + \left( 1 - \left( \frac{|z_1|}{|z_2|} \right) \right) \cdot \bar{c}_1 \quad (2-34) \]

where \( Q_r \) = equivalent sites in ion exchanger

In homovalent systems for which \( z_1 \) equals \( z_2 \), the total

moles, \( \bar{c}_{\text{total}} \), equals \( \left( \frac{Q_r}{|z_2|} \right) \) and the term, \( d\bar{c}_{\text{total}} \)
equals zero. However, in heterovalent systems such as the

system \( \text{Sr}^{++}-\text{Cs}^+ \), \( d\bar{c}_{\text{total}} \) is equal to \(-2d\bar{c}_1 \) and also
is equal to \( d\bar{c}_2 \) (since \( d\bar{c}_1 = \frac{|z_2|}{|z_1|} d\bar{c}_2 \)), and is not zero. Most investigators of ion exchange equilibria models either limit the model to homovalent exchange, or ignore the discrepancy. The residual values \( d\bar{c}_{total} \) get "buried" in the equilibrium constant or "absorbed" by the activity coefficient correlation—as was the case in this research. As will be discussed later, this works out very well in correlating binary heterovalent system equilibrium relationships, but may be responsible for the just fair ability of the correlation methods to predict ternary equilibrium concentrations from binary system parameters in such circumstances.

Two resin phase activity coefficient models will be discussed here since they led to the approach taken in this work. These are:

a. **Kielland's Equation**

Kielland's equation (53) for zeolites for homovalent exchange, neglecting water and electrolyte sorption as discussed, and assuming constant temperature and pressure:

\[
B + A \overset{\leftrightarrow}{\rightleftharpoons} BZ + A
\]  
(2-35)

The relationship of excess free energy to activity coefficients for a binary system is:

\[
\ln^o \gamma_1 = \frac{\Delta G^e_{ex}}{RT} - x_2 \left\{ \frac{\Delta G^e_{ex}}{RT} \right\}_T \delta x_2
\]  
(2-36)
In many regular systems ($\Delta^{nT}_{S}ex = 0$) the representation:

$$\frac{\Delta^{nT}_{H}ex}{RT} = A \bar{x}_1 \bar{x}_2$$

(2-37)

provides a reasonable representation of $\Delta^{nT}_{G}ex$. This is Kielland's approach and he defined:

$$\bar{x}_i = \frac{\bar{n}_{iz}}{(n_{az}n_{bz})}$$

(2-38)

where $\bar{n}_{iz}$ are moles/volume of component iz

By use of equation (2-36) and (2-37) he arrived at the truncated Redlich-Kister form of:

$$\ln n_{Y_1} = A \bar{x}_2^2$$

and

$$\ln n_{Y_2} = A \bar{x}_1^2$$

(2-39, 40)

to correlate resin phase activity coefficients in the resin phase. This approach was put on a more sound theoretical footing by Barrer (5).

b. Smith and Woodburn Model

Smith and Woodburn (116) were concerned with multicomponent ion exchange equilibria, and developed an approach to model the anionic exchange of $NO_3^-$, $Cl^-$, $SO_4^{2-}$ with Rohm and Haas Amberlite 400 resin. Although Danes and Danes (19) had investigated approaches where the excess free energy of mixing in polyionic systems was expressed as a polynomial, (Redlich-Kister type representations), Smith and Woodburn felt that this confined the relationship to regular solutions, and binary constants could not readily be utilized to predict multicomponent system activity coefficients. They noted that some features
of the ion exchange system lent themselves to the approach used by Wilson (98) in development of his equation based on non-randomness of local mole fractions in liquid mixtures for athermal solutions. As discussed above, ion exchange systems exhibit low heats of mixing and analogy can be made to non-random clustering of ion species in the exchanger phase. The Wilson equation in general form is:

\[ \ln \gamma_i = 1 - \ln \sum_{j=1}^{n} x_j G_{ij} - \sum_{j=1}^{n} \frac{x_j G_{ij}}{\sum_{k=1}^{n} G_{jk} x_k} \]  

(2-41)

where

\[ G_{ij} = \rho_{ij} \exp(-\tau_{ij}) \]  

(2-42)

and

\[ \tau_{ji} = \frac{(g_{ji} - g_{ii})}{RT} \]  

(2-43)

where

\[ g_{ji} = g_{ij}, \text{ so } \tau_{jj} = \tau_{ii} = 0 \]  

(2-44)

and

\[ \rho_{ij} = \frac{\nu_i}{\nu_j}, \text{ molar volume ratio} \]  

(2-45)

The Wilson equation was used by Smith and Woodburn to describe the resin phase interactions involved in the ternary system NO\textsubscript{3}\textsuperscript{-}, Cl\textsuperscript{-}, SO\textsubscript{4}\textsuperscript{2-}, based on predictions made from the binary pair correlations. Smith and Woodburn based the electrolyte phase ionic species activity coefficients on the extended Debye-Hückel equation given by Robinson and Stokes (103);
where $I = ionic$ $strength$

and used the Truesdell and Jones (128) parameters for $a_i$ and $b_i$. The binary pair data was regressed using the least square of normalized error criteria and they represented the system with the rational thermodynamic equilibrium treatment, $\eta_{a}^{a}$ given in equation (2-12) above.

The combinatorial rules are straight forward for multicomponent mixtures utilizing binary pair $G_{ij}$ fitted parameters, and is not subject to possible ambiguity, as is the combining of polynomial expressions derived for binary pairs into ternary representations. The results of the application of this model to predict the ternary compositions versus actual experimental results was quite good, with a standard deviation of ±4.26%.

One complication in considering the $NO_3^-, Cl^-, SO_4^-$ system is that:

$$Na^+ + SO_4^- \leftrightarrow NaSO_4^-$$

where:

$$K_{dist} = \frac{a^*-a^*}{a^*+a^*} = 5.248$$

and was considered as an additional reaction in the electrolyte phase. Since total $SO_4^-$ in the resin
phase was assayed, it was not possible to determine the amount of $\text{SO}_4^{2-}$ due to the $\text{SO}_4^{2-}$ source or the $\text{NaSO}_4^-$ source without a $\text{Na}^+$ analysis. In any event, a ternary system was modeled, not a quaternary system, and the simplest approach was to treat $\text{NaSO}_4^-$ as a non-exchanged species.

Smith and Woodburn used the chain rule:

$$\frac{n_{K^1}}{t^{2-2}} = \frac{n_{K^1}|z_2|}{n_{K^3}|z_3|}$$

(2-49)

to determine the value of $\frac{n_{K^1}}{t^{SO_4}}$ from the regression values of $\frac{n_{K^1}|Cl|}{t^{Cl}}$ and $\frac{n_{K^1}|NO|}{t^{NO}}$ for comparison to the fitted value of $\frac{n_{K^1}}{t^{SO_4}}$ and found excellent agreement.

They felt this was an indication of data consistency.

3. Tabulation of Thermodynamic Models

Table 2-1 shows a compilation of various thermodynamic representations used by other investigators in modeling either the equilibria or the dynamics of ion exchange. Dranoff and Lapidus (21, 22), Rao and his coworkers (91, 92, 93), and Omatete (78, 79, 80) are the only investigators cited here who modeled dynamic ternary exchange, and therefore required ternary equilibrium representation; however, others such as Smith and Woodburn (116), Soldatov and Bychkova (121, 122, 123), and Pieroni and Dranoff (85) have investigated ternary equilibria representation with a variety of approaches.
Table 2-1
ION EXCHANGE EQUILIBRIA - VARIOUS INVESTIGATORS

<table>
<thead>
<tr>
<th>Investigator(s)</th>
<th>System</th>
<th>Equilibrium Convention</th>
<th>Activity Coefficients</th>
<th>Electrolyte</th>
<th>Resin Phase</th>
</tr>
</thead>
<tbody>
<tr>
<td>Binary</td>
<td></td>
<td></td>
<td></td>
<td></td>
<td></td>
</tr>
<tr>
<td>(1) David &amp; Kuo (56)</td>
<td>Ba^{++}-Na^{+}</td>
<td>Mass action- Rational Selectivity</td>
<td>ideal</td>
<td>ideal</td>
<td></td>
</tr>
<tr>
<td></td>
<td>for equilibrium fit</td>
<td></td>
<td></td>
<td></td>
<td></td>
</tr>
<tr>
<td>(2) David &amp; Rao (88)</td>
<td>Cu^{++}-Na^{+}</td>
<td>Mass action- Molal Selectivity</td>
<td>literature</td>
<td>see below*</td>
<td>Harned &amp; Owen (33)</td>
</tr>
<tr>
<td></td>
<td>for equilibrium fit</td>
<td></td>
<td></td>
<td></td>
<td></td>
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<tr>
<td></td>
<td></td>
<td></td>
<td>*used version of Harned's Rule:</td>
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<tr>
<td></td>
<td></td>
<td></td>
<td>( \frac{\gamma_1}{\gamma_1} \sim \frac{\gamma_2}{\gamma_2} )</td>
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<tr>
<td></td>
<td></td>
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<td>to activity coefficients in the electrolyte</td>
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<td></td>
<td></td>
<td></td>
<td>i.e., activity</td>
<td></td>
<td></td>
</tr>
<tr>
<td>(3) Erickson (26)</td>
<td>Ethylene^{++}-diamine^{++}-NH_{4}</td>
<td>total empirical equation (see Chapter VII)</td>
<td></td>
<td></td>
<td></td>
</tr>
<tr>
<td>for dynamic simulation</td>
<td></td>
<td></td>
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<td></td>
<td></td>
</tr>
<tr>
<td>(4) Kataoka &amp; Yoshida (49)</td>
<td>Na^{+}H^{+}</td>
<td>Mass action- Molal Selectivity</td>
<td>ideal</td>
<td>ideal</td>
<td></td>
</tr>
<tr>
<td>for dynamic simulation</td>
<td></td>
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<td></td>
<td></td>
</tr>
<tr>
<td>(5) Pan (84)</td>
<td>Na^{+}H^{+}</td>
<td>Mass action- Rational Selectivity</td>
<td>ideal</td>
<td>ideal</td>
<td></td>
</tr>
<tr>
<td>for dynamic simulation</td>
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</tr>
<tr>
<td>(6) Selke &amp; Bliss (113)</td>
<td>Cu^{++}H^{+}</td>
<td>Mass action- Rational Selectivity</td>
<td>ideal</td>
<td>ideal</td>
<td></td>
</tr>
<tr>
<td>for dynamic simulation</td>
<td>Cu^{++}Mg^{+}</td>
<td></td>
<td></td>
<td></td>
<td></td>
</tr>
<tr>
<td>Investigator(s)</td>
<td>System</td>
<td>Equilibrium Convention</td>
<td>Activity Coefficients</td>
<td>Electrolyte</td>
<td>Resin Phase</td>
</tr>
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<td>---------------------------------</td>
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</tr>
<tr>
<td><strong>Ternary</strong></td>
<td></td>
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</tr>
<tr>
<td>(1) Smith &amp; Woodburn (116)</td>
<td>NO$_3^-$-Cl$^-$-SO$_4^{2-}$</td>
<td>Mass action- Rational Thermodynamic</td>
<td>extended Debye-</td>
<td>Wilson</td>
<td></td>
</tr>
<tr>
<td>for equilibrium fit</td>
<td></td>
<td></td>
<td>Huckel, Stokes</td>
<td></td>
<td>Equation</td>
</tr>
<tr>
<td></td>
<td></td>
<td></td>
<td>&amp; Robinson</td>
<td></td>
<td>(Truesdell &amp; Jones)</td>
</tr>
<tr>
<td>(2) Soldatov &amp; Bychkova (121, 122, 123)</td>
<td>K$^+$-NH$_4^+$-H$^+$</td>
<td>Mass action- Modified Rational</td>
<td>Harned's Rule</td>
<td></td>
<td></td>
</tr>
<tr>
<td>for equilibrium fit</td>
<td>Na$^+$-NH$_4^+$-H$^+$</td>
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<td></td>
</tr>
<tr>
<td></td>
<td>Ba$^{++}$-Cu$^{++}$-Co$^{++}$</td>
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<tr>
<td></td>
<td></td>
<td></td>
<td>Activity Coefficients</td>
<td>interpolated from plots for binary pairs</td>
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<tr>
<td>where $\eta_K = 2.5 + 9.3x_H + 0.2\bar{x}_K - 1.6\bar{x}_H\bar{x}_K + 28.0x_H^2\bar{x}_K$</td>
<td></td>
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<td></td>
</tr>
<tr>
<td>for system K$^+$-NH$_4^+$-H$^+$</td>
<td></td>
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<td></td>
<td></td>
<td></td>
</tr>
<tr>
<td>(3) Bajpai, Gupta, Rao (91)</td>
<td>Sr$^{++}$-Mn$^{++}$-Cs$^{++}$</td>
<td>Mass action- Thermodynamic (mol. frac. both phases)</td>
<td>see equations (a) and (b) below</td>
<td></td>
<td></td>
</tr>
<tr>
<td>for equilibrium fit</td>
<td>Mn$^{++}$-Cs$^{++}$-Na$^+$</td>
<td></td>
<td></td>
<td></td>
<td></td>
</tr>
<tr>
<td></td>
<td>Ba$^{++}$-Sr$^{++}$-Na$^+$</td>
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<td></td>
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<td></td>
</tr>
<tr>
<td>(a) $z_i\ln\bar{y}<em>j = -\bar{x}<em>i\ln K</em>{aj}^{-1} + \int</em>{\bar{x}<em>i}^{1} \ln K</em>{aj}^{-1} d\bar{x}_i$</td>
<td></td>
<td></td>
<td></td>
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</tr>
<tr>
<td>(b) $z_j\ln\bar{y}<em>j = +\bar{x}<em>j\ln K</em>{aj}^{-1} - \int</em>{\bar{x}<em>i}^{1} \ln K</em>{aj}^{-1} d\bar{x}_i$</td>
<td></td>
<td></td>
<td></td>
<td></td>
<td></td>
</tr>
<tr>
<td>$\ln K_{aj}^{-1} = \int_{0}^{1} \ln K_{aj}^{-1} d\bar{x}_i$</td>
<td></td>
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<td></td>
<td></td>
</tr>
</tbody>
</table>
Table 2-1 (continued)

<table>
<thead>
<tr>
<th>Investigator(s)</th>
<th>System</th>
<th>Equilibrium Convention</th>
<th>Activity Coefficients</th>
</tr>
</thead>
<tbody>
<tr>
<td>(3) Bajpai, Gupta, Rao (91)</td>
<td>for equilibrium fit (cont.)</td>
<td>Electrolyte</td>
<td>Resin Phase</td>
</tr>
<tr>
<td>(4) Bajpai, Gupta, Rao (93)</td>
<td>for ternary dynamic simulation</td>
<td>above</td>
<td>ideal</td>
</tr>
</tbody>
</table>

and for ternary

\[
\gamma_i = \sum_{k=1}^{n} \left( \frac{x_{k}}{n \sum_{j=1}^{n} x_{j}} \right) \gamma_{i_{1}} \quad \text{for } \quad i \neq k
\]

\[
\lambda_{1} = \frac{\bar{x}_{1}}{\bar{x}_{1}^{\text{int}}}, \quad i = 1, 2, 3 \quad \text{in eqv. fractions}
\]

binary

\[
\lambda_{1} = a + b \bar{x}_{1} + c \bar{x}_{1}^{2}
\]

\[
\lambda_{2} = \frac{(1-x_{1})}{(1-x_{1}^{\text{int}})}
\]

for ternary, where selectivity is 1,2,3

\[
\left( \frac{x_{1}^{*}}{x_{1}^{*} + x_{2}^{*}} \right) \lambda_{1} = \left( \frac{\bar{x}_{1}}{(\bar{x}_{1} + \bar{x}_{2})} \right) = \bar{x}_{1}
\]

\[
\left( \frac{x_{1}^{*}}{x_{1}^{*} + x_{3}^{*}} \right) \lambda_{1} = \left( \frac{\bar{x}_{1}}{(\bar{x}_{1} + \bar{x}_{3})} \right) = \bar{x}_{1}
\]

see equations (a), (b) and (c) below

(a), (b) and (c) below
Table 2-1 (continued)

<table>
<thead>
<tr>
<th>investigator(s)</th>
<th>system</th>
<th>equilibrium convention</th>
<th>activity coefficients</th>
<th>electrolyte</th>
<th>resin phase</th>
</tr>
</thead>
<tbody>
<tr>
<td>(5) Omatete (78) for ternary dynamic simulation</td>
<td>Ag⁺-Na⁺-H⁺</td>
<td>Separation Factor</td>
<td>ideal</td>
<td>ideal</td>
<td></td>
</tr>
<tr>
<td></td>
<td></td>
<td>(a_{ij} = \frac{x^+_i x^-_j}{x^+_i x^-_j} )</td>
<td>ternary system (ref. is also equated to n)</td>
<td></td>
<td></td>
</tr>
<tr>
<td></td>
<td></td>
<td>(\bar{x}^+<em>j = \frac{\sum</em>{k=1}^{n} a_{ref} x^-<em>k}{\sum</em>{k=1}^{n} a_{ref} x^-<em>k} ) or (x^+<em>j = \frac{\sum</em>{k=1}^{n} a</em>{ref} x^-<em>k}{\sum</em>{k=1}^{n} a_{ref} x^-_k} )</td>
<td></td>
<td></td>
<td></td>
</tr>
<tr>
<td></td>
<td></td>
<td>Rational Selectivity</td>
<td>ideal</td>
<td>ideal</td>
<td></td>
</tr>
<tr>
<td>(6) Pieroni &amp; Dranoff (85) for equilibrium fit</td>
<td>Cu²⁺-Na⁺-H⁺</td>
<td>Mass action-</td>
<td>ideal</td>
<td>ideal</td>
<td></td>
</tr>
<tr>
<td></td>
<td></td>
<td>in equivalent fractions</td>
<td></td>
<td></td>
<td></td>
</tr>
<tr>
<td></td>
<td></td>
<td>(\bar{x}^+_a = \frac{c_a}{c_a + c_b} ) ; (\bar{x}^-_a = \frac{q_a}{q_a + q_b} )</td>
<td>ternary system - use of pseudo binary</td>
<td></td>
<td></td>
</tr>
<tr>
<td></td>
<td></td>
<td>(x^+_a = \frac{c_a}{c_a + c_b} ) ; (x^-_a = \frac{q_a}{q_a + q_b} )</td>
<td></td>
<td></td>
<td></td>
</tr>
</tbody>
</table>
Table 2-1 (continued)

<table>
<thead>
<tr>
<th>Investigator(s)</th>
<th>System</th>
<th>Equilibrium Convention</th>
<th>Activity Coefficients</th>
</tr>
</thead>
<tbody>
<tr>
<td></td>
<td></td>
<td></td>
<td>Electrolyte</td>
</tr>
<tr>
<td></td>
<td></td>
<td></td>
<td>Resin Phase</td>
</tr>
</tbody>
</table>

(6) Pieroni & Dranoff (85)
for equilibrium fit (cont.)

\[
\begin{align*}
K_{\text{Cu}H} &= \frac{\left(x_{\text{Cu}}x_H^2\right)}{\left(x_H^2x_{\text{Cu}}\right)} \cdot \frac{\left(c_{\text{Cu}}^{+}c_{\text{Cu}}\right)}{\left(q_{\text{Cu}}^{+}q_{\text{Cu}}\right)} \\
K_{\text{CuNa}} &= \frac{\left(x_{\text{Cu}}x_{\text{Na}}^2\right)}{\left(x_{\text{Na}}^2x_{\text{Cu}}\right)} \cdot \frac{\left(c_{\text{Na}}^{+}c_{\text{Cu}}\right)}{\left(q_{\text{Na}}^{+}q_{\text{Cu}}\right)}
\end{align*}
\]
B. Equilibria Models Tested – This Report

1. General

Review of the literature and background provided by other investigators led to the conclusions that ternary equilibrium correlating relationships for this work should be based on:

a. The Rational Thermodynamic Selectivity relationship as demonstrated by Smith and Woodburn (116), since equivalent or mole fraction of each species in the resin phase could be directly calculated as opposed to using pseudo binary relationships. This was particularly important, since with the proposed solution method for ternary dynamic modeling the quasilinearization procedure required clear definition of the partials

$$\frac{\partial x_i^{* \text{int}}}{\partial x_j^*} \quad (i = 1,2) \text{ for } (j = 1,2) \quad (2-50)$$

In addition, if the activity coefficients of species in the resin phase could be expressed as functions of $x_i$, then the activity driving force could be investigated as an alternative to concentration driving force.

Use of the Rational Thermodynamic Selectivity allowed application of the chain rule for calculating one $N_{K_j}^i$ in terms of $N_{K_j}^i$ and $N_{K_j}^j$ as described in equation (2-19), and provided a basis for consistency between three binary pairs without ambiguity.

Finally, as discussed for heterovalent systems, some "burying" of the residual $dC_{\text{total}}$ term has to be absorbed by
the activity coefficients. Putting most of the selectivity effects into the equilibrium constant would relieve the function of the activity correlation method chosen to that of primarily representing the excess free energy relationships related to mixing and resin configurational changes.

b. Use of generalized correlations such as the Redlich-Kister three suffix equation, the Wilson equation, and the NRTL equation to relate the excess properties of the resin phase to the concentration of ionic species. Each of these correlations has combinatorial rules for the parameters so that constants derived for the three binary pairs can be combined to yield prediction of ternary system behavior. It would have been desirable to:

- be able to predict the equilibrium constant of the least favored binary pair by application of the chain rule using values of the equilibrium constants for the two most favored binary pairs;
- be able to predict ternary system behavior directly from the binary pair equilibrium constants and activity coefficient correlation parameters utilizing the mixing conventions of the method.

This would minimize the requirement to obtain experimentally many ternary data points and, in the case of the Redlich-Kister equation, obviate the need to fit parameters for ternary systems. If the predictions for ternary systems could be shown to be dependable, then preliminary computer mapping of ternary
regions of maximum interest could be accomplished from the binary parameters—with experimentation on ternary systems limited in scope to check these regions.

c. Use of a correlation package for activity coefficient prediction in the electrolyte phase for single ions in binary and ternary systems. The predictive method would have the ability to:

- predict single ion activities in multicomponent mixtures containing other cationic and anionic species;
- provide a reliable range of accuracy up to about 6 molal concentration of electrolyte salts;
- have extended Debye-Hückel relationships for which constants were available for many ubiquitous salts.

The Bromley (9) equation was selected as the prediction method for the electrolyte phase, and is discussed in Chapter III.

2. Derivation of Equilibrium Relationships

2.1 Equilibrium Equations

Derivation of the binary equilibrium relationships for the three binary pairs in a ternary system is straightforward, given the Helfferich conventions. From this point on, the shorthand symbol for the rational thermodynamic equilibrium constant will be given as \( \bar{K}_{ij} \) in this report; and \( \bar{r}_{Y_i} \) will be given as \( \bar{\gamma}_i \).

\[
\bar{K}_{ij} = \left( \frac{a_i}{a_j} \right)^{z_j} \left( \frac{a_j}{a_i} \right)^{z_i} (2-51)
\]

continued:
where:

\[ \bar{x}_i \] = mole or equivalent fraction in the resin phase of species \( i \) or \( j \) 

\[ \bar{\gamma}_j \] = activity coefficient of species \( i \) or \( j \) in the resin phase as a function of mole or equivalent fraction

\( m_{\text{int}}^* \) = molality of species \( i \) or \( j \) in the electrolyte in equilibrium with the resin

\( \gamma_j \) = activity coefficient of species \( i \) or \( j \) in the electrolyte phase

Since solutions were dilute in this work,

\( c_{i}^* \) = molarity of species (instead of molality)

Equation (2-51) was transformed so that mole fraction or equivalent fraction could be used in the electrolyte phase as well as the resin phase to simplify representation. Equivalent fraction in the electrolyte phase is given as:

\[
0 \quad x_{i \text{int}}^* = \left( \frac{|z_i| c_{i \text{int}}^*}{\text{norm}} \right) \quad \text{(2-52)}
\]
where norm = normality of coions in the electrolyte phase.

Substitution of equation (2-52) in equation (2-51) gives:

\[
K_{ij} = \left( \frac{-x_i \gamma_i}{x_i^* \gamma_i} \right)^{z_j} \cdot \left( \frac{x_j^* \gamma_j}{x_j \gamma_j} \right)^{z_j} \cdot \text{norm} \left( \frac{|z_i| - |z_j|}{|z_i|} \right) \cdot \left( \frac{|z_j|}{|z_i|} \right)
\]

for ij of 12, 13, 23

keeping in mind that \( \gamma_i \) or \( \gamma_j \) are still functions of \( m_i^* \) (or \( c_i^* \)), not \( x_i^* \), as given by the Bromley equation.

Equation (2-53) shows that if mole or equivalent fractions are used for electrolyte phase compositions, then in heterovalent systems the normality of the electrolyte has a strong influence on the ion exchanger selectivity.

2.2 Resin Phase Activity Coefficient Relationships

a. The generalized form of the Wilson equation was given above in the Smith and Woodburn discussion.

b. The NRTL equation or non-random, two liquid model developed by Renon and Prausnitz (98) is given by those investigators as:

\[
\ln \gamma_i = \frac{\sum_{j=1}^{n} \tau_{ij} G_{ij} X_j}{\sum_{k=1}^{n} G_{ki} X_k} + \frac{\sum_{j=1}^{n} X_j G_{ij}}{\sum_{k=1}^{n} G_{ki} X_k} \times \left[ \frac{\sum_{k=1}^{n} X_k \tau_{kj} X_j}{\sum_{k=1}^{n} G_{kj} X_k} \right]
\]
where
\[
\tau_{ji} = \frac{(g_{ji} - g_{ii})}{RT}
\] (2-55)

\[
G_{ji} = \exp(-\alpha_{ji} \tau_{ji})
\] (2-56)

and
\[
g_{ji} = g_{ij}, \text{ so } \tau_{jj} = \tau_{ii} = 0
\] (2-57)

\[
\alpha_{ji} = \alpha_{ij}
\] (2-58)

c. The three suffix Redlich-Kister equation is given by Smith (117) as:

\[
\ln \gamma_r = \sum_{i=1}^{n} B_{ri} X_i (X_r - X_i)^{-1} \{ X_r (1-X_r+X_i) - X_i \} - \sum_{i \neq j \neq r} B_{ij} X_i X_j (X_i - X_j)^0 + \sum_{i=1}^{n} C_{ri} X_i (X_r - X_i)^0
\]

\[
\times \{ 2X_r (1-X_r+X_i) - X_i \} - 2\sum_{i \neq j} C_{ij} X_i X_j (X_i - X_j)^1 + \sum_{i=1}^{n} D_{ri} X_i (X_r - X_i)^1 \{ 3X_r (1-X_r+X_i) - X_i \}
\]

\[
-3\sum_{i \neq j} D_{ij} X_i X_j (X_i - X_j)^2
\]

where \(B_{rr}, C_{rr}, D_{rr}\) terms are zero (2-60)

and \(B_{12} = B_{21}\)

\(C_{12} = -C_{21}\)

\(D_{12} = D_{21}\)

if a fourth suffix is included

\(E_{12} = -E_{21}\)

The constants for ternary systems are also given in this literature source, but the relationships are not
listed. Later in this chapter the discussion will include reference to regression fits of data from ternary systems made early in this research which included a fourth suffix term, E, and the three ternary constants. Along with the equilibrium constants, this required fitting 17 parameters to the experimental data. Since there were only 30 or so data points, the "principle of parsimony" was being encroached upon, and Occam's razor prevailed. This principle of the "simplest explanation which explains phenomena is probably the correct one" (paraphrased) led to working with only a three suffix expression without ternary constants.

2.3 Mole Fractions versus Equivalent Fraction

All of the systems involving heterovalent exchange were tested utilizing both mole fraction and equivalent fraction concentration to determine which system would give better prediction of ternary system behavior. One difference of convention which was introduced during the equilibrium phase of the project was that for equivalent fractions, the valence ratio adjustment term \( \frac{|z_i| |z_j|}{|z_j| |z_i|} \) collected into the equilibrium constant, (see equation 2-53), so

\[
\overline{k}_{ij}^{eq.fr.} = \overline{k}_{ij}^{eq.fr.} \cdot \left\{ \frac{|z_j| |z_i|}{|z_i| |z_j|} \right\} \tag{2-61}
\]
Since valence $i$ for the heterovalent systems investigated was 2 with $j = 1$,
\[ \bar{K}_{ij}^{eq.fr.} = \frac{1}{2} \bar{K}_{ij}^{eq.fr.} \]  
\hspace{1cm} (2-62)

During the regression of the heterovalent systems, when the binary system parameters were used to predict ternary system equilibrium data points, the equivalent fraction convention had a slight edge in standard deviation over use of mole fractions, as will be discussed later. It also was felt that it would be an advantage to use the equivalent fraction concentration units, since the dynamic models were designed to utilize equivalent fraction units. The combining of the valence relationship into the equilibrium constant caused no harm, in that in heterovalent systems the additional factor canceled with the chain rule convention. Using the system $\text{Mr}^{++}$-$\text{Cs}^+$-$\text{Na}^+$:
\[ \frac{1}{2} \bar{K}_{\text{M}n}^{\text{Mn}} = \frac{1}{2} \bar{K}_{\text{Na}}^{\text{Mn}} \frac{(1/1)}{(2/1)} \]  
\hspace{1cm} (2-63)

or with $\text{Sr}^{++}$-$\text{Mn}^{++}$-$\text{Cs}^+$
\[ \frac{1}{2} \bar{K}_{\text{Sr}}^{\text{Mn}} = \frac{1}{2} \bar{K}_{\text{Cs}}^{\text{Sr}} \frac{(2/1)}{(2/1)} \]  
\hspace{1cm} (2-64)

In comparison of equilibrium constants between the two systems, one other major difference was apparent. In heterovalent ion exchange the total number of moles in the
resin phase is a variable, whereas the number of equivalents is a constant. Mole fraction is defined as:

$$\bar{mf}_1 = \frac{c_1}{(c_1 + c_2)}$$  \hspace{1cm} (2-65)

whereas equivalent fraction is given by

$$\bar{eq.f.} = \left( \frac{c_1 |z_1|}{Q_r} \right)$$  \hspace{1cm} (2-66)

$$Q_r = \text{resin equivalent monoionic sites/volume of resin}.$$  

The relationship between mole fraction and equivalent fraction can be shown to be:

$$\bar{mf}_1 = \frac{\left( \frac{Q_r \bar{eqf}_1}{|z_1|} \right)}{\left( \frac{Q_r \bar{eqf}_1}{|z_1|} + \frac{Q_r \bar{eqf}_2}{|z_2|} \right)}$$ \hspace{1cm} (2-67)

$$= \frac{\bar{eqf}_1}{\left( \frac{1 - (|z_1|/|z_2|) \bar{eqf}_1 + (|z_1|/|z_2|) \bar{eqf}_2}{\bar{eqf}_1 + \bar{eqf}_2} \right)}$$

for a 2-1 heterovalent system, this relationship reduces to:

$$\bar{mf}_1 = \frac{\bar{eqf}_1}{(2 - \bar{eqf}_1)}$$  \hspace{1cm} (2-68)

In comparing the ratio of $\bar{R}_{12}^{eqf.}$ to $\bar{R}_{12}^{m.f.}$, keep in mind that in a 2-1 system the concentration ratios appear as:

$$\bar{R}^{eqf}_{mf} = \left( \begin{array}{c} \bar{eqf}_1 \\ \bar{eqf}_2 \\ \bar{mf}_1 \\ \bar{mf}_2 \\ \bar{mf}_1^2 \\ \bar{mf}_2^2 \\ \end{array} \right)$$, and \hspace{1cm} (2-69)
after the substitution of equation (2-68) in equation (2-69) the following ratio results:

\[ \frac{R_{\text{eqf}}}{m_f} = \frac{4}{(2-\text{eqf}_1)} \cdot (2-70) \]

Since the entire range of concentration values must be examined (from 0 to 1.0) in evaluating the operative ratio, \( R_{\text{eqf},m,f} \), expression (2-70) is integrated to yield:

\[ \frac{\bar{K}_{12}^{\text{eqf}}}{\bar{K}_{12}^{m,f}} \]

\[ = -4 \ln (2-\text{eqf}_1) \left|_0^1 \right. \]

\[ = 4 \ln 2 = 2.773 \]

Due to the effect of the change in total moles occurring within the resin during heterovalent exchange it could be expected that \( \bar{K}_{12}^{\text{eqf}} \) would be about 2.8 times greater than \( \bar{K}_{12}^{m,f} \). Since the valence ratio adjustment (equation (2-62)) had been incorporated into \( \bar{K}_{12}^{\text{eqf}} \), the observed ratio should have been about 1.4.

A relationship similar to equation (2-71) can be derived for 1-2 systems such as \( \text{NO}_3^-\text{SO}_4^= \). It can be shown that:

\[ \frac{R_{\text{eqf}}}{m_f} = \frac{(1+\text{eqf}_1)}{4} \cdot (2-72) \]
which upon integration yields:

$$R_{m.f.}^{eqf.} = 0.375 \left[ 1 + \frac{2}{4} \right]^{1}$$

Actual comparison of the ratio, $R_{m.f.}^{eqf.}$, shown in Table 2-4 gives the average for the three cationic heterovalent systems as 1.5475, with the difference from 1.4 of about 10% being absorbed into the activity coefficient ratio, $\frac{\bar{\gamma}_{1}^{2}}{\bar{\gamma}_{2}^{2}}$ eqf. vs. $\frac{\bar{\gamma}_{1}^{2}}{\bar{\gamma}_{2}^{2}}$ m.f.

In the anionic exchange, the average ratio is 0.2915 vs. 0.375, with an almost 30% difference being absorbed by the activity coefficient ratios.

C. Experimental Data for Ternary Systems

1. General

Experimental data was available for five systems, and regression analysis of the binary pair data was utilized to generate equilibrium constants and the resin phase activity coefficient parameters based on the Redlich-Kister, Wilson, and NRTL equations (the Omatate (78) system of Ag$^{+}$-Na$^{+}$-H$^{+}$ was only fitted with Redlich-Kister). These parameters were used to simulate the ternary data points and the results compared to the actual points, with standard deviations computed for each system and correlation method.

2. The systems investigated are shown in Table 2-2. The ternary system experimental data was accompanied with the experimental binary system data given either in tabular or graphical format.
### Table 2-2

<table>
<thead>
<tr>
<th>Investigator(s)</th>
<th>System</th>
<th>Experimental Method</th>
<th>No. of Ternary Data Points</th>
</tr>
</thead>
<tbody>
<tr>
<td>Omatete (78)</td>
<td>Ag⁺-Na⁺-H⁺/NO₃ on Dowex 50W X8 0.1 normal solution at 23°C.</td>
<td>Column displacement technique. Ag⁺ by Volhard titration, (±0.2%). H⁺ by titration with NaOH (±0.2%), Na⁺ by flamespectroscopy, (±7%).</td>
<td>2 points + 3 by material balance on rate runs - this project</td>
</tr>
<tr>
<td>Rao et al (91)</td>
<td>Sr⁺⁺, Mn⁺⁺, Cs⁺⁺/Cl⁻ on Dowex 50W X8. Mn⁺⁺, Cs⁺⁺/Cl⁻ on Dowex 50W X8. 0.1 normal solution at 23°C.</td>
<td>Radioactive tracer technique. Reproducibility was ±2%.</td>
<td>16 points</td>
</tr>
<tr>
<td>Smith &amp; Woodburn (116)</td>
<td>NO₃⁻-Cl⁻-SO₄⁻/Na⁺ on Rohm &amp; Haas Amberlite IRA 400 0.2 normal solution at 25 ±3°C.</td>
<td>Column displacement technique C₁⁻ by potentiometric titration with AgNO₃ solution (±1%), SO₄ gravimetrically (±2%) NO₃ by difference.</td>
<td>16 points with 2 additional replication runs</td>
</tr>
<tr>
<td>Soldatov &amp; Bychkova (121)</td>
<td>NH₄⁺-Na⁺-H⁺/Cl⁻ on Dowex 50B X12 0.1 normal solution at 25°C.</td>
<td>Equilibrium method not stated, but believed to be Column displacement. H⁺ by alkaliometry, NH₄⁺ by formaldehyde method.</td>
<td>45 points distributed along NH₄⁺ ion isotherms of 0.05, 0.10, 0.20, 0.30, 0.40, 0.50, 0.60, 0.70 in the electrolyte phase.</td>
</tr>
</tbody>
</table>
D. Correlation Model - Ion Exchange Equilibrium

1. General Background

A powerful general purpose non-linear regression program based on a variation of the Gauss-Newton method was developed by Dr. A. K. S. Murthy at Allied Corporation, and a general description is given in Appendix B. In fitting the equilibrium constants and activity coefficient parameters for the species in the resin phase, obviously one equilibrium relationship is required per binary pair, and the number of equilibrium equations required for the ternary systems is three.

The equilibrium relationship for, say the 1-2 pair, is divided into terms containing the known parameters, and terms constituting the unknown side contain the parameters to be fitted:

\[ Y(1)_{\text{calc}} = \frac{\gamma_2 |z_1|}{\gamma_1 |z_2|} \]

(2-74)

where:

\[ \bar{\gamma}_1 = \text{func} (B_{12}, C_{12}, D_{12}) \]  
Redlich-Kister

\[ \bar{\gamma}_2 = \text{func} (B_{11}, C_{12}, D_{12}) \]  
Constant

and

\[ B_{12}, C_{12}, D_{12} \text{ are func } (\bar{x}_1, \bar{x}_2) \]

and the terms containing the known parameters are:

\[ Y(1)_{\text{exp}} = \left\{ \begin{array}{c} \frac{\bar{x}_1}{x_{1\text{int}}} \\ \frac{\bar{x}_2}{x_{2\text{int}}} \end{array} \right\} \cdot \left\{ \begin{array}{c} \frac{\gamma_2}{\gamma_1} \\ \frac{x_{2\text{int}}}{x_{1\text{int}}} \end{array} \right\} \cdot \text{norm} (|z_1|-|z_2|) \]

(2-75)

where \( \bar{x}_1, \bar{x}_2, x_{1\text{int}}^*, \) and \( x_{2\text{int}}^* \) correspond to data points and, \( \gamma_1 \) plus \( \gamma_2 \) are calculated using the
Bromley equation for the electrolyte phase activity coefficients.

The $Y_1^{\text{exp}}$ terms represent the experimental knowns. The regression program then finds the numerical derivatives of $Y_1^{\text{calc}}$ with respect to each parameter, i.e.:

$$\frac{\partial Y_1^{\text{calc}}}{\partial K_{12}^{\text{eqf}}}, \frac{\partial Y_1^{\text{calc}}}{\partial B_{12}}, \frac{\partial Y_1^{\text{calc}}}{\partial C_{12}}, \frac{\partial Y_1^{\text{calc}}}{\partial D_{12}}$$

and using the weighted objective function:

$$wgt_i \left[ \frac{(Y_i^{\text{exp}} - Y_i^{\text{calc}})}{Y_i^{\text{exp}}} \right]^2 \quad \text{for } i=1 \text{ to number of dependent variables}$$

searches for global minima for all data points in each set. If some parameters are used to describe the calculation of variables in other data sets, then these sets are also included in the global minima search for that parameter. When convergence criteria have been met in terms of residual error for all parameters, the regression program prints out values for each fitted parameter along with standard deviation and cross correlation coefficients.

2. Ion Exchange Equilibrium Regression Program, RIONFT, for Ternary Systems

The general purpose ion exchange correlation program developed as part of this research was called RIONFT (Resin phase Ion exchange Fitting program). Dr. A. K. S. Murthy's generalized regression program (Appendix B) is the heart of a
program developed specifically for fitting resin phase ion exchange parameters to experimental binary and ternary equilibrium data. It step-wise performs 26 sequential correlation combinations to arrive at a best portrayal of the three binary systems and the ternary points. Using the Redlich-Kister equation for resin phase activity coefficients (parameters $B_{ij}$, $C_{ij}$, $D_{ij}$) as an example, these steps are:

- **Binary pair 1-2**
  1. fits $\bar{K}e_{12}$ and $B_{12}$
  2. fits $\bar{K}e_{12}$ and $B_{12}$, $C_{12}$ using initial estimates for $\bar{K}e_{12}$ and $B_{12}$ from 1
  3. fits $\bar{K}e_{12}$, $B_{12}$, $C_{12}$ and $D_{12}$ using initial estimates for $\bar{K}e_{12}$, $B_{12}$ and $C_{12}$ from 2

- **Binary pair 1-3**
  4-6 performs same steps for $\bar{K}e_{13}$, $B_{13}$, $C_{13}$ and $D_{13}$ as were carried out in steps 1-3

- **Binary pair 2-3**
  7-9 performs same steps for $\bar{K}e_{23}$, $B_{23}$, $C_{23}$ and $D_{23}$ as were carried out in steps 1-3

- **Binary pair 2-3**
  10 uses $\bar{K}e_{23}$ computed by the chain rule using the $\bar{K}e_{12}$ and $\bar{K}e_{13}$ fitted in steps 2 and 5, and fits $B_{23}$ and $C_{23}$
  11 uses $\bar{K}e_{23}$ computed by the chain rule using the $\bar{K}e_{12}$ and $\bar{K}e_{13}$ fitted in steps 3 and 6, and fits $B_{23}$, $C_{23}$, and $D_{23}$
Binary pair 1-3

12 uses chain rule for $\bar{K}e_{13}$ using $\bar{K}e_{12}$ and $\bar{K}e_{23}$ from fits 2 and 8 and correlates $B_{13}$ and $C_{13}$

13 uses chain rule for $\bar{K}e_{13}$ using $\bar{K}e_{12}$ and $\bar{K}e_{23}$ from fits 3 and 9 and correlates $B_{13}$, $C_{13}$ and $D_{13}$

Binary pair 1-2

14 uses chain rule for $\bar{K}e_{12}$ using $\bar{K}e_{13}$ and $\bar{K}e_{23}$ from fits 5 and 8 and correlates $B_{12}$ and $C_{12}$

15 uses chain rule for $\bar{K}e_{12}$ using $\bar{K}e_{13}$ and $\bar{K}e_{23}$ from fits 6 and 9 and correlates $B_{12}$, $C_{12}$ and $D_{12}$

Ternary data for system 1, 2, 3

16 correlates the parameters to best fit the ternary data only using the $\bar{K}e_{12}$ from 3, the $\bar{K}e_{13}$ from 6 and $\bar{K}e_{23}$ derived from those using the chain rule to fit $B_{ij}$, $C_{ij}$, and $D_{ij}$,

(i = 1, 2, 3 for j = 1, 2, 3, i = j).

Provision is made to switch $\bar{K}e_{13}$ to be calculated from $\bar{K}e_{12}$ and $\bar{K}e_{23}$ derived from 3 and 9

17 using parameters fitted in 2, 5, and 10, calculates ternary resin phase compositions using those parameters and computes standard deviation of the calculated points from the experimental data points

18 repeats 17 using parameters fitted in 3, 6, and 11

19 repeats 17 using parameters fitted in 2, 8, and 12

20 repeats 17 using parameters fitted in 3, 9, and 13

21 repeats 17 using parameters fitted in 5, 8, and 14

22 repeats 17 using parameters fitted in 6, 9, and 15
**Binary sets fit together**

23 all three binary sets then are submitted by the program for simultaneous fitting of $K_{e_{12}}$, $B_{12}$, $C_{12}$, $D_{12}$, $K_{e_{13}}$, $B_{13}$, $C_{13}$, $D_{13}$, $B_{23}$, $C_{23}$ and $D_{23}$ with $K_{e_{23}} = f (K_{e_{12}}$ and $K_{e_{13}}$)

24 same as 24 but $K_{e_{13}} = f (K_{e_{12}}, K_{e_{23}})$

**Total data sets fit, 3 binaries and ternary data**

25 similar to 23 except ternary data set is included

26 similar to 24 except ternary data set is included

The optional correlation methods for fitting the resin phase activity coefficients are the one, two or three parameter Redlich-Kister equation, the Wilson equation with $\rho_{ij} = 1.0$ (or set to some other value) to fit $\tau_{ij}$, or $\rho_{ij}$ can be fit as well.

The NRTL equation with $\alpha_{ij} = 0.3$ (or set to some other value) to fit $\tau_{ij}$, or $\alpha_{ij}$ can be fit as well.

One additional feature was built into RIONFT in order to regress the data of Smith and Woodburn involving the stable intermediate, NaSO₄⁻. The program has a subroutine which computes the equilibrium concentration of one of the stable complexes formed by one of the three ionic species, providing the stability constant is available for that complexation reaction. The use of this type of relationship is questionable in a ternary system, since it assumes that the complex is either (1) not involved in the ion exchange process or (2) has the same ion exchange equilibrium relationship as the uncomplexed ion.

It most likely is involved in the ion exchange process, and the
complexity now introduced is that a quaternary non-stoichiometric exchange process is involved, with the attendant experimental complications of measuring the amount of counterion in the exchanger due to the uncomplexed and the complexed form. As will be discussed below, the Smith Woodburn ternary system of \( \text{NO}_3^- \text{Cl}^- \text{SO}_4^- \) was equally well fit to experimental data without considering the complexation reaction to yield \( \text{NaSO}_4^- \).

3. Features of RIONFT

The operational features of the ion exchange equilibrium regression program, RIONFT, for ternary systems is given in Appendix C.

E. Correlation Results - Ternary and Binary Systems

1. Comparison of Correlation Models

The results of data regression on four ternary systems utilizing RIONFT is given in Table 2-3. If the judgment of method used were made solely on the predicted ternary results, the Wilson equation would be preferred; however, in terms of total ability to correlate binary pairs (with \( \overline{K}_{ij} \) being a function of \( \overline{K}_{ik} \) and \( \overline{K}_{jk} \)) and predict ternary systems, the Redlich-Kister three suffix equation was selected. The Omatete (78) \( \text{Ag}^+, \text{Na}^+, \text{H}^+ \) system was not included in this comparison because of the paucity of data.

2. Ternary Phase Diagrams

Utilizing the Redlich-Kister model for the prediction of resin phase activity coefficients, resin compositions were
Table 2-3

Ternary Ion Exchange Equilibrium Data Regression
- Based on Ionic Equivalent Fraction -
Resin Phase Activity Coefficient Model

Redlich-Kister - three suffix

Wilson - $\varphi_{ij} = \frac{\text{molar volume } i}{\text{molar volume } j} = 1.0$

NRTL - $\alpha_{ij} = 0.3$

Standard Deviation - % Normalized Difference

<table>
<thead>
<tr>
<th>System</th>
<th>Pair 1-2</th>
<th>Pair 1-3</th>
<th>Pair 2-3</th>
<th>Ternary Calculated from Binary Parameters</th>
</tr>
</thead>
<tbody>
<tr>
<td>1. Mn$^{++}$-Cs$^{+}$-Na$^{+}$</td>
<td></td>
<td></td>
<td></td>
<td></td>
</tr>
<tr>
<td>R-K</td>
<td>1.70</td>
<td>8.8</td>
<td>3.8</td>
<td>11.6</td>
</tr>
<tr>
<td>Wilson</td>
<td>6.9</td>
<td>17.7</td>
<td>5.1</td>
<td>12.6</td>
</tr>
<tr>
<td>NRTL</td>
<td>7.5</td>
<td>18.5</td>
<td>5.2</td>
<td>13.2</td>
</tr>
<tr>
<td>2. Sr$^{++}$-Mn$^{++}$-Cs$^{+}$</td>
<td></td>
<td></td>
<td></td>
<td></td>
</tr>
<tr>
<td>R-K</td>
<td>1.4</td>
<td>2.0</td>
<td>1.7</td>
<td>11.8</td>
</tr>
<tr>
<td>Wilson</td>
<td>1.4</td>
<td>1.7</td>
<td>7.2</td>
<td>10.5</td>
</tr>
<tr>
<td>NRTL</td>
<td>1.5</td>
<td>2.9</td>
<td>5.5</td>
<td>10.9</td>
</tr>
<tr>
<td>3. NH$_4^+$-Na$^+$-H$^+$</td>
<td></td>
<td></td>
<td></td>
<td></td>
</tr>
<tr>
<td>R-K</td>
<td>1.1</td>
<td>1.05</td>
<td>7.8</td>
<td>4.6</td>
</tr>
<tr>
<td>Wilson</td>
<td>1.05</td>
<td>3.3</td>
<td>9.5</td>
<td>3.7</td>
</tr>
<tr>
<td>NRTL</td>
<td>1.1</td>
<td>3.3</td>
<td>9.4</td>
<td>3.7</td>
</tr>
<tr>
<td>4. NO$_3^-$-Cl$^-$-SO$_4^-$</td>
<td></td>
<td></td>
<td></td>
<td></td>
</tr>
<tr>
<td>R-K</td>
<td>1.34</td>
<td>1.44</td>
<td>3.4</td>
<td>4.1</td>
</tr>
<tr>
<td>Wilson</td>
<td>1.9</td>
<td>1.8</td>
<td>3.3</td>
<td>2.7</td>
</tr>
<tr>
<td>Roth</td>
<td></td>
<td></td>
<td></td>
<td></td>
</tr>
<tr>
<td>Smith &amp; Woodburn</td>
<td>3.5</td>
<td>5.8</td>
<td>2.7</td>
<td>4.3</td>
</tr>
<tr>
<td>NRTL</td>
<td>2.3</td>
<td>3.0</td>
<td>3.3</td>
<td>5.9</td>
</tr>
</tbody>
</table>
calculated at regular electrolyte composition isotherms and plotted on triangular coordinate graphs. Figures 2-1 through 2-6 show the calculated position for the ternary resin compositions at the intersections of isotherms, and the experimental data points are superimposed. Two graphs are included for the Omatete data at normalities of 0.10 and 1.5 solution concentrations.

3. Binary Phase Diagrams

The Erickson (26) ETDA\(^{++}\) - NH\(_4\)\(^+\) system is shown on Figure 2-7 for two normalities, 0.1 and 1 solution concentrations. The percent normalized standard deviations are 5.5 and 1.3\% respectively, based on a two suffix Redlich-Kister equation. A plot of the activity coefficients is shown on Figures 2-8 showing that the system exhibits positive deviation and is almost symmetrical.

Omatete's (78) systems of Ag\(^+\) - H\(^+\), Ag\(^+\) - Na\(^+\), and Na\(^+\) - H\(^+\) are shown on Figures 2-9, 2-10, and 2-11 for 0.1 and 0.5 normal Cl\(^-\) solutions. In the Na\(^+\) - H\(^+\) system, Turner and Snowdon's (129, 130) data which was required to fill out the lower region is also plotted on Figures 2-11.

4. Discussion and Conclusions

Smith and Woodburn's (116) model for prediction of ternary ion exchange equilibria utilizing the Wilson equation has been extended to include the Redlich-Kister and NRTL activity coefficient models. The three models have been tested against the original system investigated by Smith and Woodburn, plus
Ternary Equilibrium Diagram
Resin Phase, Equivalent Fraction

$\text{Mn}^{++} - \text{Cs}^{+} - \text{Na}^{+}$

Dowex 50W - x8 Resin
$\text{Cl}^{-}$ 0.1 Normal, 23°C

○ Experimental Points
- Resin Phase

Calculated Solution Phase Isotherms
- - - - $\text{Mn}^{++}$
- - - - $\text{Cs}^{+}$ equivalent fraction
- - - - - $\text{Na}^{+}$

$\text{Mn}^{++}$ in solution

Cs$^+$ in solution

Na$^+$
Ternary Equilibrium Diagram
Resin Phase, Equivalent Fraction

\[ \text{NH}_4^+ - \text{Na}^+ - \text{H}^+ \]

Dowex 50W - 12x Resin
Cl\(^-\) ion  0.1 Normal, 25\(^\circ\)C

○ Experimental Points
- Resin Phase

Calculated Solution Phase Isotherms

- - - - - \( \text{NH}_4^+ \)
- - - - - \( \text{Na}^+ \) equivalent fraction
- - - - - \( \text{H}^+ \)

\( \text{Na}^+ \) in solution

\( \text{NH}_4^+ \) in solution
Ternary Equilibrium Diagram
Resin Phase, Equivalent Fraction

\[ \text{NO}_3^- - \text{Cl}^- - \text{SO}_4^2- \]

Amberlite 400

Na\(^+\) 0.2 Normal, 25°C

○ Experimental Points
- Resin Phase

Calculated Solution Phase Isotherms

- - - - - \( \text{NO}_3^- \) equivalent fraction
- - - - - \( \text{Cl}^- \)
- - - - - \( \text{SO}_4^2- \)

Cl\(^-\) in solution

NO\(_3^-\) in solution
Ternary Equilibrium Diagram
Resin Phase, Equivalent Fraction

\[ \text{Ag}^+ - \text{Na}^+ - \text{H}^+ / \text{NO}_3^- \]

Dowex 50W - x8 Resin
0.10 Normal, 25°C

- Experimental Points
  - Resin Phase

Calculated Solution Phase Isotherms

- - - - Ag^+
- - - - Na^+ equivalent fraction
- - - - H^+

Ag^+ in solution

Na^+ in solution
Ternary Equilibrium Diagram
Resin Phase, Equivalent Fraction

Ag⁺ - Na⁺ - H⁺/NO₃⁻
Dowex 50W - x8 Resin
1.5 Normal, 25°C

⊙ Experimental Points
- Resin Phase

Calculated Solution Phase Isotherms

- - - - Ag⁺
- - - Na⁺ equivalent fraction
- - - - H⁺

Ag⁺ in solution

Na⁺ in solution
System Ethylene Diamine/NH$_3$ - Dowex 50W x 8 Resin
Temp. 25°C

Figure 2-7

Equivalent Fraction Diamine in Resin $X_A$

Experimental Data*
- o 1 Normal Cl$^-$
- △ 0.1 Normal$^-$

$K_{ETDA}^{++} + \text{NH}_4^+ = 16.1206$

Redlich-Kister Constants
- B = 0.34626
- C = -0.02223

Bromley Constants
- $B_{ETDA\text{Cl}} \approx 0.09$
- $B_{\text{NH}_4\text{Cl}} = 0.02$

Standard Deviation
- 1 Normal: 1.3%
- 0.1 Normal: 5.5%

*K.L. Erickson
"Fixed Bed Exchange with Differing Ionic Mobilities and Non-linear Equilibria" U. Texas, 1977
System Ethylene Diamine/NH$_3$ - Dowex 50W x 8 Resin -Cl$^-$
Resin Phase Activity Coefficients

Redlich-Kister Constants
B = 0.346256
C = -0.02223
1 - Diamine
2 - Ammonia

Equivalent Fraction Diamine in Resin
System $\text{Ag}^+ / \text{H}^+ / \text{NO}_3^- / \text{Dowex 50W x 8 Resin}$
Temp. 25°C

Data
- O. Omatete "Dynamics of Ternary Ion Exchange"
  - ○ 0.1 Normal
  - □ 1.5 Normal

Calculated
- $K_{\text{Ag}^+ , \text{H}^+} = 8.8761$
- Redlich-Kister Constants
  - $B = -0.7257$
  - $C = 0.8998$
  - $D = -0.9350$
- Bromley Constants
  - $B_{\text{AgNO}_3} = -0.0828$
  - $B_{\text{HNO}_3} = 0.0776$
System $\text{Ag}^+/\text{Na}^+/\text{NO}_3^-$/Dowex 50W x 8 Resin
Temp. 25°C

Figure 2-10

Data
O. Omatete "Dynamics of Ternary Ion Exchange"
○ 0.1 Normal
□ 1.5 Normal

Calculated
0.1 Normal
1.5 Normal

$K_{\text{Ag}^+\text{Na}^+} = 5.924$

Redlich-Kister Constants
$B = -1.000$
$C = 1.2472$
$D = -0.8957$

Bromley Constants
$B_{\text{AgNO}_3} = -0.0828$
$B_{\text{NaNO}_3} = -0.0128$
System $\text{Na}^+/\text{H}^+/\text{NO}_3^-$/Dowex 50W x 8 Resin
Temp. 25°C

Figure 2-11

Data
O. Omatete "Dynamics of Ternary Ion Exchange"
○ 0.1 Normal
□ 1.5 Normal

Turner, et al.
System $\text{Cl}^-$/ZEO-Karb 8x
× 0.1 Normal

Averaged Omatete/Turner
+ 0.1 Normal

$0.05 \text{ N Same as } 0.1\text{N}$
$0.10 \text{ N}$
$1.5 \text{ N}$

$K_{\text{Na}^+/\text{H}^+} = 1.49827$

Redlich-Kister Constants, Bromley Constants
$B = -0.15021$ \hspace{1cm} $B_{\text{NaNO}_3} = -0.0128$
$C = -0.15682$ \hspace{1cm} $B_{\text{HNO}_3} = 0.0776$
$D = -0.08917$
three other systems. The results are encouraging and suggest that the Redlich-Kister model gives better overall results. The fact that the activity coefficient, whether correlated with mole fractions or equivalent fraction concentration units, must "correct" for the changing number of moles in the ion exchanger during heterovalent exchange, i.e., $d\bar{c}_{\text{total}}$, accounts for the poorer results obtained with heterovalent cationic exchange as compared to homovalent cationic exchange. The extra suffix term, or "knob," on the Redlich-Kister equation allows for more non-linear compensation than the two "suffix" Wilson and NRTL equations. In terms of extending the system to more than three ionic species, if such data were available, the Wilson equation would be more straightforward in its general application to multicomponent systems.

It should be noted that runs were made on the two heterovalent cationic systems in which $\rho_{ij}$ in the Wilson equation and $\alpha_{ij}$ in the NRTL equation were also fit with the regression model. The results were improved on some binary systems, and made worse in others, with a general degradation in the ability to predict ternary data points. In addition, physically absurd values of parameters such as $\rho_{ij}$ were obtained. Since $\rho_{ij}$ represents the molar volume ratios of the two solvated ions, some order of magnitude check against literature observations would have been encouraging. In any event, the values of $\rho_{ij} = 1.0$ and $\alpha_{ij} = 0.3$ gave overall best results for the Wilson and NRTL equations respectively.
Mole fraction and equivalent fraction concentration units were utilized in fitting the heterovalent ternary systems (obviously in the homovalent systems mol. fr. = equiv. fr.). Table 2-4 shows that overall, the equivalent fraction units yielded somewhat better results than the mole fraction system based on standard deviation of normalized difference. The same Table, 2-4, shows the mole fraction based equilibrium constants for the three binary pairs and the Redlich-Kister parameters for those systems.

The use of a four suffix Redlich-Kister equation (E_{ij} term) plus three ternary constants was evaluated initially as perhaps a more powerful activity coefficient model for both binary and ternary systems. In the case of the binary systems, experimental data for each binary pair was fit with five parameters (including $K_{ij}$). Since in some instances the number of data points may have been between 8 and 10, this was clearly a case of "overkill." In fact, in the binary systems, more of the equilibrium selectivity for a given pair became vested in the activity coefficients than in the equilibrium constant. Although the fits were slightly improved over the three suffix system, attempts to correlate the activity coefficients for a third pair based on the chain rule deriving $K_{ij}$ as a function of $K_{ik}$ and $K_{jk}$ led to poor results. This was so because the values of $K_{ik}$ and $K_{jk}$ were beginning to lose their identities as equilibrium constants when they should have represented the strongest correlatable parameter in the equilibrium
Table 2-4

Comparison of Mole Fraction vs. Equivalent Fraction
- Resin Phase Equilibria Correlation -

Heterovalent Exchange

Redlich-Kister three suffix equation utilized for resin phase activity coefficients

<table>
<thead>
<tr>
<th>System</th>
<th>Equilibrium Constant $K_{ij}$</th>
<th>Redlich-Kister Coefficients $B_{ij}$</th>
<th>$C_{ij}$</th>
<th>$D_{ij}$</th>
<th>Standard Deviation % Normalized Difference</th>
</tr>
</thead>
<tbody>
<tr>
<td>1. $\text{Sr}^{2+}\text{Mn}^{2+}\text{Cs}^+$</td>
<td></td>
<td></td>
<td></td>
<td></td>
<td></td>
</tr>
<tr>
<td></td>
<td>$\text{Sr}^{2+}\text{Mn}^{2+}$</td>
<td>$7.3112$</td>
<td>$0.0420$</td>
<td>$0.0707$</td>
<td>$-0.0679$</td>
</tr>
<tr>
<td></td>
<td>m.f.</td>
<td>$7.3111$</td>
<td>$0.0420$</td>
<td>$0.0707$</td>
<td>$-0.0679$</td>
</tr>
<tr>
<td></td>
<td>eq.f.</td>
<td>$4.4890$</td>
<td>$-0.4994$</td>
<td>$0.0170$</td>
<td>$-0.1270$</td>
</tr>
<tr>
<td></td>
<td>$\text{Sr}^{2+}\text{Cs}^+$</td>
<td>$2.8994$</td>
<td>$-0.8050$</td>
<td>$0.1689$</td>
<td>$-0.1325$</td>
</tr>
<tr>
<td></td>
<td>m.f.</td>
<td>$4.4890$</td>
<td>$-0.4994$</td>
<td>$0.0170$</td>
<td>$-0.1270$</td>
</tr>
<tr>
<td></td>
<td>eq.f.</td>
<td>$1.0723$</td>
<td>$-0.8029$</td>
<td>$0.1253$</td>
<td>$0.2425$</td>
</tr>
<tr>
<td></td>
<td>$\text{Mn}^{2+}\text{Cs}^+$</td>
<td>$1.6602$</td>
<td>$-0.4552$</td>
<td>$-0.2965$</td>
<td>$0.3943$</td>
</tr>
</tbody>
</table>

ternary m.f. 11.2  eq.f. 11.8
<table>
<thead>
<tr>
<th>System</th>
<th>Equilibrium Constant $K_{ij}$</th>
<th>Redlich-Kister Coefficients</th>
<th>Standard Deviation % Normalized Difference</th>
</tr>
</thead>
<tbody>
<tr>
<td></td>
<td></td>
<td>$B_{ij}$</td>
<td>$C_{ij}$</td>
</tr>
<tr>
<td>2. $\text{Mn}^{++}\text{Cs}^{+}\text{Na}^{+}$</td>
<td></td>
<td></td>
<td></td>
</tr>
<tr>
<td>$\text{Mn}^{++}\text{Cs}^{+}$</td>
<td>m.f.</td>
<td>1.0723</td>
<td>-0.8029</td>
</tr>
<tr>
<td></td>
<td>eq.f.</td>
<td>1.6602</td>
<td>-0.4552</td>
</tr>
<tr>
<td>$\text{Mn}^{++}\text{Na}^{+}$</td>
<td>m.f.</td>
<td>3.0236</td>
<td>0.5847</td>
</tr>
<tr>
<td></td>
<td>eq.f.</td>
<td>4.6746</td>
<td>0.4677</td>
</tr>
<tr>
<td>$\text{Cs}^{+}\text{Na}^{+}$</td>
<td>m.f.</td>
<td>1.6792</td>
<td>-0.0869</td>
</tr>
<tr>
<td></td>
<td>eq.f.</td>
<td>1.6792</td>
<td>-0.0869</td>
</tr>
<tr>
<td>3. $\text{NO}_3^{-}\text{Cl}^-\text{SO}_4^{2-}$</td>
<td></td>
<td></td>
<td></td>
</tr>
<tr>
<td>$\text{NO}_3^{-}\text{Cl}^-$</td>
<td>m.f.</td>
<td>3.9506</td>
<td>-0.4276</td>
</tr>
<tr>
<td></td>
<td>eq.f.</td>
<td>3.9506</td>
<td>-0.4276</td>
</tr>
<tr>
<td>$\text{NO}_3^{-}\text{SO}_4^{2-}$</td>
<td>m.f.</td>
<td>131.0740</td>
<td>-1.2959</td>
</tr>
<tr>
<td></td>
<td>eq.f.</td>
<td>38.2100</td>
<td>-1.2306</td>
</tr>
<tr>
<td>$\text{Cl}^-\text{SO}_4^{2-}$</td>
<td>m.f.</td>
<td>8.3990</td>
<td>-0.8172</td>
</tr>
<tr>
<td></td>
<td>eq.f.</td>
<td>2.4483</td>
<td>-0.5527</td>
</tr>
</tbody>
</table>
relationship. The standard deviations for ternary systems predicted from the binaries correlated with the four suffix Redlich-Kister equation were generally poorer than obtained with the three suffix model.

Fitting the ternary data with a four suffix Redlich-Kister model including three ternary system constants, or 17 independent parameters, for 15 ternary data points (30 independent variables) showed no important improvement over use of the three suffix model without the ternary system constants. In any case, a requirement for ternary system constants would have negated any possibility of predicting ternary equilibrium system behavior from the binary correlation constants. The four suffix Redlich-Kister model and the ternary system constants were ruled out in subsequent correlation studies in this research.

It was concluded that the three suffix Redlich-Kister equation for resin phase activity coefficients based on equivalent fraction concentration units, and coupled with the Bromley equation to predict electrolyte phase activity coefficients, generally gave excellent fits of binary systems, even in heterovalent exchange, and gave fair representation of predicted ternary equilibrium points. This system for representation of binary and ternary equilibria was selected for use in the dynamic ion exchange model.

F. Relationship of Activity Coefficients Utilized in Deriving Diffusion Coefficients - Activity Driving Force

1. Binary Systems
1.1 **Background**

The dynamics of ions exchanging within an ion exchange resin is discussed in Chapter V. As in all mass transport processes, the rate of transfer of a molecular or ionic species through a given media is governed by a factor which is proportional to the ease with which the species moves through the media (diffusion coefficient), and the chemical potential gradient of the species in the media (driving force). In the case of ion exchange it is the electrochemical potential gradient that is operational, since the movement of charged ions or molecules is influenced by electrical gradients as well as chemical concentration gradients. In Chapter V, the molar flux for binary ion exchange is given as:

\[ \overline{N}_1 = -\overline{D}_{11} \cdot \overline{v} \overline{c}_1 \quad \text{or} \quad \overline{N}_1 = -\overline{D}_{12} \text{eff} \cdot \overline{v} \overline{c}_1 \]  

from equation (5-12) and (5-13)

where:

\[ \overline{D}_{12} \text{eff} = \frac{\overline{D}_1 \overline{D}_2}{\sum \frac{c_i z_i^2 \overline{D}_i}{1}} \cdot \left( -\frac{z_1^2 + z_2^2}{2} \right) \frac{\partial n}{\partial c_1} \]  

molar

Manipulation of variables during subsequent derivations and computations can be performed more readily if the diffusional flux relationship is expressed in equivalent fractions by recognizing that:
so that in equivalent fractions equation (5-23) becomes:

$$\vec{J}_1 = - \vec{D}_{12 \text{eff}} \cdot \nabla \vec{x}_i \cdot \vec{Q}_r$$  \hspace{1cm} (5-24)

The expression for effective diffusivity becomes:

$$\vec{D}_{12 \text{eff}} = \frac{\vec{D}_1 \vec{D}_2}{\sum_{i=1}^{2} \vec{x}_i |z_i| \vec{D}_i}$$

The current chapter on equilibrium involves methods for correlating binary and ternary ion exchange systems with models involving equivalent fraction, or alternatively mole fraction, of each species in the ion exchange resin. These models are consistent in that the standard states chosen allow prediction of the equilibrium relationships, $x_{i, \text{int}}^*$, for use in the flux equation (5-23) so long as the correct concentration units are employed. As an example, if the system of interest was correlated in mole fraction resin phase concentration units then, in employing the equivalent flux equation (5-60) at the boundary condition,

$$\left. \frac{\partial \vec{x}_i}{\partial \rho} \right|_{\rho=1} = k_{L \text{eff bulk}} \cdot (x_{i, \text{int}}^* - x_{i, \text{int}}^*)$$

fraction concentration, $\vec{x}_i|_{\rho=1}$, must be converted to mole fraction, $\vec{x}_{i, \text{m.f.}}|_{\rho=1}$, in order to obtain the electrolyte
phase equilibrium value, $x_{1\text{int}}^*$, by use of equation (2-53).

1.2 Change of Standard States for Activity Driving Force

- Conversion of Molar to Equivalent Fraction Concentrations

Equation (5-21) describes the effective diffusivity for interchange of ionic species 1 and 2 in the resin. If the system can be considered to be ideal, then the term

$$\frac{a \ln \left( \frac{-\gamma_1 |z_2|}{-\gamma_2 |z_1|} \right)}{\partial x_1}$$ molar

is zero, and the equation can be used directly in terms of resin phase equivalent fraction concentration units. If the chemical potential, $\bar{\mu}_1$, is to be used as the driving force then, in deriving the effective diffusivity for interchange of species 1 and 2, the activity coefficient terms in the factor

$$\frac{a \ln \left( \frac{-\gamma_1 |z_2|}{-\gamma_2 |z_1|} \right)}{\partial x_1}$$ molar

must be computed from a model relating molar activity coefficients to molar concentration units in the resin. The models discussed in this chapter have been correlated in mole fraction or equivalent fraction concentration units, and a conversion in standard states to reflect molar concentration units must be made in order to utilize the molar flux equation correctly for non-ideal systems.

As derived in Appendix A, it was shown that the activity coefficients based on equivalent fraction concentration units
are identical to those based on molar concentration units.

The limiting values are:

\[ \bar{x}_1 \to 1.0, \quad \bar{\gamma}_1 \to 1.0, \quad \text{and} \quad \bar{a}_1 \to 1.0 \quad (2-78) \]

\[ \bar{x}_1 \to 0.0, \quad \bar{\gamma}_1 \to \frac{K}{n \bar{f}_1^{\bar{f}_0}}, \quad \text{and} \quad \bar{a}_1 \to 0.0 \quad (2-79) \]

With molar concentration units, the limiting values are:

\[ \bar{c}_1 \to \bar{c}_1 = \frac{Q_r}{|z_1|}, \quad \bar{\gamma}_1 \to 1.0, \quad \text{and} \quad \bar{a}_1 \to \frac{Q_r}{|z_1|} \quad (2-80) \]

\[ \bar{c}_1 \to 0.0, \quad \bar{\gamma}_1 \to \frac{K}{n \bar{f}_1^{\bar{f}_0}}, \quad \text{and} \quad \bar{a}_1 \to 0.0 \quad (2-81) \]

Models of activity coefficients based on molar versus equivalent fraction concentration units can be illustrated by using a truncated Redlich-Kister equation. In molar units the expression for \( \bar{\gamma}_1 \) molar would be:

\[ \ln \bar{\gamma}_1 \text{ molar} = A \cdot \bar{c}_2^2 \quad (2-82) \]

If equivalent fraction units are used,

\[ \ln \bar{\gamma}_1 = A \cdot \bar{x}_2^2 \quad (2-83) \]

and since \( \bar{x}_2 = \frac{c_2 \cdot |z_2|}{Q_r} \), and \( \bar{\gamma}_1 \) molar = \( \bar{\gamma}_1 \)

\[ A \cdot \bar{c}_2^2 = A \cdot \bar{x}_2^2 \cdot \left( \frac{|z_2|}{Q_r} \right)^2, \quad \text{so} \quad A = A \cdot \left( \frac{|z_2|}{Q_r} \right)^2 \quad (2-84) \]

and the systems are made congruent by the term \( \left( \frac{|z_2|}{Q_r} \right)^2 \).

Consequently, in utilizing the non-ideal version of the equation for effective diffusion coefficients, \((5-14)\),
activating coefficients correlated in equivalent fraction concentration units can be used directly, or:

\[
\frac{\partial \ln \left( \frac{\gamma_1 |z_2|}{\gamma_2 |z_1|} \right)}{\partial x_1} \text{ \text{ molar}} = \frac{\partial \ln \left( \frac{\gamma_1 |z_2|}{\gamma_2 |z_1|} \right)}{\partial x_1} \text{ \text{ eq.fr.}} \quad (2-85)
\]

1.3 Change of Standard States for Activity Driving Force

- Conversion of Molar to Mole Fraction Concentrations

The change of standard states involved with conversion of activity coefficients correlated in molar concentration terms to activity coefficients correlated in mole fraction terms is more complicated. The values of activity coefficients correlated in mole fractions, \( \bar{\gamma}_{i_{m.f.}} \), are not equal to the values for activity coefficients correlated in molar or equivalent fraction concentration units \( \bar{\gamma}_{i_{molar}} \) or \( \bar{\gamma}_i \), as is shown in Appendix A.

As indicated in Appendix A, the limiting values of activity coefficients correlated in mole fraction concentration units is as follows:

\[
\begin{align*}
\bar{x}_1 & \rightarrow 1.0, \quad \bar{\gamma}_{1_{m.f.}} \rightarrow 1.0, \quad \text{and} \quad \bar{a}_{1_{m.f.}} \rightarrow 1.0 \\
\bar{x}_1 & \rightarrow 0.0, \quad \bar{\gamma}_{1_{m.f.}} \rightarrow \left( \frac{K \cdot Q_r}{n_{T_1}^2 \cdot |z_2|} \right), \quad \text{and} \quad \bar{a}_{1_{m.f.}} \rightarrow 0.0
\end{align*}
\]

and for molar concentration units,

\[
\begin{align*}
\bar{c}_1 & \rightarrow c_T = |z_1|, \quad \bar{\gamma}_{1_{molar}} \rightarrow 1.0, \quad \text{and} \quad \bar{a}_{1_{molar}} \rightarrow |z_1| \\
\bar{c}_1 & \rightarrow 0.0, \quad \bar{\gamma}_{1_{molar}} \rightarrow \left( \frac{K \cdot |z_1|}{n_{T_1}^2 \cdot Q_r} \right), \quad \text{and} \quad \bar{a}_{1_{molar}} \rightarrow 0.0
\end{align*}
\]
The relationship between activity coefficients based on molar versus mole fraction concentration units is given in equation (A-18) as:

\[ n_{Y_1}^{\text{molar}} = n_{Y_1}^{\text{m.f.}} \cdot \frac{1}{(1-VR) \cdot x_1 + VR} \]  

(2-90)

noting that \( x_1 \) is the equivalent fraction

The following procedure is followed to make the standard state correction. The equilibrium relationship in molar concentration units for components 1 and 2 is:

\[ \bar{K}_{12}^{\text{molar}} = \left[ \begin{array}{c}
\frac{|z_2|}{a_1^{\text{molar}}} \\
\frac{|z_1|}{a_2^{\text{molar}}} \\
\end{array} \right] \cdot \left[ \begin{array}{c}
\left( x_2 \cdot Y_2 \right)^{\text{int}} |z_1| \\
\left( x_1 \cdot Y_1 \right)^{\text{int}} |z_2| \\
\end{array} \right] \cdot \text{norm}(z_1 - z_2) \cdot \left[ \begin{array}{c}
\frac{|z_1|}{|z_2|} \\
\frac{|z_2|}{|z_1|} \\
\end{array} \right] \]  

or simplifying:

\[ \bar{K}_{12}^{\text{molar}} = \left[ \begin{array}{c}
-\frac{|z_2|}{a_1^{\text{molar}}} \\
-\frac{|z_1|}{a_2^{\text{molar}}} \\
\end{array} \right] \cdot \text{Kelec} \]  

(2-91)

where:

\[ \bar{a}_i^{\text{molar}} = c_i \cdot \bar{Y}_i^{\text{molar}} \]  

(2-93)

The equilibrium relationship in mole fraction concentration units for components 1 and 2 is:

\[ \bar{K}_{12}^{\text{m.f.}} = \left[ \begin{array}{c}
-\frac{|z_2|}{a_1^{\text{m.f.}}} \\
-\frac{|z_1|}{a_2^{\text{m.f.}}} \\
\end{array} \right] \cdot \text{Kelec} \]  

(2-94)
where:

\[
\bar{a}_{i,m.f.} = \bar{x}_i \bar{y}_i \bar{y}_i^{*} \tag{2-95}
\]

The molar activity coefficient relationship can be given as:

\[
\Delta \ln \left( \frac{\bar{a}_1 |z_2|}{\bar{a}_2 |z_1|} \right)_{\text{molar}} = \Delta \ln \left( \frac{\bar{a}_1 |z_2|}{\bar{a}_2 |z_1|} \right)_{\text{mf}} + \Delta \ln \left( \frac{\bar{y}_1 |z_2|}{\bar{y}_2 |z_1|} \right)_{\text{molar}} + \Delta \ln \left( \frac{\bar{y}_1 |z_2|}{\bar{y}_2 |z_1|} \right)_{\text{mf}}
\]

Combining equations (2-92) and (2-94) yields:

\[
\frac{\bar{K}_{12,\text{molar}}}{\bar{K}_{12,\text{mf}}} = \left( \frac{\bar{a}_1 |z_2|}{\bar{a}_2 |z_1|} \right)_{\text{molar}} \left( \frac{\bar{a}_1 |z_2|}{\bar{a}_2 |z_1|} \right)_{\text{mf}} \tag{2-97}
\]

and in view of equations (2-93) and (2-95), equation (2-97) can be expanded to give the relationship:

\[
\frac{\bar{K}_{12,\text{molar}}}{\bar{K}_{12,\text{mf}}} = \frac{\bar{v}_{12,\text{molar}}}{\bar{v}_{12,\text{mf}}} \left( \frac{\bar{a}_1 |z_2|}{\bar{a}_2 |z_1|} \right)_{\text{mf}} \left( \frac{\bar{y}_1 |z_2|}{\bar{y}_2 |z_1|} \right)_{\text{molar}} \left( \frac{\bar{y}_1 |z_2|}{\bar{y}_2 |z_1|} \right)_{\text{mf}} \tag{2-98}
\]
The ratio expressed by the following relationship,

\[
R = \frac{x_{1 mf}}{x_{2 mf}} \left( \begin{array}{c} -z_1 \\ c'_1 \\ -z_2 \\ c'_2 \end{array} \right)
\]

(2-99)

can be transformed to an expression based on equivalent fraction concentration units by use of the relationships:

\[
\bar{x}_{1 mf} = \frac{c'_1}{\left( \frac{Q_r}{|z_2|} + (1-VR) \cdot c'_1 \right)}
\]

(2-100)

and:

\[
\bar{x}_{2 mf} = \frac{c'_2}{\left( \frac{Q_r}{|z_1|} + (1- \frac{1}{VR}) \cdot c'_2 \right)}
\]

(2-101)

where \( VR = \frac{|z_1|}{|z_2|} \)

The expression for \( R \) becomes:

\[
R = C \cdot \frac{\left( (1-VR) \cdot \bar{x}_1 + VR \right) |z_1|}{\left( (1-VR) \cdot \bar{x}_2 + VR \right) |z_2|}
\]

(2-102)

where:

\[
C = \left( \frac{|z_1|}{Q_r} \right) \cdot \left( \frac{Q_r}{|z_2|} \right) \cdot \left( \frac{1}{VR} \right)
\]
In view of equations (2-96) and (2-98), and (2-102), the following relationship obtains:

\[
\tau = \frac{\ln \left( \frac{|z_2|}{|z_1|} \right)}{\gamma_2} = \frac{\ln \left( \frac{|z_2|}{|z_1|} \right)}{\gamma_2} + \frac{\ln \left( \frac{e_{12}}{e_{13}} \right)}{\gamma_2} + \frac{\ln(C)}{\gamma_2}
\]

By carrying out the differentiation of the last two terms in equation (2-103), the final conversion equation is obtained:

\[
\tau = \frac{\ln \left( \frac{|z_2|}{|z_1|} \right)}{\gamma_2} + \frac{\ln \left( \frac{e_{12}}{e_{13}} \right)}{\gamma_2} + \frac{\ln(C)}{\gamma_2}
\]

The substitution of the right side of equation (2-104) in (5-21) corrects the activity coefficient relationship from molar units to mole fraction units. Obviously, if \(|z_1| = |z_2|\) the second right hand term is zero, and the activity coefficient relationship is the same whether molar, mole fraction or equivalent fraction concentration units are utilized.

Since it would be more convenient to take the partial
of \( \ln \left( \frac{-|z_2|}{Y_1} \right) \) with respect to mole fraction concentration units rather than equivalent fraction concentration units, the following can be arranged:

\[
\frac{\partial (\ldots)}{\partial \bar{x}_1} = \frac{\partial (\ldots)}{\partial \bar{x}_1_{mf}} \cdot \frac{\partial \bar{x}_1_{mf}}{\partial \bar{c}_1} \cdot \frac{\partial \bar{c}_1}{\partial \bar{x}_1} \tag{2-105}
\]

and since,

\[
\bar{x}_1 = \frac{\bar{c}_1 \cdot |z_1|}{Q_r}, \quad \frac{\partial \bar{c}_1}{\partial \bar{x}_1} = \frac{Q_r}{z_1} \tag{2-106}
\]

plus equation (2-100) can be arranged to give,

\[
\bar{c}_1 = \left[ \frac{\bar{x}_1_{mf} \cdot Q_r}{(|z_2|) \cdot \left( 1 + \bar{x}_1_{mf} \cdot (VR-1) \right)} \right] \tag{2-107}
\]

then,

\[
\frac{\partial \bar{c}_1}{\partial \bar{x}_1_{mf}} = \frac{Q_r}{|z_2|} \cdot \left[ \frac{1}{\left( 1 + \bar{x}_1_{mf} \cdot (VR-1) \right)^2} \right] \tag{2-108}
\]

and finally,

\[
\frac{\partial (\ldots)}{\partial \bar{x}_1} = \frac{\partial (\ldots)}{\partial \bar{x}_1_{mf}} \cdot \left[ \frac{1 + \bar{x}_1_{mf} \cdot (VR-1)}{VR} \right]^2 \tag{2-109}
\]

The resulting transformation from molar activity coefficients to mole fraction activity coefficients is then;
It is obvious from the above that computations involving determination of the effective diffusion coefficients in non-ideal systems are greatly simplified if the activity coefficients have been correlated in equivalent fraction units.

2. Ternary Systems

The use of chemical potential as a driving force, if of value in describing binary systems, must be extended to ternary dynamic models if the concept is to be proved efficacious in this research. Again, the effective diffusion coefficients derived from equations (5-27) and (5-28) involve the terms,

\[
\frac{\partial \ln \gamma_3}{\partial x_3} \text{ molar}_{k/j} \quad \text{and} \quad \frac{\partial \ln \gamma_k}{\partial x_j} \text{ molar}_{j/k} \quad \text{and} \quad 3
\]

in which the molar activity coefficients are employed. Since all of the ternary dynamic batch simulation runs were made with activity coefficients derived in equivalent fraction concentrations, there was no activity coefficient standard state correction required to satisfy equations (2-111) expressions. Equations (5-27) and (5-28) were solved by direct utilization of the equivalent fraction activity coefficients.
CHAPTER III

ELECTROLYTE PHASE - ACTIVITY COEFFICIENTS OF IONIC SPECIES
& EFFECT ON DIFFUSION COEFFICIENTS

A. Activity Coefficients - Electrolyte Phase

1. General

The various methods for describing ion exchange equilibria
the thermodynamic equilibrium constant, whether it be based on
molalities or equivalent fractions (see chapter on Equilibrium),
require knowledge of the activity coefficients of ionic species
in the electrolyte. Most investigators have dealt with binary
systems, and have either assumed ideality, or invoked Harned's
Rule for the electrolyte phase. Harned's Rule (105) states that
at constant ionic strength, I, the ratio of activity coefficients
in an electrolyte mixture is constant, and equal to the ratio for
the pure electrolyte solutions of the various ionic species at
the same ionic strength.

\[ \log \gamma_1 = \log \gamma_{1(o)} - \alpha_1 \cdot m_2^{\text{bulk}} \]  

(3-1)

where:

- \( \gamma_1 \) = activity coefficient of ionic species
  1 in a mixture of 1 and 2

- \( \gamma_{1(o)} \) = activity coefficient of ionic species
  1 in a solution containing only species
  1 at a total solution molality = \( m_{\text{bulk}} \)

- \( m_2 \) = molality of species 2 in mixture

likewise:

\[ \log \gamma_2 = \log \gamma_{2(o)} - \alpha_2 \cdot m_1^{\text{bulk}} \]  

(3-2)
Presumably this treatment could be extended to ternary systems, but the literature search for this project did not find any reference to such extension of Harned's Rule.

In any event, during exchange of ions of differing valences, the ionic strength, I, of the electrolyte phase will change. For example, during the exchange of a valence 1 cation species for a cation species having a valence of 2, at 1 normal electrolyte solution strength and with anion valence of 1:

\[
I = \frac{1}{2} \text{norm} \left\{ \sum_{i=1}^{n} x_{i_{\text{bulk}}} \cdot |z_i| + |z_{\text{anion}}| \right\} 
\]

(3-3)

where:

- \( I \) = ionic strength
- \( x_{i_{\text{bulk}}} \) = equivalent fraction of cation
- \( z_i \) = valence of ion
- \text{norm} = solution normality

For example:

- if \( x_{i_{\text{bulk}}} = 1 \) for \( z_i = 1 \)
  \[
  I = \frac{1}{2} \cdot (1 + 1) = 1
  \]

- if \( x_{i_{\text{bulk}}} = 1 \) for \( z_i = 2 \)
  \[
  I = \frac{1}{2} \cdot (2 + 1) = 1.5
  \]

So conditions for application of Harned's Rule during multivalent exchange will not hold. Reference is made to Robinson and Stokes (105) who state that Harned's Rule is valid for systems such as \( \text{HCl-BaCl}_2 \) provided the total ionic strength is kept constant, as contrasted to constant molality.
2. **Treatment of Electrolyte Activity Coefficients - This Research**

Smith and Woodburn (116), in their treatment of ternary equilibrium for the system $\text{SO}_4^2$, $\text{NO}_3^-$, & $\text{Cl}^-$ used activity coefficients for the electrolyte phase based on the extended Debye-Huckel equation given by Robinson and Stokes (103), and interaction parameters by Truesdell and Jones (128). For this work, the Bromley version of the extended Debye-Huckel equation (9) was selected. Bromley has published (9) values of his short range interaction parameters, $B$, for 175 salts in aqueous solution. Parameter $B$ can also be estimated by combining contributions of individual cations (36) with those of individual anions (40). The predictive range of the Bromley equation extends to 6 molal concentration, and is available in a form which permits the approximation of the activity coefficients of individual ionic species in mixtures of both cations and anions. The tables in Appendix C are for runs made in this research project to test this correlation against data for a variety of salts (total of 17) with molality ranging from 0.1 to 6.0, as given in Robinson and Stokes (106). The averaged root mean square of % relative difference, actual to predicted, was 2.7% for the combined total of all 17 salts.

3. **The L.A. Bromley equation is:**

For salt in aqueous solution, where $\gamma_\pm$ is the molal activity coefficient for the salt at a given solution ionic strength,  

$$
\log \gamma_\pm = \left( -0.511 \cdot |z_+z_-| \cdot I^{1/2} \right) + \left( 0.06 + 0.60 \cdot B_{12} \right) \cdot \left( |z_+z_-| \cdot I \right) + B_{12} \cdot I^{2/3}
$$
where:

\[ I = \text{ionic strength} \]
\[ I = \frac{n}{2} \sum_{i=1}^{n} m_i \cdot z_i^2 \]  (3-5)

\[ m_i = \text{molality of species } i \]
\[ z_i = \text{valence of species } i \]

\[ i = 1, 3, 5 \text{---represents cations} \]
\[ i = 2, 4, 6 \text{---represents anions} \]

\[ B_{12} = \text{experimental correlation constant for salt 12} \]

**Single Ionic Species**

Since:

\[ \log \gamma_{12} = \left( \frac{|z_2|}{(|z_1| + |z_2|)} \right) \cdot \log \gamma_1 + \left( \frac{|z_1|}{(|z_1| + |z_2|)} \right) \cdot \log \gamma_2 \]  (3-6)

it was shown that:

\[ \log 1 = \left( \frac{-0.511 \cdot z_{12}^{-2} \cdot I^{\frac{1}{2}}}{(1 + I^{\frac{1}{2}})} \right) + F_1 \]  (3-7)

where for anions 2, 4, ---; molalities are \( m_2, m_4, \ldots m_n \)

and,

\[ F_1 = \bar{B}_{12} \cdot \bar{z}_{12}^2 \cdot m_2 + \bar{B}_{14} \cdot \bar{z}_{14}^2 \cdot m_4 + \ldots \bar{B}_{1n} \cdot \bar{z}_{1n}^2 \cdot m_n \]  (3-7)

plus,

\[ \bar{z}_{12} = \frac{(|z_1| + |z_2|)}{2} \quad \bar{z}_{14} = \frac{(|z_1| + |z_4|)}{2} \]  (3-9)

so,

\[ \bar{B}_{12} \approx \left( \frac{0.06 + 0.60 \cdot B_{12} \cdot (|z_1| \cdot |z_2|)}{1 + \left(1.5 \cdot \frac{1}{(|z_1| \cdot |z_2|)}\right)^2} \right) + B_{12} \]  (3-10)

with \( \bar{B}_{14} \) described analogously to \( \bar{B}_{12} \)
B. Diffusion Coefficients - Electrolyte Phase

1. General

Development of a pseudo electric field model for ionic diffusion through the boundary separating the electrolyte solution and the ion exchange is covered in Chapter IV. However, most other investigators have utilized ionic diffusion coefficients given in the literature as limiting values (infinite dilution of the electrolyte). This work includes correction to the limiting diffusivities of ionic species by utilizing the Nernst-Hartley equation (104) in an integrated form.

2. Nernst-Hartley Equation

The Nernst-Hartley equation (104) for dilute solutions is given as:

\[ D_\pm = D_0 \pm (1 + c_\pm \cdot \frac{d \ln \gamma_\pm}{dc}) \]  

(3-11)

Two correction versions were tested, the differential form as given, and an integrated form, both utilizing the Bromley equation to provide the relationship between concentration of ionic species and its activity coefficient. The derivations are given in Appendix D.

3. Treatment of Electrolyte Diffusion Coefficients - This Research

The integrated version of the Nernst-Hartley equation was chosen since, in many electrolytes, the activity coefficient first decreases, then increases with concentration. The differential form would hold during the initial decrease, but would not represent the relationship during and after the reversal point. The results for both methods tested on data for
17 salts, as given in Robinson & Stokes (106), are included in Appendix C.

The ability of the two models to predict the experimental values of diffusivities is given as:

<table>
<thead>
<tr>
<th></th>
<th>Diffusion Coefficients - Nernst-Hartley Correction</th>
</tr>
</thead>
<tbody>
<tr>
<td></td>
<td>Averaged Root Mean Square</td>
</tr>
<tr>
<td></td>
<td>of % Relative Difference</td>
</tr>
<tr>
<td>differential</td>
<td>26.6%</td>
</tr>
<tr>
<td>integral</td>
<td>13.2%</td>
</tr>
</tbody>
</table>

The integral version was adopted for inclusion in this modeling effort.
CHAPTER IV
TREATMENT OF ELECTROLYTE PHASE FILM COEFFICIENTS

A. General Background

1. Diffusion Coefficients of Ionic Species in the Film

Most investigators of ion exchange in binary systems calculate the ionic flux through the film surrounding the resin by using the infinite dilution diffusion coefficient of a single counterion, or the average of both ionic species. As was discussed in Chapter III, the diffusion coefficients of ionic species vary considerably with increasing electrolyte concentration. This chapter deals with development of a model for diffusion of up to three ionic species through the boundary layer separating the bulk fluid from the resin outer surface. The resin surface is assumed to be in equilibrium with the boundary layer fluid in contact with it. In all cases in this research, diffusion coefficients of ionic species are corrected from their values at infinite dilution to estimated values at the average film concentration for the species in the film, based on an integrated form of the Nernst-Hartley equation as described in Chapter III.

2. Film Coefficient Correlations for Packed Beds

Film coefficients for diffusion through the boundary layer surrounding resin beads are adequately computed by equations of the form:

\[ j = k_L \cdot \frac{\varepsilon \cdot S_c^\frac{1}{2}}{v_s} = A \cdot Re^n \]  

(Lightfoot (62))  \( (4-1) \)
Pan and David (83) and this investigation found Carberry's relationship (11) satisfactory for $Re > 1$ based on superficial velocity, or $Re \geq 3$ based on actual velocity, $\frac{V_s}{\varepsilon}$.

Carberry's equation for fixed beds is given as:

$$0 \quad k_L = 1.15 \cdot \frac{Re^{\frac{1}{4}} \cdot Sc^{\frac{1}{3}} \cdot D_{eff}}{\varepsilon^{\frac{1}{4}} \cdot dp} \quad (4-2)$$

where:

- $V_s$ = superficial velocity, cm/sec
- $\varepsilon$ = bed voidage
- $\rho$ = density, grams/cc
- $\mu$ = viscosity, grams/cm/sec
- $D_{eff}$ = effective diffusivity, cm$^2$/sec
- $dp$ = particle diameter, cm
- $Re = \frac{V_s \cdot dp}{\mu} \quad (4-3)$
- $Re' = \frac{V_s \cdot dp}{\mu} \quad (4-4)$
- $Sc = \frac{\mu}{\rho \cdot D_{eff}} \quad (4-5)$
- $k_L$ = film coefficient, cm/sec

For ion exchange in fluidized beds, Turner & Snowdon (131) found the coefficient should be about 0.81 vs. 1.15 used by Carberry for packed beds. For $Re' < 1$, the correlation by Koloini and coworkers (54,55) was found satisfactory in this investigation.

$$0 \quad k_L = 1.1 \cdot \frac{Re' \cdot Sc^{\frac{1}{3}} \cdot D_{eff}}{dp} \quad (4-6)$$
Plots of the Carberry and Koloini & Zumer correlations intercept at Re=1.0, and choice of preferred correlation would be made at that Re number.

Postulating a "film" of some thickness across which ionic species migrate as a function of concentration driving force, then

\[ k_L = \frac{D_{\text{eff}}}{\delta_{\text{eff}}} \]  \hspace{1cm} (4-7)

where \( \delta_{\text{eff}} \) = effective "thickness", in cm, as in Boundary Layer theory

so:

\[ \delta_{\text{eff}} = \frac{d \rho \cdot \varepsilon^{\frac{1}{2}}}{(1.15 \cdot \text{Re}^{\frac{1}{2}} \cdot \text{Sc}^{\frac{1}{3}})} \]  \hspace{1cm} (4-8)

A problem which other investigators have researched has been to determine an effective diffusion coefficient that can be used to compute a film coefficient which properly simulates ion exchange when the film represents the major resistance to mass transfer.

3. **Nernst-Planck Models for Ionic Migration in the Film - Other Investigators**

3.1 **Description of Nernst-Planck Model**

Investigators such as Turner & Snowdon (131), Van Brocklin & David (132), Pan & David (83), and Kataoka & Yoshida (50) have examined the effect of the concentration gradient of the coion in the "stagnant film" during binary ion exchange—caused by the electric field generated in the film to satisfy (1) condition of electroneutrality and (2) that no electric current flow through the film as a result of diffusion of counterions. As will be shown, this effect
causes the effective diffusion coefficient of a species to vary across a "film," although the flux of that species does not vary across the film at any time (t). The mathematical treatment of the physics of ionic movement through electrolytes by W. Nernst and M. Planck resulted in a relation known as the Nernst-Planck equation (42), which for dilute solutions is given as:

\[ N_j = - D_i \cdot \left( \nabla c_i + \frac{c_i z_i F \nabla \phi}{RT} \right) \]  

(4-9)

Equations for Film with Electric Field - Nernst-Planck Relationship

Assumptions

- Film has thickness \( \delta_{\text{eff}} \) as described above
- All velocity components in the film are zero
- Fluid beyond the film is well mixed at composition \( C_{i \text{ bulk}} \)
- Mass transfer is in the radial direction of the resin "sphere." Since the film dimension is small compared to the resin radius, the film can be treated as a slab with thickness \( \delta_{\text{eff}} \) and surface and volume based on \( r_{\text{eff}} = r_0 + \delta_{\text{eff}} \) where \( r_0 \) is the resin radius. (4-10)
- No accumulation in film

Molar flux in film is then given as:

\[ N_i^* = - D_i^* \cdot \left( \frac{\partial c_i^*}{\partial y} + \frac{c_i^* z_i^* F \frac{\partial \phi^*}{\partial y}}{RT} \right) \]  

\( i=1,2,a(\text{coion}) \)

(4-11)
where:

\[ N_{i}^{*} = \text{molar flux of species in film, moles/cm}^2/\text{sec} \]
\[ D_{i}^{*} = \text{diffusivity of species at film concentration, cm}^2/\text{sec} \]
\[ c_{i}^{*} = \text{concentration of species in the film, molarity, moles/liter} \]
\[ z_{i} = \text{valence of species} \]
\[ F = \text{Faraday constant} \]
\[ \phi^{*} = \text{electric field in film, volts} \]
\[ y = \text{film thickness dimension, cm} \]
\[ x = \text{column length dimension, cm} \]
\[ L = \text{column length, cm} \]

The assumption of electro neutrality requires:

\[ \sum_{i=1}^{2} |z_{i}| c_{i}^{*} = |z_{a}| c_{a}^{*} \quad (a \text{ is coion}) \quad (4-12) \]

The assumption of no electric current flow, and postulating a quasi-stationary state for the coion, or \( N_{a}^{*} = 0 \), results in:

\[ \frac{\partial c_{a}^{*}}{\partial y} = - \left( \frac{z_{a} c_{a}^{*} F}{RT} \right) \frac{\partial \phi^{*}}{\partial y} \quad (4-13) \]

and it follows from equation (4-12) that:

\[ \sum_{i=1}^{2} |z_{i}| \frac{\partial c_{1}^{*}}{\partial y} = + |z_{a}| \frac{\partial c_{a}^{*}}{\partial y} \quad (4-14) \]

No accumulation in film, i.e., flux of each species, is constant through film, resulting in:

\[ \frac{\partial N_{i}^{*}}{\partial y} = 0 \text{ at } (x,t) \quad i=1,2,a \quad (4-15) \]
Initial Condition:

\[ c_i^*(x) = c_i(x,t_0) \quad \text{ bulk} \quad \begin{cases} t=0 \\ 0 \leq x \leq L \end{cases} \quad (4-16) \]

Boundary Condition:

\[ c_i^* \left( (r_0+\delta_{eff}), x, t \right) = c_i(x,t) \quad \text{bulk} \quad \begin{cases} t \geq 0 \\ 0 \leq x \leq L \end{cases} \quad (4-17) \]

Equilibrium Relationship

\[ c_i^\text{int}(r_0,x,t) = f(c_i^\text{int},x,t) \quad \begin{cases} t \geq 0 \\ 0 \leq x \leq L \end{cases} \quad (4-18) \]

\[ \bar{c}_i^\text{int} = \text{molar concentration of species in resin} \]
\[ \text{in equilibrium with } c_i^\text{int} \text{ in film.} \]

These equations have been integrated by various investigators, and the results of three of these investigators will be discussed in detail.

3.2. Model of Kataoka and Yoshida (50)

Kataoka & Yoshida for equal valence exchange of two species, developed the following equations:

\[ o N_1^* = \left( \frac{6 \cdot k^0 \cdot \beta}{d_p} \right) \cdot (c_1^\text{bulk} - c_1^\text{int}) \quad (4-19) \]

where species 1 is migrating to the resin,

and:

\[ \beta = \frac{k_L^*}{k_L^0} = \frac{\text{film coeff. - electric field}}{\text{film coeff. w/o electric field}} \quad (4-20) \]

\[ D_1^* = \text{diffusion coefficient of species 1, cm}^2/\text{sec} \]
Equilibrium is based on a mass action type relationship:

\[
\bar{K}_{12} = \left( \frac{x_1 \cdot x_2}{x_1 \cdot x_2} \right)_{\text{int}}^{\text{int}} \left( \frac{\cdot \cdot \cdot}{\cdot \cdot \cdot} \right)_{\text{int}}^{\text{int}}
\]

(4-21)

where equivalent fraction is defined as:

\[
x_1 = \frac{c_1 \cdot |z_1|}{c_a \cdot |z_a|}
\]

(4-22)

and:

- subscript int is the interface between resin/film
- superscript bar is the resin phase
- superscript star is the electrolyte film phase

Substitution and subsequent integration across the film yielded;

\[
\frac{\left[ x_1 + \alpha x_2 \right]_{\text{int}}^{\text{int}}}{\left[ (1 - \alpha) \cdot x_1 + \alpha \right]_{\text{int}}^{\text{int}}} \cdot (x_1^{*} + x_2^{*}) = 1
\]

(4-23)

where:

\[
\alpha = \frac{D_2}{D_1}
\]

(4-24)

\[
M = \frac{|z_1|}{|z_a|}; \text{ a is coion}
\]

(4-25)

and:

\[
\frac{D_2^*}{D_1} = \left( \frac{\alpha(M+1) \cdot (x_2^{*} + x_1^{*}) - 1}{(1-\alpha) \cdot (x_1^{*} - x_1^{*})} \right)
\]

(4-26)

where in equation (4-19):

\[
\beta = \frac{D_2^*}{D_1}
\]

(4-27)

Later in the chapter this treatment will be compared against a pseudo electric field treatment derived in this
research project for the extreme case of unfavorable exchange with \( K_{12} < 1 \) and \( \alpha < 1 \), specifically for the exchange of \( H^+ \) ion for \( Ag^+ \) ion on Dowex 50W-X8 resin.

Kataoka and Yoshida point out that in favorable exchange, i.e., \( K_{12} > 1 \), and with \( \alpha > 1 \), as in most industrially important cases, the breakthrough curve based on \( k_L \) calculated using the diffusion coefficient of the slower ion gives rise to errors of less than 30% in prediction of breakthrough curve, breakthrough time, and column utilization.

3.3 Model of Van Brocklin and David (132)

Van Brocklin & David for multivalent exchange of two species, developed the following equations.

The following factor was defined:

\[
0 \quad R_i = \frac{s_{i}^{*}}{N_i} = \frac{k_L}{k_L} \quad \text{for } i=1,2
\]  

\[(4-28)\]

where:

\[s = \text{area, } cm^2\]

or,

\[
R_i = \frac{\text{Nernst-Planck model}}{\text{Fick's Law model}}
\]  

\[(4-29)\]

so,

\[
0 \quad R_i = \left[ \frac{\partial c_i^{*}}{\partial n} + \frac{z_i c_i^{*} F \partial \phi^*}{RT} \right]_{\eta=0}
\]  

\[(4-30)\]
The film model was solved analytically by integration across the film. The other two models were solved by numerical methods on a computer—and use of the model involves table look up based on $D_i^*/D_j^*$ and film boundary
concentrations to obtain the $R_i$ factor. This model was used in carrying out ion exchange computations for a moving bed column (Pan & David (83)). Figures (4-1) through (4-6) from their work are shown depicting the effect on coion and counterion concentration, as well as electric field gradient, $\Phi^*$, for the various models. Pan & David (83) computed the height of a moving bed ion exchange column based on the Fick's Law model in the film and compared that to the height of a bed calculated with electric field effects, $(H/H_e)$. They conclude that "only for $H^+$ ion-multivalent ion exchange systems in dilute solution do highly unfavorable equilibrium and large effects of $R_i$ correction occur simultaneously." Normally these types of ion exchanges have no practical application, so it can be anticipated that for most simulations the $R_i$ correction factor will be small.

Pan and David (83) continue, "In practical ion exchange applications, this exchange operation is rather unusual. For exchanges with highly unfavorable equilibrium, but without $H^+$, the effect is expected to be small because of the small diffusivity ratio of the exchanging ion pair."

3.4 Model of Turner and Snowden (131)

Turner & Snowden integrated the basic Nernst-Planck film equations for two species interchanging, using the usual assumption of the coion in a quasistationary condition. Fluid bed experiments on resin with $H^+$ and $Na^+$ ions exchanging was the basis for a correlation which, though specific to this system, show agreement with the N.P. model.
Figure 4-1

Calculated $R_1$ Factors

Cu$^{++}$ – Na$^+$ Exchange
$Z_2 = -Z_3 = 1$
$Z_1 = 1$
$D_{Cu} = D_1 = 1$
$D_{Na} = D_2 = 1.87$
$X_1^* = 1.0$

Van Brocklin & Davis
Figure 4-2

Moving Ion Exchange Bed Height Correction Factor - Nernst Planck Effect vs. No Correction

$H_e$ - N.P. in Resin & Film
$H_c$ - N.P. in Resin - Ordinary Film Diffusion
$H_b$ - Ordinary Diffusion in Resin & Film

$\frac{H_b}{H_e}$ vs. $\frac{H_o}{H_e}$ vs. $C_o$ - Normality

Pan & David
Calculated Concentration Profiles - Boundary Layer Model
Na\(^+\)–H\(^+\) Exchange

Normalized Concentration and Potential

\(\eta_N\) Normalized B.L. Parameter

Van Brocklin & David
Calculated Concentration Profiles - Boundary Layer Model
Na\(^{+}\)-H\(^{+}\) Exchange

Van Brocklin & David
Calculated Concentration Profiles
Film Model
Na\textsuperscript{+}–H\textsuperscript{+} Exchange

\( \eta = y/\delta \)  
Film Thickness

Van Brocklin & David
Figure 4-6

Calculated Concentration Profiles
Film Model
$H^+ - Na^+$ Exchange

Van Brocklin & David
Their equations were:

\[
N_1^* = \frac{D_1^* D_2^* z_1^* c_1^* + z_2^* c_2^* + z_a^2 c_a^*}{D_1^* z_1^* c_1^* + D_2^* (z_2^* c_2^* + z_a^2 c_a^*)} \cdot \frac{dc_1^*}{dy} \tag{4-34}
\]

if,

\[
x_1^* = \frac{|z_1| c_1^*}{z_a c_a^*} \quad \text{and} \quad (x_1^* + x_2^*) = 1 \tag{4-35}
\]

\[
\alpha^* = \frac{D_1^*}{D_2^*} - 1 \tag{4-36}
\]

then:

\[
N_1^* = - D_1^* \left( \frac{|z_a|}{|z_1|} \right) c_a^* \left[ \frac{(|z_2| + |z_a|) + x_1^* (x_1^* - x_2^*)}{(|z_2| + |z_a|) + x_1^* (α^* - (|z_1| + |z_a|) + (|z_1| + |z_a|))} \right] \cdot \frac{dc_1^*}{dy} \tag{2-37}
\]

simplifying for \(|z_1| = |z_2| = |z_a| = 1\):

\[
c_a^* = c_{a\text{bulk}} \cdot \left( \frac{\alpha^* x_1 + 1}{\alpha^* x_1 + 1} \right) ^{-1} \tag{4-38}
\]

then by substituting (4-38) in (4-37) and integrating:

\[
N_1^* = \frac{D_1^*}{\delta_{\text{eff}}} \cdot \left[ \frac{2 \cdot c_{a\text{bulk}} \cdot (x_1^* - x_1^*)}{(1 + \alpha^* x_1^*) \sqrt{1 + (1 + \alpha^* x_1^*)^* (1 + \alpha^* x_1^* \text{int})}} \right] \tag{4-39}
\]

so if \(N_1^* = k_L \cdot c_{a\text{bulk}} \cdot (x_1^* - x_1^*)\),

\[
N_1^* = k_L \cdot c_{a\text{bulk}} \cdot (x_1^* - x_1^*) \tag{4-40}
\]
Good correlation was shown for the system investigated with the plot on Figure 4-7.

### 3.5 Summary of Film Models by Previous Investigators

In summary, previous investigators have demonstrated that in highly unfavorable exchange, particular with H⁺ because of its relatively high ionic mobility, breakthrough curve computations could be seriously in error if the electric field gradient in the film was not accounted for. They show that integrated solutions of the Nernst-Planck
Liquid Film Mass Transfer Coefficients In Ion Exchange Fluid Bed For $\text{H}^+ - \text{Na}^+$ System On Zeo-Karb 225 Nernst - Planck Relationship In Resin & Film

$$K_L \propto D_H^{2/3} F_1(y_H)$$

$$F_1(y_H) = 2(1 + 6y_H)^{1/3} + \sqrt{(1 + 6y_H^*) + (1 + 6y_H^*)(1 + 6y_H^*)}$$

$$y_H = \frac{\bar{y}_H^* + y_H^0}{2}$$

Turner & Snowden
equation for the film can correct film coefficients obtained from published correlations to better simulate "film controlled" ion exchange. The treatments have the following disadvantages:

a. The three investigators cited used ionic diffusivities measured at infinite dilution. This could be corrected using the treatment derived in this work.

b. The treatments are mainly limited to binary ion exchange and in two models the exchanging species have the same valence as the coion.

c. Numerical solution for $R_i$ factor by David and his coworkers (83, 132) has the disadvantage that $R_i$ factors must be generated off line. Modeling of ion exchange processes with this method depends on stored table look up with interpolation at each $\Delta Y$ and at $\Delta t$ increment, based on boundary condition concentrations, $x_i^{\text{bulk}}$ and $x_i^{\text{int}}$.

B. Pseudo Treatment for Electric Field Used in This Research Project

1. Assumptions

o Diffusivities of individual ions are corrected based on bulk solution normality, and as predicted by the integral form of the Nernst-Hartley equation (see Appendix D).

o The normality of the coion remains constant throughout the film; i.e., $\frac{\partial c_i^*}{\partial y} = 0$ and therefore the charges in the film coion would be analogous to the fixed charges in the resin.
All other equations in Nernst-Planck treatment of film remain as previously outlined.

The film effective boundary layer "thickness," $\delta_{\text{eff}}$, can be predicted from existing film coefficient correlations such as those of Carberry or Koloini and Zumer using an effective diffusivity $D_{\text{eff}}^*$ for computing the Schmidt No., $Sc$.

2. **Binary System - Linear Gradient**

Converting the molar flux equation to equivalent flux by:

$$J_i^* = |z_i^*| \cdot N_i^*$$

$$x_i = \frac{c_{i}^* |z_i|}{\text{norm}} \quad \text{or} \quad x_i^* = \frac{c_{i}^* |z_i|}{\text{norm}}$$

where $\text{norm} = \text{normality of coion}$

$$=[c_{a}^* |z_a|]_{\text{bulk fluid}}$$

$$=[c_{a}^* |z_a|]_{\text{film fluid}}$$

then:

$$N_i^* = - D_{i}^* \left( \frac{\partial c_i^*}{\partial y} + c_i^* z_i^* \frac{F}{RT} \frac{\partial \phi^*}{\partial y} \right)$$

$$i = 1, 2$$

transforms to:

$$\frac{J_i^*}{\text{norm}} = - D_{i}^* \left( \frac{\partial x_i^*}{\partial y} + x_i^* z_i^* \frac{F}{RT} \frac{\partial \phi^*}{\partial y} \right)$$

$$i = 1, 2$$

Electroneutrality requires:

$$\sum_{i=1}^{2} x_i^* = 1$$

(4-52)
no electric current plus \( J_a^* = 0 \), results in:

\[
\sum_{i=1}^{2} J_i^* = 0
\]  
(4-53)

in addition from equation (4-52):

\[
\sum_{i=1}^{2} \frac{\partial x_i^*}{\partial y} = 0
\]  
(4-54)

and:

\[
\frac{\partial x_a^*}{\partial y} = 0
\]  
(4-55)

Model for film

- linear gradient -

Illustration 4-1

since the concentration gradients are linear:

\[
x_{1_{\text{eff}}}^* = \left( \frac{x_{1_{\text{int}}}^* + x_{1_{\text{bulk}}}^*}{\delta_{\text{eff}}} \right)
\]  
(4-56)

\[
x_{2_{\text{eff}}}^* = \left( \frac{x_{2_{\text{int}}}^* + x_{2_{\text{bulk}}}^*}{\delta_{\text{eff}}} \right)
\]  
(4-57)
An effective diffusion coefficient, $D_{\text{eff}}^*$ can be derived if the Nernst-Planck equation is utilized and $\frac{\partial \phi}{\partial y}$ is eliminated in equation (4-51). This is shown in Chapter V in the derivation of an effective diffusion coefficient for two species exchanging in the resin phase. The "pseudo electric field" then is premised on a level distribution of coion in the film providing "fixed" opposite charges in the film analogous to the resin model.

Then:

$$D_{\text{eff}}^* = \frac{D_1^* D_2^* (x_1^* |z_1^*| + x_2^* |z_2^*|)}{(x_1^* |z_1^*| D_1^* + x_2^* |z_2^*| D_2^*)}$$  \hspace{1cm} (4-58)

and:

$$\frac{j_1^*}{\text{norm}} = + D_{\text{eff}}^* \frac{dx_1^*}{dy} = - D_{\text{eff}}^* \frac{dx_2^*}{dy} = - \frac{j_2^*}{\text{norm}}$$  \hspace{1cm} (4-59)

likewise:

$$\frac{j_1^*}{\text{norm}} = k_{\text{eff}}^* (x_1^* - x_1^*)_{\text{bulk}} \frac{dx_1^*}{dy} = D_{\text{eff}}^* \frac{dx_1^*}{dy}$$  \hspace{1cm} (4-60)

Since the concentration gradients were assumed to be linear over the film thickness, $\delta_{\text{eff}}$:

$$k_{\text{eff}}^* = \frac{D_{\text{eff}}^*}{\delta_{\text{eff}}}$$  \hspace{1cm} (4-61)
It is noted that Gopala Rao and his coworkers (92, 93) used a very similar treatment in modeling the ternary batch experiments they performed using radioactive tracer techniques. The major difference is that they calculated a fixed "hydraulic film thickness" which remained constant throughout the simulation. The derivation used in this work utilizes the "effective boundary layer thickness," which is updated for average species concentration in the film at each time step of the simulation. This treatment is closer to the previously discussed rigorous treatments.

In making the computation for $k_{\text{eff}}$, the treatment chosen was to use the concentrations $x_{i,bulk}^j$ and $x_{i,\text{int}}^j$ from the preceding or $j$th time step vs. the $j + \frac{1}{2}$th time step as prescribed by the Crank-Nicholson procedure. This circumvented the requirement to quasilinearize at the $j + 1$th time step and $k_{\text{eff}}^j$ was used in place of $k_{\text{eff}}^{j+1}$.

On binary runs, with sufficient time increments, i.e., small $\Delta j$, systematic bias on $k_{\text{eff}}$ was only in the 0.01% range.
3. Binary System - Non-Linear Gradient

Starting with the equation (4-60):

$$ \frac{J_1^*}{\text{norm}} = D_{\text{eff}}^* \frac{dx_1^*}{dy} $$

this can be put into the form:

$$ \frac{J_1^*}{\text{norm}} \int_{y=0}^{y=\Delta y} \frac{dx_{1\text{bulk}}}{x_{1\text{int}}} D_{\text{eff}}^*(x_1^*) \, dx_1^* $$

where equation (4-58) can be shown as:

$$ D_{\text{eff}}^* = \frac{D_1^* \cdot D_2^* \left( x_1^* |z_1| + (1-x_1^*) |z_2| \right)}{\left[ x_1^* |z_1| \cdot D_1^* + (1-x_1^*) |z_2| \cdot D_2^* \right]} $$

substitution of the above in equation (4-64), and integrating yields:

$$ \frac{J_1^*}{\text{norm}} \cdot \Delta y = \left[ \frac{D_1^* \cdot D_2^*}{|z_1| \cdot D_1^* - |z_2| \cdot D_2^*} \right] \cdot \left[ |z_2| - \frac{|z_2| \cdot D_2^* (|z_1| - |z_2|)}{|z_1| \cdot D_1^* - |z_2| \cdot D_2^*} \right] $$

\[ \times \ln \left[ \frac{(|z_1| \cdot D_1^* - |z_2| \cdot D_2^*) x_{1\text{bulk}} + |z_2| \cdot D_2^*}{(|z_1| \cdot D_1^* - |z_2| \cdot D_2^*) x_{1\text{int}} + |z_2| \cdot D_2^*} \right] \]

\[ + (|z_1| - |z_2|) \cdot (x_{1\text{bulk}} - x_{1\text{int}}) \]  

(4-65)

since:

$$ \frac{J_1^*}{\text{norm}} = k_{\text{eff}}^* \cdot (x_{1\text{bulk}} - x_{1\text{int}}) $$
then: \[ k_{\text{eff}}^L = \left\{ \frac{f(x_1^*_{\text{bulk}}, x_1^*_{\text{int}})}{\delta_{\text{eff}} (x_1^*_{\text{bulk}} - x_1^*_{\text{int}})} \right\} \]

(4-66)

and since:

\[ \Delta y = \delta_{\text{eff}} = \left\{ \frac{dp}{C \cdot \left( \frac{Re}{\varepsilon} \right)^n \cdot Sc_{\text{eff}}^{1/3}} \right\} \]

it follows that:

\[ k_{\text{eff}}^L = \left\{ \frac{f(x_1^*_{\text{bulk}}, x_1^*_{\text{int}})}{(x_1^*_{\text{bulk}} - x_1^*_{\text{int}})} \right\} \cdot \left\{ \frac{C \cdot \left( \frac{Re}{\varepsilon} \right)^n \cdot Sc_{\text{eff}}^{1/3}}{dp} \right\} \]

Again this value for \( k_{\text{eff}}^L \) would be calculated at the \( j^{th} \) vs. \( j^{th} + 1/2 \) step to avoid quasilinearization complications.

4. **Comparison of Binary Models - Linear versus Non-Linear Gradients**

Table 4-1 lists data from a comparison of the linear/versus the non-linear model in terms of ion flux (equivalents/cm²/sec per cm of "film thickness") at various film concentration differences, \( \Delta x_1 \), (driving force). Calculations were performed for the systems Ag⁺-H⁺, Ag⁺-Na⁺, and Sr⁺⁺-Cs⁺ in favorable exchange to determine the percent difference between models in predicted flux as the driving force was increased. As could be anticipated in univalent exchange, the Ag⁺-H⁺ system, with the largest ratio of diffusivities, gave higher differences than the Ag⁺-Na⁺ system. Multivalent exchange, represented by Sr⁺⁺-Cs⁺, at an even higher ratio of diffusivities gave the largest difference calculation of 30% at \( x_{\text{int}}^*_{\text{Sr}} = 0.10 \). In an
Table 4-1

Comparison - Linear vs Non-Linear Film Models

- **System Ag⁺ - H₂⁺**: \( D_1^* = 1 \times 10^{-6} \text{ cm}^2/\text{sec}, D_2^* = 10 \times 10^{-6} \text{ cm}^2/\text{sec} \)
  \[ \text{where } z_1 = 1 \text{ and } z_2 = 1 \]

<table>
<thead>
<tr>
<th>( x_1 \text{ bulk} )</th>
<th>( x_1^* \text{ int} )</th>
<th>( J_{1 \text{ eff}}^* \text{ equiv/cm/sec} ) (flux/cm of film thickness)</th>
<th>%diff.</th>
</tr>
</thead>
<tbody>
<tr>
<td>0.9</td>
<td>0.8</td>
<td>0.4255</td>
<td>0.4309</td>
</tr>
<tr>
<td>0.9</td>
<td>0.6</td>
<td>0.9195</td>
<td>0.9624</td>
</tr>
<tr>
<td>0.9</td>
<td>0.4</td>
<td>1.205</td>
<td>1.349</td>
</tr>
<tr>
<td>0.9</td>
<td>0.1</td>
<td>1.455</td>
<td>1.740</td>
</tr>
<tr>
<td>0.2</td>
<td>0.1</td>
<td>0.1156</td>
<td>0.1157</td>
</tr>
</tbody>
</table>

- **System Ag⁺ - Na⁺**: \( D_1^* = 1 \times 10^{-6} \text{ cm}^2/\text{sec}, D_2^* = 2.5 \times 10^{-6} \text{ cm}^2/\text{sec} \)
  \[ \text{where } z_1 = 1 \text{ and } z_2 = 1 \]

| 0.9 | 0.8 | 0.2041 | 0.2043 | nil |
| 0.9 | 0.6 | 0.5455 | 0.5504 | 0.9 |
| 0.9 | 0.4 | 0.8197 | 0.8368 | 2.1 |
| 0.9 | 0.1 | 1.143  | 1.191  | 4.2 |

- **System Sr⁺⁺ - Cs⁺⁺**: \( D_1^* = 0.14 \times 10^{-6} \text{ cm}^2/\text{sec}, D_2^* = 4.0 \times 10^{-6} \text{ cm}^2/\text{sec} \)
  \[ \text{where } z_1 = 2 \text{ and } z_2 = 1 \]

| 0.9 | 0.8 | 0.1236 | 0.1260 | 1.9 |
| 0.9 | 0.6 | 0.2430 | 0.2664 | 10.0 |
| 0.9 | 0.4 | 0.2920 | 0.3461 | 18.5 |
| 0.9 | 0.1 | 0.3140 | 0.4157 | 32   |
| 0.0 | 0.1 | -0.01542 | -0.01544 | nil (1) |

(1) In actual simulation of this system at 0.1 \( \text{m.} \) electrolyte concentration, the maximum \( \Delta x_{Sr^+} = -0.1 \), so error introduced by use of linear model was nil.
actual "batch mode" simulation of unfavorable exchange between Cs\textsuperscript{+} and Sr\textsuperscript{2+} in which both models were tested, the maximum % difference in flux between models was nil. This was true because the maximum driving force difference attained in the runs was only $\Delta x_{\text{Sr}}^* = 0.1$. This was due to the high selectivity of the resin for Sr\textsuperscript{2+} so that $x_{\text{Sr}_{\text{int}}}$, the equilibrium concentration, remained low. Since $x_{\text{Sr}_{\text{bulk}}}$ was zero, $\Delta x_{\text{Sr}}^*$ could not get very high during the exchange. Non-linearity effects are most pronounced with a high $\Delta x_i^*$ which occurs in favorable exchange, and this condition tends to be minimized due to the coupling of resin phase resistance to overall diffusion resistance. In many cases the species most favored by an ion exchanger tend to diffuse slower in the resin, so the resin phase resistance is a major factor in favorable exchange. The linear model was utilized in development of the column model since it could be extended readily to the ternary system.

5. Ternary Systems - Linear Gradients

5.1 General Development of Equations

Utilizing the assumptions made for the binary film model, the equations are:

$$
\frac{\partial J_i^*}{\partial \text{norm}} = - D_i^* \left( \frac{\partial x_i^*}{\partial y} + |z_i|x_i^* F \frac{\partial \Phi^*}{\partial y} \right) \quad i=1,2,3
$$

Electroneutrality conditions require:

$$
\sum_{i=1}^{3} x_i^* = 1
$$
condition of no electric current plus $\mathbf{J}_a^* = 0$ (coion flux) requires:

$$\sum_{i=1}^{3} \mathbf{J}_i^* = 0 \quad (4-70)$$

and based on equation (4-70):

$$\sum_{i=1}^{3} \frac{\partial x_i^*}{\partial y} = 0 \quad \text{with} \quad \frac{\partial a^*}{\partial y} = 0 \quad (4-71)$$

If (1) $\frac{\partial \phi^*}{\partial y}$ is eliminated from equation (4-48) as was done in development of the effective diffusion coefficients for the resin phase in Chapter V, (2) component 3 is eliminated in terms of components 1 and 2, and (3) concentration corrected diffusivities of the individual ionic species, $D_i^*$, $i = 1, 2, 3$, are utilized, the following equations result:

$$\frac{\mathbf{J}_1^*}{\text{norm}} = \left[ D_{11}^* \frac{\partial x_1^*}{\partial y} + \left( \frac{|z_1|}{|z_2|} \right) D_{21}^* \frac{\partial x_2^*}{\partial y} \right] \quad (4-72)$$

$$\frac{\mathbf{J}_2^*}{\text{norm}} = \left[ D_{22}^* \frac{\partial x_2^*}{\partial y} + \left( \frac{|z_2|}{|z_1|} \right) D_{12}^* \frac{\partial x_1^*}{\partial y} \right] \quad (4-73)$$

5.2 Model with Different Boundary Layer Thickness, Each Ionic Species

The coefficients $D_{ij}^*$ are calculated as with the resin derivation, and are given below in equations (4-96 to 99).
At this point, assume a linear driving force for each component across the "effective boundary layer" film thickness $\delta_{\text{eff} i}$ or $\Delta y_i$.

In order to simplify calculation complexity, it was desirable to generate effective film coefficients for each species, $k_{L1}^{\text{eff}}$ and $k_{L2}^{\text{eff}}$, since the non-variable (concentration independent) $k_{L}^{\text{eff}}$'s had been already incorporated into the first working ternary model developed in this work.

Writing the equations as follows;

$$\frac{J_{1}^*}{\text{norm}} = \frac{D_{11}^*}{\Delta y_1} \left[ \frac{x_{1_{\text{bulk}}} - x_{1_{\text{int}}}}{\Delta y_1} \right] + \frac{D_{12}^*}{\Delta y_2} \left[ \frac{x_{2_{\text{bulk}}} - x_{2_{\text{int}}}}{\Delta y_2} \right]$$  

and:

$$\frac{J_{2}^*}{\text{norm}} = \frac{D_{22}^*}{\Delta y_2} \left[ \frac{x_{2_{\text{bulk}}} - x_{2_{\text{int}}}}{\Delta y_2} \right] + \frac{D_{21}^*}{\Delta y_1} \left[ \frac{x_{1_{\text{bulk}}} - x_{1_{\text{int}}}}{\Delta y_1} \right]$$

led to equating $\frac{J_{1}^*}{\text{norm}}$ to the film coefficient form:

$$\frac{J_{1}^*}{\text{norm}} = k_{L1}^{\text{eff}} \cdot \frac{x_{1_{\text{bulk}}} - x_{1_{\text{int}}}}{\Delta y_1}$$  

and:

$$\frac{J_{2}^*}{\text{norm}} = k_{L2}^{\text{eff}} \cdot \frac{x_{2_{\text{bulk}}} - x_{2_{\text{int}}}}{\Delta y_2}$$
With this equality, the film coefficients could be computed:

\[
\frac{D_{11}^*}{\Delta y_1} + \left[ \frac{|z_1|}{|z_2|} \right] \frac{D_{12}^*}{\Delta y_2} \left( \frac{x_2^* - x_1^*}{x_2^*_{bulk} - x_1^*_{int}} \right)
\]

and:

\[
\frac{D_{22}^*}{\Delta y_2} + \left[ \frac{|z_2|}{|z_1|} \right] \frac{D_{21}^*}{\Delta y_1} \left( \frac{x_1^* - x_2^*}{x_2^*_{bulk} - x_1^*_{int}} \right)
\]

Resolution of the value of \( \Delta y_1 \) was required at this step. As shown with the binary case, boundary layer theory given in equations (4-62) and (4-61) leads to:

\[
\frac{D_i^*}{\Delta y_i} = \frac{k_{\text{L}_i}^*}{\Delta y_i} = \frac{\left( C \cdot \left( \frac{Re}{\varepsilon} \right)^n \cdot Sc_i \cdot D_i^* \right)^{\frac{1}{n}}}{dp}
\]  

for \( i = 1, 2, 3 \)

or,

\[
\frac{\Delta y_i}{dp} = \frac{\left( C \cdot \left( \frac{Re}{\varepsilon} \right)^n \cdot \frac{\mu}{\rho} \right)^{\frac{1}{n}}}{dp \cdot D_i^*}
\]

for \( i = 1, 2, 3 \)

With the binary system,

\[
D_{1\text{eff}}^* = D_{2\text{eff}}^* \quad \text{and} \quad \Delta y_1 = \Delta y_2 = \delta_{\text{eff}}
\]
In the ternary system case, there are two species moving in one direction, and the other species is moving in the opposite direction at a flux equal to the sum of the first two species:

\[ J_3^* = -(J_1^* + J_2^*) \]  \hspace{1cm} (4-82)

Since

\[ \frac{\Delta y_1}{\Delta y_2} = \frac{\text{constant} \cdot D_1^{* \frac{1}{3}}}{\text{constant} \cdot D_2^{* \frac{1}{3}}} = \left( \frac{D_1^*}{D_2^*} \right) \frac{1}{\text{eff}} \] \hspace{1cm} (4-83)

where \( D_i^{* \text{eff}} \) is an effective diffusivity for species \( i \). The values of \( D_1^{* \text{eff}}, D_2^{* \text{eff}}, \Delta y_1, \) and \( \Delta y_2 \) can be solved for by solution of a cubic equation which arises due to the \( 1/3 \) power in equation 4-80 to 83. The film coefficient for ionic species 3 is:

\[ k_{L_3}^{* \text{eff}} = -\left[ \frac{k_{L_1} \cdot (x_{1_{\text{bulk}}} - x_{1_{\text{int}}}) + k_{L_2} \cdot (x_{2_{\text{bulk}}} - x_{2_{\text{int}}})}{(x_{3_{\text{bulk}}} - x_{3_{\text{int}}})} \right] \] \hspace{1cm} (4-84)

This procedure was tried, again making the calculations for \( k_{L_i}^{* \text{eff}} \) at the \( j \)th time step versus \( j+\frac{1}{2} \)th time step to avoid quasilinearization complications, keeping \( \Delta t \) intervals small to avoid large systematic bias. Computational problems occurred when the \( J_3^* \) flux was the largest flux, since the procedure called for computation of the film coefficients and compositions of the two most favored species. Species 3 film coefficient and composition
were calculated by difference. This difficulty was caused when one of the preferred two species was at a very low concentration. Extraction of the cube root of very low values would result in arithmetic overflows or truncated computed values. Species 3 flux, calculated by difference, then gave rise to a poor flux balance thru the film. In effect three boundary layer film thicknesses were computed depending on concentration, effective diffusion coefficient and gradient of each species, and is shown in the following diagram.

Model for Film - Ternary System

Illustration 4-2

5.3 Use of Single Film Thickness - Maximum Flux

The following procedure was adopted to eliminate the computational difficulties caused by the procedure described
above. The maximum flux was determined at the \text{j}^{th} time step by checking species fluxes at the \text{j-1}^{th} time step:

\begin{equation}
\frac{J_{i}^{\text{max}}}{\text{norm}_{j}} = \frac{k_{L_{i}}^{\text{eff}_{j-1}}}{\delta_{\text{eff}_{j-1}, \text{i}}^{\text{max}}} \cdot (x_{i_{\text{bulk}_{j}}^{\text{i}}} - x_{i_{\text{int}_{j-1}}^{\text{i}}}) \tag{4-85}
\end{equation}

for \text{i}=1,2,3

The species having the maximum flux was chosen to compute the film thickness, \(\delta_{\text{eff}_{\text{max}}}^{\text{max}}\), at the \text{j}^{th} time step by the following derivation:

\begin{equation}
\frac{J_{i}^{\text{max}}}{\text{norm}} = \frac{D_{i}^{\text{eff}_{\text{max}}}}{\delta_{\text{eff}_{\text{max}}}} \cdot (x_{i_{\text{bulk}_{j}}^{\text{i}}} - x_{i_{\text{int}_{j}}^{\text{i}}}) \tag{4-86}
\end{equation}

where:

\begin{equation}
D_{i_{\text{eff}_{\text{max}}}}^{\text{eff}_{\text{max}}} = D_{i_{i}}^{\text{*}} + \frac{|z_{i_{j}}|}{|z_{i_{j}}|} \cdot D_{i_{j}}^{\text{*}} \cdot \frac{(x_{i_{\text{bulk}_{j}}^{\text{i}}} - x_{i_{\text{int}_{j}}^{\text{i}}})}{(x_{i_{\text{bulk}_{j}}^{\text{i}}} - x_{i_{\text{int}_{j}}^{\text{i}}})} \tag{4-87}
\end{equation}

then:

\begin{equation}
\delta_{\text{eff}_{\text{max}}}^{\text{max}} = \frac{\text{dp}^{\text{*}}}{C \cdot \left( \frac{\text{Re}}{\text{e}} \right) \cdot n \cdot \left( \frac{\mu}{\rho} \right)^{\frac{1}{2}}} = \Delta y_{i_{\text{max}}} \tag{4-88}
\end{equation}

and:

\begin{equation}
k_{L_{i}}^{\text{eff}_{\text{max}}} = \frac{D_{i_{\text{eff}_{\text{max}}}}^{\text{*}}}{\delta_{\text{eff}_{\text{max}}}} \tag{4-89}
\end{equation}
The equations used to calculate \( D_{ii}^{\ast} \) and \( D_{ij}^{\ast} \) were based on the assumption that since the "film thickness" was fixed at each time step and linear driving forces were assumed, the average film compositions could be calculated for each species as follows:

\[
\begin{align*}
D_{ii}^{\ast} & = \frac{D_{ii}^{\ast}}{\delta_{\text{eff-max}}} + \left( \frac{\delta_{\text{eff-max}}}{\delta_{\text{eff-max}}} \right) \\
D_{ij}^{\ast} & = \frac{(x_{i}^{\ast} - x_{j}^{\ast})}{(x_{i}^{\ast} - x_{j}^{\ast})}
\end{align*}
\]

and:

\[
\begin{align*}
k_{L_{ij}}^{\ast} & = - \left[ \frac{k_{L_{i}}^{\ast} \cdot (x_{i}^{\ast} - x_{j}^{\ast}) + k_{L_{j}}^{\ast} \cdot (x_{j}^{\ast} - x_{i}^{\ast})}{k_{L_{i}}^{\ast} \cdot k_{L_{j}}^{\ast} \cdot x_{i}^{\ast} \cdot x_{j}^{\ast}} \right] \\
& = - \left[ \frac{(x_{i}^{\ast} - x_{j}^{\ast})}{k_{L_{i}}^{\ast} \cdot k_{L_{j}}^{\ast} \cdot x_{i}^{\ast} \cdot x_{j}^{\ast}} \right]
\end{align*}
\]

where \( i \) = major flux and, \( k \) = smallest flux.

The equations used to calculate \( D_{ij}^{\ast} \) and \( D_{ij}^{\ast} \) were based on the assumption that since the "film thickness" was fixed at each time step and linear driving forces were assumed, the average film compositions could be calculated for each species as follows:

\[
x_{i}^{\ast} = \frac{x_{i}^{\ast} + x_{i}^{\ast}}{2} \quad (4-92)
\]

\[
x_{2}^{\ast} = \frac{x_{2}^{\ast} + x_{2}^{\ast}}{2} \quad (4-93)
\]

\[
x_{3}^{\ast} = \frac{x_{3}^{\ast} + x_{3}^{\ast}}{2} \quad (4-94)
\]

This procedure also gave rise to physically satisfying the requirement of equation 4-71, since:

\[
\sum_{i=1}^{3} \frac{\Delta x_{i}^{\ast}}{\delta_{\text{eff-max}}} = 0 \quad (4-95)
\]
The effective film diffusion coefficients based on species average film concentrations are:

\[
D_{ii}^* = D_i^* \left[ 1 - \frac{x_i^* \cdot |z_i| \cdot (D_i^* - D_k^*)}{3 \sum_{i=1}^{N} x_i^* \cdot |z_i| \cdot D_i^*} \right] \quad (4-96)
\]

\[
D_{jj}^* = D_j^* \left[ 1 - \frac{x_j^* \cdot |z_j| \cdot (D_j^* - D_k^*)}{3 \sum_{i=1}^{N} x_i^* \cdot |z_i| \cdot D_i^*} \right] \quad (4-97)
\]

\[
D_{ij}^* = \frac{D_i^* \cdot x_i^* \cdot |z_j| \cdot (D_j^* - D_k^*)}{3 \sum_{i=1}^{N} x_i^* \cdot |z_i| \cdot D_i^*} \quad (4-98)
\]

\[
D_{ji}^* = \frac{D_j^* \cdot x_j^* \cdot |z_i| \cdot (D_i^* - D_k^*)}{3 \sum_{i=1}^{N} x_i^* \cdot |z_i| \cdot D_i^*} \quad (4-99)
\]

where \( i \) = major flux and, \( k \) = smallest flux

The \( D_i^*, D_j^*, D_k^* \) thereby calculated are the diffusion coefficients for each ionic species in the film based on its average concentration, and corrected for electrolyte concentration by the Nernst-Hartley equation. Calculations were all carried out at \( j^{th} \) vs. \( j^{+1/2th} \) time step with minimum systematic bias as a result of the use of small time increments. Maximum dimensionless time (maximum
dimensionless time is the number of empty column volumes filled by the column volumetric feed during the run) was normally divided by 0.01 to 0.05 for computations. Systematic bias on all $k_{Li}$ for all time steps ranged from 0.1 to 0.2% on runs carried out on ternary systems.

6. Pseudo Electric Field Treatment Model Comparison to the Electric Field Gradient Model for Binary Systems

Comparison of results of the "pseudo electric" field treatment vs. Kataoka & Yoshida (50) rigorous treatment was performed for the most "difficult" system of Ag$^+$ & H$^+$, both for batch type experiments and column experiments. This is shown in Table 4-2 which lists the results of unfavorable and favorable exchange in the batch mode with the system Ag$^+$/$\text{H}^+$, both at 1.525 normal and 0.05 normal electrolyte concentrations. The concentration vs. time, film coefficients, as well as equivalent flux at the resin/electrolyte interface are practically identical for both models. Unfavorable exchanges at 0.05 normal (film control) showed the greatest differences between models after 315 sec, but these differences were only 1.5-2%.

Simulations of the favorable exchange of Ag$^+$ for H$^+$ in column experiments performed by Omatete (78) are shown in Figures 7-6 and 7-9 for Cl$^-$ normalities of 0.05 and 1.525. The simulation was performed using the pseudo electric field treatment; however, two other "S" curves are shown for which the rigorous Kataoka et al electric field model was used in the
Table 4-2

**DIFFUSION THROUGH FILM - BINARY ION EXCHANGE**

**BATCH SIMULATION RESULTS**

Calculated Comparison of Models - Electric Field Gradient (e.f.g.)
by Kataoka et al, and Pseudo Electric Field Model (p.e.f.), this work

<table>
<thead>
<tr>
<th>Time (sec)</th>
<th>Normality Interface</th>
<th>H⁺ conc. ave.</th>
<th>Flux at interface</th>
<th>Film coefficient</th>
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**Unfavorable Exchange H⁺Ag⁺-0.05 normal**

<table>
<thead>
<tr>
<th>Time (sec)</th>
<th>Ag⁺ conc. ave.</th>
<th>H⁺ conc. ave.</th>
<th>Flux at interface</th>
<th>Film coefficient</th>
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System Ag⁺-H⁺/NO₃

0.05 normal on Dowex 50W X8 resin
Superficial velocity = 0.2 cm/sec
Favorable Exchange Ag⁺ H⁺

Unfavorable Exchange H⁺ Ag⁺ 0.05 normal
<table>
<thead>
<tr>
<th>Time (sec)</th>
<th>Normality - interface</th>
<th>H⁺ conc. ave.</th>
<th>Flux at interface equiv/cm/sec</th>
<th>Film coefficient cm/sec</th>
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<tr>
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<td>1.525</td>
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<td>0.0024</td>
</tr>
</tbody>
</table>

**Unfavorable Exchange H⁺ Ag⁺**

| Time (sec) | Ag⁺ conc. ave. | |
|------------|----------------|
| 15         | 1.609 1.525    |
| 75         | 1.548 1.525    |
| 135        | 1.537 1.525    |
| 195        | 1.532 1.525    |
| 255        | 1.530 1.525    |
| 315        | 1.529 1.525    |

**System Ag⁺-H⁺/NO₃⁻**

1.525 normal on Dowex 50WX8 resin
superficial velocity = 0.08 cm/sec
Favorable Exchange Ag⁺ H⁺
simulation. The "S" curves generated for the two models are identical for all practical purposes.

It was concluded that if the pseudo electric field treatment model for worst case conditions, i.e.,

\[
\left(\frac{D_{Ag}^*}{D^*_{Ag}}\right) \approx 0.09 \text{ and } \bar{\kappa}_{H^+-Ag^+} \approx 0.12
\]

gave results essentially identical to the electric field model, the error in extending its use to other less demanding systems would be minimal—and its simplicity permitted ready extension to ternary systems, as derived above.
A. Diffusion Resin Phase Coefficients and Electrochemical Potential

1. General Background

This section deals with the effective rate of diffusion of ionic species within the ion exchanger, and development of the Nernst-Planck relationship which relates the effects of field potential within the resin to predict the effective interdiffusivity of migrating species. Additionally, the Nernst-Planck relationship will be derived with activities as potential, as compared to concentrations of the ionic species. The derivation of the Nernst-Planck model for use in predicting an effective diffusivity for two species migrating within an ion exchanger, given the "self" diffusion coefficients for both species, was first put forth by F. Helfferich and M. S. Plesset (42). As contrasted to Fick's first law, several underlying assumptions should be understood. First, that the electrochemical mobility of a species, $u_i$, can be related to the species "self" diffusion coefficient $D_i$ by the Nernst-Einstein relationship (75),

$$D_i = R \cdot T \cdot u_i$$

where:

- $D_i$ = diffusion coefficient, cm$^2$/sec
- $T$ = temperature, °K
- $R$ = gas constant (joule/mole/°K)
- $u_i$ = ionic mobility, cm$^-$mole/joule/sec
J. Newman (75) points out that the equation is strictly applicable only at infinite dilution. Helfferich (42) states, however, that experience has shown the applicability of the Nernst-Einstein equation to ion exchangers as a good approximation even at higher concentrations.

The second important assumption is that the presence of coions in the exchanger is neglected. Helfferich (42) points out that this assumption is valid if the Donnan exclusion of coions is strong. The Donnan potential, \( E_{\text{DON}} \) is defined (35) as:

\[
E_{\text{DON}} = \bar{\phi} - \phi = \frac{1}{z_i \cdot F} \left( R \cdot T \cdot \ln \left( \frac{a_i}{\bar{a}_i} \right) - \pi \cdot \nu_i \right)
\]  

(5-2)

where:

- \( \bar{\phi} - \phi \) = electric potential difference between the ion exchanger and the solution
- \( z_i \) = valence of species \( i \)
- \( F \) = Faraday's constant
- \( \bar{a}_i \) = activity of species \( i \) in solution
- \( a_i \) = activity of species \( i \) in the resin
- \( \pi \) = resin swelling pressure, bar
- \( \nu_i \) = partial molar volume of species \( i \) in the resin, cc/mole

The effect of this potential is to exclude coions from the exchanger, since migration of just a few ions of the same charge as the exchanger sites into the ion exchanger will quickly cause the electric field to build up, thereby
restricting further migration. This assumption tends to be invalid when electrolyte phase concentration is high, or the ion exchanger contains weakly dissociated active sites.

Implicit in almost all treatments of resin phase diffusion is that the resin acts as a solid solution through which the ions migrate.


In developing the general equations for ternary ion exchange in an ion exchanger, the use of irreversible thermodynamics to formulate the process will be followed, as was carried out by Barrer and Rees (6) in application to Ca\(^{++}\) and Sr\(^{++}\) exchange in chabazite, and by Rao and his coworkers (94) in their study of ternary ion exchange for the system Ba\(^{++}\), Sr\(^{++}\), Na\(^{+}\) on Dowex 50w x8 resin.

The general treatment of ternary systems can readily be reduced to use of concentration driving forces versus activities, and to a binary model if desired. The general equation describing ternary ion exchange in the resin phase is:

\[
\bar{N}_k = -\bar{c}_k \cdot \sum_{j=1}^{4} \bar{L}_{kj} \cdot \bar{\mu}_j \quad (j=1,2,3,4 \text{ is water})
\]  

(5-3)

where:

- \(\bar{N}_k\) = molar flux of component \(k\)
- \(\bar{c}_k\) = molar concentration of species \(k\), moles/liter
- \(\bar{\mu}_j\) = total electrochemical potential of species \(j\)
- \(\bar{L}_{kj}\) = phenomenological coefficients

for \(k = 1,2,3,4\)
J. Newman (76), commenting on ideas put forth by E. A. Guggenheim, states that the splitting of the electrochemical potential into a chemical potential and an electrical potential is purely a mathematical device and is arbitrary, having no physical significance. Newman (76) points out that the treatment will not produce difficulties in application to equilibrium models. In the case of ion exchange with fixed ionic sites, it appears to simplify model development. The Nernst-Planck model for transport of ionic species in dilute electrolytes is given as:

$$\nabla \bar{\mu}_j = \nabla \bar{\mu}_j^c + |z_j| \cdot F \cdot \nabla \bar{\phi}$$ \hspace{1cm} (5-4)

where:

- $\bar{\mu}_j^c$ = chemical potential of species $j$
- $z_j$ = valence of species $j$
- $F$ = Faraday's constant
- $\bar{\phi}$ = electric potential, volts

The detailed derivation of the equations describing diffusion in the resin phase is given in Appendix G, and leads to the following expression:

$$\bar{N}_k = -c_k \cdot \sum_{j=1}^{4} k_{kj} \cdot \nabla \bar{\mu}_j^c - c_k \cdot F \cdot \nabla \bar{\phi} \cdot \sum_{j=1}^{4} k_{kj} \cdot |z_j|$$ \hspace{1cm} (5-5)

for $k = 1,2,3,4$

The detailed derivation of the equations describing diffusion in the resin phase is given in Appendix G, and leads to the following expression:

$$\bar{N}_k = - \sum_{j=1}^{2} D_{kj} \cdot \nabla \bar{\mu}_j$$ \hspace{1cm} (5-6)

for $k = 1,2$
in which:

\[ D_{kk}' = c_k \frac{D_k}{c_k} \left[ 1 + \frac{\partial \ln \gamma_k}{\partial c_k} \right] - z_k^2 \left( \frac{1 + c_k \frac{\partial \ln \gamma_k}{\partial c_k}}{c_k} - D_3 \left[ 1 + c_3 \frac{\partial \ln \gamma_3}{\partial c_3} \right] \right) \]

for \( k = 1, 2 \)

and:

\[ D_{kj}' = c_k (z_k z_j) \frac{D_k}{c_k} \left[ 1 + \frac{\partial \ln \gamma_j}{\partial c_j} \right] - D_j \left[ 1 + c_j \frac{\partial \ln \gamma_j}{\partial c_j} \right] \]

\[ 3 \sum_{i=1}^{3} c_i z_i^2 D_i \]

for \( k = 1, 2 \) and \( j = 1, 2 \)

If the system is ideal, then,

\[ \frac{\partial \ln \gamma_k}{\partial c_k} = 0 \quad \text{for} \quad k = 1, 2 \]

so,

\[ D_{kk}' = D_k \left[ 1 - \frac{c_k z_k^2 (D_k - D_3)}{3 \sum_{i=1}^{3} c_i z_i^2 D_i} \right] \]

for \( k = 1, 2 \)
which represent the derivations of Rao and his coworkers (92) for an ideal ternary system.

If the system is Ideal and Binary, then:

\[ \bar{D}_{11} = -\bar{D}_{11} \cdot \nabla c_1 \]  \hspace{1cm} (5-12)

and in \( \bar{D}_{kk} = \bar{D}_1 \) and \( \bar{D}_2 = \bar{D}_3 \)

so by making the appropriate substitutions in equation (5-10), and clearing fractions the following equation results:

\[ \bar{D}_{12}^{\text{eff}} = \frac{\bar{D}_1 \cdot \bar{D}_2 \cdot (\bar{c}_1 \cdot z_1^2 + \bar{c}_2 \cdot z_2^2)}{(\bar{c}_1 \cdot z_1^2 \cdot \bar{D}_1 + \bar{c}_2 \cdot z_2^2 \cdot \bar{D}_2)} \]  \hspace{1cm} (5-13)

which is the Helfferich, Plesset (42) derivation for an ideal binary system.

If the System is Binary and Non-ideal, then by making the appropriate substitutions in equation (5-7) and clearing fractions there results:

\[ \bar{D}_{12}^{\text{eff}} = \frac{\bar{D}_1 \cdot \bar{D}_2 \cdot \left( \bar{c}_1 \cdot z_1^2 + \bar{c}_2 \cdot z_2^2 + z_1^2 \cdot \bar{c}_1 \cdot \frac{\partial \ln(\bar{Y}_2 \cdot \bar{c}_2)}{\partial \bar{c}_2} + z_2^2 \cdot \bar{c}_2 \cdot \frac{\partial \ln(\bar{Y}_1 \cdot \bar{c}_1)}{\partial \bar{c}_1} \right)}{\sum_{i=1}^{2} \bar{c}_i \cdot z_i^2 \cdot \bar{D}_i} \]  \hspace{1cm} (5-14)

where \( \bar{Y}_1 \) and \( \bar{Y}_2 \) have the molar reference state.
To simplify, equivalent fractions will be introduced, and total equivalents in the ion exchange resin is given by:

\[
-\bar{c}_1 \cdot |z_1| + -\bar{c}_2 \cdot |z_2| = Q_r (\text{equiv.-monoionic sites}) \text{ liter of resin} \tag{5-15}
\]

and \(\bar{x}_i = \frac{|z_i| \cdot -\bar{c}_i}{Q_r}\), then

\[
\frac{\partial (\_)}{\partial \bar{c}_i} = \frac{\partial (\_)}{\partial \bar{x}_i} \cdot \frac{|z_i|}{Q_r} \tag{5-16, 17}
\]

plus \(\bar{x}_1 + \bar{x}_2 = 1 \tag{5-18}\), and

\[
\frac{\partial (\_)}{\partial \bar{x}_1} = - \frac{\partial (\_)}{\partial \bar{x}_2} \tag{5-19}
\]

then by converting equation (5-14) to equivalent fraction units the following equation is obtained:

\[
\bar{D}_{12}^{\text{eff}} = \frac{\sum_{i=1}^{2} \bar{x}_i \cdot |z_i| + \bar{x}_2 \cdot |z_2| + \bar{x}_1 \cdot \bar{x}_2 \cdot \frac{\partial \ln \bar{Y}_2}{\partial \bar{x}_2} + \bar{x}_1 \cdot \bar{x}_2 \cdot \frac{\partial \ln \bar{Y}_1}{\partial \bar{x}_1}}{\Bar{D}_1 \cdot \Bar{D}_2 \cdot \left(\sum_{i=1}^{2} \bar{x}_i \cdot |z_i| \cdot \Bar{D}_i\right)} \tag{5-20}
\]

where \(\bar{Y}_1\) and \(\bar{Y}_2\) have the molar reference state.

By virtue of equation (5-19) and the properties of logarithms, the following arrangements can be made:

\[
\bar{D}_{12}^{\text{eff}} = \frac{\sum_{i=1}^{2} \bar{x}_i \cdot |z_i| + \bar{x}_2 \cdot |z_2| + \bar{x}_1 \cdot \bar{x}_2 \cdot \frac{\partial \ln \bar{Y}_2}{\partial \bar{x}_2} + \bar{x}_1 \cdot \bar{x}_2 \cdot \frac{\partial \ln \bar{Y}_1}{\partial \bar{x}_1}}{\Bar{D}_1 \cdot \Bar{D}_2 \cdot \left(\sum_{i=1}^{2} \bar{x}_i \cdot |z_i| \cdot \Bar{D}_i\right)} \tag{5-21}
\]

where \(\bar{Y}_1\) and \(\bar{Y}_2\) have the molar reference state.
and this is the Helfferich (45) derivation for the effective diffusion coefficient in non-ideal binary systems.

However, in using the equation in the convenient form of equivalent fractions, the flux equation can also be shown as,

$$\text{J}_1 = \text{N}_1 \cdot |z_1| \quad ; \quad \text{equivalents}$$

$$\text{cm}^2\text{sec}$$

and

$$\text{N}_1 \cdot |z_1| = \text{J}_1 = -\overline{D}_{12} \cdot \text{v}_{x_1} \cdot Q_r$$

$$= -\overline{D}_{12} \cdot \text{v}_{c_1} \cdot |z_1|$$

or

$$\frac{\text{J}_1}{Q_r} = -\overline{D}_{12} \cdot \text{v}_{x_1} = \overline{D}_{12} \cdot \text{v}_{x_2}$$

and

$$\text{J}_1 = -\text{J}_2 \quad (5-25), \text{ plus, } \text{v}_{x_1} = -\text{v}_{x_2} \quad (5-26)$$

It is important to note that the activity coefficient, $\gamma_1$, is a molar activity coefficient which is the same as the equivalent fraction activity coefficient. If the activity coefficient is based on mole fraction concentration units, a correction factor must be introduced to be used in the partials taken with respect to equivalent fraction concentration units. This is covered in Chapter II, part F.

Converting equations (5-7) and (5-8), which are the activity driving force effective diffusion coefficients for ternary systems, to equivalent fraction units results in:
where the \( \gamma_i \)'s have the molar reference state

These equations were derived utilizing the same relationships as were used to derive the equivalent fraction version of the non-ideal binary relationship. Again, \( \gamma_k \) or \( \gamma_j \) are molar activity coefficients which are the same as the equivalent fraction activity coefficients by definition of the standard states. Corrections must be applied to partials taken with respect to mole fraction concentration units, if the activity coefficients have been correlated in those units.
The Flux Equations for Ternary Systems are:

\[
\begin{align*}
\mathbf{J}_k &= \mathbf{N}_k \cdot |z_k| = Q_r \cdot |z_k| \cdot \left( \frac{2}{\sum_{j=1} \mathbf{D}_{kj} \cdot \frac{\mathbf{\nabla} x_j}{|z_j|}} \right) \quad (5-29) \\
\text{in equivalents per cm}^2\cdot\text{sec} \quad \text{for } k = 1, 2
\end{align*}
\]

since equivalent fraction is defined as:

\[
\bar{x}_k = \frac{|z_k| \cdot \bar{c}_k}{Q_r} \quad \text{for } k = 1, 2, 3 \quad (5-30)
\]

then,

\[
\mathbf{\nabla} \bar{c}_k = \left( \frac{Q_r}{|z_k|} \right) \cdot \mathbf{\nabla} \bar{x}_k \quad \text{for } k = 1, 2, 3 \quad (5-31)
\]

plus \( \bar{x}_1 + \bar{x}_2 + \bar{x}_3 = 1.0 \) \quad (5-32)

and \( \mathbf{\nabla} \bar{x}_3 = -\mathbf{\nabla} \bar{x}_2 - \mathbf{\nabla} \bar{x}_1 \) \quad (5-33)

and finally,

\[
\begin{align*}
\frac{
\mathbf{J}_1}{Q_r} &= - \left( \mathbf{D}_{11} \cdot \mathbf{\nabla} \bar{x}_1 + \left( \frac{|z_1|}{|z_2|} \right) \cdot \mathbf{D}_{12} \cdot \mathbf{\nabla} \bar{x}_2 \right) \quad (5-34) \\
\frac{
\mathbf{J}_2}{Q_r} &= - \left( \mathbf{D}_{22} \cdot \mathbf{\nabla} \bar{x}_2 + \left( \frac{|z_2|}{|z_1|} \right) \cdot \mathbf{D}_{21} \cdot \mathbf{\nabla} \bar{x}_1 \right) \quad (5-35)
\end{align*}
\]

The Continuity equations for the resin phase in a ternary system are:
3. Dimensionless Form and Preparation of Equations for Numerical Solutions

In setting up the diffusion and continuity equations for use in a coupled mode with the fluid phase in a column, it is useful to put the terms in dimensionless form.

Defining:

\[
\begin{align*}
BL &= \text{length of resin bed, cm} \\
1 &= \text{resin bed length dimension, cm} \\
r_o &= \text{particle radius, cm} \\
r &= \text{particle radius dimension, cm} \\
v_s &= \text{superficial velocity, cm/sec} \\
c &= \text{bed voidage} \\
t &= \text{time, sec} \\
E_D &= \text{axial dispersion coefficient, cm}^2/\text{sec} \\
k_{Li} &= \text{film coefficient, cm/sec} \\
\bar{D}_{ij}^{\text{eff}} &= \text{resin phase diffusion coefficient, cm}^2/\text{sec}
\end{align*}
\]
Svedberg (125), in his development of a numerical method for simulation of fixed-bed processes has investigated the use of a different grid spacing for intraparticle diffusion. A more accurate approximation of the derivatives would be to have a more dense grid near the particle surface where concentration gradients are greatest. Michelsen (72) and others have used the transformation $\rho = R^2$ to accomplish this alteration of particle grid density. The differential expressions with respect to $R$ become:

\[
0 \quad \frac{\partial^2 (\_)}{\partial R^2} + \frac{2}{R} \cdot \frac{\partial (\_)}{\partial R} = 4 \cdot \rho \cdot \frac{\partial^2 (\_)}{\partial \rho^2} + 6 \cdot \frac{\partial^2 (\_)}{\partial \rho} \quad (5-43)
\]

and,

\[
0 \quad \frac{\partial (\_)}{\partial R} = 2 \cdot \rho \cdot \frac{1}{2} \cdot \frac{\partial (\_)}{\partial \rho} \quad (5-44)
\]
Svedberg (125) checked this grid spacing for asymptotic constant pattern profiles which had previously been solved by Hall and his coworkers (34) which included both numerical and analytic solutions for constant pattern, self sharpening profile, ion exchange problems. Svedberg (125) showed that the results obtained with the $\rho$ transformation compared more favorably to the analytical solutions than the R grid spacings for a variety of Peclet & Biot number variations, column grid spacings, time increments, and resin grid increments, particularly for the solid diffusion model as opposed to a pore diffusion model.

The Continuity Equations Transformed become:

$$Q_r \cdot \frac{\partial x_1}{\partial T} = -\frac{4}{r_o^2 \rho^2} \cdot \frac{\partial (\rho^\frac{3}{2} \cdot \bar{J}_1)}{\partial \rho}$$  \hspace{1cm} (5-45)

or,

$$Q_r \cdot \frac{\partial x_1}{\partial T} = -\frac{\delta_s \cdot 4}{(1-\varepsilon) \rho^\frac{3}{2}} \cdot \frac{\partial (\rho^\frac{3}{2} \cdot \bar{J}_1)}{\partial \rho}$$  \hspace{1cm} (5-46)

and,

$$Q_r \cdot \frac{\partial x_2}{\partial T} = -\frac{\delta_s \cdot 4}{(1-\varepsilon) \rho^\frac{3}{2}} \cdot \frac{\partial (\rho^\frac{3}{2} \cdot \bar{J}_2)}{\partial \rho}$$  \hspace{1cm} (5-47)

Substitution of equation (5-34) and (5-35) in equation (5-46) and (5-47) yields:
\[
\frac{\partial x_1}{\partial T} = \frac{\delta_s}{(1-\epsilon)} \left( b_{11}' \left( 4\cdot \rho \cdot \frac{\partial^2 x_1}{\partial \rho^2} + 6\cdot \frac{\partial x_1}{\partial \rho} \right) + b_{12}' \left( \frac{z_1}{\partial z_2} \right) \left( 4\cdot \rho \cdot \frac{\partial^2 x_2}{\partial \rho^2} + 6\cdot \frac{\partial x_2}{\partial \rho} \right) \right)
\]

\[
+ 4\cdot \rho \cdot \frac{\partial b_{11}'}{\partial x_1} \left( \frac{\partial x_1}{\partial \rho} \right)^2 + 4\cdot \rho \cdot \frac{\partial b_{11}'}{\partial x_2} \left( \frac{\partial x_2}{\partial \rho} \right)^2
\]

\[
+ 4\rho \cdot \left( \frac{z_1}{\partial z_2} \right) \frac{\partial b_{12}'}{\partial x_2} \left( \frac{\partial x_2}{\partial \rho} \right)^2 + 4\rho \cdot \left( \frac{z_1}{\partial z_2} \right) \frac{\partial b_{12}'}{\partial x_1} \left( \frac{\partial x_1}{\partial \rho} \right) \left( \frac{\partial x_2}{\partial \rho} \right)
\]

the last four terms result from:

\[
\left( \frac{\partial b_{11}'}{\partial x_1} \right) \cdot \left( \frac{\partial x_1}{\partial \rho} \right) \text{ where:}
\]

\[
(5-49)
\]

\[
\frac{\partial b_{ij}'}{\partial x_i} = \left( \frac{\partial b_{ij}'}{\partial x_j} \right) \cdot \left( \frac{\partial x_i}{\partial \rho} \right) + \left( \frac{\partial b_{ij}'}{\partial x_j} \right) \cdot \left( \frac{\partial x_i}{\partial \rho} \right)
\]

and:

\[
(5-50)
\]

\[
\frac{\partial x_2}{\partial T} = \frac{\delta_s}{(1-\epsilon)} \left( b_{22}' \left( 4\cdot \rho \cdot \frac{\partial^2 x_2}{\partial \rho^2} + 6\cdot \frac{\partial x_2}{\partial \rho} \right) + b_{21}' \left( \frac{z_2}{\partial z_1} \right) \left( 4\cdot \rho \cdot \frac{\partial^2 x_1}{\partial \rho^2} + 6\cdot \frac{\partial x_1}{\partial \rho} \right) \right)
\]

\[
+ 4\cdot \rho \cdot \frac{\partial b_{22}'}{\partial x_2} \left( \frac{\partial x_2}{\partial \rho} \right)^2 + 4\cdot \rho \cdot \frac{\partial b_{22}'}{\partial x_1} \left( \frac{\partial x_2}{\partial \rho} \right) \left( \frac{\partial x_1}{\partial \rho} \right)
\]

\[
+ 4\rho \cdot \left( \frac{z_2}{\partial z_1} \right) \frac{\partial b_{21}'}{\partial x_1} \left( \frac{\partial x_1}{\partial \rho} \right) + 4\rho \cdot \left( \frac{z_2}{\partial z_1} \right) \frac{\partial b_{21}'}{\partial x_2} \left( \frac{\partial x_2}{\partial \rho} \right) \left( \frac{\partial x_2}{\partial \rho} \right)
\]
Boundary Conditions

flux across resin boundary where $\rho = 1$

\[ \dot{v}_1 |_{r=r_0} = k_{L_{1_{\text{eff}}}} \cdot (x_{1_{\text{bulk}}} - x_{1_{\text{int}}}) \cdot \text{norm} \]

\[ = \left( D_{11} \cdot \frac{\partial x_1}{\partial r} |_{r=r_0} + \left( \frac{|z_1|}{z_2} \right) D_{12} \cdot \frac{\partial x_2}{\partial r} |_{r=r_0} \right) \cdot Q_r \]

transformed to $\rho$ coordinates, (5-51) becomes:

\[ 2 \cdot \rho^{\frac{1}{2}} \cdot Q_r \left( D_{11} \cdot \frac{\partial x_1}{\partial \rho} |_{\rho=1} + \left( \frac{|z_1|}{|z_2|} \right) D_{12} \cdot \frac{\partial x_2}{\partial \rho} |_{\rho=1} \right) \]

\[ = k_{L_{1_{\text{eff}}}} \cdot (x_{1_{\text{bulk}}} - x_{1_{\text{int}}}) \]

since $Bi_i = 2 \cdot r_{o} \cdot k_{L_{i_{\text{eff}}}}$, then:

\[ \frac{4 \cdot Q_r}{Bi_{1_{\text{norm}}}} \left( D_{11} \cdot \frac{\partial x_1}{\partial \rho} |_{\rho=1} + \left( \frac{|z_1|}{|z_2|} \right) D_{12} \cdot \frac{\partial x_2}{\partial \rho} |_{\rho=1} \right) \]

\[ = (x_{1_{\text{bulk}}} - x_{1_{\text{int}}}) \]

and for component 2:

\[ \frac{4 \cdot Q_r}{Bi_{2_{\text{norm}}}} \left( D_{22} \cdot \frac{\partial x_2}{\partial \rho} |_{\rho=1} + \left( \frac{|z_2|}{|z_1|} \right) D_{21} \cdot \frac{\partial x_1}{\partial \rho} |_{\rho=1} \right) \]

\[ = (x_{2_{\text{bulk}}} - x_{2_{\text{int}}}) \]
gradient at resin center:

\[ \frac{\partial x_1}{\partial \rho} = 0, \quad \text{and} \quad \frac{\partial x_2}{\partial \rho} = 0 \]

\[ \begin{cases} \rho = 0 & \text{if } 0 \leq Y \leq 1 \\ T > 0 & \end{cases} \quad (5-55,56) \]

Initial Conditions

\[ \tilde{x}_1(t_0, \rho, Y) = \tilde{x}_1(t_0, Y) = \tilde{x}_2(t_0, Y) \quad \begin{cases} 0 \leq Y \leq 1 \\ 0 \leq \rho \leq 1 \end{cases} \quad (5-57,58) \]

Reduction of Ternary System to Binary System

Equation (5-48) is used and terms involving \( D_{12} \) and \( \tilde{x}_2 \) are eliminated.

\[ (5-59) \]

Boundary Conditions

flux across resin boundary where \( \rho = 1 \):

\[ \frac{4 \pi \rho \cdot Q}{Bi_{11} \cdot \text{norm}} \left. \left( \frac{\partial x_1}{\partial \rho} \right) \right|_{\rho=1} = (x_{1_{\text{bulk}}} - x_{1_{\text{int}}}) \quad (5-60) \]

gradient at resin center:

\[ \frac{\partial x_1}{\partial \rho}(0,Y,T) = 0 \quad \begin{cases} 0 \leq Y \leq 1 \\ T > 0 \end{cases} \quad (5-61) \]

Initial Conditions

\[ \tilde{x}_1(t_0, \rho, Y) = \tilde{x}_{1t_0}(\rho, Y) \quad \begin{cases} 0 \leq Y \leq 1 \\ 0 \leq \rho \leq 1 \end{cases} \quad (5-62) \]
B. Solution of the Resin Phase Equations, Numerical Methods

1. General Background

The coupled non-linear parabolic partial differential equations describing intraparticle diffusion of three species have been solved in this work by use of an implicit finite difference method of the Crank-Nicholson type (125). The following discussion will involve the resin phase only, and the solution of the equations described can be used to simulate the type of time varying concentration curves generated by Rao and his coworkers (92, 93) on several binary and ternary systems using the radioactive single particle tracer technique. Later, the two resin phase equations will be coupled with the two parabolic fluid phase equations to simulate fixed bed operations. In treating the "batch" case, it is assumed that bulk fluid phase concentrations of the species do not change (i.e., infinite reservoir); however, the concentration of each ionic species at the resin/fluid interface is in thermodynamic equilibrium. The actual thermodynamic treatment of ternary ion exchange equilibria is discussed in the Chapter II on Equilibrium. The equilibrium relationships can be treated here as:

\[ x_{1}^{*} = \phi (x_{1}^{*}, x_{2}^{*}, x_{1}^{*}, x_{2}^{*} ) \]  \hspace{1cm} (5-63)

\[ x_{2}^{*} = \theta (x_{1}^{*}, x_{2}^{*}, x_{1}^{*}, x_{2}^{*} ) \]  \hspace{1cm} (5-64)

non-linear implicit functions
Svedberg (125) linearized non-linear equilibrium expressions in his work by using a quasilinearization technique described by Lee (59), but applied it to a binary system only. The equilibrium functions for a ternary system would be iteratively solved as follows:

\[ x_1^{*} = x_1^{*} + (\overline{x}_1^{*} - \overline{x}_1^{*}) \cdot \left( \frac{\partial x_1^{*}}{\partial x_{1,\text{int}}^{*}} \right)_{n} \] (5-65)

\[ + (\overline{x}_2^{*} - \overline{x}_2^{*}) \cdot \left( \frac{\partial x_2^{*}}{\partial x_{2,\text{int}}^{*}} \right)_{n} \]

\[ x_2^{*} = x_2^{*} + (\overline{x}_2^{*} - \overline{x}_2^{*}) \cdot \left( \frac{\partial x_2^{*}}{\partial x_{2,\text{int}}^{*}} \right)_{n} \] (5-66)

\[ + (\overline{x}_1^{*} - \overline{x}_1^{*}) \cdot \left( \frac{\partial x_2^{*}}{\partial x_{1,\text{int}}^{*}} \right)_{n} \]

where \( n \) = iteration number

Use of this procedure allows \( x_i^{*} \) to be expressed as a linear function of \( \overline{x}_i^{*} \), and the system of linear algebraic equations is solved iteratively at each time step until convergence is attained. As Svedberg (125) points out, quasilinearization is basically a Newton method and has second order convergence. A serious limitation for Newton methods is that convergence is guaranteed only if the starting point lies close to the solution point. In the
technique used here, with sufficiently small time steps, the solution at time step $j$ is an excellent first guess when searching for a solution at time step $j+1$. Even with the most extreme non-linear conditions experienced in the simulations performed in this research, convergence has normally been reached in 3 to 5 iterations—except at the start of the simulation at $j=1$. If the concentration distribution and flux balances set up as initial conditions are too far from the $j=1$ solution points, convergence will not occur.

2. **Differentiation of Implicit Functions**

The thermodynamic equilibrium functions described above are implicit functions of the type:

$$f(x_{1\text{int}}, x_{2\text{int}}, x_{1\text{int}}^*, x_{2\text{int}}^*) = 0$$

$$g(x_{1\text{int}}, x_{2\text{int}}, x_{1\text{int}}^*, x_{2\text{int}}^*) = 0$$

where $x_{i\text{int}}^*$ are implicit functions of $x_{1\text{int}}$ and $x_{2\text{int}}$ and cannot be solved for explicitly. The partials

$$\frac{\partial x_{i\text{int}}^*}{\partial x_{j\text{int}}^*}, \frac{\partial x_{i\text{int}}^*}{\partial x_{j\text{int}}}$$

are obtained by the procedure described in Mickley, Sherwood & Reed (70) for differentiation of implicit functions.

3. **Quasilinearization of Other Non-Linear Terms**

Besides the equilibrium relationship, several other
terms in equation (5-48) and equation (5-50) are non linear, since multiplication of the finite difference equations approximating those terms result in algebraic variables which cannot be separated. These are given in Appendix H.

4. Treatment of the Diffusivity Functions

The functions $\bar{D}_{ij}$, and $\frac{\partial D}{\partial x_i}$, and $\frac{\partial D}{\partial x_j}$ calculated at time step $j$ for use in time step $j+1$. Since the Crank-Nicholson method uses values of variables at time step $j + \frac{1}{2}$ by averaging the $j$th & $j+1$th values, the computations carried out using this technique were subject to systematic bias. However, the algebraic complications in quasilinearizing the four diffusivities and the eight derivatives, particularly if they involved activity driving force (i.e., $\frac{\partial \ln y_i}{\partial x_i}$), were too great. This treatment was rationalized on the basis that if the time steps were reasonably small, total absolute systematic bias would be small. Table 5-1 gives the results of simulation of the seven binary system "batch" runs made by Rao and his coworkers (92, 93). The results are given in terms of total absolute systematic bias in percent, for all diffusivities grouped together, and all partials of the diffusivities grouped together, summed for all time and resin increments. At the end of the simulation, the sums were then divided by the number of resin and time increments used in the
simulation. Special attention was paid to the outermost resin increment since this increment represents the boundary condition and should have the greatest impact on solution vectors in the electrolyte phase during column simulation. The systematic bias for this increment was calculated separately. Comparison was made for two time increments, $\Delta T$, at values of 0.5 and 0.05. The time increment $\Delta T$ in the batch simulation runs is the increment into which actual time, in seconds, is divided. Most simulations were run at $\Delta T=0.01$ to 0.05. Although the systematic bias increased dramatically for $\Delta T=0.5$ in the system Na$^+$/Mn$^{2+}$, the time varying concentrations for this system simulated at the two $\Delta T$s differed by only 1-1.5% as shown in Table 5-2.

The model versatility in terms of its ability to handle ternary systems concentration dependent diffusivities and film coefficients, as well as use of activity driving forces, would justify the method used in view of the small bias introduced when the number of simulation time increments are 10-20 times the actual experimental time for the "batch" runs of Gopala Rao and his coworkers (92, 93).

C. Solution of System of Linear Equations – Seven Diagonal Array

1. Binary System

1.1 Tridiagonal Systems of Equations

The finite difference equations resulting from use of the Crank Nicholson method are based on breaking the geometry domains into grids to designate values of given
### Table 5-1

**Simulation Runs for experiments of Rao et al**

- Activity Driving Force -

\[ \Delta T = \frac{\text{total actual time for experimental run}}{\text{no. of time steps used in simulation}} \]

<table>
<thead>
<tr>
<th>System</th>
<th>( \bar{D}_{ij} )</th>
<th>( \bar{D}<em>{ij</em>{\text{max}}} )</th>
<th>( \partial \bar{D}_{ij}/\partial \bar{x}_i )</th>
<th>( \partial \bar{D}_{ij}/\partial \bar{x}_j )</th>
<th>( k_{L_{ij}}-\text{eff} )</th>
</tr>
</thead>
<tbody>
<tr>
<td>Mn\text{a} Cs\text{a}</td>
<td>0.09</td>
<td>0.009</td>
<td>0.1</td>
<td>0.01</td>
<td>0.134</td>
</tr>
<tr>
<td>Mn\text{a} Sr\text{a}</td>
<td>0.016</td>
<td>0.0016</td>
<td>0.01</td>
<td>0.001</td>
<td>-0.095</td>
</tr>
<tr>
<td>Sr\text{a} Mn\text{a}</td>
<td>0.02</td>
<td>0.002</td>
<td>0.01</td>
<td>0.001</td>
<td>0.05</td>
</tr>
<tr>
<td>Sr\text{a} Cs\text{a}</td>
<td>0.105</td>
<td>0.0105</td>
<td>0.08</td>
<td>0.008</td>
<td>0.18</td>
</tr>
<tr>
<td>Na\text{a} Cs\text{a}</td>
<td>0.083</td>
<td>0.008</td>
<td>0.08</td>
<td>0.008</td>
<td>0.120</td>
</tr>
<tr>
<td>Mn\text{a} Na\text{a}</td>
<td>0.09</td>
<td>0.009</td>
<td>0.08</td>
<td>0.008</td>
<td>0.88</td>
</tr>
<tr>
<td>Na\text{a} Mn\text{a}</td>
<td>0.32</td>
<td>0.020</td>
<td>2.7</td>
<td>0.013</td>
<td>-1.31</td>
</tr>
</tbody>
</table>
Table 5-2

SYSTEMATIC BIAS ON BINARY ION EXCHANGE SIMULATION

Comparison of Simulation Results for Rao et al

Experiment on Na\(^+\)-Mn\(^{++}\) System

\(\Delta T\) Effect

<table>
<thead>
<tr>
<th>Time - sec</th>
<th>(\bar{x}<em>{\text{Na}^+}</em>{\text{ave}}) at (\Delta T = 0.50)</th>
<th>(\bar{x}<em>{\text{Na}^+}</em>{\text{ave}}) at (\Delta T = 0.05)</th>
<th>% Difference</th>
</tr>
</thead>
<tbody>
<tr>
<td>15</td>
<td>0.802</td>
<td>0.788</td>
<td>1.8</td>
</tr>
<tr>
<td>30</td>
<td>0.689</td>
<td>0.681</td>
<td>1.2</td>
</tr>
<tr>
<td>45</td>
<td>0.611</td>
<td>0.603</td>
<td>1.3</td>
</tr>
<tr>
<td>60</td>
<td>0.550</td>
<td>0.542</td>
<td>1.5</td>
</tr>
<tr>
<td>75</td>
<td>0.498</td>
<td>0.492</td>
<td>1.2</td>
</tr>
<tr>
<td>90</td>
<td>0.454</td>
<td>0.449</td>
<td>1.1</td>
</tr>
<tr>
<td>105</td>
<td>0.415</td>
<td>0.411</td>
<td>1.0</td>
</tr>
<tr>
<td>120</td>
<td>0.381</td>
<td>0.377</td>
<td>1.0</td>
</tr>
<tr>
<td>135</td>
<td>0.350</td>
<td>0.346</td>
<td>1.1</td>
</tr>
<tr>
<td>150</td>
<td>0.322</td>
<td>0.319</td>
<td>1.0</td>
</tr>
<tr>
<td>165</td>
<td>0.297</td>
<td>0.294</td>
<td>1.0</td>
</tr>
<tr>
<td>180</td>
<td>0.274</td>
<td>0.270</td>
<td>1.5</td>
</tr>
</tbody>
</table>
variables which can then be used to approximate
derivatives at each time step. For example:

If the \( \rho \) dimension is broken into \( M \) grid points,
then:
\[
\Delta \rho = \frac{1}{(M-1)}, \text{ since } \rho = 1
\]

and,
\[
\left[ \frac{\partial x_1}{\partial \rho} \right]_j = \frac{\bar{x}_1(m+1,j) - \bar{x}_1(m-1,j)}{2\Delta \rho}
\]  \hspace{1cm} (5-69)

plus,
\[
\left( \frac{\partial^2 x_1}{\partial \rho^2} \right)_j = \begin{bmatrix}
\frac{x_1(m+1,j) - x_1(m,j)}{\Delta \rho} & \frac{x_1(m,j) - x_1(m-1,j)}{\Delta \rho} \\
\end{bmatrix}
\]  \hspace{1cm} (5-70)

\[
= \begin{bmatrix}
\frac{\bar{x}_1(m+1,j) + \bar{x}_1(m-1,j) - 2 \cdot \bar{x}_1(m,j)}{\Delta \rho^2} \\
\end{bmatrix}
\]

In the implicit Crank Nicholson method, the time
steps \( j \) and \( j + 1 \) are averaged, so that \( j+1 \)th
equivalent versions of the above partials approximations
are averaged with the \( j \)th values to represent the
partial derivative approximations as a function of time.

The linear algebraic equations developed, including
terms from the quasilinearization of non-linear
expressions, result in a coefficient matrix for the
variables which is tridiagonal since, as seen, \( m+1, m, \)
and \( m-1 \) terms are involved. Representation of the
system of equations which result from the finite difference approximation of the partial differential equation describing binary ion exchange for the resin phase would be:

\[ A_{i}^{n} \cdot \bar{x}_{1_{i}n+1}^{n} + D_{i}^{n} + E_{i}^{n} \cdot \bar{x}_{1_{n} \text{bulk}}^{n+1} (i,j+1)_{n+1} \]  

(5-71)

where:

\[ A_{i}^{n} = \begin{pmatrix} a_{1} & b_{1} \\ c_{2} & a_{2} & b_{2} \\ \vdots & \vdots & \vdots \\ c_{M} & a_{M} \end{pmatrix}_{i,n} \]

(5-72)

\[ \bar{x}_{1}^{n+1} = \begin{pmatrix} \bar{x}_{1}(1,i,j+1) \\ \bar{x}_{1}(2,i,j+1) \\ \vdots \\ \bar{x}_{1}(M,i,j+1) \end{pmatrix}_{n+1} \]

(5-73)

\[ D_{i}^{n} = \begin{pmatrix} d(1,i,j) \\ d(2,i,j) \\ d(m,i,j) \\ d(M,i,j) \end{pmatrix}_{n} \]

(5-74)

(5-75)

\[ E_{i}^{n} = \begin{pmatrix} 0 \\ 0 \\ 0 \end{pmatrix}_{n} \]

(5-75)

and \( \bar{x}_{1_{n} \text{bulk}}^{n+1} \) is the solution phase concentration at column increment \( i \).

This system of resin phase equations would be coupled with a similar system of equations describing
the solution phase concentration at each column length node, \( i \), for \( i=1 \) to \( I \). This development is treated in Chapter VII in which the column models are derived. In the "batch" case, \( x_{1 \text{ bulk}} \) is a constant and the term \( E_i \cdot x_{1 \text{ bulk}} \) can be added to the vector \( D_i \). It is noted that the vector \( D_i \) contains terms from the previous time step, \( j \), as well as the terms resulting from the quasilinearization of non-linear functions at the \( n^{th} \) iteration. The Thomas method involves an algorithm for solving systems of equations having tridiagonal coefficient matrices, and is very efficient. It was used by Svedberg (125) with satisfactory results, and is used in this research project to solve the binary systems.

1.2 Treatment of Boundary Conditions by Numerical Approximation

Two resin phase boundary conditions were approximated by finite difference equations and coupled to the general resin phase equation. The first, at the center of the resin particle is the boundary condition that was given in equation (5-61)

\[
\frac{\partial x_1}{\partial \rho}(0, Y, T) = 0 \quad \begin{cases} \rho = 0 \\ 0 \leq Y \leq 1 \\ T \geq 0 \end{cases}
\]

To solve this B.C., advantage is taken of the general equation (5-59), where \( \rho \) is set equal to zero, or:

\[
0 \times \frac{\partial x_1}{\partial T} = \frac{\delta_s}{(1-\epsilon)} \left( 6 \cdot \frac{\partial x_1}{\partial Y} \right)
\] (5-76)
and in difference terms, using a forward difference for

\[ \frac{\partial \bar{x}_i}{\partial \rho} = \left( \bar{x}_1(2,i,j+1)_{n+1} + \bar{x}_1(2,i,j) - \bar{x}_1(1,i,j+1)_{n+1} + \bar{x}_1(1,i,j) \right) \]

This treatment was used by Svedberg (125), and no problems were noted with this treatment either in his work or in the simulations carried out in this project.

The second boundary condition to be coupled to the general equation was at the resin film interface. Here, a derivative at the boundary must be replaced by a
finite difference representation. In at least one reference of many, V. G. Jenson & G. V. Jeffreys (47) suggest using a fictitious point beyond that boundary, which is eliminated by combination with the general equation. Svedberg (125) used this procedure in his work. Boundary condition equation (5-60), in finite difference representation would be:

\[
\frac{4 Q_r \, D_{11}}{B_i \, \text{norm}} \left\{ \left[ \frac{x_1(M+1,i,j+1)_{n+1} - x_1(M-1,i,j+1)_{n+1}}{4 \Delta \rho} \right] \right. \\
+ \left. \left[ \frac{x_1(M+1,i,j) - x_1(M-1,i,j)}{4 \Delta \rho} \right] \right. \\
= \left\{ \frac{x_{\text{bulk}}(i,j+1)_{n+1} - x_{\text{bulk}}(i,j)}{2} \right. \\
- \left. \left[ \frac{x_{\text{int}}^*(i,j+1)_{n+1} + x_{\text{int}}^*(i,j)}{2} \right] \right. \\
\]

where \( M = \) maximum \( m \) node, and \( M+1 \) is a fictitious point outside the resin boundary.

By setting up the general equation in a central difference notation, all \( M + 1 \) terms can be eliminated by cross multiplying coefficients and adding the B.C. equation to the general equation. The coefficients for the \( M \) and \( M-1 \) nodes are then obtained together with the solution vector for the \( M^{\text{th}} \) node which contains \( M \) and \( M-1 \) resin node elements, plus column fluid phase composition elements at the \( i \) column node. These resin
phase boundary condition treatments were used with the Ternary Column Model development as well, though the derivations were obviously more complicated algebraically because of equations for two species versus one.

2. Ternary System

When the partial differential equations describing ternary ion exchange are put into matrix notation, there results:

\[
A_{i_n} \cdot \bar{x}_{1, i_{n+1}} = D_{i_n} + R_{i_n} \cdot \bar{x}_{2, i_{n+1}} + E_{1_n} \cdot x_{1, \text{bulk}}^{(i,j+1)_{n+1}} - E_{2_n} \cdot x_{2, \text{bulk}}^{(i,j+1)_{n+1}}
\]

and:

\[
A_{i_n} \cdot \bar{x}_{2, i_{n+1}} = \Pi_{i_n} + O_{i_n} \cdot \bar{x}_{1, i_{n+1}} + \Omega_{2_n} \cdot x_{2, \text{bulk}}^{(i,j+1)_{n+1}} - \Omega_{1_n} \cdot x_{1, \text{bulk}}^{(i,j+1)_{n+1}}
\]

where

\[
A_{i_n} = \begin{pmatrix}
  a_1 & b_1 \\
  c_2 & a_2 & b_2 \\
  c_m & a_m & b_m \\
  c_M & a_M \\
\end{pmatrix}_{i,n}
\]

\[
R_{i_n} = \begin{pmatrix}
  r_1 & s_1 \\
  t_2 & r_2 & s_2 \\
  t_m & r_m & s_m \\
  t_M & r_M \\
\end{pmatrix}_{i,n}
\]

\[
\bar{x}_{1, i_{n+1}} = \begin{pmatrix}
  \bar{x}_{1, (1,i,j+1)} \\
  \bar{x}_{1, (2,i,j+1)} \\
  \bar{x}_{1, (m,i,j+1)} \\
  \bar{x}_{1, (M,i,j+1)} \\
\end{pmatrix}_{n+1}
\]

\[
\bar{x}_{2, i_{n+1}} = \begin{pmatrix}
  \bar{x}_{2, (1,i,j+1)} \\
  \bar{x}_{2, (2,i,j+1)} \\
  \bar{x}_{2, (m,i,j+1)} \\
  \bar{x}_{2, (M,i,j+1)} \\
\end{pmatrix}_{n+1}
\]
\[
D_{in} = \begin{pmatrix}
d(1,i,j) \\
n(2,i,j) \\
n(m,i,j) \\
n(M,i,j)
\end{pmatrix} \quad \text{(5-87)}
\]

\[
e_{1} = \begin{pmatrix}
o \\
o \\
o \\
e_{2}
\end{pmatrix} \quad \text{(5-88)}
\]

\[
e_{2} = \begin{pmatrix}
o \\
o \\
o \\
e_{2}
\end{pmatrix} \quad \text{(5-89)}
\]

It is noted that the vector \( D_{in} \) contains both \( x_1 \) and \( x_2 \) terms resulting from time step \( j \) or iteration \( n \) of non-linear term components resulting from quasilinearization.

Similarly

\[
\Lambda_{in} = \begin{pmatrix}
\lambda_1 & \mu_1 \\
k_2 & \lambda_2 & \mu_2 \\
k_m & \lambda_m & \mu_m \\
k_M & \lambda_M
\end{pmatrix} \quad \text{(5-90)}
\]

\[
\Lambda_{in} = \begin{pmatrix}
\lambda_1 & \mu_1 \\
k_2 & \lambda_2 & \mu_2 \\
k_m & \lambda_m & \mu_m \\
k_M & \lambda_M
\end{pmatrix} \quad \text{(5-90)}
\]

\[
\text{and } x_{2,i}^{n+1} = \begin{pmatrix}
x_2(1,i,j+1) \\
x_2(2,i,j+1) \\
x_2(m,i,j+1) \\
x_2(M,i,j+1)
\end{pmatrix} \quad \text{n+1} \quad \text{(5-91)}
\]

\[
\Theta_{in} = \begin{pmatrix}
\theta_1 & \eta_1 \\
\eta_2 & \theta_2 & \eta_2 \\
\eta_m & \theta_m & \eta_m \\
\eta_M & \theta_M
\end{pmatrix} \quad \text{(5-92)}
\]

\[
\text{and } x_{1,i}^{n+1} = \begin{pmatrix}
x_1(1,i,j+1) \\
x_1(2,i,j+1) \\
x_1(m,i,j+1) \\
x_1(M,i,j+1)
\end{pmatrix} \quad \text{n+1} \quad \text{(5-93)}
\]
terms resulting from the time step \( j \) or iteration \( n \) of non-linear term components resulting from quasilinearization.

The system of equations for the ternary case, as was shown in the binary case, is set up for coupling to two systems of equations describing the values of \( \bar{x}_1 \) \( (i,j+1)_{n+1} \) and \( \bar{x}_2 \) \( (i,j+1)_{n+1} \) in the solution phase moving through a column at nodes \( i = 1 \) to \( i = I \). In the "batch" case, \( \bar{x}_1 \) \( \text{bulk} \) and \( \bar{x}_2 \) \( \text{bulk} \) are constant, so the terms \( \Omega \bar{x}_1 \) \( \text{bulk} \), \( \Omega \bar{x}_2 \) \( \text{bulk} \), and \( \Omega^2 \bar{x}_2 \) \( \text{bulk} \) can be added to the vectors \( D_{m,n} \) and \( \Pi_{m,n} \) respectively—and subscripts \( i \) are not used.

When the system of equations described in matrix notation as equation (5-81) and equation (5-82) are combined, a 7 diagonal array results. This is illustrated in Figure 5-1 for a resin phase divided into four \( \rho \) segments, or \( M=5 \) nodes. An algorithm, analogous to the Thomas method (13, 125) for tri-diagonal arrays, was derived as part of this research to solve systems consisting of 7 diagonal coefficient matrices. The algorithm is given in Appendix E, along with an algorithm derived for a 5 diagonal
Seven Diagonal Array for Combined Ternary Resin Phase Equation Matrices

for $\rho$ divided into 4 segments, or $m = 5$ nodes

$$
\begin{bmatrix}
  a_{1,i} & -r_{1,i} & b_{1,i} & s_{1,i} & 0 & 0 & 0 & 0 & 0 \\
  -\beta_{1,i} & \lambda_{1,i} & -s_{1,i} & 0 & 0 & 0 & 0 & 0 & 0 \\
  c_{2,i} & -t_{2,i} & a_{2,i} & r_{2,i} & b_{2,i} & -s_{2,i} & 0 & 0 & 0 \\
  -n_{2,i} & \kappa_{2,i} & -\beta_{2,i} & \lambda_{2,i} & -s_{2,i} & 0 & 0 & 0 & 0 \\
  0 & 0 & c_{3,i} & -t_{3,i} & a_{3,i} & -r_{3,i} & b_{3,i} & -s_{3,i} & 0 \\
  0 & 0 & -n_{3,i} & \kappa_{3,i} & -\beta_{3,i} & \lambda_{3,i} & -s_{3,i} & 0 & 0 \\
  0 & 0 & 0 & 0 & c_{4,i} & -t_{4,i} & a_{4,i} & -r_{4,i} & b_{4,i} & -s_{4,i} \\
  0 & 0 & 0 & 0 & -n_{4,i} & \kappa_{4,i} & -\beta_{4,i} & \lambda_{4,i} & -s_{4,i} & 0 \\
  0 & 0 & 0 & 0 & 0 & 0 & c_{5,i} & -t_{5,i} & a_{5,i} & -r_{5,i} \\
  0 & 0 & 0 & 0 & 0 & 0 & -n_{5,i} & \kappa_{5,i} & -r_{5,i} & \lambda_{5,i}
\end{bmatrix}
\begin{bmatrix}
  x_{1}(1,i,j+1) \\
  x_{2}(1,i,j+1) \\
  x_{1}(2,i,j+1) \\
  x_{2}(2,i,j+1) \\
  x_{1}(3,i,j+1) \\
  x_{2}(3,i,j+1) \\
  x_{1}(4,i,j+1) \\
  x_{2}(4,i,j+1) \\
  x_{1}(5,i,j+1) \\
  x_{2}(5,i,j+1)
\end{bmatrix}
= \begin{bmatrix}
  d(1,i,j) \\
  \pi(1,i,j) \\
  d(2,i,j) \\
  \pi(2,i,j) \\
  d(3,i,j) \\
  \pi(3,i,j) \\
  d(4,i,j) \\
  \pi(4,i,j) \\
  d(5,i,j) \\
  \pi(5,i,j)
\end{bmatrix}
$$

Note that the vectors $d(5,i,j)$ and $\pi(5,i,j)$ represent the resin boundary with the fluid, and contain $x_{1,\text{bulk}}(i,j)$ and $x_{2,\text{bulk}}(i,j)$ terms from the fluid phase, as well as $n$ iteration terms from the quasilinearization of equilibrium terms. The vector terms $e_{n}^{1}x_{1,\text{bulk}}(i,j+1)_{n+1}^{n+1}$, $e_{2}^{n}x_{2,\text{bulk}}(i,j+1)_{n+1}^{n+1}$, and $\omega_{n}x_{1,\text{bulk}}(i,j+1)_{n+1}^{n+1}$ and $\omega_{n}x_{2,\text{bulk}}(i,j+1)_{n+1}^{n+1}$ are added to $d(5,k,j)$ and $\pi(5,i,j)$ respectively.
coefficient matrix, which is needed to solve the combined fluid phase equation matrices. These algorithms were tested in a program designed in this project to generate a system of equations with coefficients created by a random number generator subroutine. The solution vector terms were calculated for sequences of numbers representing values of the unknown, $x_i$. Values of $x_i$ calculated by the algorithms, using the predetermined coefficients and calculated solution vectors, were compared to the original values of the variables for matrices up to 500 by 500. Absolute precision of around 1 part in $10^{10}$ is maintained for the largest matrix; however, most simulations were run with 11 or 21 resin nodes--generating 22 x 22 or 42 x 42 matrices. Absolute precision for a 50 x 50 matrix was about 1 part in $10^{12}$. These tests were conducted in double precision on a DEC VAX 11/780 computer.

D. Programs for Simulation of Binary & Ternary Batch Systems

Two simulation programs were developed for so-called batch experiments--one for binary systems called **BMAVDM** (Batch Mass Action (Equilibrium) Variable Diffusivity Mole Fraction) and one for ternary systems called **TRYBCH** (Ternary Batch). **BMAVDM** can be run with mass action equilibrium constants and Redlich-Kister activity coefficient parameters correlated in equivalent fractions or mole fractions--hence the designation mole fraction. It can be run with or without activity driving force, and with fixed film coefficients. It also can be run with the model developed in this work for concentration dependent film coefficients, or with the Kataoka-Yoshida model (50) with electric
field in the film (equal valence exchange only). The program was
designed for use in a diffusion coefficient correlation package and,
therefore, has minimum output options. The initial conditions are set
up by matching the diffusion flux at the resin boundary using equation
(5-60):

\[
\frac{4 \cdot Q_r}{B_i \cdot \text{norm}} \left. \frac{\partial x_i}{\partial \rho} \right|_{\rho=1} = \bar{D}_{ij} = (x_{i, \text{bulk}} - x_{i, \text{int}})
\]

The term \( \frac{\partial x_i}{\partial \rho} \bigg|_{\rho=1} \) is calculated using a backward difference numerical
representation between resin layer M and M-1. This "conditions" the
program to begin at the first time step. At each time step, a
subroutine computes the diffusivity, the film coefficient, the
\( \frac{\partial \bar{D}_{ij}}{\partial x_i} \), and activity effects on the diffusivity coefficient if
that option is chosen, plus other components of the solution vector
\( D_j \), based on concentration values at time step j. Another subroutine
computes the equilibrium quasilinearization derivatives at each
iteration during time step j+1, and yet another subroutine computes the
electrolyte phase equilibrium after convergence criteria have been met
at j+1. The equilibrium subroutines are supported by subroutines which
compute resin and fluid phase activity coefficients. The key
subroutine solves the system of equations representing the resin
concentration profile at time step j+1. It is called by the main
program in successive iterations as the coefficient matrix is updated,
and the solution vector is recomputed for \( n^{th} \) values of terms
resulting from quasilinearization at each iteration. The Thomas
algorithm (13, 125) for tridiagonal arrays is utilized. The main
program continues to iterate at each time step until convergence criteria are met, then proceeds to the next time step. Criterion for convergence normally employed was:

\[
\frac{(x_{1n}^{n+1} - x_{1n}^n)}{x_{1n}^{n+1}} < 10^{-4} \text{ to } 10^{-5}
\]

Iterations averaged 2-3 per time step.

TRYBCH is obviously more complex, since it deals with components \( \bar{x}_1 \), \( \bar{x}_2 \), \( x_{1_{\text{bulk}}} \), and \( x_{2_{\text{bulk}}} \), but the solution principles are the same. The major difference is that a "scratch" seven diagonal array subroutine is used into which updated coefficient values are loaded at each time step and iteration; then the variables which have been solved are sorted into \( \bar{x}_1(m,j+1) \) & \( \bar{x}_2(m,j+1) \) values. Actually two "scratch" arrays are used--one which loads \( \bar{x}_1 \) as the first variable if \( \bar{x}_2(1,j) \) is smaller than \( \bar{x}_1(1,j) \) and vice versa, since it was found in the algorithm simulation that precision was improved if this condition was maintained. TRYBCH contains a significant number of output options, including printout of concentrations and flux balances at the resin/fluid interface. In both BMAVDM and TRYBCH average resin concentration is computed at each time step by summing the particle increments, since experimental data would represent particle average concentration.

E. Correlation of Diffusivities for Systems Studies by Rao, and his Coworkers

1. Experimental Background

Rao and his coworkers (92, 93) used single particle
radioactive tracer techniques to study the binary and ternary exchange of cations in Dowex 50w x8 resin, in addition to measuring the binary and ternary equilibrium for those ionic species. The two systems studies were Sr$^{++}$-Mn$^{++}$-Cs$^+$-Cl$^-$ and Mn$^{++}$-Cs$^+$-Na$^+$-Cl$^-$ at 0.1 normal solution concentrations at 28°C. Several resin beads about 0.1 cm. diameter were put into a column filled with glass beads of about the same diameter. Circulation from a large reservoir through the column at a high superficial velocity of 25 cm/sec insured that the bulk solution concentration was unchanged during the runs, and that film diffusion "control" was minimized. A scintillation detector, with output suitably corrected for background noise, was calibrated to read out the concentration change of the tagged species in the resin beads. Favorable and unfavorable exchanges were studied for binary & ternary compositions of the two cation systems. The equilibrium data were reproducible to ± 2%. In the rate studies, the radioactive $^{137}$Cs was evidently of poor purity, and runs in which $^{137}$Cs was exchanged were discarded.

2. Model Comparison - Rao and his Coworkers and Those Developed in This Research

Rao and his coworkers (92, 93) used the Nernst-Planck resin diffusion model to simulate the runs, in combination with a film diffusion model somewhat analogous to the model developed for this research project. Major model differences between Rao and his coworkers and this work involve:
o **Film Coefficient**

Rao and his coworkers used a fixed "hydrodynamic film thickness" and ionic species diffusivities derived from limiting (infinite dilution) ionic mobilities. This research project used a similar interdiffusivity relationship between ionic species in the film, but corrected the film "thickness" based on boundary layer treatment of the species having the highest flux, and corrected limiting diffusivities for concentration efforts.

o **Equilibrium**

Rao and his coworkers (92, 93) used polynomial expressions for the equilibrium distribution coefficients

$$\lambda_1 = \frac{x_1}{x_{1\text{int}}}$$

of the form

$$\lambda_1 = a + bx_{1\text{int}} + cx_{1\text{int}}^2.$$

For prediction of ternary behavior, the binary coefficients were combined by use of pseudo equivalent fractions obtained by elimination of one or the other of the minor concentration components in the resin. Harned's Rule (Chapter III) was invoked for the solution phase, so activity coefficients were not used in either the resin phase or solution phase in establishing equilibrium constants. The equilibrium treatment utilized by Rao and his coworkers is described in more detail in Chapter II on Equilibrium.

This work used the Bromley equation for liquid phase activity coefficients, and binary and ternary resin phase
Activity coefficients were correlated with the Redlich-Kister relationship (actually the Wilson and NRTL equations were also used in the thermodynamic correlation but found inferior to the R-K equations for ion exchange). By being able to calculate activity coefficients for the resin phase as functions of composition, activity driving forces could be tested versus concentration driving forces. As will be discussed, the use of activities gave some surprisingly positive results.

Method of Solving Partial Differential Equation

Rao and his coworkers used an integro differential equation in that:

\[
\bar{x}_{i, \text{ave}} = \frac{3}{r_o^3} \int_0^{r_o} x_i(r,t) \cdot r^2 \cdot dr
\]

was used to describe particle concentration history in combination with the basic partial differential equations describing fluxes $j_1$ & $j_2$, equations (5-34), (5-35), together with (5-10) and (5-11). A finite difference approximation with forward differencing was utilized to solve the equations.

3. Correlation Model

3.1 General Background

A powerful general purpose non-linear regression program was developed by Dr. A. K. S. Murthy at Allied Corporation several years ago.
A general description of the technique is given in Appendix B; however, Dr. Murthy developed the program in two forms. Both versions are variations of the Gauss-Newton method and are identical in all respects except for the sequence in which the elements of the matrices, $B^TB$ and $B^Te$, defining the normal equation are generated. In both cases, the objective function is $e^Te$ (least square criterion). The method allows weighting of data points or data sets appropriate to the users view of the "goodness" of one set of data versus others being correlated in the same regression analysis (default weight is unity). In the generation of the matrix elements, this weighting factor is appropriately built into the error definition, the e vector. In both versions, each parameter to be fit is perturbed by an appropriate fraction and numerical derivatives of the functions being described are generated with respect to each variable, and other associated calculations are performed. In neither version is the full B matrix ever generated, and this significantly reduces memory requirements. The first version is used to fit parameters in non-linear algebraic equations, such as those describing ion exchanger equilibria in this work, by operating on each point sequentially to generate matrix elements. The second version deals with differential equations such as those describing time varying systems, and in this work were the binary "batch" ion exchange experiments to which diffusion coefficients were fitted. In the latter case it is more efficient to
"simultaneously" generate the elements of the $B^T B$ and $B^T e$ matrices from all of the points varying with time in a given set, and in this version the sets are then treated sequentially. In both cases total sum of squares relative difference \[ \left( \frac{y_{\text{exp}} - y_{\text{calc}}}{y_{\text{exp}}} \right)^2 \]

for all points (sets) are minimized until convergence criteria have been met, and best values for the fitted parameters are obtained. Standard deviations for all parameters and cross correlation values are printed out at the end of a run. BMAVDM was appended to this second non-linear regression program, and the Rao and his coworkers data sets for seven binary systems were fitted using this regression program--called DFUSFT (DiFFUSion Coefficient FiTting Program) to yield diffusion coefficients correlated for the four species, Sr$^{++}$, Mn$^{++}$, Cs$^+$, & Na$^+$.  

3.2 Simulations Performed on Rao and his Coworkers' (92, 93) Experiments  

a. Data Sets  

Several sets of conditions were used to obtain the best total fit of the seven binary pairs of ionic species. The pairs for which data was available were:  

$\text{Sr}^{++}$-$\text{Mn}^{++}$  
$\text{Sr}^{++}$-$\text{Cs}^+$  
$\text{Mn}^{++}$-$\text{Na}^+$  
$\text{Na}^+$-$\text{Cs}^+$
If the $^{137}$Cs runs had been useful, 3 more sets would have been available, $\text{Cs}^+\text{Sr}^{++}$, $\text{Cs}^+\text{Mn}^{++}$ & $\text{Cs}^+\text{Na}^+$, but as discussed, Rao et al discarded those runs because of purity problems.

b. Experiments Conducted

The conditions which were tested included:

(i) Concentration driving force for all seven sets

(ii) Activity driving force for all seven sets

(iii) Activity driving force for the four diffusion coefficients, and a fifth parameter, FLAMDJ, which fractionally modified the constant in the Carberry film correlation from its stated value of 1.15 (as applied to all sets--not individually adjusted for each set).

(iv) Fitting of six sets which excluded what appears to be a "maverick" data set, $\text{Sr}^{++}\text{Mn}^{++}$ for both activities and concentration driving force.

(v) A correlation of just the data sets $\text{Sr}^{++}\text{Mn}^{++}$ and $\text{Mn}^{++}\text{Sr}^{++}$ to get a feel for the magnitude of changes in both $D'_{\text{Sr}^{++}}$ and $D'_{\text{Mn}^{++}}$ in order to fit the "maverick" data set.

(vi) A correlation of seven data sets with the $D'_{\text{Sr}^{++}}$ and $D'_{\text{Mn}^{++}}$ fixed at the values obtained above to fit values for $D'_{\text{Na}^+}$ and $D'_{\text{Cs}^+}$ under these conditions, and to compare total seven system error with other runs listed above.

(vii) A correlation of five data sets with a "binary"
diffusion model using concentration driving force to compare coefficients and total error with the same five data sets correlated using single diffusion coefficients with activities as the driving force. The "binary" diffusion model introduced the concept of an additional diffusion coefficient pair corrector parameter as an emulation of some recent work by E. E. Graham and J. S. Dranoff (30) on ion exchange in which the Stefan-Maxwell equations were applied to diffusion kinetics in the resin. The five additional pair corrector parameters introduced were Sr\textsuperscript{++}-Mn\textsuperscript{++}, Sr\textsuperscript{++}-Cs\textsuperscript{+}, Mn\textsuperscript{++}-Cs\textsuperscript{+}, Mn\textsuperscript{++}-Na\textsuperscript{+}, and Cs\textsuperscript{+}-Na\textsuperscript{+}. Five sets were selected to minimize computer time, since the fitting of nine parameters for five data sets containing a total of 136 data points was lengthy (6 hours in CPU time on a DEC VAX 11/780).

4. Discussion of Results

4.1 General Observations

A summary of these results is given in Table 5-3, and sample variance, is given as an indicator of goodness of fit, where

\[
\text{sample variance} = \frac{\sum_{\text{set}=1}^{n} \sum_{\text{point}=1}^{m} \left( \frac{\bar{x}_{\text{ave}}^{\text{exp}} - \bar{x}_{\text{ave}}^{\text{calc}}}{\bar{x}_{\text{ave}}^{\text{exp}}} \right)^2}{(\text{total no. of exp. points})}
\]

\[(5-98)\]
### Table 5-3

**SUMMARY – FITTED DIFFUSION COEFFICIENTS**  
**EXPERIMENTAL DATA OF G. RAO et al**

- Binary Systems of Sr\(^{2+}\)-Mn\(^{2+}\)-Cs\(^+\)-Na\(^+\)/Cl\(^-\) on Dowex 50w x8 Resin -

All values of $\bar{D}_{i} \times 10^6$ cm\(^2\)/sec

<table>
<thead>
<tr>
<th>Description of Correlation Experiment</th>
<th>$\bar{D}_{Mn}^{+}$ Value</th>
<th>$\bar{D}_{Sr}^{+}$ Value</th>
<th>$\bar{D}_{Cs}^{+}$ Value</th>
<th>$\bar{D}_{Na}^{+}$ Value</th>
<th>Sample Variance $s^2$</th>
</tr>
</thead>
<tbody>
<tr>
<td><strong>Seven Sets</strong></td>
<td></td>
<td></td>
<td></td>
<td></td>
<td></td>
</tr>
<tr>
<td>- Concentration Driving Force</td>
<td>0.330316 ± 0.00761</td>
<td>0.148729 ± 0.0050</td>
<td>10.00 ± 0.083</td>
<td>2.07116 ± 0.0519</td>
<td>0.0026</td>
</tr>
<tr>
<td>- Activity Driving Force</td>
<td>0.32240± 0.00684</td>
<td>0.139227± 0.0038</td>
<td>3.96189 ± 0.347</td>
<td>2.74910 ± 0.0977</td>
<td>0.00213</td>
</tr>
<tr>
<td>- Activity Driving Force</td>
<td>0.33877 ± 0.0103</td>
<td>0.142827 ± 0.00396</td>
<td>4.36702 ± 0.475</td>
<td>2.77647 ± 0.101</td>
<td>0.00220</td>
</tr>
<tr>
<td>- Fit $\bar{D}<em>{Sr}^{+}$, $\bar{D}</em>{Mn}^{+}$, $\bar{D}_{Sr}^{+}$ Activities</td>
<td>0.88036 ± 0.000805</td>
<td>0.084604 ± 0.0009</td>
<td>---</td>
<td>---</td>
<td>0.00044</td>
</tr>
<tr>
<td>- Fit Other Sets, $\bar{D}<em>{Sr}^{+}$, $\bar{D}</em>{Mn}^{+}$ Fixed</td>
<td>0.88036 ± 0.0000805</td>
<td>0.08460 ± 0.0009</td>
<td>2.41534 ± 0.062</td>
<td>2.0047 ± 0.119</td>
<td>0.0269</td>
</tr>
<tr>
<td><strong>Six Sets - $\bar{D}_{Sr}^{+}$-Mn(^{2+})Excluded</strong></td>
<td></td>
<td></td>
<td></td>
<td></td>
<td></td>
</tr>
<tr>
<td>- Concentration Driving Force</td>
<td>0.32907 ± 0.00426</td>
<td>0.184811 ± 0.0044</td>
<td>9.84776 ± 0.134</td>
<td>2.1170 ± 0.0308</td>
<td>0.00089</td>
</tr>
<tr>
<td>- Activity Driving Force</td>
<td>0.320027 ± 0.0039</td>
<td>0.172243 ± 0.0034</td>
<td>3.43355 ± 0.155</td>
<td>2.87574 ± 0.061</td>
<td>0.00076</td>
</tr>
<tr>
<td><strong>Five Sets - Na(^+)-Mn(^{2+})Excluded</strong></td>
<td></td>
<td></td>
<td></td>
<td></td>
<td></td>
</tr>
<tr>
<td>- Activity Driving Force</td>
<td>0.35592 ± 0.00233</td>
<td>0.178738 ± 0.0017</td>
<td>2.51790 ± 0.0525</td>
<td>3.63666 ± 0.0575</td>
<td>0.00014</td>
</tr>
<tr>
<td>- &quot;Binary&quot; Model(1)</td>
<td>0.149521</td>
<td>0.844332</td>
<td>3.82155</td>
<td>3.21048</td>
<td>0.00011</td>
</tr>
<tr>
<td>- Pair Corrector</td>
<td></td>
<td></td>
<td></td>
<td></td>
<td></td>
</tr>
<tr>
<td>- $\bar{D}<em>{Mn}^{+}$-$\bar{D}</em>{Cs}^{+}$</td>
<td>0.25878</td>
<td>---</td>
<td>0.25878</td>
<td>---</td>
<td>---</td>
</tr>
<tr>
<td>- $\bar{D}<em>{Mn}^{+}$-$\bar{D}</em>{Na}^{+}$</td>
<td>0.029884</td>
<td>---</td>
<td>---</td>
<td>---</td>
<td>0.072910</td>
</tr>
<tr>
<td>- $\bar{D}<em>{Sr}^{+}$-$\bar{D}</em>{Na}^{+}$</td>
<td>0.072910</td>
<td>---</td>
<td>-0.639182</td>
<td>-0.639182</td>
<td>---</td>
</tr>
<tr>
<td>- $\bar{D}<em>{Na}^{+}$-$\bar{D}</em>{Cs}^{+}$</td>
<td>---</td>
<td>---</td>
<td>-0.231135</td>
<td>-0.231135</td>
<td>---</td>
</tr>
<tr>
<td>- Correlated by Rao et al</td>
<td>0.22 ± N/A</td>
<td>0.195 ± N/A</td>
<td>3.0 Estimated</td>
<td>2.05 ± N/A</td>
<td>0.0083</td>
</tr>
</tbody>
</table>

(1) Std. dev. not available – these are printed out at end and this set of "correctors" was picked as representing the least total required correction out of several iterations having the same total error.
Figures 5-2 and 5-3 show the best fit of seven sets using activity driving force and the diffusion coefficients derived from that correlation, plotted against lines representing the Rao et al experimental data for the binary systems. The calculated curves in general match the experimental data very well, except for the $\text{Sr}^{++}$-$\text{Mn}^{++}$ and $\text{Mn}^{++}$-$\text{Sr}^{++}$ systems. If the $\text{Sr}^{++}$-$\text{Mn}^{++}$ system is not included in the fit, calculated points for the system $\text{Mn}^{++}$-$\text{Sr}^{++}$ fit the experimental curves very well. The film correlation constant "adjuster," FLMADJ, was kept at 1.0 for the final selected set of parameters utilized in generating the plots. A FLMADJ of $0.86 \pm 0.04$, derived from the parameter study which included FLMADJ, would indicate the Carberry coefficient of 1.15 is reasonable, and since the Re no. was not varied in the Rao and his coworkers' experiments (92, 93) no other conclusion could be drawn.

4.2 Significance of the Batch Experiment Results

The recent work of E. E. Graham and J. S. Dranoff (30) on application of the Stefan-Maxwell equations to diffusion of ionic species in ion exchange resins was undertaken because previous investigators, using the Nernst-Planck model, were unable to define so-called "self diffusion" coefficients for a given ionic species which would have the same value regardless of the other ionic species present during the exchange. Studies by
Binary Exchange for Pairs in System Sr$^{++} -$ Mn$^{+} -$ Cs$^{+}$, Dowex 50W x8, Cl$^{-}$

Binary Model – Nernst Planck/Variable Film Coefficient

Activities are Driving Force

0.10 Normal

Experimental Curves – Temp 28°C
R.K. Battpai, A.K. Gupta, M. Gapola Rao
'AlChE Journal', Sept, 1974 (Vol 20, No 5)
Figure 5-3

Binary Exchange for Pairs in System Mn$^{++} -$ Cs$^+ -$ Na$^+$, Dowex 50W x8, Cl$^-$

Binary Model – Nernst Planck/Variable Film Coefficient

Activities are Driving Force

0.10 Normal

Fractional Conversion EQF (Time)/EQF (Initial)

Elapsed Time – Sec.

Experimental Curves – Temp 28°C
R.K. Batipai, A.K. Gupta, M. Gopala Rao
‘AIChE Journal’, Sept, 1974 (Vol 20, No 5)

‘SELF’ DIFFUSION COEFFICIENTS

$D_{MN} = 0.322405 \times 10^{-6} \pm 0.0068 \times 10^{-6} \text{ cm}^2/\text{sec}$

$D_{CS} = 0.396189 \times 10^{-5} \pm 0.0347 \times 10^{-5} \text{ cm}^2/\text{sec}$

$D_{NA} = 0.274910 \times 10^{-5} \pm 0.0098 \times 10^{-5} \text{ cm}^2/\text{sec}$
Hering and Bliss (46) on various cation pairs using Dowex 50w x8 resin at 25°C for an ion electrolyte concentration of 1.0 normal showed the following results (correlated for exchange in both directions):

$$\text{Diffusion Coefficient } \times 10^6 \, \text{cm}^2/\text{sec}$$

<table>
<thead>
<tr>
<th>System</th>
<th>Nernst Planck Model</th>
<th>Fick's Law Model</th>
</tr>
</thead>
<tbody>
<tr>
<td>Na$^+$-Zn$^{2+}$</td>
<td>$D_{Na} = 2.03$</td>
<td>$D_{Na,Zn} = 0.971$</td>
</tr>
<tr>
<td></td>
<td>$D_{Zn} = 0.231$</td>
<td>$D_{Zn,Na} = 0.395$</td>
</tr>
<tr>
<td>Na$^+$-Ag$^+$</td>
<td>$D_{Na} = 2.05$</td>
<td>$D_{Na,Ag} = 1.43$</td>
</tr>
<tr>
<td></td>
<td>$D_{Ag} = 0.683$</td>
<td>$D_{Ag,Na} = 0.86$</td>
</tr>
<tr>
<td>Ag$^{2+}$-Al$^{3+}$</td>
<td>$D_{Ag} = 0.380$</td>
<td>$D_{Ag,Al} = 0.298$</td>
</tr>
<tr>
<td></td>
<td>$D_{Al} = 0.112$</td>
<td>$D_{Al,Ag} = 0.180$</td>
</tr>
<tr>
<td>Cu$^{2+}$-Zn$^{2+}$</td>
<td>$D_{Cu} = 0.281$</td>
<td>$D_{Cu,Cn} = 0.230$</td>
</tr>
<tr>
<td></td>
<td>$D_{Zn} = 0.0157$</td>
<td>$D_{Zn,Cu} = 0.181$</td>
</tr>
<tr>
<td>Zn$^{2+}$-Al$^{3+}$</td>
<td>$D_{Zn} = 0.071$</td>
<td>$D_{Zn,Al} = 0.0692$</td>
</tr>
<tr>
<td></td>
<td>$D_{Al} = 0.065$</td>
<td>$D_{Al,Zn} = 0.0648$</td>
</tr>
<tr>
<td>Ce$^{3+}$-Al$^{3+}$</td>
<td>$D_{Al} = 0.0455$</td>
<td>$D_{Ce,Al} = 0.0120$</td>
</tr>
</tbody>
</table>

*Available for Ce$^{3+}$ in exchanger only.

As Hering and Bliss (46) point out, the N-P model is preferable to Fick's Law in that a single "effective" diffusivity allows prediction of both favorable and unfavorable exchange; whereas with Fick's Law, the diffusion coefficients are different depending on direction of exchange, with the coefficient invariably higher when the faster ion is in the exchanger. Rao and
his coworkers (92) pointed out in their studies that variation of "self" diffusion coefficients of the ionic species with ionic composition of the resin was not taken into account in their analysis, adding that cross phenomenological coefficients and activity coefficients in the resin phase were neglected in development of their model. Graham and Dranoff (30), although successful in fitting the Na\(^+\)-Cs\(^+\) system with their Stefan-Maxwell model, were treating a fairly "ideal" system in terms of limiting activity coefficients determined in this work where \(\overline{\gamma}_{\text{Cs}} = 0.78\), and \(\overline{\gamma}_{\text{Na}} = 1.043\). In the case of Sr\(^{++}\)-Cs\(^+\) studies in this project, limiting activity coefficients were \(\overline{\gamma}_{\text{Sr}} = 0.526\), \(\overline{\gamma}_{\text{Mn}} = 0.544\) (or when correlated with M.F. versus Equiv. Fraction \(\overline{\gamma}_{\text{Sr}} = 0.331\), \(\overline{\gamma}_{\text{Mn}} = 0.464\)). Graham and Dranoff in their derivation have assumed that Harned's Rule applied in the exchanger phase, i.e., \[
\frac{\partial \ln \gamma_i}{\partial x_i} \Bigg|_{\text{active sites}} = \text{constant}
\]
They point out that the Helfferich definition (42) of interdiffusion coefficient for binary exchange (equation 5-13) fits limiting tracer ion data (limiting diffusion values of species 1 migrating in resin essentially loaded with species 2) if the term
\[
\frac{\overline{D}_1 \cdot \overline{D}_2}{2 \sum_{i=1}^{\overline{D}_i} |z_i|} \cdot \overline{x}_1 \cdot \overline{x}_2 \cdot \frac{\partial \ln \gamma_1}{\partial x_1} \frac{|z_2|}{\gamma_1} \frac{|z_1|}{\gamma_2} \text{molar}
\]
is included but treated as a constant, \( K \), as in Harned's Rule. They suggest that further investigation be made into the use of Gibbs-Duhem relationship so as to include the pressure gradient for water flux in the resin. This is tantamount to saying look at activity driving forces in the resin, since resin pressure volume relationships are functions of total system chemical potential for all species--including water. The final equation derived by Graham and Dranoff (30) defines "effective" Nernst-Planck diffusion coefficients as:

\[
\begin{align*}
\text{d}_1(2) &= \left[ \frac{1}{\frac{x_0}{a_{10}} - \left( \frac{z_a}{z_2} \right) \frac{x_a}{a_{12}} - \frac{1}{a_{12}}} - \left( \frac{z_2}{z_a} \right) \frac{1}{a_{1a}} \right] \\
\text{d}_2(1) &= \left[ \frac{1}{\frac{x_0}{a_{20}} - \left( \frac{z_a}{z_1} \right) \frac{x_a}{a_{21}} - \frac{1}{a_{21}}} - \left( \frac{z_1}{z_a} \right) \frac{1}{a_{2a}} \right] 
\end{align*}
\]

where:

- \( x_a \) = mole fraction of active exchanger sites
- \( z_a \) = valence of active exchanger sites
- \( x_0 \) = mole fraction of water in exchanger
- \( a_{i0} \) = Stefan Maxwell interaction coefficient between ion species \( i \) and water
- \( a_{ia} \) = Stefan Maxwell interaction coefficient between ion species \( i \) and active sites
- \( a_{ij} \) = Stefan Maxwell interaction coefficient between ionic species
This suggested to the investigator in this work that a "simple" version of $d_1(2)$ and $d_2(1)$ might only require a single pair corrector coefficient for a given ion exchange resin, i.e., $\bar{D}_{12} = \bar{D}_1 + \bar{d}_{12}$, and $\bar{D}_{21} = \bar{D}_2 + \bar{d}_{12}$ where: $\bar{D}_1$ & $\bar{D}_2$ are the "self" diffusion coefficients and $\bar{d}_{12}$ is the pair "corrector."

This was tried in fitting the set of five ion species exchange curves and was very successful. However, an equally good fit to the set of five was obtained with only four parameters, the "effective self" diffusion coefficient for each species, and correcting for non-ideality by use of the activity driving force. Although the two parameter model effectively fit the data for binaries, it is not obvious how ternary systems could be modeled. Later discussion on the Rao and his coworkers' ternary experiments will include some simulations of ternary systems with the two parameter model in which:

$$\bar{D}_{12} = \bar{D}_1 + \left( \frac{x_2}{x_2 + x_3} \right) \cdot \bar{d}_{12} + \left( \frac{x_3}{x_2 + x_3} \right) \cdot \bar{d}_{13}$$  \hspace{1cm} (5-101)

and so forth for $\bar{D}_{21}$ and $\bar{D}_{13}$.

The modeling of all of the ternary data was performed with the activity driving force model since this model is sound theoretically, and the two parameter model discussed above is not. Although Table 5-3 shows that use of concentration driving force gave a fit on the set of seven with only a somewhat increased total...
error, the diffusion coefficient for Cs\(^+\) at a value of 10.0E-06 cm\(^2\)/sec was unreasonably high in view of literature values around 2.0E-06 cm\(^2\)/sec. This was true in comparisons made in correlation of the set of six as well. In the two parameter model, if the pair correctors are added to the effective diffusion coefficient, the following values are obtained:

\[
\overline{D}_i \times 10^6 \text{ cm}^2/\text{sec}
\]

<table>
<thead>
<tr>
<th>Two Parameter Model</th>
<th>Activity Model</th>
</tr>
</thead>
<tbody>
<tr>
<td>Ion</td>
<td>Pair Ion</td>
</tr>
<tr>
<td>(D_{\text{Mn}^{++}})</td>
<td>Sr(^{++})</td>
</tr>
<tr>
<td></td>
<td>C(^+)</td>
</tr>
<tr>
<td></td>
<td>Na(^+)</td>
</tr>
<tr>
<td>(D_{\text{Sr}^{++}})</td>
<td>Mn(^{++})</td>
</tr>
<tr>
<td></td>
<td>Cs(^+)</td>
</tr>
<tr>
<td>(D_{\text{Cs}^{+}})</td>
<td>Mn(^{++})</td>
</tr>
<tr>
<td></td>
<td>Sr(^{++})</td>
</tr>
<tr>
<td></td>
<td>Na(^+)</td>
</tr>
<tr>
<td>(D_{\text{Na}^{+}})</td>
<td>Mn(^{++})</td>
</tr>
<tr>
<td></td>
<td>Cs(^+)</td>
</tr>
</tbody>
</table>

The Sr\(^{++}\)-Mn\(^{++}\) pair shows the greatest correction requirement, compared to the activity model; however, most values are not "too different" from the single coefficient activity model values.

Since the activity model was derived on a theoretically sound basis and the parameter regressions yielded diffusion coefficient values not far from previously observed experimental values, it can be said
that the activity model performed "best" in all simulations of the Rao and his coworkers' experimental data.

Finally, Rao and his coworkers simulated their experimental binary data for the set of five using the Nernst-Planck model which they derived, and which was described in previous sections of this chapter. The sample variance for 48 calculated points was 0.0083 versus 0.00014 on 136 calculated points obtained with the activity driving force model developed in this work (standard deviation approximately ±0.09 vs. ±0.012 on variable range between 0.2 and 1.0).

F. Simulation of Rao and his Coworkers' Ternary Experimental Data

1. General Discussion

Table 5-4 gives a listing of initial concentration conditions for 22 ternary rate runs made by Rao and his coworkers on the systems \( \text{Sr}^{++} - \text{Mn}^{++} - \text{Cs}^+ \) and \( \text{Mn}^{++} - \text{Cs}^+ - \text{Na}^+ \), including a subjective quality rating (10 best, 1 worst) for the runs simulated. The quality ratings are compared for simulation runs made using the activity driving force model with diffusion coefficients derived from the set of seven, set of six, plus a few runs based on the two parameter model (pair correctors). Quality ratings are given for the simulation results of Rao and his coworkers as well so that a qualitative comparison of model performance can be made. Using data from the set of seven fit, an average quality rating of 7 is somewhat better than the
### Table 5-4

**Batch Ternary Ion Exchange Experiments**

Gopala Rao et al

**System Sr⁺⁺-Mn⁺⁺-Cs⁺-Na⁺/Dowex 50W x8/Cl⁻**

<table>
<thead>
<tr>
<th>Run No.</th>
<th>Resin Composition</th>
<th>Solution Composition</th>
<th>Simulation Data Source</th>
<th>Pair Corrector Model</th>
</tr>
</thead>
<tbody>
<tr>
<td></td>
<td>Sr⁺⁺</td>
<td>Mn⁺⁺</td>
<td>Cs⁺</td>
<td>Sr⁺⁺</td>
</tr>
<tr>
<td>33B</td>
<td>0.18</td>
<td>0.82*</td>
<td>---</td>
<td>---</td>
</tr>
<tr>
<td>43B</td>
<td>0.348</td>
<td>0.652*</td>
<td>---</td>
<td>---</td>
</tr>
<tr>
<td>44B</td>
<td>1.0*</td>
<td>---</td>
<td>---</td>
<td>0.84</td>
</tr>
<tr>
<td>45B</td>
<td>1.0*</td>
<td>---</td>
<td>---</td>
<td>0.40</td>
</tr>
<tr>
<td>31B</td>
<td>---</td>
<td>0.715*</td>
<td>0.285</td>
<td>1.0</td>
</tr>
<tr>
<td>40B</td>
<td>---</td>
<td>0.639*</td>
<td>0.361</td>
<td>1.0</td>
</tr>
<tr>
<td>35B</td>
<td>0.745*</td>
<td>---</td>
<td>0.255</td>
<td>---</td>
</tr>
<tr>
<td>41B</td>
<td>0.618*</td>
<td>---</td>
<td>0.382</td>
<td>---</td>
</tr>
</tbody>
</table>

<table>
<thead>
<tr>
<th>Run No.</th>
<th>Mn⁺⁺</th>
<th>Cs⁺</th>
<th>Na⁺</th>
<th>Mn⁺⁺</th>
<th>Cs⁺</th>
<th>Na⁺</th>
<th>Rao et al</th>
<th>fit of 6</th>
<th>fit of 7</th>
<th>Pair Corrector Model</th>
</tr>
</thead>
<tbody>
<tr>
<td>28</td>
<td>---</td>
<td>0.576</td>
<td>0.424*</td>
<td>1.0</td>
<td>---</td>
<td>---</td>
<td>4</td>
<td>not fit</td>
<td>7</td>
<td>not fit</td>
</tr>
<tr>
<td>29</td>
<td>---</td>
<td>0.279</td>
<td>0.721*</td>
<td>1.0</td>
<td>---</td>
<td>---</td>
<td>5</td>
<td>not fit</td>
<td>8</td>
<td>not fit</td>
</tr>
<tr>
<td>33</td>
<td>0.393</td>
<td>---</td>
<td>0.602*</td>
<td>---</td>
<td>1.0</td>
<td>---</td>
<td>8</td>
<td>9</td>
<td>9</td>
<td>not fit</td>
</tr>
<tr>
<td>34</td>
<td>0.486</td>
<td>---</td>
<td>0.514*</td>
<td>---</td>
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<td>10</td>
<td>6</td>
<td>6</td>
<td>not fit</td>
</tr>
<tr>
<td>30</td>
<td>0.841*</td>
<td>---</td>
<td>0.159</td>
<td>---</td>
<td>1.0</td>
<td>---</td>
<td>4</td>
<td>5</td>
<td>5</td>
<td>not fit</td>
</tr>
<tr>
<td>31</td>
<td>0.632*</td>
<td>---</td>
<td>0.368</td>
<td>---</td>
<td>1.0</td>
<td>---</td>
<td>4</td>
<td>not fit</td>
<td>7</td>
<td>not fit</td>
</tr>
<tr>
<td>66</td>
<td>0.489*</td>
<td>---</td>
<td>0.511</td>
<td>---</td>
<td>1.0</td>
<td>---</td>
<td>4</td>
<td>4</td>
<td>5</td>
<td>not fit</td>
</tr>
<tr>
<td>35</td>
<td>---</td>
<td>1.0*</td>
<td>0.40</td>
<td>0.60</td>
<td>---</td>
<td>---</td>
<td>7</td>
<td>6</td>
<td>6</td>
<td>not fit</td>
</tr>
<tr>
<td>36</td>
<td>---</td>
<td>1.0*</td>
<td>0.833</td>
<td>0.167</td>
<td>---</td>
<td>---</td>
<td>6</td>
<td>10</td>
<td>8</td>
<td>not fit</td>
</tr>
<tr>
<td>37</td>
<td>0.555*</td>
<td>0.445</td>
<td>---</td>
<td>1.0</td>
<td>---</td>
<td>---</td>
<td>6</td>
<td>6</td>
<td>6</td>
<td>not fit</td>
</tr>
<tr>
<td>38</td>
<td>0.804*</td>
<td>0.196</td>
<td>---</td>
<td>1.0</td>
<td>---</td>
<td>---</td>
<td>5</td>
<td>7</td>
<td>7</td>
<td>not fit</td>
</tr>
<tr>
<td>49</td>
<td>0.604*</td>
<td>0.396</td>
<td>---</td>
<td>1.0</td>
<td>---</td>
<td>---</td>
<td>5</td>
<td>not fit</td>
<td>not fit</td>
<td>not fit</td>
</tr>
<tr>
<td>43</td>
<td>1.0*</td>
<td>---</td>
<td>---</td>
<td>0.10</td>
<td>0.90</td>
<td>---</td>
<td>6</td>
<td>8</td>
<td>8</td>
<td>9</td>
</tr>
<tr>
<td>70</td>
<td>1.0*</td>
<td>---</td>
<td>---</td>
<td>0.40</td>
<td>0.60</td>
<td>---</td>
<td>7</td>
<td>10</td>
<td>10</td>
<td>10</td>
</tr>
</tbody>
</table>

*tagged ion

**Average Quality Rating**

5.9 6.9 6.95
average rating of 6 for the Rao and his coworkers' simulation results. The results of the simulation runs are shown in Figures 5-4 through 5-8 for diffusion coefficients derived from the fit of seven, since the average quality rating was slightly better than the fit of six.

2. Comparison of Simulated Results with Experimental Data

Rao and his coworkers published the results (94) of simulated "batch" experiments for the system \( \text{Ba}^{++}\text{-Sr}^{++}\text{-Na}^{+} \) for a range of initial conditions, but limited to two basic types of exchange: (1) two ions initially in the resin and the third at the surface, (2) one ion in the resin and the other two at the surface. Plots were generated by simulation which showed the depletion rate trends for the "tagged" ion in the resin. Some actual experimental results had been obtained by Rao and his coworkers in earlier work (90) which corroborated the general computed trends. Based on those results, the following observations can be made on the simulation results obtained in this work on the systems \( \text{Mn}^{++}\text{-Cs}^{+}\text{-Na}^{+} \) and \( \text{Sr}^{++}\text{-Mn}^{++}\text{-Cs}^{+} \):

a. For cases such as exp. 35 and 36, where the least favored ion, \( \text{Na}^{+} \), is in the resin and has a diffusion coefficient higher than \( \text{Mn}^{++} \) but lower than \( \text{Cs}^{+} \), the expected trend would be for the \( \text{Na}^{+}\text{-Mn} \) binary to represent the upper boundary, and the \( \text{Na}^{+}\text{-Cs} \) binary to represent the lower boundary of ternary depletion curves, with the \( \text{Na}^{+} \) depletion curves having greater slopes than the
Ternary Exchange in System Mn$^{++}$ – Cs$^+$ – Na$^+$, Dowex 50W x 8, Cl$^-$

Ternary Model - Nernst Planck/Variable Film Coefficients

Activities As Driving Force

<table>
<thead>
<tr>
<th>System Calc.</th>
<th>Resin Comp.</th>
<th>Solution Comp.</th>
<th>Experimental**</th>
</tr>
</thead>
<tbody>
<tr>
<td></td>
<td>Mn*</td>
<td>Na  Cs</td>
<td>Binaries</td>
</tr>
<tr>
<td>○</td>
<td>1.0</td>
<td>0.90 0.10</td>
<td>R-43</td>
</tr>
<tr>
<td>×</td>
<td>1.0</td>
<td>0.60 0.40</td>
<td>R-70</td>
</tr>
<tr>
<td>△</td>
<td>Na*</td>
<td>Mn  Cs</td>
<td>R-36</td>
</tr>
<tr>
<td>□</td>
<td>1.0</td>
<td>0.833 0.167</td>
<td>R-35</td>
</tr>
<tr>
<td></td>
<td>1.0</td>
<td>0.40 0.60</td>
<td></td>
</tr>
</tbody>
</table>

"Self" Diffusion Coefficients

\[ D_{Mn} = 0.3224 \times 10^{-6} \text{ cm}^2/\text{sec} \]
\[ D_{Cs} = 0.3962 \times 10^{-5} \text{ cm}^2/\text{sec} \]
\[ D_{Na} = 0.2749 \times 10^{-5} \text{ cm}^2/\text{sec} \]

**0.1 Normal, 28°C
R.K. Bajpai, A.K. Gupta, M. Gopala Rao
"AIChE Journal," Sept. 1974 (Volume 20, No. 5)
Ternary Exchange in System Sr$^{++}$ – Mn$^{++}$ – Cs$^+$, Dowex 50W x 8, Cl$^-$

Ternary Model - Nernst Planck/Variable Film Coefficients
- Activities As Driving Force -

**"Self" Diffusion Coefficients**

\[ D_{Sr} = 0.1392 \times 10^{-6} \text{ cm}^2/\text{sec} \]
\[ D_{Mn} = 0.3224 \times 10^{-6} \text{ cm}^2/\text{sec} \]
\[ D_{Cs} = 0.3962 \times 10^{-5} \text{ cm}^2/\text{sec} \]

**0.1 Normal, 28°C**

R.K. Bajpai, A.K. Gupta, M. Gopala Rao
"AIChE Journal," Sept. 1974 (Volume 20, No. 5)
Figure 5-6

Ternary Exchange in System Sr$^{++}$—Mn$^{++}$—Cs$^+$ and Mn$^{++}$—Cs$^+$—Na$^+$ on Dowex 50W x 8/Cl$^-$

Ternary Model - Nernst Planck/Variable Film Coefficients - Activities As Driving Force -

<table>
<thead>
<tr>
<th>Systems Calc.</th>
<th>Resin Comp.</th>
<th>Solution Comp.</th>
<th>Experimental**</th>
</tr>
</thead>
<tbody>
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<td>Na* Cs</td>
<td>Mn</td>
<td>Binaries</td>
</tr>
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<td>○</td>
<td>0.721 0.279</td>
<td>1.0</td>
<td>R-29</td>
</tr>
<tr>
<td>×</td>
<td>0.424 0.576</td>
<td>1.0</td>
<td>R-28</td>
</tr>
<tr>
<td></td>
<td>Mn* Cs</td>
<td>Sr</td>
<td>R-31B</td>
</tr>
<tr>
<td>△</td>
<td>0.715 0.285</td>
<td>1.0</td>
<td>R-40B</td>
</tr>
<tr>
<td>□</td>
<td>0.639 0.361</td>
<td>1.0</td>
<td></td>
</tr>
</tbody>
</table>

“Self” Diffusion Coefficients

\[ D_{Sr} = 0.1392 \text{ E-06 cm}^2/\text{sec} \]
\[ D_{Mn} = 0.3224 \text{ E-06 cm}^2/\text{sec} \]
\[ D_{Cs} = 0.3962 \text{ E-05 cm}^2/\text{sec} \]
\[ D_{Na} = 0.2749 \text{ E-05 cm}^2/\text{sec} \]

** 0.1 Normal, 28°C

R.K. Bajpai, A.K. Gupta, M. Gopala Rao
“AICheE Journal,” Sept. 1974 (Volume 20, No. 5)
Ternary Exchange in System Mn$^{++}$ – Cs$^+$ – Na$^+$, Dowex 50W x 8, Cl$^-$

Ternary Model - Nernst Planck/Variable Film Coefficients
- Activities As Driving Force -

- **Experimental**
  - Binaries
  - R-30
  - R-31
  - R-66
  - R-33
  - R-34

- **Systems Calc.**
  - Resin Comp.
    - Mn$^+$ Na$^+$
      - $\Delta$: 0.841 0.159
      - □: 0.632 0.368
      - ◊: 0.489 0.511

- **Solution Comp.**
  - Cs
    - 1.0

- **Experimental**
  - Cs
    - 1.0

- **"Self" Diffusion Coefficients**
  - $D_{Mn} = 0.3224 \times 10^{-6}$ cm$^2$/sec
  - $D_{Cs} = 0.3962 \times 10^{-5}$ cm$^2$/sec
  - $D_{Na} = 0.2749 \times 10^{-5}$ cm$^2$/sec

**0.1 Normal, 28°C**

R.K. Bajpai, A.K. Gupta, M. Gopala Rao

"AIChE Journal," Sept. 1974 (Volume 20, No. 5)
Ternary Exchange in System Sr\(^{2+}\) – Mn\(^{2+}\) – Cs\(^+\) and Mn\(^{2+}\) – Cs\(^+\) – Na\(^+\) on Dowex 50W x 8/Cl\(^-\)

Ternary Model - Nernst Planck/Variable Film Coefficients
- Activities As Driving Force -

<table>
<thead>
<tr>
<th>Systems Calc.</th>
<th>Resin Comp.</th>
<th>Solution Comp.</th>
<th>Experimental**</th>
</tr>
</thead>
<tbody>
<tr>
<td></td>
<td>Mn (^+) Cs</td>
<td>Na</td>
<td>Binaries</td>
</tr>
<tr>
<td>□</td>
<td>0.804</td>
<td>0.196</td>
<td></td>
</tr>
<tr>
<td>◊</td>
<td>0.604</td>
<td>0.396</td>
<td>R-38</td>
</tr>
<tr>
<td>△</td>
<td>0.555</td>
<td>0.455</td>
<td>R-49</td>
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<tr>
<td>···</td>
<td></td>
<td></td>
<td></td>
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<tr>
<td>Mn (^+) Sr</td>
<td>0.820</td>
<td>0.180</td>
<td>R-37</td>
</tr>
<tr>
<td>···</td>
<td></td>
<td></td>
<td></td>
</tr>
<tr>
<td>Mn (^+) Cs</td>
<td>0.652</td>
<td>0.348</td>
<td>R-33B</td>
</tr>
<tr>
<td>···</td>
<td></td>
<td></td>
<td></td>
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<tr>
<td>MnNa &amp; R-38</td>
<td>···</td>
<td>R-43B</td>
<td></td>
</tr>
<tr>
<td>MnCs &amp; R-33B</td>
<td>···</td>
<td></td>
<td></td>
</tr>
</tbody>
</table>

Self Diffusion Coefficients

\(D_{Sr} = 0.1392 \times 10^{-6} \text{ cm}^2/\text{sec}\)
\(D_{Mn} = 0.3224 \times 10^{-6} \text{ cm}^2/\text{sec}\)
\(D_{Cs} = 0.3962 \times 10^{-5} \text{ cm}^2/\text{sec}\)
\(D_{Na} = 0.2749 \times 10^{-5} \text{ cm}^2/\text{sec}\)

Elapsed Time - Seconds

** 0.1 Normal, 28°C
R.K. Bajpai, A.K. Gupta, M. Gopala Rao
"AIChE Journal," Sept. 1974 (Volume 20, No. 5)
upper Na\(^+\)Mn binary boundary as Cs\(^+\) concentration increases in the electrolyte phase. This was observed both experimentally and in the simulation.

b. For cases such as 44B and 45B, Sr MnCs, where the most favored ion is in the resin, the selectivity is Sr>Mn>>Cs and diffusion coefficients values are Cs>>Mn>Sr, the upper envelope boundary should be the binary Sr\(^+\)Mn, and the lower boundary the binary system Sr\(^+\)Cs. The trend in Sr\(^+\) depletion curves should be to have greater slopes as Cs\(^+\) concentration increases in the electrolyte. This was the case experimentally, and was the trend in the simulated results; however, the simulation run with higher Cs\(^+\) (45B) was slightly under the lower boundary.

The system Mn\(^+\)CsNa, runs 43 and 70, where selectivity is Mn>>Cs>Na and diffusion coefficients are Cs>Na>>Mn, should have an upper boundary binary of Mn\(^+\)Na, with the lower envelope boundary being Mn\(^+\)Cs. The trend in Mn\(^+\) depletion curves should be to have greater slopes as Cs\(^+\) concentration in the electrolyte increases, and this was the case experimentally and in the simulated results.

c. For cases where the most favored ion is in the electrolyte and the other two in the resin, runs
31B and 40B for Mn\textsuperscript{*}CsSr, and runs 28 and 29 for CsNa\textsuperscript{*}Mn illustrate behavior of this type of exchange. In the Mn\textsuperscript{*}CsSr system, the upper boundary should be Mn\textsuperscript{*}Sr, and the lower binary system bounding the envelope, Cs\textsuperscript{*}Sr. Due to the bad batch of Cs\textsuperscript{137}, no experimental curves were generated by Cs\textsuperscript{*} in the resin. In this system, as the faster and least selective ion, Cs\textsuperscript{+}, is increased in the resin concentration--it exchanges with the Sr\textsuperscript{++} at a faster rate than the Mn\textsuperscript{*}, and has a retarding effect on the diffusion rate of the Mn\textsuperscript{*}. This behavior was shown experimentally, and the simulation curves show this trend, although not laying on the experimental curves. In the case of CsNa\textsuperscript{*}Mn system, the lower bound should be Cs\textsuperscript{*}Mn (again no Cs\textsuperscript{*} data) and the upper binary Na\textsuperscript{*}Mn. Increasing amounts of the more preferred ion, Cs\textsuperscript{+}, should accelerate the diffusion rate of Na\textsuperscript{*} ion in the direction of the lower boundary. This was observed experimentally, although the simulated runs gave results not much different from the Na\textsuperscript{*}Mn binary system.

d. Runs 35B and 41B for Sr\textsuperscript{*}CsMn, and runs 30, 31, 33, 34, and 66 for Mn\textsuperscript{*}NaCs and MnNa\textsuperscript{*}Cs illustrated system behavior where the resin contained both the most preferred and the least preferred ions, with the other ion constituting the
electrolyte composition. In the Sr\textsuperscript{*}-CsMn system, it would be expected that the displacement of the Cs\textsuperscript{+} ion by the slower and more favored Mn\textsuperscript{++} ion would slow the rate of diffusion of Sr\textsuperscript{*} below that observed in the Sr\textsuperscript{*}-Mn binary, as Cs\textsuperscript{+} concentration is increased. This was not observed experimentally since run 41B, having the larger Cs\textsuperscript{+} concentration, should have had a depletion curve above run 35B. In fact, the simulation of run 35B coincided quite well with the experimental values, and simulation of run 41B gave a depletion curve higher than run 35B, as expected.

The Mn\textsuperscript{*}NaCs and the MnNa\textsuperscript{*}Cs systems should have exhibited a decrease in slope of the depletion curves for Mn\textsuperscript{*} up and away from the Mn\textsuperscript{*}Cs binary as the more preferred ion Cs\textsuperscript{+} is exchanged preferentially with Na\textsuperscript{+}. The Na\textsuperscript{*} ion should have shown a trend for its depletion curves to decrease in slope down and away from the Na\textsuperscript{*}Cs binary. This was observed experimentally with two exceptions; run 34 was slightly above the Na\textsuperscript{*}Cs binary and run 30 was slightly below the Mn\textsuperscript{*}Cs binary. The simulated runs behaved as expected; however, the cases in which the ion Mn\textsuperscript{*} was diffusing yielded simulation results quite a bit higher than the experimental values.
e. Behavior of systems in which the two most preferred ions were in the exchanger and the least preferred ion constituted the electrolyte phase were exemplified by runs 33B and 43B for the system SrMn\(^{+}\)Cs and runs 37, 38 and 49 for the system Mn\(^{+}\)CsNa. In these cases it would be anticipated that the unfavorable exchange of Mn\(^{+}\) for Na\(^{+}\) ion and Sr for Cs\(^{+}\) ion would result in slower diffusion rates for Mn\(^{+}\) and Sr as Cs\(^{+}\) and Mn\(^{++}\) ion concentrations increased respectively. It should be easier for the least preferred ion to displace the second most preferred ion than the most preferred ion, resulting in retention of the most preferred species. The diffusion rate for Mn\(^{+}\) and Sr should be lower than in the respective binary systems Mn\(^{+}\)Na and Sr\(^{+}\)Cs. In the SrMn\(^{+}\)Cs system, it was the Mn\(^{+}\) which was tagged. In this case, the diffusion rate of Mn\(^{+}\) should have been higher than that of Mn\(^{+}\) in the binary system Mn\(^{+}\)Cs. Experiments validated these hypotheses for the Mn\(^{+}\)CsNa system, but not for the SrMn\(^{+}\)Cs system where run 33B lay on the Mn\(^{+}\)Cs binary curve, and run 43B was higher. The simulation of run 33B was good, and the simulation of run 43B gave a depletion curve lower than both the Mn\(^{+}\)Cs and the run 33B curve, as hypothesized.
3. Conclusions

The results of the ternary simulation runs were not nearly as good as the binary simulation results, and this was true of the Rao and his coworkers' (92, 93) ternary simulation studies as well. One problem which is quite obvious is that in a ternary rate run, two compositions should be followed experimentally in order to define the total system dynamics. Using radioactive tracer techniques, this would be difficult to carry out experimentally within any single run; however, replicate runs in which first one ion in the exchanger is tagged, then the other, would be one way to accomplish this requirement. In fact runs 34 and 66, Mn Na Cs and Mn Na Cs, at Mn of 0.488 and Na of 0.512 equivalent fraction, were carried out in this way. Based on the regression results from the fit of six and the fit of seven, the diffusion coefficient for Cs+ was determined as 0.34 x10^-5 and 0.396 x10^-5 cm^2/sec respectively; whereas, the Mn++ and Na+ values did not change appreciably. Run 34 was reasonably well simulated at both Cs+ diffusion coefficient values; however, the depletion curve for run 66 was simulated above the experimental curve, and was worse with the lower Cs+ diffusion coefficient value (higher curve). The Cs+ ion concentration in the resin has been calculated by material balance and is compared below for the experimental and the simulated data points.
<table>
<thead>
<tr>
<th>Time (Sec.)</th>
<th>Experimental</th>
<th>Simulated</th>
</tr>
</thead>
<tbody>
<tr>
<td>0</td>
<td>0</td>
<td>0</td>
</tr>
<tr>
<td>15</td>
<td>0.2336</td>
<td>0.2024</td>
</tr>
<tr>
<td>30</td>
<td>0.3202</td>
<td>0.3162</td>
</tr>
<tr>
<td>60</td>
<td>0.4390</td>
<td>0.4450</td>
</tr>
<tr>
<td>90</td>
<td>0.5216</td>
<td>0.5204</td>
</tr>
<tr>
<td>120</td>
<td>0.5694</td>
<td>0.5806</td>
</tr>
</tbody>
</table>

This shows reasonable agreement. There is no Cs Na curve to compare it to, as discussed before.

One problem encountered with the ternary program TRYBCH when running with activity driving force was that the flux balances, i.e., transfer across the film flux and resin accumulation (loss) flux, often did not balance well. This was not observed with the binary model BMAVD or RFMAVD, nor was it observed with TRYBCH when running with concentration driving force. In some cases, better flux balance could be obtained by juxtaposition of species, i.e., Sr$^{++}$-Cs$^+$-Mn$^{++}$ versus Sr$^{++}$-Mn$^{++}$-Cs$^+$ in terms of the species 1 and 2 for which the calculations are performed. In simulations of SrCsMn type systems this was helpful; however the equilibrium constants, Redlich-Kister constants, and diffusion coefficients have to be readjusted accordingly, and read in correctly.

Basically, it is believed that the calculation of $\tilde{D}_{ij}^{'}$ and $\tilde{D}_{ij}$, at the $j^{th}$ versus $j+1/2^{th}$ step caused the flux imbalance at time step $j^{1/2}$ as a result of adding the extremely non-linear activity terms,
to the diffusion coefficient expressions. Quasilinearization of all of these terms would solve the problem, since at each time step the iterative process would be carried out until convergence was attained, and the fluxes would balance. The difficulty in the derivation of matrix coefficients and solution vectors resulting from the additional terms would make this an unwarranted approach. Some other solution methods might be examined, but only if additional reliable ternary data, for which two concentrations are measured, are made available to test a revised model. In any event, the applicability of activity driving force in ion exchange should be tested further on binary systems before applying the model to additional ternary systems.
A. General Background

Axial diffusion or longitudinal dispersion of a species dissolved in a flowing fluid, or one gas mixed with another, is an important fluid mixing effect which must be accounted for in packed bed operations involving catalytic reactions or ion exchange processes, since it adversely affects concentration driving forces in the bed. In a streaming fluid, the following partial differential equation is commonly used to describe this phenomenon:

\[ \frac{\partial^2 c}{\partial z^2} - \frac{\partial c}{\partial t} = \frac{\partial c}{\partial t} \]  

where:

- \( c \) = concentration of species in bed
- \( z \) = bed length dimension
- \( U \) = interstitial velocity
- \( t \) = time
- \( E_d \) = axial dispersion coefficient or axial eddy diffusivity

Correlation of the axial dispersion coefficient are usually displayed as Reynold No. vs. Peclet No. where:

\[ Re = \frac{dp \cdot U \cdot \rho}{\nu} \]  
\[ Pe = \frac{U \cdot dp}{E_d} \left( \frac{U \cdot dp}{\varepsilon \cdot E_d} \right) \]
where:

\[ dp = \text{particle diameter} \]
\[ \varepsilon = \text{bed voidage} \]
\[ U' = \text{superficial velocity} \]
\[ U = \text{interstitial velocity} = \frac{U'}{\varepsilon} \quad (6-4) \]

Figure 6-1 is a graph on which the work of various investigators has been plotted (12, 23, 26, 27, 32, 61, 78, 115, 124), including coefficients developed in this research effort. The various investigators have used either concentration pulse analysis or frequency response analysis to characterize the axial eddy diffusivity in a variety of liquids and gases, over a range of flow rates and particle sizes. The two experimental points by Farkas & Byleveld (27) were carried out at low flow rates on Amberlite IR-120-plus ion exchange resin which was being eluted by a pulse of deionized water injected behind a flow of "city water." Conductivity measurements were used to characterize the effluent pulse dispersion.

B. Correlation of Axial Dispersion Coefficients - This Research

1. Experiments of Omatete and his Coworkers

The results obtained in this research project during the fitting of the resin loading curves for Omatete's data (78) on \( \text{Ag}^+, \text{Na}^+, \text{H}^+ \) are in the generally expected "range." This was not fortuitous, since values were chosen from this "range" when the resin phase ion diffusion coefficients were fit. It was found that the use of these eddy diffusion coefficients taken from the generalized plot, coupled with literature values for film coefficient parameters, resulted in close matching of the
Axial Dispersion in Packed Beds - Peclet No. vs. Reynolds No.
- Approximate Representation of Published Data -

Figure 6-1
experimental data solely by adjustment of resin phase diffusion coefficients. The fact that the values of the fitted resin phase diffusion coefficients were close to literature values, and that this single set of fitted diffusion coefficients for $D_{Ag^+}$, $D_{Na^+}$, & $D_H$ gave excellent simulations for 6 runs on binary systems and 3 runs on ternary systems--at minimum, suggested that the eddy diffusion coefficients taken from the generalized plot were "correct."

2. Erickson Experiments

2.1 Experimental Results

Erickson (26), on the other hand, actually ran dispersion tests in his experimental column by immersing the resin bed in 0.05 normal NH$_4^+$ solution, then followed with a step introduction of 0.1 normal NH$_4^+$ at two flow rates--and traced the effluent concentration by measuring conductivity. Plots of data at these two flow rates are given in Figures 6-2 and 6-3. The curves for various eddy diffusion coefficients are the result of an axial dispersion model derived in this research project. Erickson fitted the experimental data by solving equation 6-1 given above numerically; however, dispersion curves generated by Erickson could not account for the "tailing off" effect after the plug flow time (defined as total column voidage/volumetric flow rate). Erickson ascribed this tailing effect to holdup of 0.05 normal solution in the boundary layer around the resin beads, and chose "best"
modeled "S" curves at $E_d=0.75$ and $E_d=1.2\text{cm/sec}$ for the two flow rates.

2.2 Simulation of Erickson Data - This Research

The following model was developed in this work to test Erickson's boundary layer hypothesis.

Column

\[ \Delta z \uparrow \]

Illustration 6-1
Axial Dispersion Model Schematic

Resin Particle
Assumptions:

- No radial gradients in column
- Uniform spherical particles, $r_0$
- Uniform bed void space,
- Existence of a stagnant film around each particle with thickness, $\delta_f$, constant down column
- Resin at $r_0$ is saturated with preferred ion and is impenetrable to further ionic migration
- Diffusion coefficient in the stagnant film is $D_i^*$, constant through film and constant down column
- Axial dispersion coefficient, $E_d$, is constant down the column

**Fluid equation**

\[
\frac{\varepsilon \cdot \partial c_1 \text{bulk}}{\partial t} + \left( \frac{\text{vol}_{f} \cdot 3 \cdot (r_0 + \delta_f)^2}{(r_0 + \delta_f)^3 - r_0^3} \right) \cdot N_1^* \bigg|_{r=r_0 + \delta_f} = \frac{\varepsilon \cdot \partial^2 c_1 \text{bulk}}{\partial z^2} - \frac{v \cdot \partial c_1 \text{bulk}}{\partial z}
\]  

**Film Equation**

\[
\frac{\partial c_1^*}{\partial t} = D_i^* \left( \frac{\partial^2 c_1^*}{\partial y^2} + \frac{2}{(r_0 + y)} \frac{\partial c_1^*}{\partial y} \right)
\]

Since:

\[
r = r_0 + y ; \quad \frac{\partial}{\partial r} = \frac{\partial}{\partial y} \bigg|_{r>r_0} \quad \text{and} \quad \frac{\partial^2}{\partial r^2} = \frac{\partial^2}{\partial y^2} \bigg|_{r>r_0}
\]
where:

\[ \text{vol}_f = \frac{\text{Volume Film}}{\text{Volume Column}} = (1-\varepsilon) \cdot \left( \frac{(r_o + \delta_f)^3 - r_o^3}{r_o^3} \right), \text{ cm}^3 \]

\( \varepsilon \) = column voidage for flowing fluid
\( = (\varepsilon - \text{vol}_f), \text{ cm}^3 \)

\( c_{1,\text{bulk}}^* \) = concentration of species 1 in flowing fluid, moles/cm³

\( c_1^* \) = concentration of species 1 in film, moles/cm³

\( N_{1}^* \) = molar flux into or out of film, moles/cm²/sec

\( r_o \) = particle radius, cm

\( \delta_f \) = film thickness, cm

\( L \) = column length, cm

\( y \) = film dimension, cm

\( z \) = column length/dimension, cm

\( D_{1}^* \) = species 1 diffusivity in film, cm²/sec

\( E_d \) = species 1 axial diffusion coefficient, cm²/sec

\( v \) = superficial velocity in the column, cm/sec

\( \text{norm} \) = normality of coion, milliequiv./cm³ solution

\( z_1 \) = valence of species 1

Putting the terms into dimensionless form;

\( o \quad \text{Pe} = \frac{L \cdot v}{E_d \cdot \varepsilon} \quad (6-10), \quad x_{1,\text{bulk}}^* = \frac{c_{1,\text{bulk}}^* \cdot z_1}{\text{norm}} \quad (6-11) \)

\( o \quad x_1^* = \frac{c_1^* \cdot z_1}{\text{norm}} \quad (6-12), \quad Z = \frac{Z}{L} \quad (6-13), \quad Y = \frac{v}{\delta_f} \quad (6-14) \)

\( o \quad T = \frac{t \cdot v}{L} \quad (6-15), \quad J_{1}^* = \frac{N_{1}^* \cdot z_1}{\text{norm}} \quad (6-16) \)
\[ \lambda = \frac{3 \cdot \text{vol}_f \cdot (r_o + \delta_f)^2 \cdot L}{(r_o + \delta_f)^3 - r_o^3} \cdot v \quad (6-17), \quad \eta = \frac{D^{*} \cdot L}{\delta_f^2 \cdot v} \quad (6-18) \]

\[ \frac{\varepsilon \cdot \partial x_1^*}{\partial T} + \lambda \cdot \partial x_1^* \bigg|_{Y=1} = \frac{1}{Pe} \frac{\partial^2 x_1^*}{\partial z^2} - \frac{\partial x_1^*}{\partial z} \quad (6-19) \]

and:

\[ \frac{\partial x_1^*}{\partial T} = \eta \left[ \frac{\partial^2 x_1^*}{\partial Y^2} + \frac{2 \cdot \delta_f}{(r_o + \delta_f \cdot Y)} \cdot \frac{\partial x_1^*}{\partial Y} \right] \quad (6-20) \]

**Initial Conditions:**

\[ x_1 \mid_{T=0} \quad (6-21) \]

\[ x_1 \mid_{Z=0} = x_1 \mid_{Z=1} \quad (6-22) \]

\[ x_1 \mid_{T=0} = \begin{cases} 0 & \text{if } 0 \leq Z \leq 1 \\ \end{cases} \quad (6-23) \]

**Boundary Conditions-Column Exit**

\[ \frac{\partial x_1}{\partial z} \mid_{Z=1} = 0 \quad (6-24) \]

**Boundary Conditions-Column Inlet**

\[ x_1 \mid_{T=0} = \frac{1}{Pe} \frac{\partial x_1}{\partial z} \mid_{Z=0} = x_1 \mid_{T=0} \quad (6-25) \]
Gradient at Resin-Film Interface

\[ \frac{\partial x_1^*}{\partial Y}(0, Z, T) = 0 \quad \begin{cases} Y = 0 \\ 0 \leq Z \leq 1 \\ T \geq 0 \end{cases} \quad (6-26) \]

Boundary Condition at Film-Fluid Interface

\[ x_1(Z, T) = x_1^*(1, Z, T) \quad \begin{cases} Y = 1 \\ 0 \leq Z \leq 1 \\ T \geq 0 \end{cases} \quad (6-27) \]

so;

\[ \frac{\partial x_1^*}{\partial Y}_{\text{bulk}} = \frac{\partial x_1^*}{\partial Y} \quad \begin{cases} Y = 1 \\ 0 \leq Z \leq 1 \\ T \geq 0 \end{cases} \quad (6-28) \]

and;

\[ j_1^*(1, Z, T) = \frac{D^* \cdot \partial x_1^*}{\delta_f \cdot \partial Y} \quad \begin{cases} Y = 1 \\ 0 \leq Z \leq 1 \\ T \geq 0 \end{cases} \quad (6-29) \]

The equations were solved based on the implicit Crank-Nicholson finite difference technique. Since there are no non-linear terms, quasilinearization is not required, and solution of the coupled matrices for the fluid phase and the stagnant film phase is performed readily through use of the tridiagonal array algorithm developed by Thomas (13, 125). The simulation program developed in this work is called DISPRS, and was used to compute the concentration of
species 1 (0.1NH$_4^+$) exiting the column vs. time for various axial diffusion coefficients and "film thicknesses," as shown on Figures 6-2 & 6-3. Since the "film thickness" based on calculation of $k_L$ by Carberry's correlation (arrived at by equating $k_L$ to $\frac{D}{\delta_f}$) was in the range of 0.001 to 0.002 cm, the stagnant boundary layer could not account for more than 2.5-5% of the void volume. The model results showed this in that the calculated curves showed almost negligible differences with "film thickness" variations of from 0.001-0.004 cm. It becomes obvious in looking at the experimental points with reference to the plug flow time that the material balance on input vs. output did not reconcile.

The area under the "S" curve on the left side of the plug flow time should equal the area above the curve on the right side for proper reconciliation of material in versus out. The obvious answer was that the resin beads can absorb more 0.1 normal NH$_4^+$ ion than 0.05 normal NH$_4^+$ ion, and the assumption of impenetrable resin was incorrect (i.e., $\frac{\partial x}{\partial Y}(0,Z) = 0$).

In choosing an axial diffusion coefficient to best fit the experimental data it was therefore more important to match the concentration profile on the left side of plug flow time, representing first appearance due to dispersive effects. Axial dispersion coefficients of 3 to 7 cm$^2$/sec gave rise to Peclet No.s about an order of magnitude lower.
Residence Time Distribution - Axial Dispersion
System NH₄Cl and Dowex 50W x 8 Resin
- Stagnant Film, Impenetrable Sphere Model -

Experimental Data - Ave. Runs 5 & 6*
Temp. 26°C
Column Normality = 0.0519
Feed Normality = 0.1038
Superficial Velocity = 0.820 cm/sec
Resin Bed Depth = 55.6 cm
Column Dia. = 3.98 cm
Particle Radius = 0.039 cm
Plug Flow Residence Time = 23.9 sec
Calculated $E_D = 5.0 \text{ cm}^2/\text{sec}$
Calculated $E_D = 3.0 \text{ cm}^2/\text{sec}$
Calculated $E_D = 1.2 \text{ cm}^2/\text{sec}$
$D_{NH_4^+} = 1.994 \times 10^{-5} \text{ cm}^2/\text{sec}$
$K_f = 0.011 \text{ cm/sec}$
$\delta = 0.0019 \text{ cm (film "thickness")}$

Residence Time Distribution - Axial Dispersion
System NH\textsubscript{4}Cl and Dowex 50W x 8 Resin
- Stagnant Film, Impenetrable Sphere Model -

- Experimental Data - Ave. Runs 1, 2, 7 & 8*
  Temp. 26°C
  Column Normality = 0.0519
  Feed Normality = 0.1038
  Superficial Velocity = 1.682 cm/sec
  Resin Bed Depth = 55.6 cm
  Column Dia. = 3.98 cm
  Particle Radius = 0.039 cm
  Plug Flow Residence Time = 11.7 sec

- Calculated $E_D = 3.0 \text{ cm}^2/\text{sec}$
- Calculated $E_D = 5.0 \text{ cm}^2/\text{sec}$
- Calculated $E_D = 7.0 \text{ cm}^2/\text{sec}$

$D_{NH_3}^+ = 1.994 \times 10^{-5} \text{ cm}^2/\text{sec}$

$K_f = 0.015 \text{ cm/sec}$

$\delta = 0.00133 \text{ cm (Film "Thickness")}$

than other correlations on the generalized plot (Figure 6-1). A value of $E_d=4.0 \text{ cm}^2/\text{sec}$ was used in subsequent ion exchange simulations with the Erickson experimental exchange data (26). Values in the range of previous correlations shown on the generalized plot were used in the Omatete simulations, as stated above.
A. Development of General Ternary Model

The development of a ternary diffusion model for the resin phase was described in Chapter V; moreover, the treatment was general so that activity driving force could be used as opposed to concentration driving force. A program, TRYBCH, was developed to simulate radioactive tracer "batch" experiments on ternary systems. Likewise a binary version, BMAVDM, was developed to be used in conjunction with a general purpose non-linear regression program, DFUSFT, which was used to correlate diffusion coefficients and film coefficients utilizing "batch" data obtained from radioactive tracer experiments. Chapter IV on treatment of diffusion of ionic species through the boundary layer film for binary and ternary exchange included the development of a "pseudo electric field" model for use in the TRYBCH and BMAVDM simulation programs.

Additionally, development of a general purpose non-linear regression program, RIONFT, was described in the chapter on Equilibrium. This program was utilized to correlate the equilibrium constants relating the mass action between ionic species in electrolytes in equilibrium with ion exchangers, and the activity coefficient parameters for ionic species in the resin phase for both binary and ternary systems. Based on correlation of data on four ternary systems, the Redlich-Kister equation was chosen versus the Wilson and NRTL equations as having greater
ability to portray resin phase activity coefficient relationships in ion exchange systems, and yielded better predictions of equilibrium data on ternary systems based on equilibrium constants derived from binary systems and activity coefficient parameters.

All of these previous chapters have dealt with the data gathering phase which precedes any modeling effort. As was discussed in the preceding chapters, some of the data can be obtained from existing correlations, such as those describing film coefficients, electrolyte ionic species activity and diffusion coefficients, with appropriate correction for effects of electrolyte concentration. Correlations for axial dispersive effects are less well developed, and the treatment in this work compares the existing correlations, along with results derived from a simulation program, DISPRS, which can be used to simulate suitably designed experiments in columns. The results from two such axial diffusion simulations were discussed in Chapter VI.

Experimental data are required to correlate resin phase equilibrium parameters for the 3 binary pairs in a ternary system since literature background is sparse in this area. It has been shown in this work that binary correlation parameters can be combined to estimate ternary equilibrium data with fair reliability, although it is desirable to have experimental data points on ternary systems in the region of interest so that material balance requirements of column operation can be satisfied.

Similarly, shallow bed experiments using radioactive tracers are desirable to establish the effective diffusion coefficients
for ionic species in the particular resin grade which will be utilized in column operation. As demonstrated in this work, it is possible to correlate single value effective diffusivities for multicomponent systems by using activity driving forces. As will be shown subsequently in this chapter, the effective diffusion coefficients can be derived from column effluent concentration histories (ECH), but the fitting of diffusion parameters involves a trial and error procedure through simulation of column experiments. A starting point is to use best available literature values of the resin phase diffusion coefficients for the species involved, and then "tuning" the values to get a best fit.

The development of the general "batch" equations for both ternary and binary ion exchange in the resin phase included a column dimension, i, which implied varying bulk fluid ion species concentration as a function of time and column position. The i dimension was dropped out for the less complicated case developed for "infinite reservoir" type batch simulations. The fluid phase equations describing column operation will be developed in this chapter, and then coupled to the resin phase equations developed in Chapter V to yield two simulation models. **RFMAVD** (Revised Final Mass Action (Equilibrium) Variable (Solid Phase) Diffusivity) is the model for binary ion exchange column simulation, and is utilized to fit diffusion coefficients from the experimental column effluent concentration history (ECH). These coefficients can be used in the ternary simulation model. **TERNEX** (Ternary Ion Exchange) is the ternary column simulation model for ternary ion exchange. Both
have extensive diagnostics and write out options which will be described later.

B. Fluid Phase - Ternary Ion Exchange, Column Operation

1. Development of Fluid Phase Equations

The general ternary ion exchange model for column operation will be developed first, since the binary ion exchange model for column operation is a simplified subset of the ternary equations.

Volumetric Flow = \( v(\text{velocity}) \times \text{empty column area} \)

Assumptions:

- No radial concentration gradients in column
- Uniform spherical resin particles
- Uniform bed voidage
- No swelling of resin, i.e., no bed volume change with time
Isothermal behavior throughout column

Stoichiometric exchange, i.e., no adsorption of undissociated salts

No electric field gradients in column streaming fluid

Axial dispersion coefficient, $E_d$, is constant down the column

Fluid Equations

\[
\frac{\partial c_{i,bulk}}{\partial t} + \frac{(1-\varepsilon)}{\varepsilon} \frac{3}{r_0} \cdot k_{Li} \cdot (c_{i,bulk} - c_{i,\text{int}}^*) = E_D \cdot \frac{\partial^2 c_{i,bulk}}{\partial \ell^2} - \frac{v}{\varepsilon} \cdot \frac{\partial c_{i,bulk}}{\partial \ell}
\]

for $(i=1,2,3)$

where from Chapter IV, Treatment of Electrolyte Phase Film Coefficients, equation (4-90) is given in molar units as:

\[
k_{Li,\text{eff}} = \frac{D_{ii}^*}{\delta_{\text{eff max}}} + \frac{D_{ij}^*}{\delta_{\text{eff max}}} \cdot \left( \frac{(c_{j,bulk} - c_{j,\text{int}}^*)}{(c_i,bulk - c_i,\text{int})} \right)
\]

\[c_i = \text{concentration of species } i \text{ in flowing fluid, moles/cc}
\]

\[c_i^* = \text{concentration of species } i \text{ in film at interface with resin, and in equilibrium, moles/cc}
\]

above for $(i = 1,2,3)$

\[\varepsilon = \text{column voidage for flowing fluid, cc void}
\]

\[\ell = \text{resin bed length dimension, cm}
\]

\[BL = \text{resin bed length, cm}
\]

\[r_0 = \text{resin particle radius, cm}
\]
\[ E_D = \text{axial diffusion coefficient (all species) - cm}^2/\text{sec} \]
\[ v = \text{superficial velocity of fluid in column, cm/sec} \]
\[ \text{norm} = \text{normality of coion, equiv/cc} \]
\[ z_i = \text{valence of species } i \ (i = 1, 2, 3, Y; \text{ where } Y = \text{coion}) \]
\[ t = \text{time, sec} \]
\[ 3/r_0 = \text{area/volume ratio for a sphere} \]
\[ Q_r = \text{equivalents of active sites in resin/cc} \]
\[ D^*_i, D^*_{ij} = \text{effective diffusion coefficients in the film} \]
\[ \text{as defined by equations (4-96), (4-97), (4-98), and (4-99)} \]
\[ \delta_{\text{eff max}} = \text{"effective boundary layer thickness" as defined by equation (4-88)} \]

Putting the terms into dimensionless form.

\[ Y = \frac{\dot{q}}{BL}, \quad R = \frac{r}{r_0}, \quad T = t\frac{v}{BL} \quad (7-2, 3, 4) \]
\[ Pe = \frac{BL\cdot v}{E_D\cdot \varepsilon}, \quad \delta_s = \frac{BL\cdot (1-\varepsilon)}{r_0^2\cdot v}, \text{bed length parameter} \]
\[ x_{i_{\text{bulk}}} = \frac{c_i^*\cdot (|z_i|)}{\text{norm}}, \quad x_{i_{\text{bulk}}}^* = \frac{c_i^*\cdot (|z_i|)}{\text{norm}} \quad (7-7, 8) \]
\[ B_{i} = 2\cdot k_L\cdot r_0; \text{ Biot no.} \quad (7-9) \]

where \( i = 1, 2, 3 \) for above equations
In Fick's Law treatment where $\bar{D}_i = \text{constant}$ ($i = 1, 2, 3$)

\[
\begin{align*}
2r_0k_i \frac{2r_0k_i}{\bar{D}_i} = \text{Bi}_i \text{eff}, \quad \text{and} \quad \delta_s = \frac{BL(1-\varepsilon)\bar{D}_i}{r_0^2\nu}
\end{align*}
\]

so

\[
\text{Bi}_i \delta_s = \left( \frac{2k_i \text{eff BL}(1-\varepsilon)}{r_0^2 \nu} \right)
\]

(7-12)

however $\bar{D}_i = f(x_i)$, and must be kept inside the partials

\[
\frac{\partial^2(x)}{\partial x_1^2} \quad \text{and} \quad \frac{\partial^2(x)}{\partial x_1^2}
\]

Using the transformation $\rho = R^2$ as discussed in Chapter V, Diffusion in the Resin Phase, the column equations are derived below.

Setting up the equations for the two most favored species, 1 and 2:

\[
\begin{align*}
\varepsilon \frac{\partial x_1}{\partial t} \text{bulk} + \frac{3}{2} \delta_s \text{Bi}_1(x_1 - x_1^{*}) &= \frac{1}{\text{Pe}} \frac{\partial^2 x_1}{\partial Y^2} - \frac{\partial x_1}{\partial Y} \\
\varepsilon \frac{\partial x_2}{\partial t} \text{bulk} + \frac{3}{2} \delta_s \text{Bi}_2(x_2 - x_2^{*}) &= \frac{1}{\text{Pe}} \frac{\partial^2 x_2}{\partial Y^2} - \frac{\partial x_2}{\partial Y}
\end{align*}
\]

(7-14)

Initial Conditions

\[
\begin{align*}
\text{Bi}_1 \text{bulk} & \quad x_1 - \text{in} \quad (T) \quad T \geq 0 \\
\text{Bi}_2 \text{bulk} & \quad x_2 - \text{in} \quad (T) \quad T \geq 0
\end{align*}
\]

(7-15)

(7-16)
\[ x_{1\text{bulk}}(Y,0) = x_{1\text{bulk}}(Y) \quad 0 \leq Y \leq 1 \]  \hspace{1cm} (7-17)

\[ x_{2\text{bulk}}(Y,0) = x_{2\text{bulk}}(Y) \quad 0 \leq Y \leq 1 \]  \hspace{1cm} (7-18)

**Boundary Conditions**

\[ \frac{\partial x_{1\text{bulk}}}{\partial Y}(1,T) = 0 \quad T \geq 0 \]  \hspace{1cm} (7-19)

\[ \frac{\partial x_{2\text{bulk}}}{\partial Y}(1,T) = 0 \quad T \geq 0 \]  \hspace{1cm} (7-20)

Note: this B.C. gave rise to instability and will be discussed in a subsequent section of this Chapter.

\[ x_{1\text{bulk}}(0,T) - \frac{1}{Pe} \cdot \frac{\partial x_{1\text{bulk}}}{\partial Y} = x_{1\text{int}}(T) \]

\[ T \geq 0 \]  \hspace{1cm} (7-21)

\[ x_{2\text{bulk}}(0,T) - \frac{1}{Pe} \cdot \frac{\partial x_{2\text{bulk}}}{\partial Y} = x_{2\text{int}}(T) \]

\[ T \geq 0 \]  \hspace{1cm} (7-22)

**Resin - Film Interface where \( \rho = 1 \) at conditions of**

\[ 0 \leq Y \leq 1 \]

\[ T \geq 0 \]

\[ \frac{Q_r}{\text{norm } Bi_1} \left( \frac{\partial x_1}{\partial \rho} \bigg|_{\rho=1} + \frac{\partial x_2}{\partial \rho} \bigg|_{\rho=1} \right) = (x_{1\text{bulk}} - x_{1\text{int}}) \]  \hspace{1cm} (7-23)

\[ \frac{Q_r}{\text{norm } Bi_2} \left( \frac{\partial x_2}{\partial \rho} \bigg|_{\rho=1} + \frac{\partial x_1}{\partial \rho} \bigg|_{\rho=1} \right) = (x_{2\text{bulk}} - x_{2\text{int}}) \]  \hspace{1cm} (7-24)

Equilibrium is maintained between \( x^*_i \) and \( x^*_{i\text{int}} \).
Chapter V, Diffusion in the Resin Phase, contains equations (5-65) and (5-66) which deal with the quasilinearization of $x_i^{*\text{int}}$ as an implicit function of $(x_i^{*\text{int}}, x_i^{\text{int}})$, and differentiation of implicit functions is discussed as a method to obtain the partial derivatives required for quasilinearization.

As with the treatment of the non-linear parabolic partial differential equations describing diffusion of ionic species in the resin phase, the non-linear parabolic partial differential equations describing the fluid phase in column operation were solved using the Crank-Nicholson implicit method (125) with quasilinearization of non-linear terms (59). It is the addition of the axial dispersion terms to the hyperbolic equations describing the concentration changes in the fluid phase which convert them to parabolic partial differential equations. As such, they cannot be decoupled from the resin phase equations in a straightforward manner--since finite difference representations in both phases involve nodes n, n+1, & n-1. The solution of the system of finite difference equations, including the resin phase equations, will be covered below.

**Matrix Equations - Fluid Phase**

The setting up of finite difference equations for the two ionic species, where column length $Y (=1)$ is divided into I nodes, or I-1 increments ($\Delta Y=1/(I-1)$), involves a matrix representation, since the terms in the partial differential equations are approximated as:
In the Crank-Nicholson method, the time steps \( j \) and \( j+1 \) are averaged so that the \( j+1 \)th equivalent versions of the above partials are averaged with the \( j \)th versions to give the partial derivative approximations as a function of time.

Matrix equations to be solved at iteration \( n+1 \) and time step \( j+1 \) are:

\[
B_1 \cdot x_{1, \text{bulk}}_{n+1} = U_1 \cdot \left[ \frac{dx_1}{dx_1 \text{ int}} \right]_{n} (i,j+1) \cdot \bar{x}_1(M,i,j+1)_{n+1} + V_1 \cdot \left[ \frac{dx_1}{dx_2 \text{ int}} \right]_{n} (i,j+1) \cdot \bar{x}_2(M,i,j+1)_{n+1}
\]

\[
B_2 \cdot x_{2, \text{bulk}}_{n+1} = U_2 \cdot \left[ \frac{dx_2}{dx_1 \text{ int}} \right]_{n} (i,j+1) \cdot \bar{x}_1(M,i,j+1)_{n+1} + V_2 \cdot \left[ \frac{dx_2}{dx_2 \text{ int}} \right]_{n} (i,j+1) \cdot \bar{x}_2(M,i,j+1)_{n+1}
\]

where equation solution vectors \( U_1 \) and \( U_2 \) contain terms from time step \( j \), as well as the values of resulting terms qualilinearization of non-linear terms at the previous iteration, \( n \).
When the matrices and vectors are expanded:

\[
\begin{align*}
B_1 &= \begin{pmatrix}
\alpha_1 & \beta_1 \\
\gamma_1 & \alpha_2 & \beta_2 \\
\gamma_1 & \alpha_1 & \beta_1 \\
\gamma_1 & a_1 & \beta_1 \\
\end{pmatrix} \\
V_1 &= \begin{pmatrix}
v_1 \\
v_1 \\
v_1 \\
v_1 \\
\end{pmatrix}
\end{align*}
\]

and

\[
\begin{align*}
x_1 &= \begin{pmatrix}
\left. x_1 \right|_{i,j+1} \\
\left. x_1 \right|_{2,j+1} \\
\left. x_1 \right|_{(i,j+1)} \\
\left. x_1 \right|_{(I,j+1)} \\
\end{pmatrix} \\
U_1 &= \begin{pmatrix}
u_1(1,d) \\
u_1(i,j) \\
u_1(I,j) \\
\end{pmatrix}
\end{align*}
\]

\[
\begin{align*}
B_2 &= \begin{pmatrix}
\alpha_2 & \beta_2 \\
\gamma_2 & \alpha_2 & \beta_2 \\
\gamma_2 & \alpha_2 & \beta_2 \\
\gamma_2 & a_2 & \beta_2 \\
\end{pmatrix} \\
V_2 &= \begin{pmatrix}
v_2 \\
v_2 \\
v_2 \\
v_2 \\
\end{pmatrix}
\end{align*}
\]

and

\[
\begin{align*}
x_2 &= \begin{pmatrix}
\left. x_2 \right|_{1,j+1} \\
\left. x_2 \right|_{2,j+1} \\
\left. x_2 \right|_{(i,j+1)} \\
\left. x_2 \right|_{(I,j+1)} \\
\end{pmatrix} \\
U_2 &= \begin{pmatrix}
u_2(1,j) \\
u_2(i,j) \\
u_2(I,j) \\
\end{pmatrix}
\end{align*}
\]
2. Coupling of Resin Phase Equations to Fluid Phase Equations

The coupled set of two resin phase matrix equations (equations (5-81) and (5-82)) with the two fluid phase matrix equations can be solved through use of algorithms developed in this project for inverting seven diagonal and five diagonal arrays (Appendices E&F) following the Thomas method (13, 125) for tridiagonal arrays. In order to solve the fluid phase matrix equations, values of $\bar{x}_1(M, i, j+1)_{n+1}$ and $\bar{x}_2(M, i, j+1)_{n+1}$ derived from the solution of the resin phase algorithm are expressed as functions of $x_1(i, j+1)_{n+1}$ and $x_2(i, j+1)_{n+1}$, since the fluid phase concentrations of species 1 and 2 appear in the resin phase equations (5-81) and (5-82) at the resin-fluid interface.

Equations (7-27) and (7-28) then have the following matrix notation when the substitution is made:

\[
0 = -\frac{dx_1}{dx_1^{\text{int}}} (i, j+1)_{n+1} \cdot g_1(i, j+1)_{n+1} \cdot x_1(i, j+1)_{n+1}^{\text{bulk}} + \frac{dx_1}{dx_1^{\text{int}}} (i, j+1)_{n+1} \cdot g_2(i, j+1)_{n+1} \cdot x_2(i, j+1)_{n+1}^{\text{bulk}}
\]

where

\[
0 \bar{x}_1(M, i, j+1)_{n+1} = g_1(i, j+1)_{n+1} \cdot x_1(i, j+1)_{n+1}^{\text{bulk}}
\]

\[
0 \bar{x}_2(M, i, j+1)_{n+1} = g_2(i, j+1)_{n+1} \cdot x_2(i, j+1)_{n+1}^{\text{bulk}}
\]
The equations can then be depicted in terms of a five diagonal array as shown in Figure 7-1, for a simple five node example. The values of $x_1(i,j+1)_{n+1}$ and $x_2(i,j+1)_{n+1}$ are calculated through use of the five diagonal array algorithm, then these values are used to compute values of $\bar{x}_1(M,i,j+1)_{n+1}$ and $\bar{x}_2(M,i,j+1)_{n+1}$ which represent the concentration of

\begin{equation}
\begin{aligned}
B_{1i} \cdot x_{1 \text{ bulk}_{n+1}} &= U_{1i} + V_{1i} \left( \frac{d\bar{x}_1}{dx_1} \right)_{n+1} + g_1(i,j+1) \cdot x_1(i,j+1)_{n+1} \\
&\quad + \frac{d\bar{x}_2}{dx_2} (i,j+1) \cdot h_2(i,j+1) \cdot x_2(i,j+1)_{n+1} \\
&\quad + \frac{d\bar{x}_1}{dx_1} (i,j+1) \cdot h_1(i,j+1) \cdot x_1(i,j+1)_{n+1}
\end{aligned}
\end{equation}

where:

\begin{equation}
\begin{aligned}
\bar{x}_2(M,i,j+1)_{n+1} &= h_2(i,j+1) \cdot x_2(i,j+1)_{n+1} \\
\bar{x}_1(M,i,j+1)_{n+1} &= h_1(i,j+1) \cdot x_1(i,j+1)_{n+1}
\end{aligned}
\end{equation}

or to simplify:

\begin{equation}
\begin{aligned}
B_{1i} \cdot x_{1 \text{ bulk}_{n+1}} &= U_{1i} + g_1(i,j+1) \cdot x_1(i,j+1)_{n+1} + g_2(i,j+1) \cdot x_2(i,j+1)_{n+1} \\
&\quad + h_1(i,j+1) \cdot x_1(i,j+1)_{n+1} \\
B_{2i} \cdot x_{2 \text{ bulk}_{n+1}} &= U_{2i} + h_2(i,j+1) \cdot x_2(i,j+1)_{n+1} + h_1(i,j+1) \cdot x_1(i,j+1)_{n+1}
\end{aligned}
\end{equation}

The equations can then be depicted in terms of a five diagonal array as shown in Figure 7-1, for a simple five node example. The values of $x_1(i,j+1)_{n+1}$ and $x_2(i,j+1)_{n+1}$ are calculated through use of the five diagonal array algorithm, then these values are used to compute values of $\bar{x}_1(M,i,j+1)_{n+1}$ and $\bar{x}_2(M,i,j+1)_{n+1}$ which represent the concentration of $\bar{x}_1$.
Figure 7-1

Five Diagonal Array - Fluid Phase Equations
Matrix combining 2 fluid equations
to be solved for $x_1^{\text{bulk}}$ and $x_2^{\text{bulk}}$ at I column nodes

$$
\begin{bmatrix}
  a_{11} - g_{11} & \beta_1 & 0 & 0 & 0 & 0 & 0 & 0 \\
  -h_{11} & a_{21} & 0 & \beta_{21} & 0 & 0 & 0 & 0 \\
  \gamma & 0 & a_{12} - g_{22} & \beta & 0 & 0 & 0 & 0 \\
  0 & \gamma & -h_{12} & a_{22} & 0 & \beta & 0 & 0 \\
  0 & 0 & \gamma & 0 & a_{13} - g_{23} & \beta & 0 & 0 \\
  0 & 0 & 0 & \gamma & -h_{13} & a_{23} & 0 & \beta \\
  0 & 0 & 0 & 0 & \gamma & 0 & a_{14} - g_{24} & \beta \\
  0 & 0 & 0 & 0 & 0 & \gamma & -h_{14} & a_{24} & 0 & \beta \\
  0 & 0 & 0 & 0 & 0 & 0 & \gamma & 0 & a_{15} - g_{25} \\
  0 & 0 & 0 & 0 & 0 & 0 & 0 & \gamma & -h_{15} & a_{25} \\
\end{bmatrix}

\begin{bmatrix}
x_1^{(1,j+1)} \\
x_2^{(1,j+1)} \\
x_1^{(2,j+1)} \\
x_2^{(2,j+1)} \\
x_1^{(3,j+1)} \\
x_2^{(3,j+1)} \\
x_1^{(4,j+1)} \\
x_2^{(4,j+1)} \\
x_1^{(5,j+1)} \\
x_2^{(5,j+1)} \\
\end{bmatrix}

= \begin{bmatrix}
x_1^{(1,j)} \\
x_2^{(1,j)} \\
x_1^{(2,j)} \\
x_2^{(2,j)} \\
x_1^{(3,j)} \\
x_2^{(3,j)} \\
x_1^{(4,j)} \\
x_2^{(4,j)} \\
x_1^{(5,j)} \\
x_2^{(5,j)} \\
\end{bmatrix}
$$

where $a_{11} = a_{1} - g_{11}$

$a_{21} = a_{2} - h_{21}$
and $\bar{x}_2$ at the resin-film interface at each column length node. The remaining values of $\bar{x}_1(m,i,j+1)_{n+1}$ and $\bar{x}_2(m,i,j+1)_{n+1}$, $m=1$ to $M-1$, are solved with the seven diagonal array algorithm, and the process continues at each time step $j+1$ until the iterative process brings all values of $\bar{x}_1$ and $x_i$ bulk to within the solution tolerance limits--then proceeds to the next time step. As with the seven diagonal array, a scratch five diagonal array is used to load $n^{th}$ iteration values of coefficients and solution vectors for both $x_1$ bulk and $x_2$ bulk at each time step $j+1$, solves for values of $x$, and sorts them alternately into $x_1$ bulk and $x_2$ bulk values for the $n+1^{th}$ iteration.

Appendix I covers the detailed development of the coupling of the resin phase matrix equations with those of the fluid phase for the Ternary Model.

C. Solution for Column Ion Exchange Operation - Binary Systems

1. Finite Difference Equations and General Solution

Treatment of the development of the binary column model are covered in some detail in Appendix J, since it is quite easy to see how the solution of the resin phase equation is coupled to the fluid phase equation, in that both result in tridiagonal arrays. The ternary column model, although more complex, is solved using the same general procedure. In addition, treatment of the boundary conditions by finite difference equations in both the resin phase and fluid phase are also discussed in Appendix J.
2. Initialization Procedures

The proper "conditioning" of the column is essentially the same as described by eq 5-60 in the chapter V, Diffusion in the Resin Phase. That is, the initial conditions in the resin are set up by matching the diffusion flux at the resin boundary:

\[
\left( \frac{4 \cdot Q_r}{B_{i1}(1,0) \cdot \text{norm}} \right) \cdot \bar{D}_{i1}(1,0) \cdot \frac{\partial \bar{x}_1}{\partial p} \Bigg|_{\rho=1} (i,0) = (x_1(i,0) - x_{1 \text{ int}}^*(i,0))
\]

where \( \frac{\partial \bar{x}_1}{\partial p} \Bigg|_{\rho=1}(i,0) \) is calculated using a backward difference numerical representation between resin node \( M \) and \( M-1 \). In the "batch case," a trial and error procedure (false positioning) is used to solve the flux balance equations, since the equilibrium expression relating the resin surface concentration \( \bar{x}_{1 \text{ int}} \), and the fluid phase concentration, \( x_{1 \text{ int}}^* \), is non-linear. However, the relationship is monotonic and this procedure works very quickly. In the column model, some distribution of the bulk fluid concentration down the column is required at time equal 0 (\( j = 0 \)), since a step input at \( j = 1 \) generally will not result in convergence for the solution to the coupled resin and fluid phase equations. Svedberg (125) suggested an exponential decay type of relationship of the form:

\[
\begin{align*}
\text{bulk} \quad x_1(i,0) \ &= \ x_{1 \text{ int}}(i,0) \cdot \exp\left\{ -\frac{(i-1)}{\text{disfac}_{\text{bulk}}} \right\} \\
&= x_{1 \text{ int}}(i,0) \cdot \exp\left\{ -\frac{(i-1)}{\text{disfac}_{\text{bulk}}} \right\}
\end{align*}
\]

\( (i = 1 \text{ to } 1) \)
In this work, three revisions were made to the Svedberg approach:

- A constant, diskon, was multiplied times disfac to allow a read-in input adjustment to the initial distribution. This was required, since under certain combinations of the elements of disfac the column, including the resin phase, could be "preloaded" up to 8% by the initialization procedure. In almost all runs made in this work, a diskon factor could be used such that only 0.1 to 0.5% column "preloading" occurred, consistent with a smooth start. The amount of "preloading" is calculated and printed out by RFMAVD before starting the simulation so that without much difficulty the appropriate diskon factor can be used for each case.

- Since most of the column capacity is in the resin phase, it required fewer adjustments to minimize column "preloading" if the resin phase was treated according to:

\[
disfac = 3\Delta Y_k(1,0) \cdot (1-\varepsilon) \cdot \frac{BL_{\text{eff}}}{v \cdot r_0} \quad (7-47)
\]

The flux balance then determined the appropriate

\[
\bar{x}_i(M, i, 0) = \bar{x}_i(M, 1, 0) \cdot \exp \left\{ -(i-1) \cdot \text{disfac}_{\text{initial}} \right\} \quad (i=1 \text{ to } I)
\]

\[
x_i(m, i, 0) \quad \text{for } i=1 \text{ to } I
\]
As a further refinement for column operation, \( x_1(M,1,0) \) was defined as the composition of resin surface in equilibrium with \( x_{1}^{\text{bulk}}(1,0) \), node 1 or \( Y = 0 \). As noted by B.C. eq 21, \( x_{1}^{\text{bulk}}(1,0) \) is not \( x_{1}^{\text{feed}}(1,0) \) but the resultant of the amount of species 1 moving into the column by convection minus the amount moving away due to the axial dispersion flux.

These refinements are included in RFMAVD, and invariably lead to a smooth start with minimum column "preloading" due to initiation. The trial and error procedure (false positioning) is used to solve for \( x_{1}^{\text{bulk}}(M,1,0) \), from which all other column compositions at \( x_{1}^{\text{bulk}}(M,i,0) \) are calculated, \( (i=2 \text{ to } I) \) by equation (7-48). Finally, \( x_{1}^{\text{int}}(i,0) \) values are computed, \( (i=1 \text{ to } I) \), and \( x_{1}^{\text{bulk}}(i,0) \) values are given by equation (7-45), \( (i=1 \text{ to } I) \).

3. Treatment of Boundary Conditions by Numerical Approximation

Two fluid phase boundary conditions were approximated by finite difference equations, and coupled to the general fluid phase equation. The first, at the column inlet, transforms eq 7-21 into a central difference approximation using a "fictitious" point outside the column inlet.

\[
\frac{1}{2} \left[ x_1^{\text{bulk}}(1,j+1)_{n+1} + x_1^{\text{bulk}}(1,j) \right] - \frac{1}{4\cdot Pe\cdot \Delta Y} \left[ x_1^{\text{bulk}}(2,j+1)_{n+1} \right. \\
\left. - x_1^{\text{bulk}}(0,j+1)_{n+1} + x_1^{\text{bulk}}(2,j) - x_1^{\text{bulk}}(0,j) \right] = x_1 \text{-in}_{\text{bulk}}
\]

The numerical approximation of the general equation for the fluid phase is also set up around \( i \) nodes 0, 1, 2, and through cross
multiplication of coefficients, \( x_i(0,j+1)_{n+1} \) and \( x_i(0,j)_{\text{bulk}} \) are eliminated. This gives the coefficients and solution vector for node \( i = 1 \) in terms of column nodes \( i = 1 \) and 2.

This representation of the column inlet boundary condition presented no problems in the binary model for the simulations performed in conjunction with this research. Svedberg notes (125) however that in his studies of the numerical simulation of fixed bed processes that if the concentration profile is too steep as caused by a low Peclet No. (high axial dispersion), and the column length grid is too "coarse," the composition at the bed inlet will oscillate. He found that the oscillations are local and not transferred into the bed. This problem was not observed with RFMAVD on binary simulations; however in the ternary model, TERNEX, this problem was severe and was corrected with an initializing procedure to be discussed later.

The second boundary condition, at the column outlet, was represented initially in this work as the method Svedberg (125) had used, i.e.:

\[
\frac{\partial x_i}{\partial Y} \big|_{1_{\text{bulk}}} = 0 , \text{ where at node } I, Y = 1 \quad (7-50)
\]

Svedberg represented this with a forward difference approximation instead of a central difference, since he found the latter approximation gave incorrect solutions for high Peclet Numbers. He stated that the central difference gave
too much weight to the concentration value at the boundary. The forward difference, using a "fictitious" point outside of the column could represent equation (7-50) as:

\[
x_{1bulk}(i,j+1)_{n+1} + x_{1bulk}(i,j) = x_{1bulk}(i+1,j+1) + x_{1bulk}(i+1,j)
\]

The "fictitious" point, I+1, is eliminated in the usual way by combination with a central differenced general equation for node i = I. The coefficients and solution vector for this node are then obtained. Even so, Svedberg noted that for high Peclet no. values, oscillations in concentration occurred at the column exit as the mass transfer zone passed through this boundary. He noted that this was probably due to the "stiff" component introduced into the solution by the boundary condition, and that more column length nodes (smaller increments) decreased the oscillations. In this work, these oscillations were not observed, probably because (1) liquid Peclet numbers are an order of magnitude lower than Peclet Nos. for gases, and Svedberg was investigating both gas adsorption as well as liquid ion exchange type processes; and (2) dimensionless bed length increments were small (0.01). Use of RFMAVD for simulation of binary ion exchange gave no problem in this regard; however TERNEX, in ternary ion exchange modeling gave rise to a "piling up" of concentration at the column exit. The following treatment was used to minimize this effect in TERNEX, and was therefore adopted for RFMAVD, and will be
discussed for the binary system since it is easy to draw comparisons.

Assume the bottom column increment, $\Delta Y$, is a CFSTR, then a material balance on that section gives:

\begin{equation}
-x_{1}^{*}(I-1) + x_{1}(I) + \frac{3}{2}\delta_{s}Bi_{1}(I) \cdot \left[x_{1}^{*}(I) - x_{1}(I)\right] \Delta Y
+ \frac{1}{Pe} \frac{\partial x_{1}(I-1)}{\partial Y} = -\varepsilon \cdot \Delta Y \cdot \frac{\partial x_{1}(I)}{\partial T}
\end{equation}

that is,

\begin{equation}
\text{convection (out-in) + transfer to resin + in from diffusion = accumulation}
\end{equation}

in finite difference terms:

\begin{equation}
\frac{1}{2} \left[ x_{1}^{*}(I,j+1)_{n+1} + x_{1}(I,j) \right] - \frac{1}{2} \left[ x_{1}^{*}(I-1,j+1)_{n+1} - x_{1}(I-1,j) \right]
+ \frac{3}{4}\delta_{s}Bi_{1}(I) \cdot \Delta Y \cdot \left[ x_{1}^{*}(I,j+1)_{n+1} + x_{1}(I,j) - x_{1}^{*}(I,j+1)_{n+1} \right]
+ x_{1}(I,j) - x_{1}(I-1,j)
\end{equation}

This leads to coefficients and solution vector for the $i^{th}$ node, where $\alpha_{i}$ is the $x_{1}(I)$ coefficient and $\gamma_{i}$ is the $x_{1}^{*}(I-1)$ coefficient. Comparison of these coefficients for the various methods are:
Forward difference

\[ \alpha_I = 1 + \frac{3 \delta_s \cdot \text{Bi}_I(I) \cdot \Delta T}{4 \epsilon} + \frac{\Delta T}{4 \epsilon \Delta Y} + \frac{1}{2} \frac{\Delta T}{\epsilon \text{Pe} \Delta Y^2} \]  \hspace{1cm} (7-55)

\[ \gamma_I = - \frac{\Delta T}{4 \epsilon \Delta Y} - \frac{\Delta T}{2 \epsilon \text{Pe} \Delta Y^2} \]  \hspace{1cm} (7-56)

CFSTR

\[ \alpha_I = 1 + \frac{3 \delta_s \cdot \text{Bi}_I(I) \cdot \Delta T}{4 \epsilon} + \frac{\Delta T}{2 \epsilon \Delta Y} + \frac{1}{2} \frac{\Delta T}{\epsilon \text{Pe} \Delta Y^2} \]  \hspace{1cm} (7-57)

\[ \gamma_I = - \frac{\Delta T}{2 \epsilon \Delta Y} - \frac{\Delta T}{2 \epsilon \text{Pe} \Delta Y^2} \]  \hspace{1cm} (7-58)

Central Difference

\[ \alpha_I = 1 + \frac{3 \delta_s \cdot \text{Bi}_I(I) \cdot \Delta T}{4 \epsilon} \]  \hspace{1cm} (7-59)

\[ \gamma_I = - \frac{\Delta T}{\epsilon \text{Pe} \Delta Y^2} \]  \hspace{1cm} (7-60)

Inspection would suggest that increase in the Pe no. term is moderated in the CSFTR case by giving more weight to the convective term. It also suggests why the central difference at this B.C. gave trouble, since neither coefficient, \( \alpha_I \) or \( \gamma_I \), contained a convective term. A high Pe no. would reduce the coefficients in this case, which for a given value of \( x_I(I-I) \), would tend to have a pronounced upward leveraging effect on the value of \( x_I(I) \). In any event, the CSFTR model applied to \( \text{bulk}_\text{TERNEX} \) ameliorated the concentration "pile up" for particular species at the column exit. The CSFTR model also is, in effect, the treatment for the center spherical node of the resin, as discussed before.
4. Convergence and Stability of the Simulation Model

It would be useful at this point to discuss the convergence and stability performance of the implicit Crank-Nicholson method for approximating solutions to parabolic partial differential equations. Sherwood, Mickley & Reed (71) indicate that the accuracy of the results obtained by numerical solution methods is difficult to estimate; however, it is known that if the method used is both "convergent" and "stable" for the equations being solved then the accuracy is determined by the number of increments used in the grids. Estimate of accuracy for a given solution can be tested by rerunning solutions with finer grid spacings until the difference between results is within limits tolerable to the user. The convergence criterion deals with the approach of the approximation solution to the exact solution in the limit as number of grid nodes approaches infinity. Stability deals with the growth of error in the numerical approximation as the solution proceeds. Error is due to the fact that numerical representation is an approximation and, in a practical sense, a finite number of increments must be used, and in any arithmetic computation roundoff error is present. If the errors do not increase in magnitude as the solution proceeds, the method is said to be stable. Jenson and Jeffreys, (48) in discussing stability and convergence criteria recommend the Crank-Nicholson methods for solution of initial value equations such as the
They state the Crank-Nicholson method for this type of equation is stable for all values of $M$, and recommend it for initial value problems with a modulus $M < 1$. R. E. Carlile and B. E. Gillett, (14) in discussing numerical methods for solution of both initial value and boundary value (hyperbolic equations) problems, recommend implicit methods such as Crank-Nicholson because of their stability and improved accuracy over explicit methods.

Choi, Fan, and Hsu (16), in developing a simulation model for a diabatic absorption, used the Crank-Nicolson implicit finite difference method to solve their system of parabolic partial differential equations. They had chosen this method after reviewing the collocation method and the method of characteristics. Specifically, Choi, Fan, and Hsu cite the less stringent stability and convergence restrictions offered by the Crank-Nicolson method than the explicit methods, and less computing time than the method of characteristics, and particular applicability of the method to solution of parabolic partial differential equations.

In a study of fixed bed adsorption modeling with non-linear equilibria, Brown, Mullins and Melsheimer (10) investigated solution of equations identical to those describing the fluid phase in this research. In their investigation the equation was:

$$\frac{\partial T}{\partial t} = \alpha \frac{\partial^2 T}{\partial x^2}$$

for which a finite difference equation modulus $M = \alpha \frac{\Delta t}{(\Delta x)^2}$.
deleted, the equation is of hyperbolic form, and they cite commentary by Von Rosenberg (137) on the nonapplicability of the Crank-Nicholson method to solution of hyperbolic forms of partial differential equations. Brown and his coworkers state that when the dispersive term is small relative to the convection term, \( v \frac{\partial c}{\partial z} \), the equation takes on many of the characteristics of hyperbolic systems, and difficulty may be encountered in applying the Crank-Nicholson method. McCracken and his coworkers (65) in one paper; and Meyer and Weber (69) in another study, indicate that in these cases, rather stringent restrictions on size of grid increments must be observed to avoid instability and to obtain convergence with the exact solution. Brown and his coworkers compared results of simulation of adsorption experiments they performed using the Crank-Nicholson methods with various \( \Delta t \) and \( \Delta z \) increments, to results obtained by using the State Variable method of McCrackin et al (65). In the State Variable method, \( X = \frac{\partial c}{\partial z} \), and equation (7-61) becomes:

\[
0 \quad D \frac{\partial^2 c}{\partial z^2} - v \frac{\partial c}{\partial z} = \frac{\partial c}{\partial t} + k^*(c-c^*)
\]

(7-61)

They noted that second order parabolic equations of this type are commonly solved by the Crank-Nicholson method. However, as Svedberg noted (125), if the \( D \frac{\partial^2 c}{\partial z^2} \) dispersive term is deleted, the equation is of hyperbolic form, and they cite thereby reducing a second order equation to a pair of first order equations. The result of the study by Brown and his
coworkers indicated that at the same $\Delta z$ and $\Delta t$, if the increments were sufficiently small, both models converged to the same numerical solution. However, as the grid increments were "coarsened," the Crank-Nicholson method did not converge and the ECH curves smeared" out around the experimental curves in a lazy "S", whereas the State Variable method continued to converge and simulate the experimental points. Obtaining convergent solutions with a coarser grid spacing using the State Variable method used 67.5% less computation time than that required by the Crank-Nicholson method when utilizing "finer" convergent grid increments.

Future work on ternary ion exchange modeling should examine the State Variable method as an alternate solution method--mainly to shorten computation time, since in all of the simulations performed in this research the grid spacing and time intervals were decreased until the calculated values were essentially identical, as suggested in Sherwood, Mickley & Reed (71). Examples of this can be seen in the ECH curve for $\text{Ag}^+$ exchange for $\text{H}^+$ at 1.5 normal solution concentration, Figure 7-10, where the number of column and resin increments were systematically increased. As the number of column nodes were increased from 101 to 201, no further change in ECH was observed and, indeed, this curve closely fits the experimental data utilizing the resin phase diffusion coefficients fitted in runs on other $\text{Ag}^+\text{-Na}^+\text{-H}^+$ systems. The effect on the general shape of the ECH curves in the ternary run P-15, is
shown in Figure 7-13, as the column nodes are increased from 51 to 101 and the number of time steps is increased from 800 to 2100. The initial simulated ECH curve showing appearance of H\(^+\) and disappearance of Na\(^+\) versus the experimental data is definitely improved by finer grid spacings, although the appearance of Ag\(^+\) is 250 sec late. However this difference represents less than a 1% effect on the total material balance for the run.

5. Activity Effects - Resin Phase

A few column runs were made using activity driving force to simulate the ECH curves of Erickson (26) and Omatete (78) experiments. As discussed in Chapter V on Diffusion in the Resin Phase, using activity driving force vs. concentration driving force is more elegant because a single value of an ionic specie's diffusion coefficient can be used in describing interchange with many other ionic species. This precludes the need for multi-parameter or multi-value methods such as the Stefan-Maxwell model or ion-species "pair corrector" coefficients. It was observed in making these runs that use of activity driving force did not alter the shape of the ECH curves appreciably in column operation unless the system was highly non-ideal, and even then the effect was not dramatic. The "batch" type of experiment is clearly best able to discriminate between activity and concentration models. The starting points for ECH fitting in this research were literature values for resin phase diffusion coefficients, when
available, which had been derived by the various investigators using concentration driving force analysis. It was decided, therefore, to perform all column simulations in this research project using concentration driving force—and to test a few extreme cases for effect of activity driving force.

It is interesting to analyze how the activity correction would alter the ECH curve. As discussed, the activity correction for effective diffusion coefficient, $D_i$ involves the addition of the following term to the equation for calculating the effective diffusion coefficient for an ideal binary pair (equation 5-13):

\[
\frac{D_1 \cdot D_2}{2} \cdot \frac{\partial \ln \left( \frac{|z_2|}{\gamma_1 z_1} \right)}{\partial x_1} \frac{\partial \ln \left( \frac{|z_1|}{\gamma_2} \right)}{\partial x_1} \text{molar}
\]

(7-63)

where $\gamma_1$ and $\gamma_2$ have the molar reference state

For a positive deviation system, as $x_1$ increases from zero concentration, the following term is negative:

\[
\frac{\partial \ln \left( \frac{|z_2|}{\gamma_1 z_1} \right)}{\partial x_1} \text{molar}
\]

and the effective diffusion coefficient is decreased. Therefore as $x_{1\text{bulk}}$ displaces $x_{2\text{bulk}}$, the ECH curve will show appearance of $x_{1\text{bulk}}$ earlier than the concentration model since it cannot penetrate the resin as quickly, and this was observed with the
positive deviation system NH$_4^+$-ETDA$^{++}$, and is shown in Figure 7-3. With negative deviation systems such as Ag$^+$-Na$^+$ the reverse is true, and the effective diffusivity is increased, leading to a delayed appearance of Ag$^+$ in the ECH compared with the concentration model. This was observed, as is shown in Figure 7-9.

One other way of examining this effect is to look at a limiting case where $\bar{x}_1$ is departing from zero concentration to higher values. The equivalent fluxes can be shown as:

$$\bar{J}_1 = -D_{\text{eff}} \frac{\partial \bar{a}_1}{\partial \rho} = -\bar{J}_2 = D_{\text{eff}} \frac{\partial \bar{a}_2}{\partial \rho}$$  (7-64)

However, as $\bar{x}_2$ approaches 1.0, the activity, $\bar{a}_2$, approaches $\bar{x}_2$, and the following relationship obtains:

$$\bar{J}_1 = -D_{\text{eff}} \frac{\partial (\bar{Y}_1 \bar{x}_1)}{\partial \rho} = -\bar{J}_2 = D_{\text{eff}} \frac{\partial \bar{x}_2}{\partial \rho} = -D_{\text{eff}} \frac{\partial \bar{x}_1}{\partial \rho}$$  (7-65)

Electroneutrality in the resin requires that $\frac{\partial \bar{a}_2}{\partial \rho} = -\frac{\partial \bar{x}_1}{\partial \rho}$ so that the following relationship results from expansion of equation (7-65):

$$\bar{J}_1 = -D_{\text{eff}} \left\{ \frac{\partial \bar{Y}_1 \bar{x}_1}{\partial \rho} \frac{\partial \bar{x}_1}{\partial \rho} + \bar{Y}_1 \frac{\partial \bar{x}_1}{\partial \rho} \right\} = -D_{\text{eff}} \frac{\partial \bar{x}_1}{\partial \rho}$$  (7-66)

The following equality can be derived from equation (7-66):

$$-D_{\text{eff}} \left\{ \frac{\partial \bar{Y}_1 \bar{x}_1}{\partial \rho} + \bar{Y}_1 \right\} \frac{\partial \bar{x}_1}{\partial \rho} = -D_{\text{eff}} \frac{\partial \bar{x}_1}{\partial \rho}$$  (7-67)

as $\bar{x}_2 \rightarrow 1.0$
The relationship defining the effective diffusion coefficient as \( x_2 \) approaches 1, or as \( x_1 \) is departing from zero concentration is:

\[
D_{\text{eff}}' = D_{\text{eff}} \left( \frac{\partial \gamma_1}{\partial x_1} \cdot x_1 + \gamma_1 \right) \tag{7-68}
\]

If the system is ideal, \( D_{\text{eff}}' \) is equal to \( D_{\text{eff}} \) since \( \frac{\partial \gamma_1}{\partial x_1} \) is equal to zero and \( \gamma_1 \) has a value of one. If the system has a positive deviation, then the effective diffusion coefficient, \( D_{\text{eff}}' \) used in conjunction with the concentration gradient \( \frac{\partial \gamma_1}{\partial \rho} \), is smaller than the effective diffusion coefficient \( D_{\text{eff}} \) defined for use in ideal systems. This is derived from equation (7-68) where as \( x_1 \) approaches zero, the term \( \frac{\partial \gamma_1}{\partial x_1} \cdot x_1 \) approaches zero, and the term \( \gamma_1 \) is greater than one and increasing. Exactly the opposite occurs with negative deviation systems. In these systems, the term \( D_{\text{eff}}' \) is greater than the ideal effective diffusion coefficient, \( D_{\text{eff}} \), since the expression \( \frac{\partial \gamma_1}{\partial x_1} \cdot x_1 \) is approaching zero and the activity coefficient, \( \gamma_1 \), is less than one and decreasing.


The discussion above led to development of a model which utilizes the implicit Crank-Nicholson method to approximate the coupled equations describing the operation of a fixed bed binary ion exchange process. Chapter V on Diffusion in the
Resin Phase dealt with development of the Nernst-Planck (N.P.) diffusion model including use of activity driving force, and Chapter IV on Diffusion Through the Film treated the development of a pseudo electric field model which provided film coefficients based on N.P. type interdiffusivity relationships between migrating ionic species. This chapter has provided the development of the features of coupling the resin phase equations to those describing the fluid phase, with quasilinearization of non-linear terms using the quasilinearization technique developed by Lee (59). The reduction of the ternary general case to the simpler binary case illustrated the use of the Thomas algorithm (13, 125) to invert tridiagonal matrices, and showed how the two matrix equations were coupled at the resin-fluid interface. An iterative process at each time step resolved non-linear terms. The treatment of boundary conditions was discussed, as well as the initialization procedure. The details of other features built into program RFMAVD are given in Appendix K, Features of RFMAVD.

D. Binary Simulation of Column Operation

1. General

Nine binary simulations were performed using experimental effluent concentration histories (ECH) from the work of K. L. Erickson (26) on NH$_4$-Ethylene Diamine$^{++}$ETDA$^{++}$ and O. O. Omatete (78) on the systems comprised by Ag$^+$-Na$^+$-H$. The equilibrium constants and
Redlich-Kister activity coefficient parameters for these systems were correlated with RIONFT, and this is covered in Chapter II on equilibrium. In general, the binary model was able to provide a close match to the experimental ECH curves by making only modest adjustments to previously observed resin phase ionic diffusion coefficients. The modeling effort was based on (a) previously fitted equilibrium parameters, (b) the pseudo electric field film coefficients based on the Carberry (11) and the Koloini et al (54, 55) film coefficient correlations for packed beds, (c) electrolyte phase activity coefficients based on the Bromley equation (9), and (d) the use of generalized correlations for axial diffusion coefficients (Fig. 6-1). Normally, 10 increments or 11 nodes were used for resin grid spacing, and 100 increments or 101 nodes were used for column grid spacing--although the program is dimensioned for a 21 x 301 grid spacing. Maximum dimensionless time is defined as the number of empty column displacements (based on superficial velocity) to completion of the run. Normally, this maximum dimensionless time was divided into 100 to 500 increments to establish \( j_{\text{max}} \), or the total number of time steps increments used in the computation. The number of increments used was based on the shape of the ECH curve, and for a very steep curve, such as that of \( \text{Ag}^+-\text{H}^+ \) in favorable exchange, more time increments were required to provide adequate solution stability. This was tested by
trying a series of different time increments until no change was observed (figure 7-10).

2. Research of K. L. Erickson (26) System NH$_4^+$ - ETDA$^{2+}$/Cl$^-$ on DOWEX 50Wx8 Resin

2.1 Discussion - Erickson Experiments

Erickson conducted two batch experiments by very quickly dropping a 1 normal solution of ETDA (500 cc) into a well stirred flask of resin (33 cc) suspended in 1 normal NH$_4^+$ solution (20 cc), then quickly withdrawing samples at timed intervals. Total solution volume due to sampling dropped to 70% of the initial volume, and the total resin capacity in equivalents was about 25% of the total initial solution equivalents. Erickson set up material balance equations, including the following equation for the resin, to fit the $\bar{D}_{NH_4^+}$ and $\bar{D}_{ETDA^{2+}}$ values which best represented the data:

$$
\bar{W}_1(t) = \frac{3 \cdot Q \cdot Vol}{r_0} \int_0^{r_0} r^2 \bar{x}_1(r,t) \cdot dr
$$

(7-69)

where:

$\bar{W}_1(t)$ = total equivalents of species 1 in the resin utilized in the batch experiment at time=t

Q = equivalents of active sites in resin/cc resin

Vol = total volume of resin utilized in batch experiments
This equation was integrated by utilizing Simpson's Rule.

In order to define $\bar{x}_1(r,t)$ in the above equation, the following expression was solved:

$$\frac{\partial \bar{x}_1(r,t)}{\partial t} = \frac{1}{r^2} \left\{ \frac{\partial (r^2 J_1)}{\partial r} \right\}$$  \hspace{1cm} (7-70)

where $J_1$ is defined by the Nernst-Planck model described in Chapter V, equations 5-24 and 5-13.

Erickson used the Crank-Nicholson method to solve for the resin phase concentration profile, then used that profile in conjunction with equation (7-69). The batch material balance equation was given as:

$$x_{1, \text{bulk}}(t+\Delta t) = \frac{V(t)}{V(t+\Delta t)} \cdot x_{1, \text{bulk}}(t) - \frac{1}{N_t \cdot V(t+\Delta t)} \left\{ \bar{W}_1(t+\Delta t) - \bar{W}_1(t) \right\}$$ \hspace{1cm} (7-71)

where $V(t) = \text{total solution volume at time } t$

$N_t = \text{normality of solution}$

It was assumed because of high agitation speed that the film resistance was low, so the B.C. utilized was:

$$\bar{x}_1(t) = g(\bar{x}_1(1,t))$$  \hspace{1cm} (7-72)

relating equilibrium between the bulk solution and the outer surface of the resin.

Erickson felt the data from one experiment was somewhat more reliable than the other, and the following pairs of coefficients were found to fit the 6 data
points with excellent accuracy for the experimental run:

<table>
<thead>
<tr>
<th>$D'_{ETDA^{++}}$</th>
<th>$D'_{NH_4^+}$</th>
<th>Sample variance $-s^2$</th>
</tr>
</thead>
<tbody>
<tr>
<td>cm$^2$/sec</td>
<td>cm$^2$/sec</td>
<td></td>
</tr>
<tr>
<td>1.2 x 10^{-6}</td>
<td>4.0 x 10^{-6}</td>
<td>0.00016</td>
</tr>
<tr>
<td>1.3 x 10^{-6}</td>
<td>3.9 x 10^{-6}</td>
<td>0.00018</td>
</tr>
<tr>
<td>2.0 x 10^{-6}</td>
<td>2.9 x 10^{-6}</td>
<td>0.00023</td>
</tr>
<tr>
<td>1.8 x 10^{-6}</td>
<td>2.9 x 10^{-6}</td>
<td>0.00045</td>
</tr>
</tbody>
</table>

Erickson, in simulating the ECH of his column experiments found little difference in goodness of fit using diffusion coefficient ratios, $D'_{ETDA^{++}}/D'_{NH_4^+}$ of 1.2/4.0 versus 2.0/2.9.

In order to define the equilibrium relationship between the resin and the electrolyte, Erickson derived the following empirical expression for the 1 normal system (he did not attempt to fit the 0.1 normal data):

for $0.012 \leq x^*_{eq} \leq 1.0$

$$x^*_{eq} = x^*_{eq} \times 0.33 - (1-x^*_{eq}) \times 0.1 \times 0.17 \times (1-x^*_{eq}) \times 0.17$$

$$- 0.034 (x^*_{eq} - x^*_{eq}) + 0.03 (1-x^*_{eq}) (0.25) (|\sin(2\pi x^*_{eq})|)$$

for $0.0 \leq x^*_{eq} \leq 0.012$

$$x^*_{eq} = 8.333 x^*_{eq}$$

where:

$x^*_{eq}$ = ETDA$^{++}$ concentration in resin in equilibrium with electrolyte in equiv. fraction

$x^*_{eq}$ = ETDA$^{++}$ concentration in electrolyte in equilibrium with resin, equiv. fraction
Whereas the expression represented the experimental data extremely well, the relationship has no theoretical merit, and cannot be extended to other concentrations such as the 0.1 normal data for same system. Other parameters measured by Erickson included the axial dispersion effects, and these experimental results were discussed in Chapter VI. Additionally, Erickson found the uptake of coion Cl⁻ by the resin to be only 0.023 mequiv/mequiv of total capacity. This indicated that the Donnan potential was strong enough to minimize coion intrusion for this system. Resin volume change in the transition from the NH₄⁺ form to the ETDA ++ form was minus 8%. The average bed void volume, ε, was measured as 0.352.

The column effluent histories (ECH) for six runs were included in Erickson's report. These were:

Runs 13 and 15 for favorable exchange of ETDA++ for NH₄⁺ at 0.94 cm/sec superficial velocity

Runs 17 and 18 for favorable exchange of ETDA++ of NH₄⁺ at 1.85 cm/sec superficial velocity

Runs 14 and 16 for unfavorable exchange of NH₄⁺ for ETDA++ at 1.85 cm/sec superficial velocity

The ECH data points for experiments 17-18 (1.85 cm/sec, favorable) were so close at each time interval so that they
are represented together in Figure 7-2 of this report as averaged data points. The ECH data points for experiments 13 and 15 were sufficiently different that they are both plotted in Figure 7-3. The two unfavorable exchanges, experiments 14 and 16, had somewhat different velocities, 1.75 cm/sec and 1.85 cm/sec and could not be combined. Consequently only experiment 16 was used in this current work, primarily because it had Reynolds no. criteria (41.0) identical to favorable exchange experiments 13 and 15, and could be compared directly with those runs.

2.2 Erickson Simulation Model Development & Simulation Results

Erickson's Column Model was premised on the equations for the resin phase given above. The diffusion across the stagnant layer from the bulk fluid to the resin was modeled by using the conventional equation:

$$\overline{J}_i^* = k_f^* (x_i^* - x_i^{\text{int}})$$  \hspace{1cm} (7-75)

where

$$k_f = \frac{D^*_{\text{eff}}}{\delta}$$  \hspace{1cm} (7-76)

$D^*_{\text{eff}}$ = average diffusion coefficient of two ionic species at infinite dilution

$\delta$ = "film thickness"

The film thickness, $\delta$, was a parameter modeled by Erickson to fit the initial portion of the ECH curve. He found $\delta = 0.0065$ cm for $Re = 41$, and $\delta = 0.0035$ cm for $Re = 20$, gave the best fit to his data.

The Column equations utilized were:
However, the finite difference method employed was explicit in character, since the partials were expressed as forward differences in both the time and column dimension domain. The method by which the difference equations were solved was to guess forward values at $t+\Delta t$ and $x+\Delta x$, then use those values to resolve the equations at length step $x+\Delta x$ and time step $t+\Delta t$ until a reasonable tolerance criterion was met. The process then proceeded to the next time increment.

Quasilinearization of non-linear terms was not performed, and as discussed earlier, explicit methods tend to instability if the moduli are not carefully chosen. Indeed, in order to fit the experimental data, Erickson had to include $\Delta t$, the finite difference time increment, as one of the "variables." Higher or lower values of time increment would not fit the experimental ECH curves. With this stipulation, and the film "thickness" values previously cited, Erickson obtained reasonable fits to the experimental data.

Erickson's experimental work was carefully performed, and the batch experimental technique he utilized to measure resin phase diffusion coefficients is an alternative to radioactive tracer experiments. The latter tend to be more costly in terms of equipment and materials and require higher experimental skill levels. The experimental work on axial diffusion gave rise to values an order of magnitude higher...
than general correlations (figure 6-1) would predict for Re numbers between 20-40. One observation by the researcher on this current work is that within the time constraints of a doctoral research program for a process of this kind, it is difficult to be thorough on both the experimental side as well as the modeling component of the work.

2.3 Simulation of Erickson Experiments - This Research Project

The results of simulation of Erickson's column experiments are shown in Figures 7-2, 7-3, & 7-4. Two favorable exchange runs in which ETDA\textsuperscript{++} displaced NH\textsubscript{4}\textsuperscript{+} at two flow rates--1.85 cm/sec and 0.94 cm/sec, and one unfavorable exchange in which NH\textsubscript{4}\textsuperscript{+} displaced ETDA\textsuperscript{++} at a velocity of 1.85 cm/sec, have been simulated and the results closely match the experimental data. The parameters used in the simulation were (1) equilibrium constant and Redlich-Kister parameters for resin phase activity coefficients correlated with RIONFT, plus the Bromley equation for electrolyte species activity coefficients to predict equilibrium between the resin surface and the fluid film, (2) film coefficients based on the Carberry correlation--using the pseudo electric field model with electrolyte species diffusivity corrected for concentration, (3) axial diffusion coefficients obtained by simulating Erickson's axial dispersion experiments, and (4) resin phase diffusion coefficient values of:

\[
\begin{align*}
D_{\text{EDTA}^{++}}' &= 2.0 \cdot 10^{-6} \text{ cm}^2/\text{sec} \\
D_{\text{NH}_4^+}' &= 2.9 \cdot 10^{-6} \text{ cm}^2/\text{sec}
\end{align*}
\]
Ethylene Diamine Exchange with NH$_3$ on Dowex 50W x 8 - Cl$^-$
Binary Model - Nernst Planck Relationship
For Resin Diffusivities
- Favorable Equilibrium -

Equilibrium Data
$K_{ETDA^{++}NH_4^+} = 16.1206$
Redlich - Kister Constants
$B = 0.34626$
$C = 0.02223$
Bromley Constants
$B_{ETDACL_2} = 0.09$
$B_{NH_4Cl} = 0.02$

Experiments 17 & 18\(^*\)
1. Data at 28°C
   - $v = 0.94$ cm/sec (superficial)
   - $L = 53.5$ cm (effective bed length)
   - Normality = 1.019 (pH = 3.5)
   - Equivalents in Resin = 3.0 eq./liter
   - Void Fraction = 0.352 ($\bar{\epsilon}$)

   Calculated N.P. Model
   - $D_{ETDA^{++}} = 2.0 \times 10^{-6}$ cm$^2$/sec
   - $D_{NH_4^+} = 2.9 \times 10^{-6}$ cm$^2$/sec
   - $E_D = 4.0$ cm$^2$/sec (experimental fit)
   - $K_f$ varies with soln. conc.$^{(1)}$
     - $0.004 \rightarrow 0.009$ cm/sec
     - Re = 20.5 (Based on $V/\bar{\epsilon}$)

\(^{(1)}\) J.J. Carberry, Re > 1.0 $K_f = 1.15 \text{Re}^{1/2} \text{Sc}^{1/3} D_{eff}/dp$

Ethylene Diamine Exchange with NH$_3$ on Dowex 50W x 8 - Cl$^-$

Binary Model - Nernst Planck Relationship
For Resin Diffusivities
- Favorable Equilibrium -

Figure 7-3

Equilibrium Data
$K_{ETDA}^{++}NH_4^+ = 16.1206$

Redlich-Kister Constants
$B = 0.34626$
$C = -0.02223$

Bromley Constants
$B_{ETDACL_2} = 0.09$
$B_{NH_4Cl} = 0.02$

Experiments 13 & 15$^*$

Data at 28°C
$V = 1.85$ cm/sec (superficial)
$L = 52.7$ cm (effective bed length)
Normality = 1.025 (pH = 3.5)
Equivalents in Resin = 3.08 eq/liter
Void Fraction = 0.352 ($\epsilon$)

--- = Calculated N.P. Model

$D_{ETDA}^{++} = 2.0 \times 10^{-6}$ cm$^2$/sec
$D_{NH_4^+} = 2.9 \times 10^{-6}$ cm$^2$/sec

$E_D = 4.0$ cm$^2$/sec (experimental fit)

$K_i$ varies with soln. conc.$^{(1)}$

$0.006 - 0.013$ cm/sec

Re$ = 41.0$ ($V/\epsilon$ as velocity)

$^{(1)}$ J.J. Carberry, Re$ > 1.0$  $K_i = 1.15 \ Re^{1/2}Sc^{1/3} \ D_{eff}/dp$

Ethylene Diamine Exchange with NH$_3$ on Dowex 50W x 8 - Cl$^-$

Binary Model - Nernst Planck Relationship
For Resin Diffusivities
- Unfavorable Equilibrium -

Figure 7-4

Equilibrium Data

$K_{ETDA}^{+} + NH_4^+ = 16.1206$

Redlich Kister Constants

$B = 0.34626$
$C = -0.02223$

Bromley Constants

$B_{ETDACL_2} \approx 0.09$
$B_{NH_4Cl} = 0.02$

Experiment 16*

Data at 28°C
$v = 1.85$ cm/sec (superficial)
$L = 51.3$ cm (effective bed depth)
Normality = 1.022 (pH = 3.5)
Equivalents in Resin = 3.55 eq./liter
Void Fraction = 0.352 ($\epsilon$)

Calculated N.P. Model

$D_{ETDA}^{+} + = 2.0 \times 10^{-6}$ cm$^2$/sec
$D_{NH_4}^+ = 2.9 \times 10^{-6}$ cm$^2$/sec
$E_D = 4.0$ cm$^2$/sec (experimental fit)
$K_t$ varies with soln. conc.$^{(1)}$

0.006 - 0.014 cm/sec
$Re = 41.0$ (based on $V/\epsilon$)

$\text{Elapsed Time - Seconds}$

$X_B - \text{Equiv. Fraction NH}_3 \text{ Exiting Column}$

$^{(1)}$ J.J. Carberry, $Re > 1.0$ $K_t = 1.15 Re^{1/2}Sc^{1/3}D_{eff/dp}$

Other diffusion coefficients were tried in trial and error parameter searches but they did not give as satisfactory a total fit to the effluent concentration histories (ECH) of the three experimental runs. Admittedly, this was an "eyeball" judgment; however the diffusion pair selected was one of the pairs which Erickson had fitted in his batch experiments.

In fitting the predicted ECH curve to the experimental curve, the experimentally determined resin capacity of 3.24 milliequiv./cc of resin had to be adjusted downward by 5 to 8% to "center" the plug flow time for the favorable exchange runs, and upward by 9% for the unfavorable exchange run. This "centering" made the area under the ECH curve before plug flow time equal to that above the ECH curve after plug flow time, thereby insuring reconciliation of the material balance. Of all of the quantities measured in ion exchange experiments, the least reliable is resin capacity/volume of resin "in use"--since the capacity will be somewhat different for each of the species involved. The resin swells to different degrees depending on the various ion species concentrations in the resin, and the resin volume changes during the course of a run. Flow rate, time, and electrolyte phase ionic species concentration measurements are normally quite accurate. It is not unreasonable,
therefore, to adjust the resin capacity within reasonable limits from that measured to insure reconciliation of the material balance. This was done in the other systems simulated as well.

The Carberry film coefficient correlation for packed beds has the form,

\[ k_L = b \cdot \frac{Re^{1/2} \cdot Sc^{1/2} \cdot D^*_{\text{eff}}}{\varepsilon^{1/2} \cdot dp} \text{ for } Re > 1 \] (7-78)

so the film coefficient values should differ by about 40% if the Re numbers differ by a factor of two. Reynolds' numbers differed by a factor of two in Erickson's favorable exchange experiments, run at velocities of 0.94 and 1.85 cm/sec. If the value of "b" were chosen to obtain a good simulation of the experimental ECH at one velocity, the fit was poor at the other velocity. Two values of "b", different from Carberry's recommended value of 1.15, were used in early simulations to obtain satisfactory matching of the experimental results at the two velocities. Use of the pseudo electric field model brought the simulated ECH curves for both velocities into remarkable agreement with the experimental data, using the Carberry film correlation \((b=1.15)\) to estimate "boundary layer effective film thickness" as a function of the average concentration of ionic species in the film. At the lower velocity, gradients are steeper, and the effect is to increase the average concentration of the slower ion,
ETDA$^{++}$, in the film—thereby reducing flux. The "apparent" film coefficient is lower than would be predicted by $Re^{1/2}$ adjustment alone, since now the effective diffusion coefficient is also lower.

In the unfavorable exchange, the initial appearance and early steep slope of the ECH was "insensitive" to all parametric variations, with the initial appearance time being equal to the column length divided by the interstitial velocity, or 9 seconds, i.e., the inlet concentration of the least favored species runs at convection velocity directly to the column bottom without interchange with the resin. The fit from two "column displacement times" (18 sec) onward was good using the same parameters used to fit the experimental data for the two favorable exchange systems.

The simulation model allows fractional adjustment (FLMADJ) of the constant "b" in either the Carberry (11) or the Koloini et al (54, 55) correlations to test the effect on simulated ECH profile (Appendix K). As discussed in Chapter V, Diffusion in the Resin Phase, the Rao and coworkers' binary batch run results (92, 93) were regressed in one set with FLMADJ as a parameter, yielding a value of 0.86 ±0.04. The parametric study of FLMADJ was made in all of the nine sets of binary ECH curves fit (Erickson and Omatete), using values used of 0.7, 1.0 and 1.2. Overall, the most consistent results were obtained when FLMADJ was unity, as portrayed in the original correlations.
The Erickson equilibrium data were regressed with the Redlich-Kister activity coefficient correlation using both mole fraction and equivalent fraction concentration units, with equally good results (discussed in Chapter II on equilibrium). Experiments 13 and 15 were simulated with RFMAVD utilizing equilibrium constants and activity coefficient parameters both derived for mole fraction and equivalent fraction concentration units, and the results were identical to within ±0.5%.

3. Research of O. O. Omatete - Ag⁺-Na⁺-H⁺ (78)

3.1 Discussion - Omatete Experiments

Omatete carried out favorable exchange experiments for the system Ag⁺,Na⁺,H⁺/NO₃⁻ on Dowex 50W x8 resin having an average particle diameter of 0.71 mm. The columns used were 1" i.d. glass with lengths ranging from one to four feet. Ag⁺ and H⁺ concentrations were determined by titration to ± 0.2% accuracy and Na⁺ concentration was determined by material balance, with check by flame spectroscopy (±7%). Flow rates varied from 3 to 60 ml/min, and equal effluent volumes were collected with an automatic fraction collector. Feed rate and column length were selected to insure that the concentration profiles in the favorable exchange runs had reached a stable configuration, or constant pattern breakthrough ECHs. A total of eighteen runs were made to determine the binary equilibrium relationships for the three binary pairs at 0.1 normal
concentration, with ten runs out of the eighteen for the Na\textsuperscript{+}-H\textsuperscript{+} system alone. Six additional points were measured at 1.5 normal concentrations, one each for the Ag\textsuperscript{+}-Na\textsuperscript{+} and Ag\textsuperscript{+}-H\textsuperscript{+} systems, and four for the Na\textsuperscript{+}-H\textsuperscript{+} system. Only two ternary equilibrium points were measured. In order to simulate the three ternary column experiments, the closing of a satisfactory material balance in this research effort resulted in the addition of three additional ternary equilibrium points which were consistent with the equilibrium parameters correlated from the experimental data. Needless to say, this represented far too few data points since, as will be shown, the equilibrium relationships in "azeotrope" type (selectivity reversal) systems, such as that of Na\textsuperscript{+}-H\textsuperscript{+}, largely determine the shape of the ECH curves. Indeed, additional literature data had to be utilized in the Na\textsuperscript{+}-H\textsuperscript{+} system to better define the "azeotrope" cross over point versus concentration, since at 0.1 normal the upper region "pinches" in a "pseudo azeotrope" manner and at 1.5 normal the system "azeotropes." In addition, the lower part of the "x-y" curve for the Na\textsuperscript{+}-H\textsuperscript{+} system tends to a lower "pinch" region.

Experimental rate runs were made for favorable exchange of each binary system at normalities of 0.05 and 1.5 to demonstrate film diffusion control at the low
normality, and resin phase diffusion control at the higher concentration. Actually, this was done since Omatete's model was a single resistance model (it could simulate film or resin phase transport resistance, but not both). In addition, two favorable ternary runs were made, one at each normality (0.05 and 1.5), and one unfavorable ternary run was made at 1.5 normal solution concentration. In a paper presenting Omatete's work by Omatete, Clazie and Vermeulen (79, 80), only one ternary (0.05 normal) was presented—stating that the "other" rate run did not have fully developed concentration plateaus. Omatete's dissertation work performed six years earlier contained three ternary system runs as described above. All three were simulated in this research project with good results by TERNEX program, utilizing equilibrium parameters correlated from the experimental data with RIONFT for the binary and ternary systems.

3.2 Omatete Simulation Model Development

The Omatete model of fixed bed ion exchange involving binary and ternary systems had the following features which, in effect, represent considerable simplification and "bending" of theory.

a. Equilibrium

(i) Binary equilibrium was based on a separation factor which is similar to the K value in distillation,
As with non-ideal vapor-liquid system equilibrium, \( \alpha_{ij} \) (equivalent to \( K_{ij} \)) values are not constant over the composition range.

(ii) Ternary equilibrium was represented by the following general multicomponent relationship:

\[
\alpha_{ij} = \frac{1}{\alpha_{ji}} = \frac{\bar{x}_i}{\bar{x}_j} = \frac{x_i^{\text{int}}}{x_j^{\text{int}}} ,
\]

where bar indicates the resin phase, 'int' the resin-fluid interface, and the star depicts an equilibrium condition.

\[
\frac{x_j^*}{x_j^{\text{int}}} = \frac{\alpha_{\text{ref},j} x_j}{\sum_{k=1}^{n} \alpha_{\text{ref},k} x_k}
\]  \( \text{for } j = 1, 2, \ldots, n \)  \( (7-80) \)

b. Dynamics - Binary and Ternary Systems

(i) Diffusion between the bulk fluid and the resin was based on a "film" in one phase or the other.

\[
\frac{\partial \bar{x}_i}{\partial t} = \sum_{j=1}^{n} k_{L_{ij}} (x_j^{*} - x_j^{\text{bulk ave}})
\]

\[
\text{where}
\]

\[
k_{L_{ij}} = \frac{\alpha_{D} D_{ij}^*}{\delta_{F}} ; \delta_{F} \text{ is film 'thickness'}
\]  \( \text{(7-82)} \)
o resin phase control

assumes an "effective constant solid phase film", $\bar{\delta}_F$, in the resin phase:

$$\left(\frac{\partial \bar{x}_{ij}^{\text{av}}}{\partial t}\right)_v = \sum_{j=1}^{n} k_{ij}^{*} (\bar{x}_{j}^{\text{av}} - \bar{x}_{j}^{*}) \quad (7-83)$$

where $i = 1, 2, \ldots, n$

and:

$$k_{ij}^{*} = \frac{a_p}{(1-\varepsilon) \cdot \bar{\delta}_F} \cdot D_{ij} \quad (7-84)$$

(ii) Film Coefficient Representation - Fick's

Law Solution Phase

o Simple Fick's Law

$$k_{ij}^{*} = \frac{k_{ij}^{L}}{1 - \varepsilon} \left\{ \left( x_{j}^{\text{bulk}} + x_{i}^{\text{bulk}} \right) \cdot k_{Lij}^{L} + x_{i}^{\text{bulk}} \cdot k_{Lij}^{L} \right\}$$

$$k_{ij}^{L} = \frac{x_{i}^{\text{bulk}} \cdot k_{Lij}^{L} + x_{j}^{\text{bulk}} \cdot k_{Lij}^{L} + x_{k}^{\text{bulk}} \cdot k_{Lij}^{L}}{x_{i}^{\text{bulk}} \cdot k_{Lij}^{L} + x_{j}^{\text{bulk}} \cdot k_{Lij}^{L} + x_{k}^{\text{bulk}} \cdot k_{Lij}^{L}}$$

where $k_{Lij}^{*}$'s (unprimed) are binary Fick's Law rate coefficients.

o An improved Fick's Law was also evaluated in which:

$$k_{ij}^{L} = (b_{0})_{ij} + (b_{1})_{ij} \cdot x_{i}^{\text{bulk}} + (b_{2})_{ij} \cdot x_{i}^{2}$$

i.e., the Fick's Law rate coefficients are fit to a polynomial expression.
(iii) Film Coefficient Representation - Fick's Law

Resin Phase

- Fick's Law applied to the resin phase "film"

The equations describing diffusion in the resin phase "film" are similar to (7-85) and (7-86), except that $\bar{k}_{ij}$ is a function of:

$$
\bar{x}_{i,ave}, \bar{x}_{j,ave}, \bar{x}_{k,ave} \text{ and the individual } \bar{k}_{L_{ij}}'s.
$$

(iv) Film Coefficient Representation - Nernst-Planck, Solution Phase

- Simple Nernst Planck representation

$$
k_{L_{ij}}' = k_1\left(1 + \frac{z_i x_{i, bulk} (k_j' - k_i')}{\sum_{i=1}^{n} z_i x_{i, bulk} k_i'}\right)
$$

(7-88)

for $i=1,2,\ldots,n$ and $j=1,2,\ldots,n$

- Improved Nernst-Planck representation

$$
k_i' = \sum_{j=1}^{n} x_{j, bulk} (k_j')_j
$$

(7-89)

for $i=1,2,\ldots,n$

$(k_i')_j$ is the rate coefficient for species $i$ in "pure" component $j$
(v) Film Coefficient Representation - Nernst Planck,

Resin Phase

\( k_{ij} \) has the same functional representation as (7-88) and (7-89) but is a function of the variables.

\[
\bar{x}_{i_{ave}}, \bar{x}_{j_{ave}}, \bar{x}_{k_{ave}}
\]

plus the individual \( k_i \)'s

c. Material Balance equations

Using a dimensionless throughput parameter, \( \bar{T} \)

where:

\[
\bar{T} = T \cdot v - \bar{T} \cdot v
\]

(7-91)

and \( \bar{T} \) is the reciprocal of the dimensionless velocity at which the center of the "S" curve moves, the following can be shown:

\[
0 \quad R_i = - \left\{ \frac{\partial x_{i_{bulk_{ave}}}}{\partial v} \right\}_{T \cdot v} = \left\{ \frac{\partial \bar{x}_{i_{ave}}}{\partial (\bar{T} \cdot v)} \right\}_v
\]

(7-92)

Since a constant concentration profile was assumed to be traveling through the column, all concentrations travel together and are time independent functions of the column volume coordinate. This leads to:

\[
0 \quad \left\{ \frac{dx_{i_{bulk_{ave}}}}{d\bar{T}} \right\} = \bar{T} \cdot \left\{ \frac{dx_{i_{ave}}}{d\bar{T}} \right\}
\]

(7-93)

For a binary system with column presaturation entirely by one component, and feed containing only the more favored component, \( \bar{T} = 1 \). Integration of equation (7-93) yields,

\[
x_{i_{bulk_{ave}}} = \bar{x}_{i_{ave}}
\]

(7-94)
Using the method of characteristics, equation (7-92) is uncoupled into two independent variables yielding ordinary differential equations in each variable i.e.,

\[
\frac{dx_{i,\text{bulk ave}}}{dv} \bigg|_{\bar{T} \cdot v} = -R_i, \quad i=1,2,\ldots,n-1 \tag{7-95}
\]

\[
\frac{d\bar{x}_{i,\text{ave}}}{d(\bar{T} \cdot v)} \bigg|_{v} = R_i, \quad i=1,2,\ldots,n-1 \tag{7-96}
\]

also the continuity equations become:

\[
\frac{d\bar{x}_{i,\text{ave}}}{d\bar{\tau}} = \frac{k_{Li}'}{F} \cdot (x_{i,\text{bulk ave}} - x_{i,\text{bulk ave}}^*) \tag{7-97}
\]

\[i=1,2,\ldots,n-1\]

Following a constant characteristic, \(R_i\), the equations are numerically integrated to solve for species concentration profiles versus dimensionless time, \(\bar{\tau}\).

d. \textbf{Determining rate constants from experimental data.}

For a binary system which meets the criteria that allows \(\bar{T} = 1\), equation (7-94) can be approximated as:

\[
k_{Li}' = \left( \frac{F \cdot \bar{T} \cdot \Delta x_{i,\text{bulk ave}}}{\Delta \bar{\tau} \cdot (x_{i,\text{bulk ave}}^* - x_{i,\text{bulk ave}}^*)} \right)_{i=1,2,\ldots,n-1} \tag{7-98}
\]
so by selecting points on the "S" curve \( k'_{L_{ij}} \) can be approximated as a function of \( x_{i_{ave}} \) (or \( k'_{f_{ij}} \) as a function of \( x_{f_{ave}} \)), and fit to a polynomial for Fick's Law. The same procedure is used to obtain \( k'_{L_{ij}} \) or \( k'_{f_{ij}} \) for N.P. treatment.

3.3 Simulation Results - Omatete

In general, the six binary column effluent concentration histories (ECHs) were fitted well with the polynomial version of Fick's Law, using a common set of nine parameters \( (b_{0_{ij}}, b_{1_{ij}}, b_{2_{ij}}) \), for \( i,j,k = 1,2,3 \). Unfortunately, correlated values of neither these nor the twelve N.P. parameters \( (k_{i_{ij}}, k_{j_{ij}}, k_{i_{ij}}, k_{j_{ij}}) \), for \( i,j,k = 1,2,3 \), bear any relationship to the film coefficients or diffusion coefficients measured by other investigators. The model should be considered semi-empirical due to the simplifications made in both the equilibrium relationship and the diffusion mechanism. The "transportability" of the parameters to other ion exchange systems should be regarded as low.

In the three binary cases in which fluid film control was assumed, the fits using the Improved Fick's Law were by far the best— and this should be no surprise since a polynomial fit was specific to the ECH curves. In general, all models gave reasonable representation in the 0.05 normal runs, except Simple Fick's Law in the \( \text{Ag}^+ - \text{Na}^+ \) run, and Simple Nernst Planck in the \( \text{Na}^+ - \text{H}^+ \) run.
The three resin "film" control runs were more difficult to simulate, with only the polynomial version of Fick's Law fitting all three runs well, as could be anticipated. The Simple Nernst Planck Model, followed by the Improved Nernst Planck Model, gave the poorest fits to the "resin film" controlled binary rate run ECH curves. This was due to the attempt to represent the complex dynamics of resin diffusion, having space and time varying compositions, with a simple film representation. Ternary simulation of the two favorable exchanges was good, but the ternary simulation of the unfavorable exchange was only fair, and will be discussed below.

One complication which has to be taken into account in any simulation of the Ag⁺ systems is the high degree of association of Ag⁺ to form AgNO₃, particularly at high normalities. The resin then adsorbs the AgNO₃, taking up NO₃⁻, which makes the exchange non-stoichiometric during the uptake period. This effect can be offset to a large degree by adjusting the resin equivalent ion exchange site concentration upward to include the adsorbed AgNO₃ as Ag⁺ equivalents as well as the ion exchanged Ag⁺ in any exchange process involving Ag⁺. Omatete's throughput parameter \(\bar{T}\), at a value of one, is essentially the plug flow time used in this research. That is, all of the equivalents of counterion in the solution fed into the column up to the plug flow time (\(t_{p.f.}\)), displaces
all of the equivalents of the other counterions in the column resin plus solution in the voids. Omatete adjusted his throughput parameter upward by including the adsorbed Ag⁺ in the column equivalent capacity. This approach works provided the adsorption process is faster than the ion exchange process, and all of the material balance transactions occur before the ECH curve leaves the column, with its pattern established primarily due to the ion exchange process.

3.4 Simulation of Omatete's Binary Experiments - This Research Project

a. General

The results of the simulation of the six favorable binary exchanges are shown in Figures 7-5 through 7-10. These runs are in contrast to the Erickson experiments in that the superficial velocities were much lower, with Re numbers of 2 to 4 compared to values of 20 to 40 in Erickson's work. This is an important distinction in that the axial diffusion flux in the region of maximum mass transfer for Omatete's work was about 1-2 orders of magnitude lower than that observed in simulating Erickson's experiments (values of $0.5 \times 10^{-4}$ equiv/cm²/sec versus $0.1 \times 10^{-2}$ equiv/cm²/sec). In the very favorable exchange of Ag⁺ for H⁺, gradients were steep and, as discussed earlier, if the grid increments chosen were too "coarse," the Crank-Nicholson method would not converge to the exact solutions. As discussed, Figure 7-10 for the depicts 1.5 normal
Na⁺ Exchange with H⁺ on Dowex 50W x 8 Resin - NO₃⁻
Binary Model - Nernst Planck
- Favorable Equilibrium -
0.055 Normal

Equilibrium Data
K_{Na⁺/H⁺} = 1.49827
Redlich-Kister Constants
B = -0.15021
C = -0.15682
D = -0.08917
Bromley Constants
B_{NaNO₃} = -0.0128
B_{HNO₃} = 0.0776

Run S - 2*

○ Experimental Data at 23 ± 2°C
r = 0.036 cm (ave. particle radius)
v = 0.1885 cm/sec (superficial)
L = 60.32 cm (effective bed depth)
Equivalents in Resin = 2.19 eq./liter
 Void Fraction = 0.324 (ε)

Calculated N.P. Model
D_{Na⁺} = 2.5x10⁻⁶ cm²/sec
D_{H⁺} = 10.0x10⁻⁶ cm²/sec
Kₜ varies with soln. conc.
0.0038 – 0.014 cm/sec
Eₜ = 0.05 cm²/sec
Re = 4.2 (based on V/Ε)

*Omatete, "Column Dynamics of Ternary Ion Exchange" U. CA. 1971
Ag⁺ Exchange with Na⁺ on Dowex 50W x 8 Resin - NO₃⁻
Binary Model - Nernst Planck
- Favorable Equilibrium -
0.053 Normal

Figure 7-6

Equilibrium Data
$K_{Ag^+ Na^+} = 5.924$
Redlich - Kister Constants
B = −1.000
C = 1.2472
D = −0.8957
Bromley Constants
$B_{AgNO_3} = −0.0828$
$B_{NaNO_3} = −0.0128$

Run S - 3*
○ Experimental Data at 23 ± 2°C
r = 0.036 (ave. particle radius)
v = 0.183 cm/sec (superficial)
L = 60.32 cm (effective bed depth)
Equivalents in Resin = 2.12 eq./liter
Void Fraction = 0.324 ($\epsilon$)
— Calculated N.P. Model
$D_{Ag^+} = 0.9 \times 10^{-6}$ cm²/sec
$D_{Na^+} = 2.5 \times 10^{-6}$ cm²/sec
$K_f$ varies with soln. conc.
0.0037–0.0043 cm/sec
$E_D = 0.05$ cm²/sec
Re = 4.07 (based on V/\epsilon)

*Omatete "Column Dynamics of Ternary Ion Exchange" U. CA. 1971
Ag⁺ Exchange with H⁺ on Dowex 50W x 8 Resin - NO₃⁻
Binary Model - Nernst Planck
- Favorable Equilibrium -
0.05 Normal

Equilibrium Data
\[ K_{Ag^+ H^+} = 8.8761 \]
Redlich-Kister Constants
\[ B = -0.7257 \]
\[ C = 0.8988 \]
\[ D = -0.9350 \]
Bromley Constants
\[ B_{AgNO₃} = -0.0828 \]
\[ B_{HNO₃} = 0.0776 \]

Run S·5*
- Experimental Data at 23±2°C
- \( r = 0.036 \) cm (ave. particle radius)
- \( v = 0.198 \) cm/sec (superficial)
- \( L = 60.32 \) cm (effective bed depth)
- Equivalents in Resin = 1.97 eq./liter
- Void Fraction = 0.324 (\( \epsilon \))
- Calculated N.P. Model
  \[ D_{Ag^+} = 0.9 \times 10^{-6} \text{ cm}^2/\text{sec} \]
  \[ D_{H^+} = 10.0 \times 10^{-6} \text{ cm}^2/\text{sec} \]
  \( K_v \) varies with soln. conc.
  \[ 0.0045-0.014 \text{ cm/sec} \]
  \[ E_D = 0.05 \text{ cm}^2/\text{sec} \]

Cross calculated using Kataoka, Yoshida
Relationship for Electric Field in Film
Re = 4.4 (based on \( V/\epsilon \))

*Omatete, "Column Dynamics of Ternary Ion Exchange" U. CA. 1971
Na⁺ Exchange with H⁺ on Dowex 50W x 8 Resin - NO₃⁻  
Binary Model - Nernst Planck Relationship  
- Favorable Equilibrium -  
1.54 Normal

Equilibrium Data
K_{Na⁺ H⁺} = 1.4983

Redlich-Kister Constants
B = -0.15021
C = -0.15682
D = -0.08917

Bromley Constants
B_{NaNO₃} = -0.0128
B_{HNO₃} = 0.0776

Run P - 11*
○ Experimental Data at 23 ± 2°C
r = 0.036 cm (ave. particle radius)
v = 0.079 cm/sec (superficial)
L = 121.65 cm (effective bed depth)
Equivalents in Resin = 3.92 eq./liter
Void Fraction = 0.324 (ε)

--- Calculated N.P. Model ---
D_{Na⁺} = 2.5 \times 10^{-6} \text{ cm}^2/\text{sec}
D_{H⁺} = 10.0 \times 10^{-6} \text{ cm}^2/\text{sec}
K_r varies with soln. conc.
0.0023 - 0.0094 cm/sec
Eₚ = 0.025 \text{ cm}^2/\text{sec}
Re = 1.89 (based on V/ε)

*Omatete, “Column Dynamics of Ternary Ion Exchange” U. CA. 1971
Ag⁺ Exchange with Na⁺ on Dowex 50W x 8 Resin - NO₃⁻
Binary Model - Nernst Planck
- Favorable Equilibrium -

1.5 Normal

Equilibrium Data
K_{Ag⁺Na⁺} = 5.924

Redlich - Kister Constants
B = -1.000
C = 1.2472
D = -0.8957

Bromley Constants
B_{AgNO₃} = -0.0828
B_{NaNO₃} = -0.0128

Run P - 12
○ Experimental Data at 23±2°C
r = 0.036 (ave. particle radius)
v = 0.079 cm/sec (superficial)
L = 121.65 cm (effective bed length)
Equivalents in Resin = 4.37 eq./liter
Void Fraction = 0.301 (ε)

Calculated N.P. Model
D_{Ag⁺} = 0.9 × 10⁻⁶ cm²/sec
D_{Na⁺} = 2.5 × 10⁻⁶ cm²/sec
K_F varies with soln. conc.
0.00235–0.00245 cm/sec
E_D = 0.025 cm²/sec
Re = 1.9 (based on V/ε)

activity driving force

Omatete "Column Dynamics of Ternary Ion Exchange" U. CA. 1971
Ag+ Exchange with H+ on Dowex 50W x 8 Resin - NO₃⁻
Binary Model - Nernst Planck
- Favorable Equilibrium -
1.525 Normal

Equilibrium Data
K_{Ag+H+} = 8.8761

Redlich-Kister Constants
B = -0.7257
C = 0.8998
D = -0.9350

Bromley Constants
B_{AgNO_3} = -0.0828
B_{HNO_3} = 0.0776

Run P - 4*
- Experimental Data at 23±2°C
- r = 0.036 cm (ave. particle radius)
- \nu = 0.079 cm/sec (superficial)
- L = 121.65 cm (effective bed length)
- Equivalents in Resin = 4.67 eq./liter
- Void Fraction = 0.324 (ε)

Calculated N.P. Model
- \rho_{Ag+} = 0.9x10^{-6} cm²/sec
- \rho_{H+} = 10.0x10^{-6} cm²/sec
- K_i varies with soln. conc.
  0.0024–0.0094 cm/sec
- E_D = 0.025 cm²/sec
- Re = 1.42 (based on V/ε)

Film Coefficient Calculated Using
Kataoka-Yoshida Relationship for
Electric Field in Film (l = 101, M = 11)

*Omatete, "Column Dynamics of Ternary
Ion Exchange" U. CA. 1971
exchange of Ag$^+$ for H$^+$ and shows the effect of decreasing resin and column grid spacing on solution convergence. Utilizing 11 resin nodes and 101 column nodes provided a convergent solution, since no further improvement was made by increasing column nodes to 201. In general, increasing resin nodes from 11 to 21 had little effect on improvement in convergence. This intuitively makes sense since the ratio of actual column dimension to resin dimension was 500-1500 to 1.

b. Literature Values of Diffusion Coefficient - Resin Phase

The simulation program RFMAVD was used with (1) equilibrium parameters derived from the program RIONFT, (2) the pseudo electric field film coefficient derived using the Carberry correlation for "effective boundary layer film thickness," (3) axial diffusion coefficients derived from Figure 6-1, and (4) starting values of resin phase diffusion coefficients derived from literature sources. These literature sources gave the following experimental values:

Hering & Bliss, (46) using Dowex 50w x8 resin found the N.P. effective resin diffusion coefficients for Na$^+$-Ag$^+$ to be

\[ \bar{D}_{Na^+} = 2.05 \times 10^{-6} \text{ cm}^2/\text{sec} \]
\[ \bar{D}_{Ag^+} = 0.68 \times 10^{-6} \text{ cm}^2/\text{sec} \]
Graham & Dranoff (30) in studying the exchange of \( \text{Na}^+ - \text{Cs}^+ \) on Dowex 50w x8 found the effective diffusion coefficient of \( \text{Na}^+ \) to be \( D_{\text{Na}} = 2.04 \times 10^{-6} \text{ cm}^2/\text{sec} \).

Boyd & Soldano (7) studied isotopic exchange of ions to obtain self diffusion coefficients, and for Dowex 50w x8 resin obtained:

\[
D_{\text{Na}^+} = 0.944 \times 10^{-6} \text{ cm}^2/\text{sec} \\
D_{\text{Ag}^+} = 0.642 \times 10^{-6} \text{ cm}^2/\text{sec}
\]

Reichenberg, (97) using Dowex 50w x5, found the effective Fick's Law diffusion coefficient for \( \text{Na}^+ \) displacing \( \text{H}^+ \) to be

\[ D_{\text{Na}^+\text{H}^+} = 7.3 \times 10^{-6} \]

Although the value of \( D_{\text{H}^+} \) is "buried" in the effective overall coefficient, a value of \( 10.0 \times 10^{-6} \text{ cm}^2/\text{sec} \) for \( D_{\text{H}^+} \) relative to \( D_{\text{Na}^+} \) of 2.0-2.5 \( \text{ cm}^2/\text{sec} \) would not seem unreasonable. No other value for \( D_{\text{H}^+} \) was found in the literature for the Dowex 50w X8 resin system. In electrolytes at infinite dilution, the diffusion coefficient ratio, \( D_{\text{H}^+}/D_{\text{Na}^+} \) is about 7.

Gopala Rao and his coworkers (92, 93) measured \( D_{\text{Na}^+} \) to be \( 2.04 \times 10^{-6} \text{ cm}^2/\text{sec} \) in their tracer experiments on Dowex 50w x8 resin. The research in this work on Rao et al data yielded a correlated value of \( D_{\text{Na}^+} \) to be \( 2.11 \times 10^{-6} \text{ cm}^2/\text{sec} \).
using concentration driving force, and 2.88 x 10^{-6} \text{ cm}^2/\text{sec} using activity driving force.

c. Diffusion Coefficients - Resin Phase

Diffusion coefficient values were obtained through repeated simulation of the experimental ECH curves in order to obtain a single set of values for $D_{Ag^+}$, $D_{Na^+}$, & $D_{H^+}$. This single set of values had to reproduce the three binary systems at two concentration levels, and give good representation of the three ternary systems as well.

The following ranges of diffusion coefficients were investigated:

\[
\begin{array}{|c|c|}
\hline
& \text{cm}^2/\text{sec} \times 10^6 \\
\hline
D_{Ag^+} & 0.7 - 1.1 \\
D_{Na^+} & 1.0 - 3.5 \\
D_{H^+} & 5.0 - 15.0 \\
\hline
\end{array}
\]

The set of values finally selected to fit the 9 systems were

\[
\begin{align*}
D_{Ag^+} &= 0.9 \times 10^{-6} \text{ cm}^2/\text{sec} \\
D_{Na^+} &= 2.5 \times 10^{-6} \text{ cm}^2/\text{sec} \\
D_{H^+} &= 10.0 \times 10^{-6} \text{ cm}^2/\text{sec}
\end{align*}
\]

As can be noted, the value for $D_{Ag^+}$ was not greatly different from the Hering & Bliss value of 0.68 x 10^{-6} \text{ cm}^2/\text{sec} or the Boyd & Soldano value of 0.64 x 10^{-6} \text{ cm}^2/\text{sec}. The value of $D_{Na^+}$ at 2.5 \text{ cm}^2/\text{sec} was higher than the measurements of
most investigators who found $D_{Na^+}$ to be about $2.05 \times 10^{-6}$ cm$^2$/sec. Dranoff & Graham found $D_{Na^+}$ to be $2.4 \times 10^{-6}$ cm$^2$/sec with the Stefan-Maxwell model, and this current work found a value of $D_{Na^+}$ to be $2.87 \times 10^{-6}$ cm$^2$/sec using activity driving force. No comment can be made on the value of $10.0 \times 10^{-6}$ cm$^2$/sec for $D_{H^+}$, since no literature values were found for exchange in Dowex 50w X8 ion exchange resin. Turner and his coworkers (129) found the effective diffusion coefficient for $D_{H^+}$ to be $5.7 \times 10^{-6}$ cm$^2$ in Zeokarb 225-x8 resin based on the Nernst Planck Model. The other ion exchanged was Na$^+$ in a 0.1N Cl$^-$ solution and the value obtained for $D_{Na^+}$ was $1.21 \times 10^{-6}$ cm$^2$/sec. On a straight ratio basis, using $D_{Na^+}$ as the key, $D_{H^+}$ would be about $11.4 \times 10^{-6}$ cm$^2$/sec.

d. Axial Dispersion Coefficients

Although the values of axial diffusion coefficients were taken from Figure 6-1, several runs were made in which axial diffusion coefficient values ten times larger than those given in Figure 6-1 were tried. Considerable "slewing" of the simulated ECHs resulted, with obviously poorer fit. The effective resin diffusion coefficients would have had to be adjusted upward considerably to match the ECH curves in the Ag$^+-H^+$ and
Ag⁺-Na⁺ systems, utilizing the exaggerated axial diffusion coefficients. This was particularly true at a normality of 1.5 where resin phase flux rate tends to control the overall transport of ions. The effect in the more dilute systems, where the film tends to control the diffusion rate, is that smaller film "thicknesses", $\delta_{F,\text{eff}}$, would have had to be employed. These film coefficients would have been higher than and outside of the range of coefficients predicted by the Carberry correlation. It was for this reason that the values of $E_D$ (axial diffusion coefficients) from Figure 6-1 were judged to be about the correct order of magnitude.

e. Equilibrium Effects

Considerable effort was expended in attempting to fit the Na⁺-H⁺ binary systems by utilizing various combinations of diffusion coefficients with little initial success. While a given $\overline{D}_{Na^+}/\overline{D}_{H^+}$ ratio might provide a better fit, the ratio selected invariably "ruined" the fit on the other four binary systems. Examination of Figure 7-5 showing the ECH curve for Na⁺-H⁺ exchange at 0.05 normality shows a "lazy S" shape. Using equilibrium parameters correlated from the experimental data developed by Omatete, fits of this system tended to be late in first appearance
of the Na\(^+\) ion, and would tend to undershoot the experimental points at the end of the run. This suggested that the driving force was too high at low values of Na\(^+\) ion, and too low at the high concentration end. This same problem was encountered with the 1.5 normal Na\(^+\)-H\(^+\) run as well, shown on Figure 7-8. Indeed, the equilibrium data of Omatete gave correlation parameters which, when used to predict the full binary equilibrium curves, showed upper "azeotropes" for both normalities, and no "pinch point" at the lower Na\(^+\) concentrations. Turner and his coworkers (129) had developed Na\(^+\)-H\(^+\) system data at 0.1 normality in their work on ion exchange mass transfer mechanisms. Use of eleven additional points from their work, particularly in the upper and lower regions, allowed refitting of the Na\(^+\)-H equilibrium data. The equilibrium curves calculated from the new correlation parameters showed pronounced "pinching" in the region of 0 to 0.1 equivalent fraction Na\(^+\) for both normalities. The 0.1 normal curve now "pinched" at the 0.9 to 1.0 region but did not show a reversal of selectivity. The "azeotrope" point for the 1.5 normal system was pushed up from around 0.7 to 0.82 Na\(^+\) equivalent fraction, with the selectivity reversal being not so pronounced.
Upon resimulating the ion exchange dynamics for the Na-H systems, satisfactory fits could now be obtained using reasonable values (by literature standards) of $D_{Na^+}$ and $D_{H^+}$ without disturbing the fits on the other four systems. The lower "pinch" point reduced driving force in the transfer region where Na$^+$ concentration was low, so the Na$^+$ ion appeared early, matching the experimental points. The driving force at both normalities was increased in the region where Na$^+$ predominated, and the previously observed undershoot of the experimental ECH curves by the simulation model was no longer evident, since the driving forces had been increased.

This merely illustrates the obvious point that before dynamic processes can be modeled, the equilibrium values representing driving force potential should be experimentally measured and correlated or estimated as precisely as possible, particularly when any physical-chemical process exhibits dramatic selectivity changes as a function of composition.

4. Conclusions - Binary Ion Exchange Simulation - Column Operation

The binary model developed based on the features discussed in this chapter gave excellent simulated representation of the experimental effluent concentration histories for nine systems, with literature correlations for all important parameters except
for resin phase diffusion coefficients and equilibrium data. The equilibrium parameters used to provide equilibrium driving force potential were obtained by correlating the experimental data of Erickson and Omatete using the program RIONFT, except as supplemented from other literature sources as cited above. The values of the fitted resin phase diffusion coefficients were close to previously cited values where such information was available in the literature.

D. Ternary Solution for Column Ion Exchange Operation

1. General

Section B of this chapter covered the development of the fluid phase equations describing the diffusional processes and material balance for ternary ion exchange. The two resulting fluid phase non-linear partial differential equations were approximated by finite difference equations using the Crank-Nicholson approach, and these finite difference equations were expressed in matrix notation in eq (7-27) & (7-28). Section B also described how the two fluid phase matrix equations were coupled to the two resin phase matrix equations which were developed in Chapter V, Section C, describing the diffusional processes and material balance for ternary ion exchange.

The detailed development of the solution of the four coupled matrix equations is given in Appendix G, and involves the use of seven and five diagonal array algorithms developed in this research to solve the coupled sets of equations. The resulting solution of the finite difference equations at each time step, with resolution of non-linear terms through the
quasilinearization procedure, leads to an approximation of the concentration profiles in both the resin phase and the fluid phase, along with diffusional fluxes if desired. Appropriate adjustment of standard parameters such as film coefficients, axial diffusion coefficients, and resin phase diffusion coefficients should result in profiles which match the experimental effluent concentration histories (ECH) for column operation.

The binary solution for column ion exchange operations was covered in Section C of this chapter, since it provided a simpler treatment of initialization procedures and approximation by finite difference equations of the relevant boundary conditions, along with attendant model operating problems and their solutions. The performance of the binary model in simulating nine experimental column ion exchange effluent concentration histories, under widely varying conditions, was judged to be good. This section will deal with the same initialization procedures and representation of boundary conditions as they apply to the ternary exchange. The essential features of the ternary simulation program, TERNEX, will be described in this section, as well as the results of simulation of the three Omatete experimental ternary runs using TERNEX.

2. Initialization Procedures

The initialization procedure for the binary model was described in Section C, para. 2. In the ternary model, a similar "conditioning" of the column is essential to obtain a smooth start to the solution of the finite difference equations. In the
ternary model, the procedure is also based on balancing the ionic species flux at the fluid/resin interface, using a backward difference numerical representation of

\[
\frac{\partial \bar{x}_i}{\partial \rho} \Bigg|_{\rho=1} (i,0) \quad (7-99)
\]

The equations describing flux at the interface where \( \rho = 1 \) are:

\[
\frac{2 \cdot Q_r}{k_f \cdot r_o \cdot \text{norm} \cdot \Delta \rho} \left\{ \begin{array}{c} \bar{D}_{i1} \cdot \frac{\partial \bar{x}_i}{\partial \rho} \Bigg|_{\rho=1} + \left( \frac{|z_i|}{|z_j|} \right) \cdot \bar{D}_{ij} \cdot \frac{\partial \bar{x}_j}{\partial \rho} \Bigg|_{\rho=1} \end{array} \right\} = (x_i^* - x_i)_{\text{bulk int}}
\]

for \( i = 1, j = 2 \), and \( i = 2, j = 1 \)

and \( x_i^* \) is an implicit function of \( (\bar{x}_1, x_i^* \text{ int}) \)

The equations are approximated in finite difference terms for column length node 1 (column inlet) as:

\[
\frac{2 \cdot Q_r}{k_f(1) \cdot r_o \cdot \text{norm} \cdot \Delta \rho} \left\{ \begin{array}{c} \bar{D}_{i1}(M,1) \cdot (\bar{x}_1(M,1) - \bar{x}_1(M-1,1)) \\
+ \left( \frac{|z_i|}{|z_j|} \right) \cdot \bar{D}_{ij}(M,1) \cdot (\bar{x}_j(M,1) - \bar{x}_j(M-1,1)) \end{array} \right\} = \left\{ \begin{array}{c} x_i(1) - x_i^*(1) \\
\end{array} \right\}
\]

for \( i = 1, j = 2 \), and \( i = 2, j = 1 \)

Note \( \bar{x}_i(M-1,1) \) and \( \bar{x}_j(M-1,1) \) equal \( \bar{x}_i(\text{initial}) \) and \( \bar{x}_j(\text{initial}) \) respectively.
Since the equations are non-linear, i.e., equilibrium relationship and diffusion coefficients are functions of $\bar{x}_i(M,1)$ and $\bar{x}_j(M,1)$, the Newton-Raphson procedure is used to solve for $\bar{x}_i(M,1)$ and $\bar{x}_j(M,1)$.

Having determined $\bar{x}_i(M,1)$ and $\bar{x}_j(M,1)$, the distribution of concentrations of $\bar{x}_i(M,\lambda)$ and $\bar{x}_j(M,\lambda)$ are determined using the exponential decay function described in Section C, para. 2. In the ternary case, however, there are four potential starting distributions:

1. $\bar{x}_1(M,1)$ and $\bar{x}_2(M,1)$ are $> \bar{x}_1(\text{initial})$ and $\bar{x}_2(\text{initial})$
2. $\bar{x}_1(M,1)$ and $\bar{x}_2(M,1)$ are $< \bar{x}_1(\text{initial})$ and $\bar{x}_2(\text{initial})$
3. $\bar{x}_1(M,1) > \bar{x}_1(\text{initial})$ and $\bar{x}_2(M,1) < \bar{x}_2(\text{initial})$
4. $\bar{x}_1(M,1) < \bar{x}_1(\text{initial})$ and $\bar{x}_2(M,1) > \bar{x}_2(\text{initial})$

The initial value of ionic species concentration in the bed is model input data, selected by the user to represent the case to be simulated. In the three Omatete experiments, two were based on the resin bed completely in the $H^+$ form, or case 1, and one was based on the resin bed completely in the $Na^+$ form, or case 3 (since $Ag^+ = 1, Na^+ = 2, H^+ = 3$ in order of selectivity). Once the program tests for type of distribution, the appropriate exponential decrease or increase is made for each component, starting with $\bar{x}_i(M,1)$ and $\bar{x}_j(M,1)$, down the length of the column at each column node.

The concentration of electrolyte ionic species in equilibrium with $\bar{x}_i$ or $j(M,\lambda)$, ($\lambda = 1, I$) or $\bar{x}_i^*$ ($\lambda$), ($\lambda = 1, I$), is calculated next. Then, using the flux balance equations
described above, equation (7-101), the fluid phase concentrations are calculated for \( x_{i\text{orj}}(\ell), \) \( (\ell = 1, I). \) This technique was used to provide starting compositions prior to time step \( j = 1, \) which would "condition" the column for a smooth start with minimum "presaturation" due to initialization, and indeed only 0.5-1.0% initial preloading was experienced.

In discussing the finite difference representation of the Boundary condition at the column inlet for the binary case, Section C, para. 3, it was noted that the representation given in equation (7-49) worked well for the binary model RFMAVD. It was further noted that Svedberg (125) had found in cases where the concentration gradients were too steep that the composition at the bed inlet tended to oscillate, but that the oscillations were not transferred into the bed. In simulating ternary ion exchange under the experimental conditions used by Omatete, i.e., very low electrolyte convection flux relative to resin takeup flux, it was found that the concentration gradients were steep. Even using a fine grid spacing \( (\Delta Y = 0.01), \) in the simulation, oscillations in electrolyte concentration, \( x_{i\text{bulk}}(\ell), \) and \( x_{j\text{bulk}}(\ell) \) would pass down the column in a "ripple" effect. A revised initialization procedure corrected this problem, so it will be described here under Initialization rather than under the section on Boundary Conditions.

The revised procedure was to allow the solution to the equations describing the ternary ion exchange to proceed
through the first ten dimensionless time steps, but at each step

\[ \bar{x}_{i}(m,1)_{t+1} = \bar{x}_{i}(m,1)_{t+1} \cdot \text{wgt} + \bar{x}_{i}(m,1)_{t} \cdot (1 - \text{wgt}) \]

or

\[ \bar{x}_{i}(m,1)_{t+1} = \bar{x}_{i}(m,1)_{t} \cdot \text{wgt} + \bar{x}_{i}(m,1)_{t} \cdot (1 - \text{wgt}) \]

for \( m = 1, M \) and \( i = 1, I \)

It was found that a "wgt" (weight) of about 0.50 minimized the perturbations after running the calculations through \( j = 10 \). The "clock" is then set back to \( j = 1 \), and the solution restarted using the distribution of concentrations in the resin and solution phases resulting from the solution at \( j = 10 \) (start). After the new start, the weighting relationship described above is "turned off" for the remainder of the computation. Invariably this procedure led to a smooth initial second start, and although some bed "presaturation" was a natural consequence of the procedure, the "preloading" rarely exceeded 2-3%. This initialization procedure is used in TERNEX, the ternary ion exchange simulation program.

3. Treatment of Boundary Conditions by Numerical Approximation

The numerical approximation of Boundary Condition (B.C.) at the column inlet was described in Section C, para. 3, for the binary model, along with a discussion of potential instability problems. The revised initialization procedure ameliorated the problem of composition oscillation at this B.C. for the ternary model.

The approximation of the second boundary condition, at
the column exit, was described in Section C, para 3, for the binary model. As discussed, the column increment between column nodes \( i = I-1 \), and \( I \) was approximated by a CFSTR model, and this was based on experience gained from results of early simulation trials with the ternary model. Use of the CFSTR model reduced the "stiff" condition at the column exit, and allowed the "S" curve to pass through in a normal manner. Prior to the new initialization procedure described above, the ripple caused by the inlet oscillation would reach the column exit and the concentration "pile up" was even more severe. The combination of smoothing the initial operation enhanced the effect of using the CFSTR model at the column exit in terms of producing reasonable ECHs for the three ternary cases simulated.

4. **Features of Ternary Modeling Program - TERNEX**

4.1 The ternary model is based on the coupled matrix equations describing the Nernst-Planck diffusion model for three ionic species in the resin phase, along with the coupling of the ternary fluid phase matrix equations to the resin phase matrix equations to describe the effluent concentration history of fixed bed ion exchange operation. Diffusion through the "film," or stagnant layer separating the resin phase from the fluid phase, is based on a pseudo electric field model, which again utilizes the Nernst-Planck relationship describing interdiffusivity under the influence of an electric
field—in this case assumed constant through the "film," as is coion concentration. Axial dispersion for all species is accounted for by a single value for the axial diffusion coefficient, and the ternary equilibrium is described by mass action type of relationships, using the Redlich-Kister activity coefficient model to predict resin phase activity coefficients for each ionic species.

Discussion in Sections A and B of this chapter described the use of the quasilinearization technique to "linearize" non-linear terms and, through use of seven and five diagonal algorithms, the four coupled matrix equations are solved at each time step.

4.2 The ternary model, TERNEX, unlike the two binary models, BMAVD AND RFMAVD, or the batch ternary model, TRYBCH, does not have an option to use activity driving force instead of concentration driving force. The case for use of activities versus concentrations in ion exchange processes is best tested by simulating batch type experiments based on, for example, radioactive tracer techniques. Unless the deviation from ideality is large, the activity effect on column effluent history is difficult to discriminate. As was shown in the discussion of the simulation of Erickson's experiments on the system ETDA-NH$_3$, use of equilibrium parameters correlated in terms of either mole fraction or equivalent fraction give identical results for all
practical purposes. Consequently the ternary model, TERNEX, utilizes equilibrium parameters correlated in equivalent fraction only.

4.3 The detailed features of TERNEX are given in Appendix L.

E. Ternary Simulation of Column Operations

1. Discussion - Omatete Experiments and Simulation Results

1.1 Experimental Results and Model Development

The experimental technique used by Omatete in both the binary and ternary runs is described in Section D, para. 3.1 of this chapter. In developing experimental ECH data on the film controlled ternary systems (0.05 normal), Omatete performed two runs, S-17 and S-18, but only S-17 was used. The plateau concentration in run S-18 was not fully developed due either to a flow rate which was too high, or the column (2 ft.) was too short for the conditions imposed (feed $x_{Ag} = 0.75, x_{Na} = 0.25$; column preloading $\bar{x}_{Ag} = 0.250, \bar{x}_{Na} = 0.240, \bar{x}_H = 0.510$).

Experiments in the region where resin diffusion rates were controlling (1.5 normal) were performed using four different combinations of feed and column conditions, labeled runs P-7, P-8, P-15 and P-16. Of these, only runs P-8 and P-15 had fully developed concentration plateaus and so were simulated. The two runs not used were for conditions in which the column was presaturated with all three species, rather than a single species as in runs P-8 and P-15. No explanation
was given for the lack of development of the concentration plateaus in the two rejected experiments; however, in retrospect, the rejected experiments may have been more useful if only one species, say $H^+$, was in the elution feed thereby simulating column regeneration (unfavorable exchange).

Omatete's model development is covered in detail in Section D, para. 3.2 and, as discussed, was based on constant separation factors to describe equilibrium, and rate laws in which either a fluid interface film controlled at low normalities, or a pseudo resin "film" controlled at high concentrations of electrolyte. Four models were described: (1) Simple Fick's Law, (2) Improved Fick's Law, (3) Nernst Planck, and (4) Improved Nernst Planck. The Improved Fick's Law, with polynomial fitted constants, gave the best all around representation of the six binary runs. The same binary models were extended to the ternary case by use of the combinatorial rules given in Section D, para. 3.2.

The ternary ECH curves were fit equally well by all four models. Two ECH curves, S-17 (0.05 normal) and P-8 (1.5 normal), were fit quite well, and P-15 (1.5 normal - unfavorable) would have been reasonably good if an adjustment had been made in the throuput parameter (resin equivalents) to account for less adsorption of AgNO$_3$ than had been allowed for by Omatete in this run. As will be discussed below, equilibrium considerations and material balance are extremely important in ternary ion exchange in defining the height and
length of plateaus, so even if the dynamic considerations are not correctly predicted, the general shape of the ternary ECH curves will appear to be satisfactory. It is likely that these considerations made model discrimination difficult in the ternary cases, and therefore the binary rate runs are more useful in determining ion exchange parameters from simulation of experimental data.

2.0 Simulation of Omatete's Ternary Experiments - This Research Project

2.1 General

The results of simulation of the three ternary runs, S-17, P-8, and P-15, are shown on Figures 7-11, 7-12, and 7-13. TERNEX was used to perform the computations leading to the simulated ECH curves for the ternary cases. In all cases, the ternary equilibrium concentrations were calculated from the fitted equilibrium constants, and resin phase activity coefficients derived from the correlated experimental binary data including the two ternary data points. The electrolyte phase activity coefficients were calculated from the Bromley Equation. The film coefficients were derived based on corrected single ion diffusion coefficients plus the Carberry film coefficient correlation using the pseudo electric field film model. The axial diffusion coefficients were obtained from the generalized correlations, Figure 6-1. The resin phase "self" diffusion coefficients were derived from "best" fits of the six binary rate runs, and used in program TERNEX with concentration driving forces (vs. activities). The literature values of the "self" diffusion coefficients were
Ag⁺ and Na⁺ Displacing H⁺ on Dowex 50W x 8 Resin - NO₃⁻
Ternary Model - Nernst Planck
- Favorable Equilibrium -
0.0533 Normal

Run S - 17*
○ Experimental Data at 23±2°C
r = 0.036 cm (ave. particle radius)
v = 0.0967 cm/sec (superficial)
L = 60.365 cm (effective bed length)
Equivalents in Resin = 2.38 eq./liter
Void Fraction = 0.356 (€)

Calculated N.P. Model

D_{Ag⁺} = 0.9x10⁻⁶ cm²/sec
D_{Na⁺} = 2.5x10⁻⁶ cm²/sec
D_{H⁺} = 10.0x10⁻⁶ cm²/sec

*Omatete "Column Dynamics of Ternary Ion Exchange" U. CA. 1971
Figure 7-12

Ag⁺ and Na⁺ Displacing H⁺ on Dowex 50W x 8 Resin - NO₃⁻
Ternary Model - Nernst Planck
-Favorable Equilibrium -
1.51 Normal

Run R - P8*
- Experimental Data at 23 ± 2°C
- r = 0.036 cm (ave. particle radius)
- v = 0.03455 cm/sec (superficial)
- L = 121.28 cm (effective bed length)
- Equivalents in Resin = 3.7 eq./liter
- Void Fraction = 0.356 (ε)
- Calculated N.P. Model

\[ \text{Elapsed Time - Seconds} \]

\[ X, \text{Equivalent Fraction of Species Exiting Column} \]

- \( D_{Ag^+} = 0.9 \times 10^{-6} \text{ cm}^2/\text{sec} \)
- \( D_{Na^+} = 2.5 \times 10^{-6} \text{ cm}^2/\text{sec} \)
- \( D_{H^+} = 10.0 \times 10^{-6} \text{ cm}^2/\text{sec} \)

*Omatete "Column Dynamics of Ternary Ion Exchange" U. CA. 1971
Ag⁺ and H⁺ Displacing Na⁺ on Dowex 50W x 8 Resin - NO₃⁻
Ternary Model - Nernst Planck
- Unfavorable Equilibrium -

1.5 Normal

Key
I = Column Nodes, M = Resin Nodes
J = Dimensionless Time Steps

I = 101, M = 11, J = 2100
I = 51, M = 11, J = 800

Run R - P15 *

Experimental Data at 23±2°C
r = 0.036 cm (ave. particle radius)
ν = 0.0461 cm/sec (superficial)
L = 121.28 cm (effective bed length)
Equivalents in Resin = 3.15 eq./liter
Void Fraction = 0.356 (ε)

Calculated N.P. Model

\[ \begin{align*}
D_{Ag^+} &= 0.9 \times 10^{-6} \text{ cm}^2/\text{sec} \\
D_{Na^+} &= 2.5 \times 10^{-6} \text{ cm}^2/\text{sec} \\
D_{H^+} &= 10.0 \times 10^{-6} \text{ cm}^2/\text{sec}
\end{align*} \]

* Omatete "Column Dynamics of Ternary Ion Exchange" U. CA. 1971
cited in Section D, para. 3.0, subsection 3.4, and the values utilized in the simulation of 6 binary systems and 3 ternary systems were not far different from those values. The results, as seen in the plots, are reasonably good, particularly since all of the information used provided good simulations of the six binary cases as well.

2.2 Equilibrium and Material Balance Considerations

This section will deal with the problems encountered in representing the period of "dwell" for plateaus in the ternary simulations. As can be seen, in a favorable exchange of a feed consisting of Ag\(^+\) and Na\(^+\) with a column containing resin presaturated with the least favored species, H\(^+\), the first breakthrough is Na\(^+\) ion at a concentration level which exceeds the Na\(^+\) ion inlet concentration. H\(^+\) ion level decreases, but to a plateau level higher than its feed concentration. The total exchange is completed when the Ag\(^+\) breakthrough is observed, and all concentrations reach their feed values.

This phenomena can be best visualized by assuming instantaneous ion exchange of Ag\(^+\) for H\(^+\), allowing Na\(^+\) and H\(^+\) released by the ion exchanger to proceed ahead in a "binary" mode. The resin ahead of the Ag\(^+\) front soaks up Na\(^+\) ion, in effect storing it, until the Ag\(^+\) ion rich feed front gets to that point. The excess Na\(^+\) is then "chased" and lends its concentration to the forward front--thereby building Na\(^+\) concentration to a higher level than the Na\(^+\) concentration in the feed. The Na\(^+\) in the fluid phase has
concentration gradients allowing it to attempt to diffuse up and down the column away from the plateau. In the down column direction, the gradient is gradual as some Na\(^+\) is ion exchanged with H\(^+\) in the resin. Up column, the gradient is steep, since there is no place for Na\(^+\) to go except to satisfy the ternary equilibrium requirements of the feed with the resin. Keep in mind, when looking at an ECH curve, that as an observer in the column you would view the concentration history as occurring in the reverse direction from the time plot. First Na\(^+\) arrives displacing H\(^+\), then the Na\(^+\) concentration drops as it is exchanged to satisfy the feed equilibrium, and Ag\(^+\) ion appears in the electrolyte phase for the first time as the resin is saturated.

The concentration height of the Na\(^+\) ion above inlet value can be approximated simply by material balance as follows:

assumee:

- wave fronts are square
- the most preferred ion displaces all species 2 and 3 ions ahead of it
- the concentration of species 2 and 3 ions represents a pseudo binary in equilibrium with the resin in their end of the column
- in the diagram below, excess species 2 in area B = deficit of species 3 in area A.
Checking this approximate method

bulk concentrations

<table>
<thead>
<tr>
<th></th>
<th>$x_{1,\text{in}}$</th>
<th>$x_{2,\text{in}}$</th>
<th>$x_{3,\text{in}}$</th>
<th>calc. $x_1$</th>
<th>obs. $x_1$</th>
<th>calc. $x_2$</th>
<th>obs. $x_2$</th>
<th>calc. $x_3$</th>
<th>obs. $x_3$</th>
</tr>
</thead>
<tbody>
<tr>
<td>S-17</td>
<td>0.25</td>
<td>0.24</td>
<td>0.51</td>
<td>0.365</td>
<td>0.365</td>
<td>0.635</td>
<td>0.635</td>
<td></td>
<td></td>
</tr>
<tr>
<td>P-8</td>
<td>0.350</td>
<td>0.24</td>
<td>0.41</td>
<td>0.415</td>
<td>0.39</td>
<td>0.585</td>
<td>0.61</td>
<td></td>
<td></td>
</tr>
<tr>
<td>P-15*</td>
<td>0.187</td>
<td>0.133</td>
<td>0.680</td>
<td>0.22</td>
<td>0.175</td>
<td>0.78</td>
<td>0.83</td>
<td></td>
<td></td>
</tr>
</tbody>
</table>

*unfavorable non-sharpening "S" curve so area $A = area B$

The next question is, what determines the approximate "dwell" for the species 2 and 3 plateaus? Again, a material balance gives the approximation, but this time it involves the ternary resin composition in equilibrium with the feed composition. Time $T$ depicted in Figure 7-14 can be computed based on favored species 1 by balancing equivalents into the column against the holdup of species 1 in the column up to time $T$, i.e.:
where:

\[ x_1 = \text{resin concentration of species 1 in equilibrium with the ternary system feed composition} \]

In the case of run S-17 time \( T \) works out to 46,700 seconds versus about 46,600 seconds observed. It is by this mechanism that the resin equivalents were adjusted in this work to account for adsorption of complexed AgNO\(_3\) (Omatete adjusted his Thruput Parameter in an analogous manner). Time \( T \) is thereby anchored to the observed value by material balance. In the case of the two Ag\(^+\) binary systems, this correction was about 15%. In the case of the ternary systems, the AgNO\(_3\) adjustments would be less because the Ag\(^+\) feed compositions were lower (between 0.187-0.350 equivalent ionic fraction), and the adjustment represented approximately 7% for the worst case, P-8 (100 milliequiv/1500 milliequiv).

Similarly, the initial plateau time for Na\(^+\), or species 2, can be approximated by a material balance on species 2 and knowledge of the ternary phase equilibrium diagram.

\[
T = \frac{\left( \text{equiv.-resin} \times x_1^* + \text{equiv.-col. voids} \times x_1 \right)_{\text{in}}}{\left( \text{total equiv. into column} \times x_1 \right)_{\text{bulk}}} 
\]

where:

\[ x_1^* = \text{resin concentration of species 1 in equilibrium with the ternary system feed composition} \]

(7-104)

Species 2 balance

\[
(\text{total equiv. flow in}) \times \left[ t \times x_2 \right]_{\text{in}} - (T-t) \times (x_2 - x_2)_{\text{in}} = \left( \text{equiv.-resin} \times x_2^* + \text{equiv.-col. voids} \times x_2 \right)_{\text{bulk}} 
\]

where:

\[ x_2^* = \text{resin concentration of species 2 in equilibrium with the ternary system feed composition} \]
This simple approach, using values from S-17, gave a \( t' \) from Figure 7-14 equal to 23,860 seconds vs. 24,000 observed. In this case, the equilibrium concentration value of Na\( ^+ \) taken from Figure 2-5 was 0.16, and represented one of the ternary data points resulting from closing the material balance on the three ternary runs. The initial correlation of Omatete equilibrium data using program RIONFT, not including the material balance points, gave \( x_{Na}^* \) for S-17 as 0.185—resulting in a simulation result with \( t' = 25,100 \) seconds, or 5% less dwell time. Run P-15 would have had an 8% shorter dwell for the H\( ^+ \) plateau based on correlation of the Omatete initial equilibrium data. The ternary equilibrium point correlated for P-8 conditions was close to the material balance requirement for that run, and no change was observed during recorrelation with the material balance points.

The previous discussion was detailed, not so much to provide a simple method of predicting ternary ECH curves, but to show the rationale for engaging in the somewhat time consuming procedure required to reconcile material balance and equilibrium data by a procedure consisting of:

(a) approximating the ternary equilibrium points from the correlation of the original data without the material balance points

(b) simulating the ternary runs and observing from material balance considerations, based on the plateau dwell, what the ternary equilibria concentrations should be to give a better fit
(c) refitting the binary and ternary equilibrium data with program RIONFT, using the new material balance equilibrium points in addition to the original data

(d) resimulating the six binary rate runs to make sure that the influence of the new material balance points on total data regression to obtain new equilibrium parameters did not alter those ECH curves significantly

(e) resimulating the three ternary systems to check out the effect of the new equilibrium parameters on the general material balances and Na\(^+\) and H\(^+\) plateau "dwell"s.

(f) repeating above, to "fine tune" the equilibrium correlation. This procedure was time consuming, and future investigators are encouraged to add a few more measured equilibrium data points in order to reduce correlation effort. The first ternary phase diagrams generated with the ternary phase diagram regression program, RIONFT, would have provided an excellent first "map" to select regions for additional experimental equilibrium data points.

2.3 Stability and Convergence of TERNEX

The general considerations of convergence and stability of the Crank-Nicholson method were discussed in Section C, para. 4, for binary model simulation results. The solution time for TERNEX, with resin mode grid spacings of 11, and column nodes of 50, with maximum dimensionless time (number of empty columns full displaced by the feed
solution during the run) divided by 800 required about 2.5-3.0 CPU hours. Therefore the number of tests at varying increment spacings during the ternary modeling was limited, in view of the iterations required to resolve the equilibrium and material balance tolerance as described above. In the case of experiment S-17, runs of $M = 7, I = 51, T = 1200$ were identical with $M = 11, I = 101, T = 1200$ (resin nodes = $M$, column nodes = $I$, time increments = $T$) and convergence and stability were assumed. In experiment P-8, runs of $M = 7, I = 51, T = 1850$ were identical to $M = 11, I = 101, T = 1850$, and convergence and stability were assumed.

Run P-15, Figure 7-13, an unfavorable exchange was fitted at $M = 11, I = 51$, and $T = 800$ with good results except for the initial appearance of $Na^+$ and $H^+$ ions. A final check at $M = 11, I = 101$, and $T = 2100$ (7 CPU hours on a DEC VAX 11/780) showed some tightening up of ECH profile for this unfavorable (nonsharpening "S" curve profile) exchange. However, the final appearance of the $Ag^+$ ion in the electrolyte phase was delayed by 250 seconds (less than 1% of the total material balance), and it was deemed proper not to "retune" the $H^+$ equilibrium point to compensate for this difference. It would suggest that in ever spreading elution ECH curves, where unfavorable exchange exists, that a "fine" grid should be utilized with the Crank-Nicholson method to maintain
stability and convergence. Whether the State Variable method (65) or Collocation method (29) would be better (less computer time) than Crank-Nicholson remains to be tested.

3. Conclusions

The ability of the binary model, RFMAVD, and its ternary version, program TERNEX, to simulate ion exchange in columns under wide ranging conditions has been demonstrated, as verified against experimental data on actual column runs. The use of literature information on all input parameters, except equilibrium data and ionic species "self" diffusion coefficients makes the models "powerful." The non-linear regression packages accompanying the dynamic models has improved the overall model capability, since with minimum experimental data if necessary, thermodynamically sound equilibrium parameters can be determined. Additionally, "self" diffusion coefficients found in literature sources for ubiquitous ions in the more common ion exchange resins (Dowex 50w) can provide a starting point for diffusion coefficient parameter estimation utilizing experimental ECH curves. Future studies should be concentrated on use of activity driving force versus concentration driving force, but this can best be explored with "batch" type experimentation on a variety of ion pair combinations. Finally, other numerical methods should be investigated for inherent convergence, stability and potential for reduction in computation time. Industrially, CPU solution times of 6 hours
are acceptable given the fact that the numbers of VAX 11/780 type systems are growing in industry at an accelerated pace. Even with existing installations, CPU utilization is normally limited to 6 out of 21 shifts/week, so access for longer running programs is not a problem. The major advantage of faster running methods which are both convergent and stable would be to couple the binary column ion exchange model based on those methods to a non-linear regression package. This was done with program BMAVD, the batch binary model, to develop program DFUSFT, the non-linear regression model designed to fit diffusion coefficients based on data generated in batch type ion exchange experiments. It should be possible to use experimental ECH data for a variety of binary ionic pairs as input to a regression routine from which diffusion and film coefficients could be derived. Using program RFMAVD as a subroutine, solution times of 20 minutes/ECH curve would be needed to yield a practical regression model of this type. This would require a 4-5 fold reduction from current solution times on a DEC VAX 11/780. This next step is a logical refinement of the currently described ion exchange modeling package.
A. Background

The development of a model to simulate the exchange of two or three counterions in an ion exchange resin selective for a desired ionic species has wide application in hydrometallurgical processes, waste treatment operations, winning of precious metals, and by analogous mechanism, to future biotechnology applications. Past treatments of ion exchange models have been concerned primarily with binary systems. These models represent phase equilibria with simple linear isotherms and the dynamics have been simulated using dominant film theory. The dominant film models are based on treatment of ionic diffusion in the resin phase as through a fictitious solid "film" surrounding the resin particle.

In all cases, in order to arrive at analytical solutions to the system of equations describing binary ion exchange in a fixed bed, the axial diffusion term has been neglected, linear equilibrium isotherms have been assumed, and diffusion parameters have been assumed to be constant. Numerical methods treating binary ion exchange kinetics have been a natural evolution in ion exchange theory over the past ten years. It was recognized that the axial dispersive effects cannot be ignored and that the equilibrium relationships are non-linear, so that numerical methods must be employed in order to obtain solutions.

The Nernst-Planck model has been shown to be effective in describing interdiffusivity of ions within the resin phase and also
in the film surrounding the resin particle, and these additional nonlinear effects mandate numerical solution methods. Investigators have examined separately the application of the Nernst-Planck model to the resin phase and to the film around the resin in order to determine the applicability of the model in each domain. Models of ion exchange dynamics in binary systems inclusive of both effects have been investigated by Turner and Snowdon in shallow fluidized beds (131) and Rao and his coworkers (92, 93) in batch type single particle experiments. Pan and David (83) developed a model inclusive of both effects for moving beds, but without axial diffusion terms.

Ternary ion exchange systems are more important in industrial applications than binary systems. For example, in selectively separating one cation from another, the mixed cations may be dissolved in dilute acid, or the selectively recovered cation may be eluted with strong acid, so H\(^+\) ion normally is involved with the two cations being separated. Two recent models were developed for simulating ternary ion exchange systems. Rao and his coworkers (92, 93) incorporated the Nernst-Planck diffusion model in the resin phase together with a pseudo Nernst-Planck treatment in the film to simulate batch experiments, and Omatete (78) developed a single film kinetic approach to fixed bed simulation. Neither model utilized thermodynamic equilibrium relationships to describe the distribution of ionic species between the electrolyte phase and the resin phase, and consequently neither characterizes phase equilibria adequately.

The present status of ion exchange in industrial process applications is similar to that of distillation technology twenty years ago. Distillation theory then treated the equilibrium stages
required, with dynamics of tray efficiency left to experiments in standard laboratory devices such as Oldershaw columns, plus generalized correlations which could be used to estimate the tray efficiencies as first approximations. Predictive methods for phase equilibria have advanced enormously during the period including the development of the NRTL and UNIFAC equations for prediction and/or correlation of liquid phase activity coefficients. Through the efforts of Fractionation Research Institute (FRI) and other similar agencies, tray efficiencies now can be predicted with good confidence. Systems for component separation by distillation can now be designed reliably without resort to experimental confirmation in many instances.

B. Objective

The objective of this research work was to likewise advance the engineering of ion exchange systems. Specifically, the objective was to develop a simulation model for ion exchange in ternary systems which would embody all of the currently proven factors applying to prediction of phase equilibria and the dynamics of ion exchange in fixed beds. A further objective was to develop a model requiring a minimum of experimentation for parameter estimation. A ternary system simulation model has been developed and, together with parameter estimation support programs, computer based optimization of the design and scale up of ion exchange systems can be carried out with reasonable confidence. The time and cost required to optimize and develop scale-up parameters through laboratory and pilot plant investigations should be reduced considerably.
C. General Approach to Development of Model and Parameter Estimation Programs

1. Overview

Figure 8-1 provides a "road map" of the development of the general purpose Ion Exchange Model for binary and ternary systems, along with the support regression analysis programs for parameter estimation. Along with Figure 1-1, which depicts the summary approach to model development and application discussed in the introduction, Figure 8-1 can be utilized to describe the necessary steps required to develop a model of any physical process. The discussion to follow will highlight only those activities considered to be the key advancements or improvements in the present work over other ion exchange models. The general approach utilized in the model development was to treat (1) the Fluid Phase Material Balance, (2) the Phase Equilibrium connecting the electrolyte phase to the ion exchange phase, and (3) Transport Resistance to ionic diffusion through the film and within the resin phase, all as separate phenomena to be researched and developed before combining them into the framework of an overall ion exchange model. The Figure 8-1 "roadmap" should be referred to as each major model element is discussed in the subsequent Part C paragraphs.

2. Fluid Phase Material Balance

The major consideration in the development of the electrolyte phase material balance equations was the addition of an axial dispersion term to the basic equations describing the concentration dependence of each ionic species as it moves
through a fixed ion exchange bed. This feature is obviously not a new concept and has been examined by most investigators of fixed bed operation involving adsorption or ion exchange. Due to the mathematical complications involved, it has been assumed to be negligible in most analytical solutions. Binary system models based on numerical solution methods which have been developed by other investigators normally have included the axial diffusion term for simulation of adsorption or ion exchange in fixed beds. These investigators found through simulation of experimental results that the axial dispersive effects were significant and could not be neglected. The fixed bed ion exchange model for ternary systems developed and tested by Omatete (78) did not include axial dispersive effects in order to simplify his numerical method solution procedure. An alternative material balance case shown on Figure 8-1 is that for the moving bed. The moving bed has not had wide industrial application due to practical problems involved with the handling of fragile resin beads without damage. In any event it represents a simple revision to the fixed bed model, since the fluid phase material balance reflects a steady state relationship to the resin phase at all points in the column.

The "batch" model listed on Figure 8-1 requires no fluid phase material balance, since its use in this work assumed no change in electrolyte composition during the exchange of ionic species within the resin.

A search of the literature for correlations of axial diffusion coefficients in packed beds led to development of
Figure 6-1 (12, 23, 27, 32, 61, 78, 115, 124). A model for simulating axial dispersion in packed beds was developed in this research. This model was used to determine the axial dispersion coefficients at two Reynolds numbers from the experimental data developed by K. L. Erickson (26). These points were added to Figure 6-1.

The material balance module for the electrolyte phase was prepared in the form of partial differential equations, including appropriate Initial and Boundary Conditions, to be included in the overall model.

3. Phase Equilibria

The phase equilibria relationships describing the distribution of three counterions between the electrolyte phase and the ion exchange resin phase had to be developed in order to obtain reliable predictable driving force potentials for the dynamic model. Utilization of a standard format known as the rational thermodynamic equilibrium constant satisfied the Gibbs free energy equilibrium constraints in that the activity relationship of each species, in both the resin and electrolyte phases were related by a constant at a given temperature. The equilibrium constants in a ternary system are therefore related by the chain rule (equation (2-19)),

\[
\overline{k}_{ij} = \left(\frac{|z_j|}{|z_i|}\right) \left(\frac{|z_k|}{|z_j|}\right) \overline{k}_{ik} \overline{k}_{jk}
\]

for species i, j, and k

This relationship is useful in the estimation of ternary ion exchange system equilibrium parameters since the degree to which it is satisfied is a measure of the consistency of the experimental data.
A survey of literature on correlation of activity coefficients of single ionic species in single salt and mixed salt electrolyte solutions led to selection of the Bromley version (9) of the extended Debye-Huckel equation (equation (3-4)). The Bromley equation was tested in this research against experimental values given by Robinson and Stokes (106) for activity coefficients on 17 salts at concentrations up to 5-6 molal, and was found to yield satisfactory predictions. Bromley parameters (9) which extend the predictive concentration range to the 5-6 molal region are available for a large number of salts, and the Bromley equation format has been arranged to allow prediction of single ion activity coefficients in mixed salt electrolyte solutions.

The equilibrium model could not be completed until a satisfactory model was found to describe the activity coefficient relationship of three ionic species within an ion exchange resin as a function of concentration. Three models were selected based on a review of the literature, including the work of Smith and Woodburn (116) on the NO₃⁻-Cl⁻-SO₄²⁻ system. These models were the three suffix Redlich-Kister, the Wilson, and the NRTL equations.

The rational thermodynamic equilibrium relationship, including the Bromley equation to describe electrolyte phase activity coefficients, was tested with each of the resin phase activity coefficient models described above. The models were also tested with regard to improved predictive capability depending on whether mole fraction or equivalent fraction
concentration units were used in the resin phase. Use of A. K. S. Murthy's non-linear regression program to estimate unknown parameters in explicit algebraic equations allowed testing of the models against experimental data available in the literature for four ternary ion exchange systems. These were (1) Mn$^{++}$, Cs$^+$, Na$^+$, (2) Sr$^{++}$, Mn$^{++}$, Cs$^+$, (3) NH$_4^+$, Na$^+$, H$^+$, and (4) NO$_3^-$, Cl$^-$, SO$_4^{2-}$.

The selected equilibrium model included the three suffix Redlich-Kister equation utilizing equivalent fraction concentration units to describe activity coefficient relationships in the resin phase. The selection was based on overall ability of that model to correlate binary systems and to predict ternary data points from the binary system parameters.

4. Transport Resistance

All models of diffusion processes must deal with the phenomena related to movement of physical entities through a medium, whether the entities are electrons, ions, or molecules. The rate of movement or diffusion can be characterized by (1) driving forces and (2) parameters related to the ease of movement of the entities, the diffusion coefficients. Movement of ions within an ion exchange resin, or intraparticle diffusion, has been characterized successfully with the Nernst-Planck model (equation (5-5)). The general equation has been derived for ternary systems in terms of chemical potential (equations (5-6), (5-7), and (5-8), and has been reduced to the ideal case (equations (5-10) and (5-11)) and to the binary non-ideal and ideal cases (equations (5-21) and (5-13)).
Inclusion of the Nernst-Planck model in the continuity equations for the resin phase is shown in equations (5-48) and (5-50). These equations, including appropriate initial and boundary conditions, constituted formulation of the resin phase diffusion model.

Many correlations for prediction of resistance to diffusion of molecular species through a film have been produced over the years, premised for the most part on boundary layer theory (equation (4-1)). Carberry's correlation (11) for fixed beds has been found satisfactory by many investigators and was adopted in this research (equation (4-2)). Defining the effective diffusion coefficient for use in such film resistance correlations as applied to ion exchange has attracted the interest of many investigators (26, 50, 78, 83, 131, 132), who variably recommend use of the value for the slowest ion, the average value, or the harmonic mean value. In most cases, the limiting value (infinite dilution) diffusion coefficients of single ions have been used regardless of the convention used in the choice of diffusion coefficient. By combining the Nernst-Hartley (104) equation, which relates the effect of the activity of an ionic species on its diffusion coefficient (equation (3-11)), with the Bromley equation which predicts the effect of solution Ionic Strength on ionic species activity, a relationship was derived between ionic diffusion coefficients and electrolyte concentration. Both an integrated and a differential form of that relationship was derived in this research. The two forms were tested against experimental
diffusion coefficient values given by Robinson and Stokes (106) on 17 salts involving concentrations up to 4 molal, and the integrated form was found to yield reasonable results. This correction was applied to all electrolyte phase diffusion coefficients in this work.

Migration of ions through the film surrounding each resin particle was recognized in previous work as being subject to an interdiffusivity dependency based on the Nernst-Planck model, the coions in the film electrolyte being analogous to the fixed charges in the resin. In this instance, however, the coions are free to move and do so in such a way that the concentration gradient of the coions in the film are counterbalanced by the electric charge gradient in the film. This phenomenon is interrelated with the relative diffusion coefficients of the counterions diffusing through the film.

The effect of an electric field in the film for binary systems had been treated rigorously by several investigators (43, 50, 131, 132); however the complexity in applying these rigorous derivations to ternary systems led this investigator to adopt a model somewhat similar to that adopted by Rao and his coworkers (92, 93). If the coion gradient in the film is assumed to be small, i.e., \( \frac{\partial c_i^*}{\partial y} = 0 \), then the coion charges of sign opposite to those of the counterions are distributed uniformly thru the film, in a manner analogous to the fixed charges in the resin. The Nernst-Planck equations derived for the resin phase system can then be applied to the film phase to interrelate the diffusivities of the exchanging ions. Assuming
the concentration gradients of all species to be linear across
the film, average electrolyte phase ionic concentrations can be
used in the model to yield concentration corrected diffusion
coefficients for each species. The Nernst-Planck equations then
are used to predict the flux of each species through the film.
The film "thickness," $\delta_{\text{eff}}$, is computed for the major flux
based on the following equation:

$$k_{\text{eff-max}} = \frac{D_{\text{eff-max}}}{\delta_{\text{eff-max}}} \quad \text{as taken from equation (4-89)}$$

The effective diffusion coefficient $D_{\text{eff-max}}^*$ is derived
from the Nernst-Planck relationship and the film coefficient,
$k_{\text{eff-max}}$, is calculated from the Carberry film coefficient
$D_{\text{eff-max}}^*$ correlation using $D_{\text{eff-max}}$ for the major flux species.
This film "thickness" is utilized to compute the film
coefficients of the other ionic species using equation (4-90)
and (4-91). The pseudo electric field model was tested against
the rigorously derived Kataoka and Yoshida correlation (50) for
homovalent exchange and found to yield comparable results in
subsequent simulations. The pseudo electric field film model
was judged adequate for incorporation into the overall
simulation models for both batch and fixed bed cases.

5. **Overall Simulation Model - Partial Differential Equations**

The general ternary system equations were developed for the
resin phase (equations (5-48) and (5-50)) and the electrolyte
phase (equations (7-13) and (7-14)), along with the appropriate boundary conditions described in Chapters V and VII. The equations describing flux of species through the film linked the two non-linear parabolic equations describing the electrolyte phase with the two non-linear parabolic equations describing the resin phase through the phase equilibria relationships at the resin-film interface. The models could then be reduced to numerical difference equations and subsequently solved.

6. Numerical Methods and Reduction of Equations to Finite Difference Format

6.1 General Background

Svedberg (125), in his doctoral dissertation involving numerical methods for simulation of fixed bed processes, found that the implicit Crank-Nicholson finite difference method, combined with Lee's quasilinearization technique (59) for resolution of non-linear terms, simulated fixed bed operations successfully for a wide variety of conditions. Svedberg used the Thomas algorithm (13, 125) for solution of algebraic equations having tridiagonal matrices, since the quasilinearization technique reduced the system to a set of linear algebraic equations. Solutions were found to be convergent and stable, providing that grid increments were made fine enough. This method was adopted for this research project.

6.2 Binary and Ternary System Batch Model

The batch model involved solving the resin phase
equations only, and in ternary systems, the combining of 
the continuity equations for two species resulted in 
algebraic coefficient matrices having 7-diagonal 
regularity. An algorithm was derived and tested to invert 
the seven-diagonal matrix in order to obtain solutions for 
the unknown resin phase concentration values. The two 
equations describing the ternary system phase equilibrium 
involved an implicit relationship between the electrolyte 
phase concentrations and the resin surface concentrations. 
Since the particular derivatives were required for 
quasilinearization of terms nonlinear in concentration, 
the use of techniques for differentiation of implicit 
functions (70) was required.

To reduce the quasilinearization complexity, values 
for $D_{ii}$ and $D_{ij}$ and their eight partial derivatives 
with respect to species 1 and 2 concentrations were 
performed at time step $j$ versus $j+1/2$. It was 
subsequently shown that systematic bias introduced by the 
simplification was very small (0.01%). The binary system 
batch model developed in this work was used as a subroutine 
of Dr. A. K. S. Murthy's non-linear regression program for 
time varying system parameter estimation. Several resin 
phase models, (1) concentration driving force, (2) activity 
driving force, and (3) multi-suffix diffusion coefficients 
(Stefan-Maxwell model), were used to simulate experimental 
batch data of Rao and his coworkers (92, 93), with ionic 
species diffusion coefficients estimated for a variety of
cases. The model utilizing the activity driving force expressions performed best in these verification trials.

Subsequent verification of the resin phase batch models against prediction of data generated by Rao and his coworkers on ternary system ion exchange gave qualitative indication that the activity driving force model performed best.

6.3 Binary and Ternary System Fixed Bed Model

The equations describing the fluid phase concentration change in ternary systems could be reduced to finite difference equations which, when combined, resulted in matrix equations yielding a regular 5-diagonal coefficient matrix. An algorithm for inversion of this type of matrix equation was derived and tested. Coupling of the resin phase matrix equations to the fluid phase matrix equations was performed through the film boundary condition representing equilibrium with the resin surface. The 7-diagonal algorithm factors containing fluid phase concentration vectors in terms of resin surface concentration vectors could be carried over into the 5-diagonal algorithm containing resin surface concentration vectors, and substituted to eliminate resin phase concentrations at each column node. The fluid phase equations could now be solved for fluid phase concentrations at each node down the packed bed. These fluid phase concentrations could then be used back in the 7-diagonal algorithm to solve for the resin phase
concentrations at each column node and at each resin node. This mode of coupling permitted solution for all concentrations at each iteration, and was repeated after updates on derivatives for non-linear expressions until convergence of non-linear terms had been achieved at each time step.

At this stage, complete fixed bed effluent concentration histories (ECH) could be simulated, and this was performed for the Erickson (26) experimental work on the system NH$_4^+$-ETDA$^{++}$, Dowex 50W X8/Cl$^-$ in order to verify the binary system fixed bed model. In addition, comparison of the rigorous derivation of film field effects by Kataoka and Yoshida (50) against the pseudo electric field model was performed to confirm the efficacy of the latter model.

The diffusion coefficient parameters for the system Ag$^+$-Na$^+$-H$^+$/Dowex 50W X8 researched by Omatete (78) were determined by matching the ECH curves on the six binary system runs. These diffusion coefficients were utilized to simulate the three ternary system runs. Representation of the column exit boundary condition (B.C.) as $\frac{dx_i}{dx} = 0$ gave rise to oscillation in concentration of the column exit. Representation of this B.C. as a CFSTR at the last column increment ameliorated the problem. The CFSTR representation was added to the binary model as well, and all binary simulations were rerun to confirm that no unforeseen problems occurred. The three Omatete ternary
systems were successfully simulated.

Other studies of Crank-Nicholson method stability as a function of time and geometry increment size were performed to provide guidelines in the use of the model.

The fixed bed model for ternary ion exchange and the regression analysis support programs had now been proven and tested against a wide variety of experiment conditions and systems.

D. Results

1. Fluid Phase Material Balance

The generation of the fluid phase material balance equations for ion exchange in ternary systems was straightforward. Addition of axial dispersive terms required information on values of Axial Diffusion Coefficients versus Reynolds number. A number of correlations are plotted on Figure 6-1. However, Erickson (26) had obtained data on dispersion of NH$_4^+$ ion in aqueous solutions in fixed beds by pulsing concentration, and a model was developed in this work to fit the Erickson experimental dispersion curves. Values of axial dispersion coefficients of 3-7 cm$^2$/sec obtained from the parameter fitting process were plotted on Figure 6-1 and are an order of magnitude higher than values derived from other correlations. A value of 4 cm$^2$/sec was used in simulating the Erickson binary system fixed bed experimental ion exchange ECH curves with good results.

Use of the axial dispersion coefficient values within the range indicated on Figure 6-1 for the Omatete (78) experimental
runs gave satisfactory fits to the 9 ECH curves which he generated experimentally. Resin phase diffusion coefficients which were close to literature values for Ag⁺, Na⁺, and H⁺ were used in the simulations. Use of higher values for axial dispersion coefficients than those derived from Figure 6-1 resulted in poor simulations of the effluent concentration histories. Consequently, this investigator recommends that Figure 6-1 correlations be used when actual data are not available for simulation with the dispersion model developed in this work.

2. Phase Equilibria

2.1 Gibbs Free Energy Equilibrium Constraints

There are many equilibrium conventions which have been used in defining ion exchange equilibrium. The rational thermodynamic equilibrium constant was selected for this work. The resin phase concentrations can be represented as mole fraction or equivalent fraction concentration units and, in this form, most of the ion exchanger selectivity is vested in the equilibrium constant since the standard states for the resin and electrolyte phases are different. The activity coefficients for species in the resin phase "trim up" the equilibrium correlations. In systems for which the total number of moles is a constant, the Gibb-Duhem relationship is given for constant temperature and pressure as:

\[
\bar{c}_1 \cdot \Delta n_1 + \bar{c}_2 \cdot \Delta n_2 + \ldots \bar{c}_n \cdot \Delta n_n + \frac{\Delta \Gamma}{RT} = 0
\]
In heterovalent ion exchange, \( \Delta C_T \neq 0 \), and the activity coefficients estimated for such systems must "absorb" this effect, since all of the equilibrium models for ion exchange assume \( \Delta C_T \) to be zero. The activity relationships resulting from Gibbs free energy constraints in a ternary system provide that one of the equilibrium constants is derivable from the other two. In correlation of ternary system data, this relationship (chain rule given by equation (2-19)) was used to provide an indication of the data consistency.

2.2 Electrolyte Phase Activity Coefficient Model

The Bromley equation (9) for the extended Debye-Huckel equation for prediction of activity coefficients of ionic species in aqueous solutions of salts was selected. It was tested in this work against literature values given by Robinson and Stokes (106) for activity coefficients on 17 salts at concentrations up to 5-6 molal. The averaged root mean square of % relative difference, actual to predicted, was 2.7% for the combined total of all 17 salts. In addition, parameters to extend the concentration range have been estimated by Bromley (9) for a wide variety of salts, and can be utilized to predict single ion activities in mixed salts, obviating the need to use Harned's Rule (105) as an approximation in this research.

2.3 Resin Phase Activity Coefficient Models

Three models were tested for ability to model the resin phase activity coefficient as a function of
composition. These were the three suffix Redlich-Kister, the Wilson, and the NRTL equations. Each model has the ability to predict ternary system data from the binary system parameters.

2.4 Development of Equilibrium Models and Model Evaluation

The three models described above were incorporated into the rational thermodynamic equilibrium constant along with the Bromley equation (9) for the electrolyte phase activity coefficients. A non-linear regression program created by Dr. A. K. S. Murthy allowed parameters to be estimated for explicit algebraic relationships utilizing as the objective function the global minimum of the sum of squares of normalized difference between experimental and observed data points. Experimental data on thirteen binary systems (26, 91, 116, 121) and four ternary systems (91, 116, 121) were regressed to yield equilibrium parameter estimates. The three suffix Redlich-Kister equation, using equivalent fraction concentration units, was selected as the most accurate model based on its ability to predict ternary data points using binary system parameters. In some regression runs using the Wilson equation the molar volume ratio \( \rho_{ij} \) was included as a parameter to be estimated. In the case of regression runs using the NRTL equation, the constant \( \alpha_{ij} \) was included as an estimated parameter. Fixed values of \( \rho_{ij} = 1.0 \) and \( \alpha_{ij} = 0.3 \) gave overall better results for those two models than were obtained when these constants were included as estimated.
parameters. The results for the Erickson experiments (26) are shown in Figure 2-7. Table 2-2 compiles the summary information relevant to the experiments performed to obtain equilibrium data on ternary ion exchange systems, and the associated binary systems. Omatete's data (78) was not used in the equilibrium model selection process, so the experimental data used for model validation involved 4 ternary systems and 12 associated binary systems. Table 2-3 gives the comparative results of the equilibrium model simulations performed in this work using the three suffix Redlich-Kister, the Wilson, and the NRTL equations. Standard deviations based on % normalized difference between the experimental and calculated values are tabulated for each case.

2.5 Preparation of Equilibrium Model for Use in Overall Ion Exchange Model

The expressions representing equilibria in ternary systems are implicit in nature, and particular derivatives are required in the quasilinearization of these expressions, so methods (70) for differentiation of implicit functions were employed to meet this requirement. Additionally, a Newton-Raphson method was developed to solve simultaneously the two implicit non-linear equilibrium expressions to yield the composition of each ionic species in the electrolyte phase in equilibrium with the composition of species at the resin surface, based on the fitted parameters.
3. **Transport Resistance**

3.1 **Film Diffusion**

Most investigators (26, 50, 78, 83, 131, 132) utilize the infinite dilution values of diffusion coefficients of single ions in electrolyte solutions. Using the Nernst-Hartley expression (equation (3-11)) in both an integral and differential form, and utilizing the Bromley equation to predict activity coefficient behavior as a function of concentration, diffusion coefficient values at concentration levels up to 3-4 molal were predicted and compared to experimental data given by Robinson and Stokes (106) for 17 salts. The integral version was found to be best, yielding a root mean square of % normalized difference of 13.2%, averaged for all 17 salts, as compared to 26.6% for the differential version. The integral version was even better in the range up to 1.5 molal which was the upper limit for ion exchange simulations performed in this work. The integrated version of the Nernst-Hartley equation was used in the dynamic models to correct individual ionic species diffusion coefficients for concentration effects.

As described in Section C, part 4, of this chapter, a pseudo electric field model was adopted for both the binary and ternary system models based on Carberry's correlation (11) for film coefficient in packed beds above Reynold's number equal to 1. Integration of the ionic flux equation across a film of "thickness", $\delta_{\text{eff}}$, yields concentration
gradients in the film which are nonlinear, as was derived for the case of binary systems in Chapter IV (equation (4-65)). Table 4-1 indicates maximum possible error in using the linear gradient model versus the more rigorous nonlinear case. These errors could approach 30% in heterovalent exchange if the concentration difference across the film was very high. In actual simulation of ion exchange in heterovalent binary systems, the film concentration difference values across the film were so low as to yield negligible error in applying the simpler linear driving force model. The pseudo electric field model based on linear driving force was tested against the rigorous electric field model derived by Kataoka and Yoshida (50) for homovalent exchange and was found to give comparable results. This is shown in Table 4-2 and Figures 7-7 and 7-10 for the Ag⁺-H⁺ system. The calculated data given in Table 4-2 for concentration profiles and mass flux values, compared over time, show little difference between the two models.

The effectiveness of the pseudo electric field model was demonstrated further in the two Erickson (26) experiments involving the favorable exchange of ETDA⁺⁺ for NH₄⁺ on Dowex 50W X8 resin. Figures 7-2 and 7-3 show that the simulated ECH (effluent concentration history) curves fit the experimental data very well, although the two runs had Reynolds numbers differing by a ratio of two (20 and 40). Prior to use of the pseudo
electric field model, film coefficients derived from all cited literature correlations had failed to allow successful simulation of the two ECH curves with any given film coefficient relationship. This was true whether the diffusion coefficient used in any of the film correlations was based on the slowest ion, the average, the harmonic mean, or the fastest diffusing ion. The pseudo electric field model using Carberry's (11) film coefficient correlation for packed beds resulted in the successful simulations shown in Figures 7-2 and 7-3, all parameters being the same except for the Reynolds numbers.

3.2 Intraparticle Diffusion

The inclusion of activity driving force was a key feature of the development of the Nernst-Planck model for binary and ternary exchange systems in this research. The results are described below. The ionic flux equations describing the transport of ions in the resin phase were derived in molar concentration units, then transformed to equivalent fraction concentration units for ease of mathematical manipulation through the relationship

$$\bar{x}_i = \frac{\bar{c}_i \cdot |z_i|}{Q_r}$$

where \( \bar{x}_i \) for species \( i \) is in equivalent fraction units, \( \bar{c}_i \) is the molar concentration of species \( i \), \( |z_i| \) represents the species valence, and \( Q_r \) is the equivalent concentration of fixed charge moieties in the resin/liter. If the chemical potential, \( \bar{u}_i^c \), is used as driving force instead of concentration then, in non-ideal systems, the effective diffusion
coefficient which describes ion exchange in binary systems contains the additional term

\[ \frac{D_1 \cdot D_2}{2} \cdot \sum_{i=1}^{\frac{z_1}{x_1}} \frac{\bar{D}_i}{x_i \cdot x_{i+1}} \left( \frac{z_2}{\bar{Y}_2} \right) \]

Since all of the equilibrium parameter correlations were performed using activity coefficient models in which the concentration units were either mole fractions or equivalent fractions, activity coefficient corrections are required for mole fraction concentration units to reflect the change in standard state to the molar basis as given in the expression above. These transformations are discussed in Chapter II, section F.

4. Numerical Representation of the Ternary System Ion Exchange Model

The selection of the Implicit Crank-Nicholson Finite Difference method, along with quasilinearization of non-linear terms, is discussed in section C, part 6, of this chapter. As described, use of a non-linear regression program developed by Dr. A. K. S. Murthy for time varying relationships allowed regression estimation of diffusion coefficients for Sr\(^{++}\), Mn\(^{++}\), Cs\(^{+}\) and Na\(^{+}\), utilizing the data of Rao and his coworkers (92, 93). Table 5-3 compares the normalized sample variance, \(s^2\), for models involving (1) concentration or (2) activity driving force and (3) a two suffix diffusion coefficient similar to the Stefan-Maxwell model (30). These
regressions were performed for data sets consisting of 5, 6, and 7 binary systems, and the fitted values of diffusion coefficients with their standard deviations are listed. The activity driving force model is elegant in its ability to use single values for so called "self diffusion" coefficients and yet yield the best overall results. Heretofore, investigators (46, 89) have found that the "self diffusion" coefficient for a given counterion species has a different value depending on the nature of the other counterion being exchanged. Figures 5-2 and 5-3 show the plots of experimental data versus the predicted values, using the activity driving force model.

5. Ternary Model - Batch System

The ternary ion exchange data for tagged single particle experiments conducted by Rao and his coworkers (92, 93) were simulated with the Ternary Batch Model using activity driving force, concentration driving force, and some limited work with the two suffix diffusion coefficient. Table 5-4 gives a subjective quality rating to the observed results, and the activity driving force model was rated somewhat better than the others. The plots of experimental versus simulated time varying concentration curves are shown in Figures 5-4 to 5-8. This is the first ternary ion exchange model in which activity driving forces have been utilized, and the results are gratifying.

6. Binary and Ternary Fixed Bed Models

The results of simulating nine binary ion exchange experiments conducted in fixed beds are shown on Figures 7-2 to 7-4 and Figures 7-5 to 7-10. The runs were made utilizing
concentration driving force, since the ternary system batch model which included activity driving force has not been extended to the fixed bed model. The activity driving force was used in the model for a fixed bed in binary systems, and Figure 7-3 shows the effect of use of activities in a positive deviation system (\(\text{NH}_4^+-\text{ETDA}^{4+}\)), while Figure 7-9 shows the effect in a negative deviation system (\(\text{Ag}^+-\text{Na}^+\)). Generally the fixed bed simulation results are quite good, given the use of literature correlations for all parameters except for resin phase diffusion coefficients. The "ground rule," with respect to the fit of the latter parameter, was that the values must be not far different than reported literature values, and that the same values for each species be used in all simulations involving that species.

The simulation results of three ternary system experimental runs by Omatete (78) are shown in Figures 7-11, 7-12, and 7-13. The equilibrium data and diffusion coefficient values derived from the work on the binary systems were used unchanged in simulating the ternary experiments. Values for diffusion coefficients which yielded best ECH curve fits during simulations performed on this system were

\[
\bar{D}_{\text{Ag}} = 0.9 \times 10^{-6} \text{ cm}^2/\text{sec}, \quad \bar{D}_{\text{Na}} = 2.5 \times 10^{-6} \text{ cm}^2/\text{sec}, \text{ and} \\
\bar{D}_{\text{H}} = 10 \times 10^{-6} \text{ cm}^2/\text{sec}. 
\]

The coefficients for \(\text{Ag}^+\) and \(\text{Na}^+\) are close to literature values of \(0.68 \times 10^{-6}\) and \(2.05 \times 10^{-6}\) respectively on Dowex 50W X8, as reported by Hering and Bliss (46).
A. Conclusions

1. Phase Equilibria

It has been shown in this research that ion exchange equilibria can be correlated very well in homovalent or heterovalent binary systems using the rational thermodynamic equilibrium constant in combination with the following activity coefficient models. Electrolyte phase molal activity coefficients can be estimated adequately for most salts in aqueous solution by the Bromley (9) version of the extended Debye-Huckel equation. Bromley (9) has published values of his short range interaction parameter, $B$, for 175 salts. In tests performed in this work the model proved to provide accurate estimates of electrolyte phase activity coefficients for salts in concentrations up to 6 molal. The Bromley equation (9) is available in a form which permits estimation of the molal activity coefficients of single ionic species in mixed electrolytes. The incorporation of the Bromley equation into the phase equilibria models, developed and tested in this work, precluded any need to include electrolyte phase activity coefficient parameters in the correlation of ion exchange equilibrium data.

Three models were used to represent the relationship of activity coefficient of each ionic species in an ion exchange
resin as a function of its concentration: the three suffix Redlich-Kister, the Wilson, and the NRTL equations. All three models can give adequate representation of a single binary system. However, in a ternary system, the Gibbs free energy constraints require that one of the three rational thermodynamic equilibrium constants be predictable from the other two. With this restriction, the three suffix Redlich-Kister equation was clearly superior to the others, as shown in Table 2-3.

The ability of the three models to predict equilibrium compositions for ternary systems based on parameters derived from the binary systems was good in the one cationic homovalent ternary system used in this work, and only fair in the two cationic heterovalent ternary systems. This is indicated by the standard deviations of % relative difference shown in Table 2-3 for these systems and models. Correlation results were also good with the one heterovalent anionic ternary system tested, as shown in Table 2-3. The three suffix Redlich-Kister model, utilizing equivalent fraction concentration units, yielded overall best results in predicting ternary system equilibria from binary system parameters. The poorer results with heterovalent systems are believed to be caused by a non-zero d\(\bar{c}\)\(_{\text{total}}\) (\(\bar{c}\)\(_{\text{total}}\)=total moles of ionic species per unit volume of ion exchange resin) term in the Gibbs-Duhem expression. This residual would then be "absorbed" into the equilibrium constant and activity
coefficient parameters during regression analysis. Recombination of these binary system parameters to predict ternary equilibria in heterovalent systems will lead to poorer estimations in areas of the phase diagram close to the homovalent ionic pair (where the "absorbed factors" are not needed to describe the system).

Equivalent fraction concentration units gave better results than mole fraction units in the prediction of ternary system equilibria, as is shown in Table 2-4. The number of equivalents in an ion exchange resin is constant, and the rational equilibrium constant based on equivalent fractions reflects a linear concentration pattern history during heterovalent exchange. The mole fraction of a species in a heterovalent exchange is not a linear function of the number of moles of that species exchanged during the process. A rational equilibrium constant based on mole fraction units would represent a system having a greater degree of non-linearity when contrasted to equivalent fraction representation. This non-linearity effect must be compensated for in the resin phase activity coefficient parameters. Recombination of such parameters would give poorer predictions of ternary system equilibria, as was indicated in Table 2-4.

2. **Fluid Phase Material Balance**

   It is concluded from the research performed in this work that the axial dispersive term should be included in
equations describing the electrolyte phase in order to adequately represent the unsteady state concentration profiles in fixed beds. The generalized correlations of axial diffusion coefficients, as depicted in Figure 6-1, should yield reasonable predictions of ECH (effluent concentration history) curves over a wide range of Reynolds number for liquid systems. If experimental axial dispersion data is available for the ion exchange system under investigation, the axial dispersion model developed in this research should be used to obtain values of axial dispersion coefficient for use in overall system modeling.

3. Diffusion in the Film

The pseudo electric field model based on the Nernst-Planck equation, as derived in this work compares favorably with the rigorously derived electric field model. This was the case even when the ratio of diffusion coefficients for exchanging species was large, and under conditions of unfavorable exchange. As Pan and David (83) point out, the conditions described above should produce a high electric field gradient, a commensurate high coion gradient in the film, and a large deviation from film mass transfer coefficients calculated without consideration of electric field effects. Kataoka and Yoshida (50) indicate that errors of up to 30% can occur in prediction of breakthrough curve (ECH) and column utilization efficiency if a film coefficient model is used which does not account for
electric field effects.

Although unfavorable heterovalent exchanges, such as those involving $H^+$, are unlikely to occur in industrial practice, simulations of the types of exchange which would give rise to these deviations were performed in this work. Diffusion in the resin phase controls the magnitude of ionic species flux throughout most ion exchange processes (except for very dilute electrolyte concentrations). The ionic species which diffuses most slowly through the resin phase is normally the species most preferred by the ion exchanger. Therefore in unfavorable exchange in a heterovalent system, the resin phase flux is influenced by the slower more favored ion, and its interfacial equilibrium concentration in the electrolyte film would be low due to the favorable selectivity the ion exchanger has for it. Since the film flux must equal the resin phase flux, the concentration gradients across the film are low along with commensurately lower coion and electric field gradients.

Simulations carried out in this work showed that film concentration gradients were low in systems such as the exchange of $\text{Sr}^{++}$ for $\text{Cs}^+$ as shown in Table 4-2. If the counterion gradients are small, the coion gradient is low, and the pseudo electric field model yields results equivalent to rigorous models such as developed by Kataoka and Yoshida (50) for homovalent exchange. This result can be seen in Figures 7-7 and 7-10. The pseudo electric field model can be
extended readily to mulicomponent systems, whereas the rigorous models cannot.

4. **Diffusion in the Resin Phase**

   The use of the Nernst-Planck model to predict interdiffusivity of species within the ion exchange resin is now well established. Investigators have performed experiments designed to isolate this effect from coupled film diffusion. Two mechanisms are normally beneficial: (1) use of concentrated solutions to increase gradients, and (2) high electrolyte phase velocities to minimize film "thickness."

   The experiments have generally been run in "batch" type apparatus, i.e., where the change in electrolyte phase concentration remains "unchanged" during the exchange. Studies performed in fixed beds do not adequately discriminate between the Fick's law models and the Nernst-Planck model since ECH or "S" curves can be represented readily by many mathematical models.

   The clear advantage which the Nernst-Planck model has over Fick's Law is that a single pair of "self" diffusion coefficients can be used to predict both favorable and unfavorable exchange (whereas with Fick's law a different diffusion coefficient pair is required to model each exchange direction).

   The use of activity driving force in conjunction with the Nernst-Planck model gave the best representation of the ion exchange results obtained by Rao and his coworkers, as
given in Table 5-3 and shown in Figures 5-2 and 5-3. Activity driving force, used with the Nernst-Planck model, may be shown to be a unifying ion exchange theorem. Single and unique "self" diffusion coefficients for ionic species could be used to characterize ion exchange in a given resin without need for ion pair specific values or ion pair corrector models, such as the Stefan-Maxwell model (30).

5. Overall Ternary Ion Exchange Model

This research project has shown that a practical model has been developed for simulation of ternary ion exchange in fixed beds which can be used to optimize process design applications. The model utilizes recognized published correlations for estimation of all parameters required, except those specific to the ion exchange system being investigated. Parameters which must be determined by experimental measurement include the equilibrium parameters and the resin phase "self" diffusion coefficients for each species being exchanged.

Simple laboratory experiments utilizing tagged particle techniques or other experimental methods as outlined in Helfferich's publication, "Ion Exchange," (44) can yield data for use in the regression analysis programs developed in this research project. These regression programs can yield parameter estimates for design, to map areas for further experimentation to fill in gaps, or to resolve questionable experimental results.
The use of the Implicit Crank-Nicholson finite difference method, using Lee's quasilinearization technique (59) for resolving non-linear terms, is a robust solution technique in terms of convergence and stability. Faster solution times may be desirable in order to tie the ion exchange model into a larger optimization model for process plant design, such as ASPEN, or use it as a subroutine in regression analysis for parameter estimation based on ECH curve experimental data, and recommendations will be made below to address this issue.

The model as now constituted is for stoichiometric exchange. However it was shown in this work that non-stoichiometric exchange resulting from complex formation can be simulated. In the systems involving Ag\(^+\)-NO\(_3^–\), the stable complex AgNO\(_3\) is formed and adsorbs into the resin phase. Assuming that the physical adsorption of AgNO\(_3\) takes place as fast as the ion exchange process, satisfactory system simulations can be realized by adjusting the equivalent capacity of the ion exchange resin upward to account for the additional Ag\(^+\) uptake as AgNO\(_3\).

The ion exchange dynamic model could be modified to include physical adsorption of complexed species, but the quasilinearization algebra, size of matrices, and computational time presently would limit this to a single complex for practical purposes. The same conclusion could be drawn with regard to addition of a reaction term to the resin
phase material balance, or enthalpy terms to handle adiabatic ion exchange processes for which there is a significant heat of reaction. The latter case would occur when the ion exchanged reacts with coion in the electrolyte phase accompanied by a substantial heat of reaction (ex., H\(^+\) and OH\(^-\)), particularly when there has been significant coion intrusion into the resin. The model could be readily modified to a pore diffusion model if that feature were desirable to model ion exchange in macroreticular resins (relatively large pore openings).

B. Recommendations

1. Phase Equilibria

   Although the phase equilibria model used in correlating binary and ternary ion exchange system data was satisfactory for binary systems, the ability of the model to predict ternary equilibria from binary parameters was only fair in heterovalent systems. As pointed out by Soldatov (119), the Gibbs-Duhem expression leading to thermodynamic consistency in models of excess free energy must take into account the net change of moles in heterovalent exchange. This should be examined in future work.

   Other excess free energy models for equilibrium relationships in ion exchange resins should be proposed and examined, including a Debye-Hückel type model with terms to extend its applicability to higher concentrations. At higher concentrations of electrolyte in a resin, more resin swelling occurs and the resin osmotic pressure changes are greater.
during the ion exchange. Most investigators have assumed that the resin volume and pressure changes are small during ion exchange and is neglected by setting the Gibbs-Duhem equation equal to zero \( \partial(\pi V)_T = 0 \). In fact, significant resin volume changes occur and the osmotic pressure build up can be quite large. Although satisfactory correlation of binary system phase equilibria was possible with the above stated assumption, this assumption may be in part responsible for the poorer ability of the equilibria models to predict ternary equilibria. The impact of the constant pressure volume assumption should be assessed.

The phase equilibrium correlation package used in this work allowed for one charged ionic complex to exist in the electrolyte phase. However, the exchange selectivity of the charged complexed species was assumed to be zero since the existence in the resin of alternate complex forms of an ionic species is not distinguished readily by experimental procedures. Models which would correlate the activities of three ionic species plus a charged complex form of one of the species would be a quaternary model. It would be valuable to know whether any of the models reviewed and tested in this research would provide adequate prediction of quaternary systems. This should be examined in future phase equilibria model development for ion exchange systems.

Finally, resins are available with varying degrees of cross linking, and this effect has a pronounced influence on ionic species selectivity. It would be important to see if
predictive trends can be established in the parameters which characterize equilibrium ion exchange, i.e., $K_{ij}$ and the activity coefficient parameters, which would portray them as functions of the degree of crosslinking in the resin.

2. Dynamics of Ion Exchange

One key observation which requires follow-on verification is that, by use of activity driving force, single "self" diffusion coefficients can characterize ion exchange in binary systems without recourse to multisuffix diffusion coefficients or use of different values for various ionic pair combinations. If activity "corrects" for hydration shells around specific ions and other charge potential effects, the verification of this observation on a wider spectrum of ionic systems is required. As discussed above in connection with equilibria, it would be valuable to determine if the same "self" diffusion coefficient could be used for resins with increasing degrees of crosslinking, providing that the activity differences were known for each ionic species within the various resins having different degrees of crosslinking. Based on concentration driving force models, the measured diffusion coefficients decrease dramatically as crosslinking is increased, and perhaps this primarily is an activity related effect.

The model developed in this work utilized the Implicit rank-Nicholson numerical method. The State Variable (65) and Collocation (29) methods should be evaluated for potential improvement in computation speed commensurate with equivalent
convergence and stability performance. The method of Characteristics (2, 22, 58) is an attractive alternative, but its use limits the user to asymptotic solutions which infer stable self-sharpening elution or loading patterns (not both) which have existed within the column long enough to have reached a stable configuration. Effects of uneven bed presaturation which exist in many practical applications could not be evaluated with the above method, nor could unfavorable exchange processes.

Finally, the model developed in this research simulated fixed bed operations successfully based on data derived from 1" diameter glass columns. Obviously, scale-up effectiveness has yet to be proven in an industrial application. This will most likely not be within the purview of future academic investigators, but result from actual industrial applications.

3. Other Potential Applications

With modification, the model could be utilized to simulate isothermal adsorption processes. The required changes would include a pore diffusion model to describe diffusion in the particle, and equilibrium relationships such as the Langmuir or Freundlich models. Substantial changes would be required, but the basic partial differential equations and numerical solution techniques should be reviewed for applicability to biotechnology operations involving reactions induced by immobilized enzymes or bacterial cultures within porous beads or gels.
Appendix A

EFFECT OF CONCENTRATION UNITS ON ACTIVITY COEFFICIENT RELATIONSHIPS IN AN ION EXCHANGER

A. Molar versus Equivalent Fraction Concentration Units

The free energy relationship for the exchange of ionic species in an ion exchange resin are as follows for the concentration units of molar versus equivalent fraction:

\[ \bar{G}^\circ_1 = \bar{G}^\circ_{\text{molar}} + RT \ln(c_1 \cdot \bar{n}_{1\text{molar}}) \] \hspace{1cm} (A-1)

\[ \bar{G}^\circ_1 = \bar{G}^\circ_{\text{eqf}} + RT \ln(x_1 \cdot \bar{n}_{1\text{eqf}}) \] \hspace{1cm} (A-2)

Setting (A-1) and (A-2) equal to each other results in:

\[ \ln \left( \frac{c_1}{x_1} \right) + \frac{\left( \bar{G}^\circ_{\text{molar}} - \bar{G}^\circ_{\text{eqf}} \right)}{RT} = \ln \left( \frac{\bar{n}_{1\text{eqf}}}{\bar{n}_{1\text{molar}}} \right) \] \hspace{1cm} (A-3)

From equation (2-14), the following limits are observed:

as \( \bar{x}_1 \rightarrow 1.0 \), \( \bar{n}_{1\text{eqf}} \rightarrow 1.0 \), and \( \bar{n}_{1\text{molar}} \rightarrow 1.0 \) (A-4)

Likewise, as \( \bar{c}_1 \rightarrow |z_{1\text{eqf}}| = \bar{c}_T \), \( \bar{n}_{1\text{molar}} \rightarrow 1.0 \), and \( \bar{n}_{1\text{molar}} \rightarrow |z_{1\text{molar}}| \)

Solving equation (A-3) at the limits gives the free energy difference due to concentration units.

\[ \ln \left( \frac{Q_r}{|z_{1\text{molar}}|} \right) = - \frac{\left( \bar{G}^\circ_{\text{molar}} - \bar{G}^\circ_{\text{eqf}} \right)}{RT} + \ln \left( \frac{1}{1} \right) \]

and,

\[ \ln \left( \frac{Q_r}{|z_{1\text{eqf}}|} \right) = - \frac{\left( \bar{G}^\circ_{\text{molar}} - \bar{G}^\circ_{\text{eqf}} \right)}{RT} \] \hspace{1cm} (A-5)
substituting (A-5) into (A-3) yields:

\[
\ln \left( \frac{n_{1}}{x_{1}} \right) - \ln \left( \frac{Q_{r}}{z_{1}} \right) = \ln \left( \frac{n_{Y_{1}}}{n_{Y_{1}} \text{ molar}} \right)
\]

(A-6)

but \( x_{1} = \frac{c_{1} \cdot |z_{1}|}{Q_{r}} \)

so, it is obvious from (A-6) that:

\[
\left( \frac{n_{Y_{1}}}{n_{Y_{1}} \text{ molar}} \right) = 1
\]

(A-7)

This relationship applies at all concentrations. This can be proven at the other concentration extreme where \( x_{1} \) and \( c_{1} \) approach zero.

Looking at Figure 2-2, and equations (2-15) and (2-16), the following construct is possible:

Then for \( x_{1} \), as in (2-15) and (2-16),

\[
\frac{n_{Y_{1}}}{n_{Y_{1}} + c_{1}} = \frac{n_{Y_{1}}}{n_{Y_{1}} + c_{1}} = \frac{K}{n_{f_{1} + o}}
\]

Using the same treatment for molar units at the identical \( n_{f_{1} + o} \) point,

\[
K = \frac{\left( \frac{n_{f_{1} + o}}{c_{1} + o \left( \frac{Q_{r}}{z_{1}} \right)} - 0 \right)}{c_{1} + o \left( \frac{Q_{r}}{z_{1}} \right) - 0} = \frac{n_{f_{1} + o} \cdot Q_{r}}{c_{1} + o \cdot |z_{1}|}
\]

(A-8)
and,
\[
\bar{n}_{f_{1}+0} = n_{f_{1}+0} \left( \frac{-c_{1} + 0}{c_{1 \text{ total}}} \right) = n_{f_{1}+0} \left( \frac{-c_{1} + 0}{Q_{r} \frac{z_{1}}{|z_{1}|}} \right) \tag{A-9}
\]

so,
\[
\bar{n}_{f_{1}+0} \text{ molar} = \frac{K \cdot \bar{n}_{f_{1}+0} \cdot \frac{|z_{1}|}{Q_{r}}}{n_{f_{1}+0}} = \frac{K}{n_{f_{1}+0}} \tag{A-10}
\]

Substitution of \( \bar{n}_{f_{1}+0} \) and \( \bar{n}_{f_{1}+0} \text{ molar} \) in equation (A-3) leads to equation (A-5) since:
\[
\ln \left( \frac{\bar{n}_{f_{1}+0}}{\bar{n}_{f_{1}+0} \text{ molar}} \right) = \ln \left( \frac{K}{n_{f_{1}+0}} \right) = 0 \text{ , and the proof of } \bar{n}_{f_{1}+0} = \bar{n}_{f_{1}+0} \text{ molar}
\]
is made at the other concentration extreme.

B. Molar versus Mole Fraction Concentration Units

The relationship between molar units and mole fraction units is given as:
\[
\bar{x}_{f_{1} \text{ mf}} = \frac{c_{1}}{Q_{r} \left( \frac{|z_{1}|}{z_{1}} \right) + (1 - VR) \cdot c_{1}} \tag{A-11}
\]

where \( VR = \frac{|z_{1}|}{|z_{2}|} \)

Using the free energy relationship given by equation (A-1) in conjunction with:
\[
\bar{G}^{0} = \bar{G}_{f_{1} \text{ mf}}^{0} + RT \ln(\bar{x}_{f_{1} \text{ mf}} \cdot \bar{n}_{f_{1}+0}) \tag{A-12}
\]
then,
\[
\ln \left( \frac{\overline{c}_1}{x_{1\text{mf}}} \right) = - \frac{\left( \bar{G}_1^\prime - \bar{G}_{1\text{mf}}^\prime \right)}{RT} + \ln \left( \frac{n_{\overline{Y}_1\text{mf}}^-}{n_{\overline{Y}_1\text{molar}}} \right) \tag{A-13}
\]

Defining the limits as was done in equation (A-4),
\[
\overline{x}_{1\text{mf}} \rightarrow 1.0, \quad n_{\overline{Y}_1\text{mf}}^- \rightarrow 1.0, \quad \text{and} \quad n_{a_1\text{mf}}^- \rightarrow 1.0 \quad (A-14)
\]
and equation (A-4) gives the limits in molar units. Substitution in equation (A-13) gives:
\[
\ln \left( \frac{Q_r}{z_1} \right) = - \frac{\left( \bar{G}_1^\prime - \bar{G}_{1\text{mf}}^\prime \right)}{RT} \quad (A-15)
\]
Substitution of equation (A-15) in equation (A-13) yields:
\[
\ln \left( \frac{\overline{c}_1}{x_{1\text{mf}}} \right) - \ln \left( \frac{Q_r}{z_1} \right) = \ln \left( \frac{n_{\overline{Y}_1\text{mf}}^-}{n_{\overline{Y}_1\text{molar}}} \right) \quad (A-16)
\]
but,
\[
\overline{x}_{1\text{mf}} = \frac{\overline{c}_1}{\left( \frac{Q_r}{z_1} + (1-VR)\overline{c}_1 \right)}
\]
so,
\[
\left( \frac{\overline{c}_1}{c_1} \right) \cdot \left( \frac{Q_r}{z_1} \right) + (1-VR)\overline{c}_1 \cdot \left( \frac{z_1}{Q_r} \right) = \left( \frac{n_{\overline{Y}_1\text{mf}}^-}{n_{\overline{Y}_1\text{molar}}} \right) \quad (A-17)
\]
therefore,
\[
n_{\overline{Y}_1\text{molar}}^- = n_{\overline{Y}_1\text{mf}}^- \cdot \left( \frac{1}{VR + (1-VR)\overline{x}_1} \right) \quad (A-18)
\]
noting that \( \overline{x}_1 \) is equivalent fraction since
\[
\overline{x}_1 = \frac{\overline{c}_1 \cdot |z_1|}{Q_r}
\]
A like derivation can be made for $n^{-}_{Y_{2\ molar}}$ versus $n^{-}_{Y_{2\ mf}}$ in that:

$$\frac{-x_{2\ m}}{z_{2\ m}} = \frac{c_{2}}{\left(\frac{Q_{r}}{|z_{2\ m}|} + (1-\frac{1}{VR})c_{2}\right)} \quad (A-19)$$

so,

$$\ln \left(\frac{-c_{2}}{-x_{2\ m}}\right) - \ln \left(\frac{Q_{r}}{|z_{2\ m}|}\right) = \ln \left(\frac{n^{-}_{Y_{2\ mf}}}{n^{-}_{Y_{2\ molar}}}\right) \quad (A-20)$$

Substituting equation (A-19) in (A-20) gives:

$$\left(\frac{Q_{r}}{|z_{2\ m}|} + (1-\frac{1}{VR})c_{2}\right) \cdot \left(\frac{|z_{2\ m}|}{Q_{r}}\right) = \frac{n^{-}_{Y_{2\ mf}}}{n^{-}_{Y_{2\ molar}}} \quad (A-21)$$

which leads to:

$$n^{-}_{Y_{2\ molar}} = n^{-}_{Y_{2\ mf}} \cdot \left(\frac{1}{VR + (1-\frac{1}{VR})x_{2}}\right) \quad (A-22)$$

Checking the relationship at the limit where $x_{1\ mf} \rightarrow 0$

The lower limits are then defined as:

$$n^{-}_{Y_{1\ molar}}+o = n^{-}_{Y_{1\ molar}}+o = \frac{K}{n^{-}_{f_{1}} \cdot Q_{r}} \left(\frac{|z_{1\ m}|}{z_{2\ m}}\right) \quad (A-23)$$
since,

\[
k' = \frac{n_{f_1}'}{c_1 + o}, \quad \text{and} \quad n_{f_1}'' = \left( \frac{c_1}{c_1 + o} \right) \cdot n_{f_1}''
\]

it follows that:

\[
n_{Y_1}''_{\text{molar}} = \left( \frac{K_{\text{mf}}}{n_{f_1} + o} \right) \cdot \left( \frac{Q_r}{l_{z_1} \cdot |z_2|} \right) = \frac{K_{\text{mf}}}{n_{f_1} + o} \cdot \left( \frac{Q_r}{l_{z_1} \cdot |z_2|} \right) \quad (A-24)
\]

Performing the same procedure in mole fraction units,

\[
k_{\text{mf}} = n_{f_1} + o \cdot \left( \frac{|z_2|}{x_{1 \text{mf}} + o} \right) \cdot \left( 1 + x_{1 \text{mf}} + o \cdot (VR-1) \right) = \frac{n_{f_1} + o}{x_{1 \text{mf}} + o} \cdot \left( \frac{|z_2|}{Q_r} \right)
\]

since \( x_{1 \text{mf}} + o = 0 \) as it approaches 0

and,

\[
n_{f_1}'' = n_{f_1}'' \cdot x_{1 \text{mf}} + o
\]

it similarly follows that:

\[
n_{Y_1}''_{\text{mf}} = \frac{n_{f_1}''}{n_{f_1}''} = \frac{K_{\text{mf}}}{n_{f_1}''} \cdot \left( \frac{Q_r}{l_{z_2}} \right) \quad (A-25)
\]

Substitution of equation (A-24) and (A-25) in equation (A-13) leads to:

\[
\ln \left( \frac{x_{1 \text{mf}} + o \cdot Q_r}{(1 + x_{1 \text{mf}} + o \cdot (VR-1))} \right) = - \frac{\bar{g}_{\text{molar}} - \bar{G}_{\text{mf}}}{RT} \left( \frac{K_{\text{mf}}}{n_{f_1}''} \right) \cdot \left( \frac{Q_r}{l_{z_1} \cdot |z_2|} \right)
\]

\[
+ \ln \left( \frac{K_{\text{mf}}}{n_{f_1}''} \right) \cdot \left( \frac{Q_r}{l_{z_1} \cdot |z_2|} \right)
\]
Since \( \bar{x}_{1mf}^{+o} = 0 \), and \( K_{mf} = K \) as the concentration of species 1 approaches zero, then the following expression is derived:

\[
- \left( \frac{\bar{G}_{1molar} - \bar{G}_{1mf}}{RT} \right) = \ln \left( \frac{Q_r}{|z_2|} \right) - \ln \left( \frac{|z_1|}{|z_2|} \right)
\]  

(A-26)

Substitution of equation (A-26) in (A-13) leads to,

\[
\ln \left( \frac{\bar{c}_1}{x_{1mf}} \right) - \ln \left( \frac{Q_r}{|z_1|} \right) = \ln \left( \frac{n_{\gamma 1mf}^{-}}{n_{\gamma 1molar}^{-}} \right)
\]

or,

\[
n_{\gamma 1molar}^{-} = n_{\gamma 1mf}^{-} \cdot \left( \frac{\bar{c}_1 \cdot |z_1|}{x_{1mf} \cdot Q_r} \right)
\]

(A-27)

The expression for \( \bar{x}_{1mf} \) in terms of \( \bar{c}_1 \) is,

\[
\bar{x}_{1mf} = \frac{\bar{c}_1}{\left( \frac{Q_r}{|z_2|} + (1-VR) \cdot \bar{c}_1 \right)}
\]

Substitution for \( \bar{x}_{1mf} \) leads to:

\[
n_{\gamma 1molar}^{-} = n_{\gamma 1mf}^{-} \cdot \left( \frac{1}{VR + (1-VR) \cdot \bar{x}_1} \right)
\]

(A-28)

where \( \bar{x}_1 \) is the equivalent fraction unit.

This expression is identical to equation (A-18).
A. Introduction

The problem of establishing mathematical relationships between a set of dependent variables and a set of independent variables using experimental data is frequently encountered in applied science. For example, it may be necessary to obtain expressions for the exit composition of a chemical reactor in terms of the inlet composition. The functional form of the mathematical relationships, whether they are derived based on fundamental physical and chemical principles or simply selected to give sufficient mathematical flexibility, may be written as:

$$ y_i = f_i(b_k; x_j) $$

(B-1)

where, $x_j$ are the independent variables, $y_i$ the dependent variables, and $b_k$ are parameters. The functional relationships, although shown above as explicit algebraic expressions, can also be implicitly defined in the form of algebraic, differential or integral equations, or even a complex series of calculations performed by computer simulation programs. Parameters are constants whose numerical values change from system to system and are usually not directly measurable. The mathematical process of determining the values of the parameters using experimental x-y data is known as regression analysis. When the functions $f_i$ are such that the partial derivatives ($\partial f_i / \partial b_k$) are independent of $b_k$, linear regression techniques can be used to obtain the values of the parameters. However, many chemical engineering models (mathematical relationships)
are nonlinear with respect to the parameters and require iterative techniques for parameter estimation. The purpose of this report is to describe an algorithm (mathematical recipe) for nonlinear regression and to provide instructions for using three general purpose computer programs based on that algorithm. One program handles explicit algebraic models and the other two differential equation models. In the latter case, one version has built-in integration procedures for ordinary differential equations with initial conditions. The other version requires the user to provide an integration routine for the partial or ordinary differential equations.

B. The Weighted Least Squares Procedure

If \( y_{j\ell}^* \) are experimentally observed values corresponding to a set of independent variables \( x_{j\ell} \) in the \( \ell \)-th observation, the object of regression analysis is to determine the values of the parameters such that the difference between \( y_{j\ell}^* \) and the values of \( y_i \) calculated using \( x_{j\ell} \) in equation (B-1) is a minimum according to some statistically valid criterion. The use of a statistically valid criterion is important since, due to random errors, the exact \( x-y \) relationship cannot be experimentally observed. Consequently, the parameter values obtained from such measurements are subject to uncertainty. A statistical procedure for obtaining the best estimates of the parameters is the method of maximum likelihood (Draper and Smith 1966). The least squares procedure developed by Gauss is the most frequently used for parameter estimation and is the maximum likelihood procedure when the following assumptions are satisfied. Only the dependent variable measurements are subject to error. The
errors are purely random and obey Gaussian statistics with zero mean and known variance.

The least squares method calls for the minimization of the following objective function.

\[ S(b) = \frac{1}{s_i} \sum_{i=1}^{m} \left( \sum_{\ell=1}^{L} (y_{i\ell}^* - y_{i\ell})^2 \right) / s_i^2 \]  

\[ = \sum_{i=1}^{m} \left( \sum_{\ell=1}^{L} \left[ y_{i\ell}^* - f_i(b; x_{j\ell}) \right]^2 \right) / s_i^2 \]

where \( s_i \) is the variance associated with the measurement of \( y_i \), \( L \) is the total number of data points, and \( m \) is the number of dependent variables measured experimentally, or derived from those measurements.

C. The Gauss–Newton Method

An efficient method for finding the minimum of a multidimensional function is due to Newton. His method consists of guessing the location of the minimum, approximating the function by the first three terms of the Taylor series expansion about the guess point, finding the minimum of the approximation and using that minimum point as the next guess. This procedure is repeated until the true minimum is found. If \( b_0 \) is the guess point, the quadratic approximation is

\[ S = S(b_0) + \sum_{k=1}^{P} \frac{\partial S}{\partial b_k} \Delta b_k + 1/2 \sum_{i=1}^{P} \sum_{j=1}^{P} \frac{\partial^2 S}{\partial b_i \partial b_j} \Delta b_i \Delta b_j \]  

(B-3)

where, the function and the derivatives are evaluated at \( b=b_0 \). A set of corrections, \( \Delta b \), which should be added to \( b_0 \) to obtain the minimum of the above approximation is given by

\[ \sum_{j=1}^{P} \frac{\partial^2 S}{\partial b_i \partial b_j} \Delta b_j = -\frac{\partial S}{\partial b_i} \quad i = 1, P \]  

(B-4)
where, \( P \) is the number of parameters in the model. It can be shown that,

\[
\frac{\partial S}{\partial b_k} = -2 \sum_{i=1}^{m} \frac{1}{s_i^2} \sum_{l=1}^{L} [y_{i:l} - f(x_{i:l})] \frac{\partial f_i}{\partial b_k} \quad ; k=1,P \tag{B-5}
\]

and

\[
\frac{\partial^2 S}{\partial b_k \partial b_n} = -2 \left\{ \sum_{i=1}^{m} \frac{1}{s_i^2} \sum_{l=1}^{L} \left[ \frac{\partial f_i}{\partial b_k} \frac{\partial^2 f_i}{\partial b_k \partial b_n} - \frac{\partial f_i}{\partial b_k} \frac{\partial f_i}{\partial b_n} \right] \right\} \quad ; k=1,P \tag{B-6}
\]

If the guess values are good, \((y_{i:l} - f_i)\) should be small and hence

\[
\frac{\partial^2 S}{\partial b_k \partial b_n} = 2 \sum_{i=1}^{m} \frac{1}{s_i^2} \sum_{l=1}^{L} \frac{\partial f_i}{\partial b_k} \frac{\partial f_i}{\partial b_n} \quad ; k=1,P \tag{B-7}
\]

Equation (B-7) is the approximation by Gauss leading to the Gauss-Newton method. When the guess value is near the minimum, this method gives quadratic convergence like Newton's method, but does not require the evaluation of the second derivatives. Gauss-Newton corrections are given by

\[
\sum_{i=1}^{m} \frac{1}{s_i^2} \sum_{l=1}^{L} \frac{\partial f_i}{\partial b_k} \Delta b_n = \sum_{i=1}^{m} \frac{1}{s_i^2} \sum_{l=1}^{L} (y_{i:l} - f_i) \frac{\partial f_i}{\partial b_k} \quad ; k=1,P \tag{B-8}
\]

If \( B_i \) is an \((L \times P)\) matrix who \( k\)-th element is \((\partial f_i / \partial b_k)\) for the \( l\)-th data point, the above equation may be written as

\[
\left\{ \sum_{i=1}^{m} \frac{(B_i^T B_i)}{s_i^2} \right\} \Delta b = \sum_{i=1}^{m} B_i^T e_i / s_i^2 \tag{B-9}
\]

\( m \) = number of dependent variables measured or derived from those measurements,

where, \( e_i \) is a vector whose \( k\)-th element is \((y_{i:k} - f_i)\). Let

\[
G = \sum_{i=1}^{m} B_i / s_i \quad \text{and} \quad r = \sum_{i=1}^{m} e_i / s_i
\]
Then equation (B-9) may be written as
\[ G^T G \Delta b = G^T r \]  

(B-10)

Sometimes it is necessary to analyze several sets of data, obeying different functional relationships but having a common set of parameters. For example, one may attempt to estimate the parameters in a liquid phase excess free energy model using vapor-liquid as well as liquid-liquid equilibrium data. In such cases equation (B-10) becomes
\[
\left( \sum_{i=1}^{N} W_i G_i^T G_i \right) \Delta b = \sum_{i=1}^{N} W_i G_i^T r_i
\]

where \( W_i \) is the weighting factor for the \( i \)-th set and \( G_i \) and \( r_i \) are the \( G \) and \( r \) matrices associated with that set. As defined in the computer program, weight = \( W_i / s_i^2 \).

Both the Newton's method and the Gauss-Newton method are known to be the most efficient algorithms when good guesses at the minimum can be made. They are unreliable if good guesses cannot be made. The author's experience is that this unreliability is due to a few poorly determined directions in applying the Newton's method, and can be rectified by a quadratic surface analysis of the Gauss-Newton matrix \( (G^T G) \) as shown in the next section.

D. A Modified Gauss-Newton Method

The Gauss-Newton method uses the following quadratic approximation of the objective function about the guess point to obtain the corrections to be added to the guess values.
\[
S = S(b_0) - 2 (G^T r) \Delta b = \Delta b^T (G^T G) \Delta b
\]

(B-12)
The nature of the quadratic surface given by equation (B-12) can be determined by the eigenvalues and the eigenvectors of the $G^T G$ matrix. It can be shown that $G^T G$ is a positive definite matrix (all positive eigenvalues) and hence equation (B-12) represents a hyperellipsoid. The eigenvectors give the directions of the axes of the ellipsoid. The lengths of the axes are proportional to the reciprocals of the eigenvalues. Directions corresponding to very small eigenvalues are directions in which the function does not change very rapidly [according to the local approximation given by (B-12)] and hence large changes would be recommended by the Gauss-Newton method. Large corrections given by any local approximation method are likely to be unreliable. The modification of the Gauss-Newton algorithm presented in this report is based on this heuristic argument. The Gauss-Newton corrections are given by

$$\Delta b = (G^T G)^{-1} G^T r$$

(B-13)

If $Q$ is a matrix whose columns are eigenvectors of $G^T G$, then $Q$ is an orthonormal matrix and

$$Q^T (G^T G) Q = L$$

(B-14)

where $L$ is a diagonal matrix containing the eigenvalues of $G^T G$. Readers not familiar with these properties should refer to any textbook on matrix theory covering quadratic forms and unitary transformation. From equations (B-13) and (B-14) it can be seen that

$$\Delta b = (QL^{-1} Q^T) G^T r$$

or

$$\Delta b = \left( \sum_{i=1}^{n} \frac{1}{\nu_i} \nu_i V_i V_i^T \right) G^T r$$

(B-15)
where, $V_i$ is the $i$-th eigenvector and $\lambda_i$ the $i$-th eigenvalue. The above equation gives the total Gauss-Newton correction as a sum of corrections along the eigenvector.

The modified method consists of summing the terms of equation (B-15) in the order of decreasing eigenvalues and, after the addition of each term, checking the cumulative corrections to determine if any of the corrections exceed the allowable step size. When the step size is exceeded, no additional terms are included in the summation. Only a fraction of the last term is used so that no correction exceeds the allowable step size. User supplied scale factors are used to scale the parameters and the corrections to comparable magnitude so that the same step size can be used for all the variables. The step size starts with a user specified value and increases by 10% for every successful step (steps leading to decrease in the objective function) and is cut to half for every unsuccessful step. It is possible to conceive of several other alternate strategies similar to the strategy outlined in this section. A systematic study to determine the best strategy has not been carried out, primarily because the present method has been very satisfactory in a number of practical problems.
RESIN PHASE ION EXCHANGE FITTING PROGRAM

RIONFT (AKM 62)

RIONFT (AKM 62) is a non-linear regression program for estimation of the equilibrium parameters in binary and ternary ion exchange systems, and has the following features:

A. Input module - subroutine M62A

1.0 Input 1 - title

2.0 Input 2 gives the following information in free format:

2.1 the total number of data sets

2.2 the total number of parameters involved

2.3 debug options selected

2.4 non-default values, if desired, for parameter search step size, numerical Jacobian perturbation values, and run time limit

2.5 print control on a variety of computed values giving full, partial, or minimum information

2.6 switching options on selection of $K_{ij}$ to be computed by the chain rule from $K_{ik}$ and $K_{jk}$ values

2.7 choice of resin phase activity coefficient correlation method (i.e., Redlich-Kister, Wilson, or NRTL)

3.0 Input 3

A fit string which allows selection, 0 or 1, of those parameters which are to be fit in a given run. Those parameters not fit can be either zero, or be used in the regression analysis with a preset estimate value.
4.0 Inputs 4 and 5
Involve scale factors and initial estimates for parameters

5.0 Input 6
Allows individual parameters to be bounded by upper and lower values if known

6.0 The data sets are then read in giving, for each set,
6.1 Input 1 gives no. of data points, no. of constants, no. of variables (y and x), no. of dependent variables (y)
6.2 Input 2 - labels for constants
6.3 Input 3 gives the relative weight to be given to each dependent variable (default = 1), and constants
6.4 Input 4 - labels for the variables
6.5 Input 5 - data points (y and x)

B. The subroutine CONTROL performs the sequencing of regression runs 1 through 26 as outlined in Chapter II, para. D, section 2.

C. The evaluation subroutine, M62G, is specifically designed by the user of program AKM62 to compute the dependent and independent variables for user specific equations in response to calls from AKM62 subroutines. The independent variables are computed for each data point utilizing the constants, and the dependent variables are computed for the latest values of the fitted parameters. Equations (2-73) and (2-75) given in Chapter II represent the dependent and independent variables computed at each call. If the data set being regressed is a ternary system, the M62G, as designed for RIONFT, will switch to another part
of the subroutine where the $Y(1)_{\text{exp}}$, $Y(2)_{\text{exp}}$, and $Y(3)_{\text{exp}}$

independent variables and the $Y(1)_{\text{calc}}$, $Y(2)_{\text{calc}}$, and $Y(3)_{\text{calc}}$
dependent variables are calculated, the latter representing the
dependent portions of the three binary pair equilibrium
relationships.

D. **Subroutines** supporting M62G are the resin phase activity
coefficient generators, REDKIS, WILSON, and NRTL, and the
electrolyte phase activity coefficient generator, LIQION. LIQION
also computes the equilibrium composition of a stable complex,
if there is one, along with its effect on ionic strength
including the activity coefficient for the complex species and
the coion.

E. When the regression has met convergence criteria on all fitted
parameters for all sets within one of the 26 controlled runs,
**subroutine M62C** is called, which provides optional printout of the
best value of parameters along with standard deviation and
correlation coefficients between parameters. **M62C** stores the
best parameter values fitted, to be used by CONTROL in setting the
initial estimates for the next run, or in subsequent runs where
these values are required. It optionally will print out a variety
of information related to that run, including the activity
coefficients for each species in the resin and electrolyte phase,
ionic strength, and fraction of complex present if any, for each
data point.

F. Two major **subroutines**, called by the main program after each
of the 26 controlled runs, are TERION and BINION. These
subroutines, TERION for ternary systems, and BINION for binary
systems, compute the equilibrium resin phase composition for each experimental electrolyte composition data point for all sets within a run, on a set-by-set basis. They compute the standard sample deviation of the calculated equilibrium composition from the experimental values for each set within a run, on a % difference and a % relative difference basis. These routines print out the standard deviation information along with optional information including actual data points alongside the calculated points for all species. The resin and electrolyte phase activity coefficients are printed out, along with ionic strength.

The equations describing either binary or ternary equilibria are non-linear because of the relationships describing the activity coefficient correlations. In BINION, a false positioning method is utilized to solve the single equilibrium relationship describing a binary pair, since the function is monotonic. With TERION, a simultaneous solution of two of the equilibrium relationships is required to define the equilibrium concentrations of $x_i$ and $x_j$, with $x_k = 1-x_i-x_j$. The Newton-Raphson method is used in that subroutine to provide solution to the coupled non-linear algebraic equations.
Appendix D

CONCENTRATION EFFECTS ON DIFFUSION COEFFICIENTS OF IONIC SPECIES
IN AQUEOUS SOLUTIONS

A. Diffusion Coefficients of Salts in Aqueous Solution—Infinite Dilution

\[
D_0^+ = 8.931 \cdot 10^{-10} \cdot T \cdot \left( \frac{L_0^+}{L_0^-} \right) \cdot \left( \frac{|z_+| + |z_-|}{z_+ \cdot z_-} \right)
\]  

\((D-1)\)

where:

- \(D_0^+\) = diffusion coefficient of salt as concentration \(\to 0\), cm\(^2\)/sec
- \(L_0^+\) = conductance of cation, mhos/equiv.
- \(L_0^-\) = conductance of anion, mhos/equiv.
- \(\Lambda\) = \(L_0^+ + L_0^-\), salt conductance, mhos/equiv.
- \(T\) = \(K\)
- \(z_+ \cdot z_-\) = charge on ion

\[
R^2 = \frac{8.931}{2} \cdot 10^{-6}, \text{ gas constant/Faraday constant, joule/} K\text{/mole/Coulomb/equiv.}
\]

B. Diffusion Coefficients of Ions in Aqueous Solution—Infinite Dilution

\[
D_0^+ = 8.931 \cdot 10^{-10} \cdot T \cdot \left( \frac{L_0^+}{|z_+|} \right)
\]  

\((D-2)\)

\[
D_0^- = 8.931 \cdot 10^{-10} \cdot T \cdot \left( \frac{L_0^-}{|z_-|} \right)
\]  

\((D-3)\)

combining (D-2) and (D-3) into (D-1) yields:

\[
D_0^+ = D_0^+ \cdot D_0^- \cdot \left( \frac{|z_+ \cdot z_-|}{(|z_+| \cdot D_0^+ + |z_-| \cdot D_0^-)} \right)
\]  

\((D-4)\)
C. Diffusion Coefficient Variation with Electrolyte Concentration

Nernst-Hartley Equation for Dilute Solutions:

\[ D_\pm = D_\pm^0 \left( 1 + c_\pm \cdot \frac{d\ln n_\pm}{dc_\pm} \right) \]  \hspace{2cm} (D-5)

where:

\[ c_\pm = \text{molality (assume equal to molarity at specific gravity of one)} \]

\[ n_\pm = \text{molar activity coefficient of salt} \]

If molality is equal to molarity, then

\[ D_\pm = D_\pm^0 \left( 1 + c_\pm \cdot \frac{d\ln \gamma_\pm}{dc_\pm} \right) \]  \hspace{2cm} (D-6)

where:

\[ \gamma_\pm = \text{molal activity coefficient as given by the Bromley equation.} \]

D. Differential Form of Correction to Diffusion Coefficient for Concentration

If the viscous drag effects on hydrated ions which occur at high electrolyte concentration are ignored, the following treatment can be used to correct \( D_\pm^0 \) for the effect of electrolyte concentration. Using the Bromley equation to relate \( \gamma_\pm (c_\pm) \) to concentration, \( c_\pm \), the Bromley equation can be differentiated with respect to \( c_\pm \), molarity (molality).

1.0 Single Salt in Aqueous Solution

Substitution of Ionic Strength, \( I \), in the Bromley and Nernst-Hartley equations leads to:
If \( \text{norm} = \text{electrolyte normality} \)
\[
\text{norm} = c_-(|z_-|) \quad (D-8)
\]

And,
\[
c_+ = \frac{c_-(|z_-|)}{z_+} = \frac{\text{norm}}{z_+} \quad (D-9)
\]

The molarity of the salt is given by:
\[
c_\pm = \frac{\text{norm}}{|z_+ \cdot z_-|} \quad (D-10)
\]

Differentiating (D-10) with respect to \( c_\pm \) and
substituting in equation (D-6) leads to:
\[
D_\pm = D_\pm^0 \left(1 + \text{norm} \frac{d \ln \gamma_\pm}{d(\text{norm})}\right) \quad (D-11)
\]

By differentiating the Bromley equation, the following
expression is obtained:
\[
\frac{d \ln \gamma_\pm}{d(\text{norm})} = 2.303 \left[ -0.511 \cdot (|z_+ \cdot z_-|) \cdot \frac{(|z_+| + |z_-|)}{2} \right]^{\frac{1}{2}} \times
\frac{2 \cdot \text{norm}^{\frac{1}{2}} \left[ 1 + \left( \frac{1}{2} \cdot \text{norm} \cdot (|z_+| + |z_-|) \right)^{\frac{1}{2}} \right]}{2}
\]

\[
(0.06 + 0.6 \cdot \text{Bia}) \cdot (|z_+ \cdot z_-|) \cdot \left[ 1 - \frac{3}{4} \cdot \left( \frac{|z_+| + |z_-| \cdot \text{norm}}{|z_+ \cdot z_-|} \right) \right] \cdot \frac{(|z_+| + |z_-|)}{2}
\]

\[
+ \left[ 1 + \left( \frac{3}{4} \cdot \text{norm} \cdot (|z_+| + |z_-|) \right)^{\frac{3}{2}} \right]^{\frac{1}{3}} \cdot \text{Bia}
\]

This relationship can be substituted into equation (D-11)
to correct the Diffusion Coefficient of a salt for
electrolyte concentration.
2.0 Cation Relationship with a Single Salt in Aqueous Solution

\[
\frac{d \ln \gamma_+}{d(\text{norm})} = 2.303 \left\{ \frac{-0.511 \cdot z_+^2 \left( \frac{|z_+| + |z_-|}{2} \right)^{1/2}}{2 \cdot \text{norm}^{1/2} \left[ 1 + \left( \frac{1}{2} \cdot \text{norm} \cdot (|z_+| + |z_-|) \right)^{1/2} \right]^2} \right. \\
+ (0.06 + 0.6 \cdot \text{Bia}) \left( |z_+^* \cdot z_-^*| \right) \left[ 1 - \left( \frac{\frac{3}{4} \cdot (|z_+^*| + |z_-^*|) \cdot \text{norm}}{|z_+^* \cdot z_-^*|} \right) \cdot \left( \frac{|z_+| + |z_-|}{2} \right)^2 \cdot \left( \frac{1}{|z_-|} \right) \right] \\
\left. + \text{Bia} \cdot \left( \frac{|z_+| + |z_-|}{2} \right)^2 \cdot \left( \frac{1}{|z_-|} \right) \right\} 
\]

The Nernst-Hartley equation for a single cation is,

\[
D_+ = D_+^0 \left[ 1 + c_+ \cdot \frac{d \ln \gamma_+}{d c_+} \right] 
\]

where:

\[
c_+ = \frac{\text{norm}}{z_+} \quad \text{(D-15), and} \quad \frac{d(\text{norm})}{d c_+} = z_+ \quad \text{(D-16)}
\]

The substitution of equations (D-15) and (D-16) in equation (D-14) yields:

\[
D_+ = D_+^0 \left[ 1 + \text{norm} \cdot \frac{d \ln \gamma_+}{d(\text{norm})} \right] 
\]

Substitution of equation (D-13) in (D-17) gives the relationship for dependency of a single cation diffusion coefficient on electrolyte normality.

Analogous treatment can be used for the anion diffusion coefficient.
3.0 Three Cations in Aqueous Solution with a Single Anion

3.1 The ionic strength is given as,

\[ I = \frac{1}{2} \left[ \frac{3}{\sum_{i=1}^{3} c_+^i \cdot z_+^i} + \frac{2}{c_-^A \cdot z_-^2} \right] \]  

or in equivalent fractions,

\[ x_i = \frac{3}{\sum_{j=1}^{3} c_+^j \cdot z_+^j} = \frac{c_+^i \cdot z_+^i}{c_-^A \cdot (|z_-|)} = \frac{c_+^i \cdot z_+^i}{\text{norm}} \]

for \( i = 1, 2, 3 \)

and equation (D-18) becomes:

\[ I = \frac{1}{2} \cdot \text{norm} \cdot \left[ \frac{3}{\sum_{i=1}^{3} x_i \cdot |z_+^i|} + |z_-^i| \right] \]

for \( i = 1, 2, 3 \)

3.2 Differentiating equation (D-19) yields:

\[ \frac{dx_+^i}{dc_+^i} = \frac{|z_+^i|}{\text{norm}} \]

for \( i = 1, 2, 3 \)

and the Nernst-Hartley equation becomes,

\[ D_+^i = D_+^0 \cdot \left( 1 + x_+^i \cdot \frac{1}{dx_+^i} \right) \]

for \( i = 1, 2, 3 \)

3.3 The Bromley Equation in terms of Equivalent Fraction versus Ionic Strength becomes upon substitution of equation (D-20) in equation (3-4):
It is noted that the major variable in computing $Y_{+i}$ is $I$, the ionic strength, so even if a cation species equivalent fraction approaches zero, its activity coefficient will remain relatively constant in mixed cation solutions. For example, if a system has $z_{+1} = z_{+2} = z_{+3}$, $I$ is constant, and $Y_{+i}$ is a function of norm only. In another example, if a system has $z_{+1} = z_{+2} = 2$ and $z_{+3} = 1$, then relative values of $I$ will vary only between 1 and 1.5 as $x_{+3}$ goes from a concentration value of one to zero.

\[ \kappa n Y_{+i} = 2.303 \times \frac{-0.511 \cdot z_{+i}^2 \cdot \left\{ \frac{1}{2} \cdot \left( \sum_{j=1}^{3} x_{+j} \cdot |z_{+j}| \right) + |z_{-A}| \right\}^{1/2}}{1 + \left( \frac{\text{norm} \cdot \left( \sum_{j=1}^{3} x_{+j} \cdot |z_{+j}| \right) + |z_{-A}|}{\text{norm}} \right)^{1/2}} \]  

\[ + (0.06 + 0.6 \cdot \text{Bia}) \cdot (|z_{+i} \cdot z_{-i} - z_{-A}|) \times \left( \frac{\left( |z_{+i} + |z_{-i}| \right)^2}{2} \right) \times \left( \frac{\text{norm}}{|z_{-A}|} \right) \]

\[ + \frac{3}{4} \cdot \text{norm} \left( \frac{\sum_{j=1}^{3} x_{+j} \cdot |z_{+j}| + |z_{-A}|}{|z_{+i} \cdot z_{-i} - z_{-A}|} \right)^2 \]

\[ + \text{Bia} \cdot \left( \frac{\left( |z_{+i} + |z_{-i}| \right)^2}{2} \right) \times \left( \frac{\text{norm}}{|z_{-A}|} \right) \]  

for $i = 1, 2, 3$
It is noted that the major variable in computing $\gamma_{+i}$ is $I$, the ionic strength, so even if a cation species equivalent fraction approaches zero, its activity coefficient will remain relatively constant in mixed cation solutions. For example, if a system has $z_{+1} = z_{+2} = z_{+3}$, $I$ is constant, and $\gamma_{+1}$ is a function of $\text{norm}$ only. In another example, if a system has $z_{+1} = z_{+2} = 2$ and $z_{+3} = 1$, then relative values of $I$ will vary only between 1 and 1.5 as $x_{+3}$ goes from a concentration value of one to zero.

\[
\ln \gamma_{+i} = 2.303 \left[ -0.511 \cdot z_{+i}^2 \cdot \left( \frac{1}{2} \cdot \frac{\sum_{j=1}^{3} x_{+j} \cdot |z_{+j}| + |z_{-A}|}{\text{norm}} \right)^{\frac{1}{2}} \right] \cdot \text{norm}^{\frac{1}{2}}
\]

\[
+ \frac{(0.06 + 0.6 \cdot \text{Bia}) \cdot \left| z_{+i} \cdot z_{-A} \right| \cdot \left( \frac{|z_{+i}| + |z_{-A}|}{2} \right)^2 \cdot \left( \frac{\text{norm}}{|z_{-A}|} \right)}{\left[ 1 + \frac{3 \cdot \text{norm} \cdot \left( \sum_{j=1}^{3} x_{+j} \cdot |z_{+j}| + |z_{-A}| \right)}{(|z_{+i}| - |z_{-A}|)^2} \right]^{\frac{1}{1}}}
\]

\[
+ \text{Bia} \cdot \left( \frac{|z_{+i}| + |z_{-A}|}{2} \right)^2 \cdot \left( \frac{\text{norm}}{|z_{-A}|} \right)
\]

for $i = 1, 2, 3$.
3.4 Equation (D-23) must be differentiated in order to use the relationship in the Nernst—Hartley equation. This expression is:

\[
\frac{d\ln N_+^i}{d(norm)} = 2.303 \cdot \left( -0.511 \cdot z_{+i}^2 \cdot \frac{1}{2} \left[ \sum_{j=1}^{3} x_{+j} \cdot |z_{+j}| + |z_{-A}| \right] \right)^{1/2} \left( 1 + \frac{\text{norm} \cdot \left[ \sum_{j=1}^{3} x_{+j} \cdot |z_{+j}| + |z_{-A}| \right]}{2} \right)^{1/2}
\]

\[
(0.06 + 0.6 \cdot \text{Bia}) \cdot (|z_{+i}| \cdot |z_{-A}|) \cdot \left( 1 - \frac{3}{4} \cdot \text{norm} \cdot \left[ \sum_{j=1}^{3} x_{+j} \cdot |z_{+j}| + |z_{-A}| \right] \right) \left( |z_{+i}| \cdot |z_{-A}| \right)^3 + \text{Bia}
\]

\[
\cdot \left( \frac{|z_{+i}|^2 + |z_{-A}|^2}{2} \right) \cdot \left( \frac{1}{|z_{-A}|^2} \right)
\]

for \( i = 1, 2, 3 \)

3.5 Equation (D-19) can be differentiated to give:

\[
\frac{d(norm)}{d c_{+i}} = \frac{|z_{+i}|}{x_{+i}} \tag{D-25}
\]

Then the Nernst—Hartley equation becomes:

\[
D_{+i} = D_{+i}^0 \cdot \left( 1 + \text{norm} \cdot \frac{d\gamma_{+i}}{d(norm)} \right) \tag{D-26}
\]

for \( i = 1, 2, 3 \)
E. Integral Form of Correction to Diffusion Coefficients for Concentration

Most salts show the following variation of mean activity coefficient $\gamma_\pm$, with $c_\pm$ (molarity = molality):

$$\gamma_\pm$$

The differential form of the Nernst-Hartley equation would hold up to point a, but not at higher concentrations.

1.0 Single Salt in Aqueous Solution (D-27)

$$D_\pm = D_\pm^0 \left[ 1 + \ln \gamma_\pm - \left( \frac{2 \cdot B \cdot (|z_+| \cdot |z_-|)}{A^2 \cdot c_\pm} \right) \cdot \left( \frac{1}{2} \left( 1 + A \cdot c_\pm^{\frac{1}{2}} \right)^2 \right) \right.
- 2 \cdot \left( 1 + A \cdot c_\pm^{\frac{1}{2}} \right) + \ln \left( 1 + A \cdot c_\pm^{\frac{1}{2}} \right) + 1.5 \left. \right]$$

$$- \left( \frac{F}{G^2 \cdot c_\pm} \right) \cdot \left( \frac{1}{(1 + G \cdot c_\pm)} + \ln \left( 1 + G \cdot c_\pm \right) - 1 \right) - \frac{E \cdot c_\pm}{2}$$

where:

- $c_\pm = \text{molality} \approx \text{molarity of salt}$
- $A = \left( \frac{1}{2} \cdot (|z_+| \cdot |z_-|) \cdot (|z_+| + |z_-|) \right)^{\frac{1}{2}}$
- $B = -2.303 \cdot 0.511 = -1.1768$

continued
2.0 Three Cations in Aqueous Solution with a Single Anion  

\[ D_{+i} = D_{+i}^0 \left( 1 + \gamma_{+i} - \left( \frac{2 \cdot B \cdot z_{+i}^2}{2 \cdot \ln(z_{+i} + 1)} \right) \cdot \left( 1 + A \cdot \text{norm}^2 \right) \right) \]

\[ - 2 \cdot \left( 1 + A \cdot \text{norm}^2 \right) + \ln\left( 1 + A \cdot \text{norm}^2 \right) + 1.5 \]

\[ - \left( \frac{F}{G \cdot \text{norm}} \right) \cdot \left( \frac{1}{1 + G \cdot \text{norm}} + \ln(1 + G \cdot \text{norm}) - 1 \right) \]

\[ - \frac{E \cdot \text{norm}}{2} \]

for \( i = 1, 2, 3 \)

where:

\[ \text{norm} = c_\text{A} \cdot \left( |z_{-\text{A}}| \right) \]

\[ A = \left( \frac{1}{2} \cdot \left( \sum_{j=1}^{3} x_{+j} \cdot |z_{+j}| + |z_{-\text{A}}| \right) \right)^{1/2} \]

\[ B = -1.1768 \]

\[ F = (0.034 + 0.34 \cdot \text{Bia}) \cdot (z_{+i} \cdot z_{-\text{A}}) \cdot (|z_{+i}| + |z_{-\text{A}}|)^2 \cdot \frac{1}{|z_{-\text{A}}|} \]

continued
Equation (D-28) can be used for three anions with a single cation by obvious juxtaposition of the terms.

F. Summary of Results

Calculations were performed to predict the activity coefficients and the diffusion coefficients for 17 salts, and the results were compared to experimental data given by Robinson and Stokes(106). The molalities of the solutions ranged from 0 to 6.0, and the resulting output from the computations is shown in the following pages. However, a summary of results is as follows:

\[
E = 0.5758 \cdot \text{Bio} \cdot (|z_+| + |z_-|)^2 \cdot \left(\frac{1}{|z_A|}\right)
\]

\[
G = 0.75 \cdot \left[ \left( \sum_{i=1}^{3} x_i \cdot |z_{+i}| \right) + |z_{-A}| \right] \left( |z_{+i} \cdot z_{-A} | \right)
\]

Equation (D-28) can be used for three anions with a single cation by obvious juxtaposition of the terms.

<table>
<thead>
<tr>
<th>% Relative Difference</th>
<th>Activity Coefficients, Bromley Equation</th>
<th>Diffusion Coefficients, Nernst-Hartley</th>
</tr>
</thead>
<tbody>
<tr>
<td></td>
<td>2.7 %</td>
<td>26.6 %</td>
</tr>
<tr>
<td></td>
<td></td>
<td>13.2 %</td>
</tr>
<tr>
<td></td>
<td>- differential form</td>
<td></td>
</tr>
<tr>
<td></td>
<td>- integral form</td>
<td></td>
</tr>
</tbody>
</table>
This program evaluates the effect of concentration on the activity and diffusion coefficients of salts using the Bromley equation and the Nernst-Hartley equation for diffusivity. 

\[ D(c) = D(0) \cdot (1 + c \cdot \frac{d(\ln(\gamma))}{dc}) \]  

Dilute solutions

<table>
<thead>
<tr>
<th>Concentration</th>
<th>Activity Coeff.</th>
<th>Diffusion Coeff.</th>
</tr>
</thead>
<tbody>
<tr>
<td>0.100000</td>
<td>0.778000</td>
<td>0.148300E-04</td>
</tr>
<tr>
<td>0.200000</td>
<td>0.735000</td>
<td>0.147500E-04</td>
</tr>
<tr>
<td>0.300000</td>
<td>0.710000</td>
<td>0.147500E-04</td>
</tr>
<tr>
<td>0.500000</td>
<td>0.681000</td>
<td>0.147400E-04</td>
</tr>
<tr>
<td>0.700000</td>
<td>0.667000</td>
<td>0.147500E-04</td>
</tr>
<tr>
<td>1.000000</td>
<td>0.657000</td>
<td>0.148400E-04</td>
</tr>
<tr>
<td>1.500000</td>
<td>0.656000</td>
<td>0.149500E-04</td>
</tr>
<tr>
<td>2.000000</td>
<td>0.668000</td>
<td>0.151600E-04</td>
</tr>
<tr>
<td>3.000000</td>
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<tr>
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</tr>
<tr>
<td>5.000000</td>
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<td>0.159000E-04</td>
</tr>
</tbody>
</table>

Aqueous solution of compound NaCl

Comparison of activity coefficients: actual vs. predicted by Bromley equation

<table>
<thead>
<tr>
<th>Concentration</th>
<th>Actual</th>
<th>Predicted</th>
<th>Difference-%</th>
</tr>
</thead>
<tbody>
<tr>
<td>0.100000</td>
<td>0.778000</td>
<td>0.776447</td>
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<tr>
<td>0.200000</td>
<td>0.735000</td>
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<tr>
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<td>0.500000</td>
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<tr>
<td>1.500000</td>
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<tr>
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<td>0.871853</td>
<td>0.359644</td>
</tr>
</tbody>
</table>

Average difference = 0.0028  
Root mean square-difference = 0.0032

Average percent difference = 0.3812  
Root mean square-percent difference = 0.4443
### COMPARISON OF DIFFUSION COEFFICIENTS—ACTUAL VS. PREDICTED BY INTEGRAL VERSION OF Nernst-Hartley EQUATION

<table>
<thead>
<tr>
<th>CONCENTRATION</th>
<th>ACTUAL</th>
<th>PREDICTED</th>
<th>DIFFERENCE-%</th>
</tr>
</thead>
<tbody>
<tr>
<td>0.100000</td>
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<td>0.857423</td>
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<td>0.148312E-04</td>
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</tr>
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AVERAGE DIFFERENCE= 0.78941E-06  
AVERAGE PERCENT DIFFERENCE= 5.0516  
ROOTMEAN SQUARE-DIFFERENCE= 0.12427E-05  
ROOTMEAN SQUARE-PERCENT DIFFERENCE= 7.8495

### COMPARISON OF DIFFUSION COEFFICIENTS—ACTUAL VS. PREDICTED BY DIFFERENTIAL VERSION OF Nernst-Hartley EQUATION

<table>
<thead>
<tr>
<th>CONCENTRATION</th>
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<th>PREDICTED</th>
<th>DIFFERENCE-%</th>
</tr>
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<tbody>
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</table>

AVERAGE DIFFERENCE= 0.23251E-05  
AVERAGE PERCENT DIFFERENCE= 14.8663  
ROOTMEAN SQUARE-DIFFERENCE= 0.36440E-05  
ROOTMEAN SQUARE-PERCENT DIFFERENCE= 23.0543
COMPOUND NABR

This program evaluates the effect of concentration on the activity and diffusion coefficients of salts using the Bromley equation and the Nernst-Hartley equation for diffusivity:

\[ D(c) = D(0)*(1+c*d(ln(GAMA))/dc); \]

Dilute solutions

VALENCE OF COMP.A  
VALENCE OF ANION  
CATION A CONDUCTANCE-Mhos/Equiv.-298 K, INFINITE DILUTION 50.1000  
ANION CONDUCTANCE-Mhos/Equiv.-298 K, INFINITE DILUTION 78.1400  
BROMLEY EQ. SALT PARAMETER - CATION A, COMMON ANION 0.074900

Nernst LIMITING DIFFUSIVITY-CM**2/SEC 0.16250E-04

DIFFUSIVITY CALCULATED FROM CONDUCTANCES-CM**2/SEC 0.16280E-04

NUMBER OF DEPENDENT VARIABLES 3

NUMBER OF DATA POINTS 9

<table>
<thead>
<tr>
<th>CONCENTRATION</th>
<th>ACTIVITY COEFF.</th>
<th>DIFFUSION COEFF.</th>
</tr>
</thead>
<tbody>
<tr>
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<td>0.151700E-04</td>
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<td>0.200000</td>
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<tr>
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<td>0.170200E-04</td>
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</table>

AQUEOUS SOLUTION OF COMPOUND NABR

COMPARISON OF ACTIVITY COEFFICIENTS ACTUAL VS. PREDICTED BY BROMLEY EQUATION

<table>
<thead>
<tr>
<th>CONCENTRATION</th>
<th>ACTUAL</th>
<th>PREDICTED</th>
<th>DIFFERENCE-%</th>
</tr>
</thead>
<tbody>
<tr>
<td>0.100000</td>
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AVERAGE DIFFERENCE= 0.0009   AVERAGE PERCENT DIFFERENCE= 0.1280

ROOT MEAN SQUARE-DIFFERENCE= 0.0010   ROOT MEAN SQUARE-PERCENT DIFFERENCE= 0.1470
### Comparison of Diffusion Coefficients - Actual vs. Predicted by Integral Version of Nernst-Hartley Equation

<table>
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<th>Actual</th>
<th>Predicted</th>
<th>Difference-%</th>
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<td>0.170200E-04</td>
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**Average Difference** = 0.19741E-06  
**Average Percent Difference** = 1.2331

**Root Mean Square Difference** = 0.25073E-06  
**Root Mean Square Percent Difference** = 1.5626

### Comparison of Diffusion Coefficients - Actual vs. Predicted by Differential Version of Nernst-Hartley Equation

<table>
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<tr>
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<th>Predicted</th>
<th>Difference-%</th>
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**Average Difference** = 0.98608E-06  
**Average Percent Difference** = 5.9275

**Root Mean Square Difference** = 0.16165E-05  
**Root Mean Square Percent Difference** = 9.6116
COMPOUND NAI

THIS PROGRAM EVALUATES THE EFFECT OF CONCENTRATION ON THE ACTIVITY AND DIFFUSION COEFFICIENTS OF SALTS USING THE BROMLEY EQUATION AND THE Nernst-Hartley EQUATION FOR DIFFUSIVITY:

\[ D(c) = D(0) \cdot (1 + c \cdot \frac{d\ln(GAMA)}{dc}) \]; DILUTE SOLUTIONS

VALENCE OF COMP.A 1
VALENCE OF ANION 1
CATION A CONDUCTANCE-MHOS/EQIV.-298 K,INFINITE DILUTION 50.1000
ANION CONDUCTANCE-MHOS/EQIV.-298 K,INFINITE DILUTION 76.8000
BROMLEY EQ. SALT PARAMETER - CATION A, COMMON ANION 0.099400
Nernst LIMITING DIFFUSIVITY-CM**2/SEC 0.16140E-04
DIFFUSIVITY CALCULATED FROM CONDUCTANCES-CM**2/SEC 0.16255E-04
NUMBER OF DEPENDENT VARIABLES 3
NUMBER OF DATA POINTS 10

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<th>DIFFUSION COEFF.</th>
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AQUEOUS SOLUTION OF COMPOUND NAI

COMPARISON OF ACTIVITY COEFFICIENTS-ACTUAL VS. PREDICTED BY BROMLEY EQUATION

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<th>DIFFERENCE-%</th>
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AVERAGE DIFFERENCE= 0.0032 AVERAGE PERCENT DIFFERENCE= 0.4026
ROOT MEAN SQUARE-DIFFERENCE= 0.0039 ROOT MEAN SQUARE-PERCENT DIFFERENCE= 0.4927
### COMPARISON OF DIFFUSION COEFFICIENTS-ACTUAL VS. PREDICTED BY INTEGRAL VERSION OF Nernst-Hartley EQUATION

<table>
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<tr>
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AVERAGE PERCENT DIFFERENCE= 3.2328
ROOTMEAN SQUARE-DIFFERENCE= 0.63188E-06
ROOTMEAN SQUARE-PERCENT DIFFERENCE= 3.5654

### COMPARISON OF DIFFUSION COEFFICIENTS-ACTUAL VS. PREDICTED BY DIFFERENTIAL VERSION OF Nernst-Hartley EQUATION

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AVERAGE PERCENT DIFFERENCE= 7.4404
ROOTMEAN SQUARE-DIFFERENCE= 0.21310E-05
ROOTMEAN SQUARE-PERCENT DIFFERENCE= 11.0415
COMPOUND KCL

THIS PROGRAM EVALUATES THE EFFECT OF CONCENTRATION ON THE ACTIVITY AND DIFFUSION COEFFICIENTS OF SALTS USING THE BROMLEY EQUATION AND THE Nernst-Hartley EQUATION FOR DIFFUSIVITY

\[ D(c) = D(0) \times (1 + c \times d(\ln(\gamma))/dc) \]; DILUTE SOLUTIONS

<table>
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<tr>
<th>VALENCE OF COMP. A</th>
<th>VALENCE OF ANION</th>
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<th>ANION CONDUCTANCE-MHOS/EQIV.-298 K, INFINITE DILUTION</th>
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BROMLEY EQ. SALT PARAMETER - CATION A, COMMON ANION

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DIFFUSIVITY CALCULATED FROM CONDUCTANCES-CM**2/SEC

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<th>NUMBER OF DATA POINTS</th>
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AQUEOUS SOLUTION OF COMPOUND KCL

COMPARISON OF ACTIVITY COEFFICIENTS - ACTUAL VS. PREDICTED BY BROMLEY EQUATION

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<th>DIFFERENCE-%</th>
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AVERAGE DIFFERENCE= 0.0006
ROOT MEAN SQUARE-DIFFERENCE= 0.0008

AVERAGE PERCENT DIFFERENCE= 0.0924
ROOT MEAN SQUARE-PERCENT DIFFERENCE= 0.1150
### COMPARISON OF DIFFUSION COEFFICIENTS-ACTUAL VS. PREDICTED BY INTEGRAL VERSION OF Nernst-Hartley EQUATION

<table>
<thead>
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<th>CONCENTRATION</th>
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<th>DIFFERENCE-%</th>
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<tr>
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<tr>
<td>4.000000</td>
<td>0.219600E-04</td>
<td>0.192482E-04</td>
<td>12.348989</td>
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</table>

**Average Difference:** 0.13338E-05  
**Average Percent Difference:** 6.5115  
**Root Mean Square Difference:** 0.16365E-05  
**Root Mean Square Percent Difference:** 7.8328

### COMPARISON OF DIFFUSION COEFFICIENTS-ACTUAL VS. PREDICTED BY DIFFERENTIAL VERSION OF Nernst-Hartley EQUATION

<table>
<thead>
<tr>
<th>CONCENTRATION</th>
<th>ACTUAL</th>
<th>PREDICTED</th>
<th>DIFFERENCE-%</th>
</tr>
</thead>
<tbody>
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**Average Difference:** 0.72945E-06  
**Average Percent Difference:** 3.7223  
**Root Mean Square Difference:** 0.75595E-06  
**Root Mean Square Percent Difference:** 3.8625
COMPOUND KBR

THIS PROGRAM EVALUATES THE EFFECT OF CONCENTRATION ON THE ACTIVITY AND DIFFUSION COEFFICIENTS OF SALTS USING THE BROMLEY EQUATION AND THE Nernst-Hartley EQUATION FOR DIFFUSIVITY

\[ D(c) = D(0) \times (1 + c \times d(\ln(\Gamma)) / dc) \]; DILUTE SOLUTIONS

| VALENCE OF COMP.A | 1 |
| VALENCE OF ANION | 1 |
| CATION A CONDUCTANCE-MHOS/EQIV.-298 K, INFINITE DILUTION | 73.5000 |
| ANION CONDUCTANCE-MHOS/EQIV.-298 K, INFINITE DILUTION | 78.1400 |
| BROMLEY EQ. SALT PARAMETER - CATION A, COMMON ANION | 0.02960 |
| Nernst LIMITING DIFFUSIVITY-CM**2/SEC | 0.20160E-04 |
| DIFFUSIVITY CALCULATED FROM CONDUCTANCES-CM**2/SEC | 0.20246E-04 |
| NUMBER OF DEPENDENT VARIABLES | 3 |
| NUMBER OF DATA POINTS | 12 |

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<tr>
<th>CONCENTRATION</th>
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<th>DIFFUSION COEFF.</th>
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<tr>
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<td>0.693000</td>
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<td>0.188500E-04</td>
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<tr>
<td>0.700000</td>
<td>0.636000</td>
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AQUEOUS SOLUTION OF COMPOUND KBR

COMPARISON OF ACTIVITY COEFFICIENTS-ACTUAL VS. PREDICTED BY BROMLEY EQUATION

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<tr>
<th>CONCENTRATION</th>
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<th>PREDICTED</th>
<th>DIFFERENCE-%</th>
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AVERAGE DIFFERENCE= 0.0032  AVERAGE PERCENT DIFFERENCE= 0.5009
ROOT MEAN SQUARE-DIFFERENCE = 0.0035  ROOT MEAN SQUARE-PERCENT DIFFERENCE = 0.5724
### COMPARISON OF DIFFUSION COEFFICIENTS—ACTUAL VS. PREDICTED BY INTEGRAL VERSION OF Nernst-Hartley EQUATION

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<table>
<thead>
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**AVERAGE DIFFERENCE=** 0.19522E-05  
**ROOTMEAN SQUARE-DIFFERENCE=** 0.24363E-05  
**AVERAGE PERCENT DIFFERENCE=** 8.8688  
**ROOTMEAN SQUARE-PERCENT DIFFERENCE=** 10.7663

### COMPARISON OF DIFFUSION COEFFICIENTS—ACTUAL VS. PREDICTED BY DIFFERENTIAL VERSION OF Nernst-Hartley EQUATION

<table>
<thead>
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<table>
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**AVERAGE DIFFERENCE=** 0.11413E-05  
**ROOTMEAN SQUARE-DIFFERENCE=** 0.12122E-05  
**AVERAGE PERCENT DIFFERENCE=** 5.4063  
**ROOTMEAN SQUARE-PERCENT DIFFERENCE=** 5.6569
**COMPOUND KI**

This program evaluates the effect of concentration on the activity and diffusion coefficients of salts using the Bromley equation and the Nernst-Hartley equation for diffusivity:

\[ D(c) = D(0) \times (1 + c \times d(\ln(\Gamma)) / dc) \]

**Dilute Solutions**

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<th>Activity Coeff.</th>
<th>Diffusion Coeff.</th>
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Aqueous solution of Compound KI

Comparison of Activity Coefficients - Actual vs. Predicted by Bromley Equation

<table>
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<tr>
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<th>Predicted</th>
<th>Difference-%</th>
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Average Difference = 0.0085
Average Percent Difference = 1.2806
Root Mean Square Difference = 0.0091
Root Mean Square Percent Difference = 1.3838
### COMPARISON OF DIFFUSION COEFFICIENTS - ACTUAL VS. PREDICTED BY INTEGRAL VERSION OF Nernst-Hartley EQUATION

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AVERAGE DIFFERENCE = 0.23307E-05  
AVERAGE PERCENT DIFFERENCE = 10.2982

ROOTMEAN SQUARE-DIFFERENCE = 0.28352E-05  
ROOTMEAN SQUARE-PERCENT DIFFERENCE = 12.1770

### COMPARISON OF DIFFUSION COEFFICIENTS - ACTUAL VS. PREDICTED BY DIFFERENTIAL VERSION OF Nernst-Hartley EQUATION

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<th>DIFFERENCE-%</th>
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AVERAGE PERCENT DIFFERENCE = 6.0002

ROOTMEAN SQUARE-DIFFERENCE = 0.13863E-05  
ROOTMEAN SQUARE-PERCENT DIFFERENCE = 6.3834
COMPOUND LICL

THIS PROGRAM EVALUATES THE EFFECT OF CONCENTRATION ON THE ACTIVITY AND DIFFUSION COEFFICIENTS OF SALTS USING THE BROMLEY EQUATION AND THE Nernst-Hartley EQUATION FOR DIFFUSIVITY

\[ D(c) = D(0) \times (1 + c \times d(ln(GAMA))/dc) \]; DILUTE SOLUTIONS

VALENCE OF COMP.A 1
VALENCE OF ANION 1
CATION A CONDUCTANCE-MHOS/EQIV.-298 K, INFINITE DILUTION 38.680
ANION CONDUCTANCE-MHOS/EQIV.-298 K, INFINITE DILUTION 76.350
BROMLEY EQ. SALT PARAMETER - CATION A, COMMON ANION 0.128300
Nernst LIMITING DIFFUSIVITY-CM**2/SEC 0.13660E-04
DIFFUSIVITY CALCULATED FROM CONDUCTANCES-CM**2/SEC 0.13669E-04
NUMBER OF DEPENDENT VARIABLES 3
NUMBER OF DATA POINTS 11

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AQUEOUS SOLUTION OF COMPOUND LICL

COMPARISON OF ACTIVITY COEFFICIENTS - ACTUAL VS. PREDICTED BY BROMLEY EQUATION

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ROOT MEAN SQUARE-DIFFERENCE= 0.0183  ROOT MEAN SQUARE-PERCENT DIFFERENCE= 1.8264
COMPARISON OF DIFFUSION COEFFICIENTS—ACTUAL VS. PREDICTED BY INTEGRAL VERSION OF Nernst-Hartley EQUATION

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AVERAGE PERCENT DIFFERENCE= 10.7078
ROOTMEAN SQUARE-PERCENT DIFFERENCE= 13.9512

COMPARISON OF DIFFUSION COEFFICIENTS—ACTUAL VS. PREDICTED BY DIFFERENTIAL VERSION OF Nernst-Hartley EQUATION

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ROOTMEAN SQUARE-PERCENT DIFFERENCE= 38.6105
COMPOUND LIBR

THIS PROGRAM EVALUATES THE EFFECT OF CONCENTRATION ON THE ACTIVITY AND DIFFUSION COEFFICIENTS OF SALTS USING THE BROMLEY EQUATION AND THE Nernst-Hartley EQUATION FOR DIFFUSIVITY

\[ D(c) = D(0)(1 + c \cdot \frac{d(\ln(GAMA))}{dc}) \]; DILUTE SOLUTIONS

VALENCE OF COMP.A 1
VALENCE OF ANION 1
CATION CONDUCTANCE-MHOS/EQIV.-298 K, INFINITE DILUTION 38.6800
ANION CONDUCTANCE-MHOS/EQIV.-298 K, INFINITE DILUTION 78.1400
BROMLEY EQ. SALT PARAMETER - CATION A, COMMON ANION 0.152700
Nernst LIMITING DIFFUSIVITY-CM**2/SEC 0.13770E-04
DIFFUSIVITY CALCULATED FROM CONDUCTANCES-CM**2/SEC 0.13869E-04
NUMBER OF DEPENDENT VARIABLES 3
NUMBER OF DATA POINTS 11

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AQUEOUS SOLUTION OF COMPOUND LIBR

COMPARISON OF ACTIVITY COEFFICIENTS - ACTUAL VS. PREDICTED BY BROMLEY EQUATION

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AVERAGE PERCENT DIFFERENCE= 3.3102
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### COMPARISON OF DIFFUSION COEFFICIENTS-ACTUAL VS. PREDICTED BY INTEGRAL VERSION OF Nernst-Hartley EQUATION

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### COMPARISON OF DIFFUSION COEFFICIENTS-ACTUAL VS. PREDICTED BY DIFFERENTIAL VERSION OF Nernst-Hartley EQUATION

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ROOTMEAN SQUARE-DIFFERENCE= 0.56140E-05  ROOTMEAN SQUARE-PERCENT DIFFERENCE= 34.5350
THIS PROGRAM EVALUATES THE EFFECT OF CONCENTRATION ON THE ACTIVITY AND DIFFUSION COEFFICIENTS OF SALTS USING THE BROMLEY EQUATION AND THE Nernst-Hartley EQUATION FOR DIFFUSIVITY

\[ D(c) = D(0)*(1 + c*d(ln(GAMA))/dc) \]; DILUTE SOLUTIONS

| VALENCE OF COMP.A | 1 |
| VALENCE OF ANION | 1 |
| CATION CONDUCTANCE-MHOS/EQIV.-298 K, INFINITE DILUTION | 38.6800 |
| ANION CONDUCTANCE-MHOS/EQIV.-298 K, INFINITE DILUTION | 71.4600 |
| BROMLEY EQ. SALT PARAMETER - CATION A, COMMON ANION | 0.093800 |
| Nernst LIMITING DIFFUSIVITY-CM**2/SEC | 0.13350E-04 |
| DIFFUSIVITY CALCULATED FROM CONDUCTANCES-CM**2/SEC | 0.13375E-04 |
| NUMBER OF DEPENDENT VARIABLES | 3 |
| NUMBER OF DATA POINTS | 13 |

### Concentration, Activity Coeff., Diffusion Coeff.

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<thead>
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<th>Activity Coeff.</th>
<th>Diffusion Coeff.</th>
</tr>
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<tr>
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</tr>
<tr>
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<td>0.124800E-04</td>
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<tr>
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AQUEOUS SOLUTION OF COMPOUND LINO3

### Comparison of Activity Coefficients-Actual vs. Predicted by Bromley Equation

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AVERAGE PERCENT DIFFERENCE= 2.6522
ROOT MEAN SQUARE-DIFFERENCE= 0.0366
ROOT MEAN SQUARE-PERCENT DIFFERENCE= 3.2316
### Comparison of Diffusion Coefficients - Actual vs. Predicted by Integral Version of Nernst-Hartley Equation

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<th>Difference-%</th>
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**Average Difference:** 0.19185E-05  
**Average Percent Difference:** 15.5083  
**Root Mean Square Difference:** 0.32647E-05  
**Root Mean Square Percent Difference:** 27.0921  

### Comparison of Diffusion Coefficients - Actual vs. Predicted by Differential Version of Nernst-Hartley Equation

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**Average Difference:** 0.47091E-05  
**Average Percent Difference:** 37.7062  
**Root Mean Square Difference:** 0.72089E-05  
**Root Mean Square Percent Difference:** 59.2064
COMPOUND CSCL

THIS PROGRAM EVALUATES THE EFFECT OF CONCENTRATION ON THE ACTIVITY AND DIFFUSION COEFFICIENTS OF SALTS USING THE BROMLEY EQUATION AND THE Nernst-Hartley EQUATION FOR DIFFUSIVITY

\[ D(c) = D(0) \left[1 + c \cdot \frac{d}{dc} \ln(\gamma)\right] \]

DILUTE SOLUTIONS

VALENCE OF COMP.A 1
VALENCE OF ANION 1
CATION CONDUCTANCE-MHOS/EQUIV.-298 K, INFINITE DILUTION 77.2000
ANION CONDUCTANCE-MHOS/EQUIV.-298 K, INFINITE DILUTION 76.3500
BROMLEY EQ. SALT PARAMETER -CATION A, COMMON ANION 0.002500
Nernst LIMITING DIFFUSIVITY-CM**2/SEC 0.20440E-04
DIFFUSIVITY CALCULATED FROM CONDUCTANCES-CM**2/SEC 0.20506E-04
NUMBER OF DEPENDENT VARIABLES 3
NUMBER OF DATA POINTS 11

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<th>DIFFUSION COEFF.</th>
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AQUEOUS SOLUTION OF COMPOUND CSCL

COMPARISON OF ACTIVITY COEFFICIENTS ACTUAL VS. PREDICTED BY BROMLEY EQUATION

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<th>DIFFERENCE-%</th>
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AVERAGE DIFFERENCE= 0.0173
AVERAGE PERCENT DIFFERENCE= 3.1402
ROOT MEAN SQUARE DIFFERENCE= 0.0193
ROOT MEAN SQUARE PERCENT DIFFERENCE= 3.5447
### COMPARISON OF DIFFUSION COEFFICIENTS-ACTUAL VS. PREDICTED BY INTEGRAL VERSION OF Nernst-Hartley EQUATION

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**Average Difference**: 0.24038E-05  
**Average Percent Difference**: 10.9016

**Root Mean Square Difference**: 0.32493E-05  
**Root Mean Square Percent Difference**: 14.3004

### COMPARISON OF DIFFUSION COEFFICIENTS-ACTUAL VS. PREDICTED BY DIFFERENTIAL VERSION OF Nernst-Hartley EQUATION

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**Average Difference**: 0.26309E-05  
**Average Percent Difference**: 11.1844

**Root Mean Square Difference**: 0.30659E-05  
**Root Mean Square Percent Difference**: 13.6252
COMPOUND NH4CL

THIS PROGRAM EVALUATES THE EFFECT OF CONCENTRATION ON THE ACTIVITY AND DIFFUSION COEFFICIENTS
OF SALTS USING THE BROMLEY EQUATION AND THE Nernst-Hartley EQUATION FOR DIFFUSIVITY

\[ D(c) = D(0)(1+c\delta \ln(GAMA)/dc); \]

DILUTE SOLUTIONS

VALENCE OF COMP.A 1
VALENCE OF ANION 1
CATION A CONDUCTANCE-MHOS/EQIV.-298 K, INFINITE DILUTION 73.5000
ANION CONDUCTANCE-MHOS/EQIV.-298 K, INFINITE DILUTION 76.3500
BROMLEY EQ. SALT PARAMETER -CATION A, COMMON ANION 0.020000

Nernst LIMITING DIFFUSIVITY-CM**2/SEC 0.19940E-04
DIFFUSIVITY CALCULATED FROM CONDUCTANCES-CM**2/SEC 0.20047E-04
NUMBER OF DEPENDENT VARIABLES 3
NUMBER OF DATA POINTS 14

COMPARISON OF ACTIVITY COEFFICIENTS-ACTUAL VS. PREDICTED BY BROMLEY EQUATION

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AVERAGE PERCENT DIFFERENCE= 0.7901
ROOT MEAN SQUARE-DIFFERENCE= 0.0052
ROOT MEAN SQUARE-PERCENT DIFFERENCE= 0.8962
### COMPARISON OF DIFFUSION COEFFICIENTS—ACTUAL VS. PREDICTED BY INTEGRAL VERSION OF Nernst-Hartley EQUATION

<table>
<thead>
<tr>
<th>CONCENTRATION</th>
<th>ACTUAL</th>
<th>PREDICTED</th>
<th>DIFFERENCE-%</th>
</tr>
</thead>
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<td>0.180575E-04</td>
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**AVERAGE DIFFERENCE= 0.19892E-05**

**AVERAGE PERCENT DIFFERENCE= 9.3513**

**ROOTMEAN SQUARE-DIFFERENCE= 0.23652E-05**

**ROOTMEAN SQUARE-PERCENT DIFFERENCE= 10.9260**

### COMPARISON OF DIFFUSION COEFFICIENTS—ACTUAL VS. PREDICTED BY DIFFERENTIAL VERSION OF Nernst-Hartley EQUATION

<table>
<thead>
<tr>
<th>CONCENTRATION</th>
<th>ACTUAL</th>
<th>PREDICTED</th>
<th>DIFFERENCE-%</th>
</tr>
</thead>
<tbody>
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</table>

**AVERAGE DIFFERENCE= 0.12540E-05**

**AVERAGE PERCENT DIFFERENCE= 6.0939**
**COMPOUND AMNO3**

This program evaluates the effect of concentration on the activity and diffusion coefficients of salts using the Bromley equation and the Nernst-Hartley equation for diffusivity:

\[ D(c) = D(0) \times (1 + \frac{d(\ln GAMA)}{dc}) \]; dilute solutions

| VALENCE OF COMP. A | 1 |
| VALENCE OF ANION | 1 |
| CATION & CONDUCTANCE-MHO/EQIV.-298 K, INFINITE DILUTION | 73.5000 |
| ANION CONDUCTANCE-MHO/EQIV.-298 K, INFINITE DILUTION | 71.4600 |
| BROMLEY EQ. SALT PARAMETER - CATION A, COMMON ANION | -0.035800 |
| Nernst LIMITING DIFFUSIVITY-CM**2/SEC | 0.19290E-04 |
| DIFFUSIVITY CALCULATED FROM CONDUCTANCES-CM**2/SEC | 0.19415E-04 |
| NUMBER OF DEPENDENT VARIABLES | 3 |
| NUMBER OF DATA POINTS | 13 |

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<thead>
<tr>
<th>CONCENTRATION</th>
<th>ACTIVITY COEFF.</th>
<th>DIFFUSION COEFF.</th>
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</thead>
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Aqueous solution of compound AMNO3

**Comparison of activity coefficients—actual vs. predicted by Bromley equation**

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<th>ACTUAL</th>
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Average difference = 0.0113  
Average percent difference = 2.3744  
Root mean square difference = 0.0121  
Root mean square percent difference = 2.5718
### Comparison of Diffusion Coefficients—Actual vs. Predicted by Integral Version of Nernst-Hartley Equation

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<th>Actual</th>
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<th>Difference-%</th>
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Average Difference = 0.11420E-05
Average Percent Difference = 7.2948
Root Mean Square Difference = 0.13434E-05
Root Mean Square Percent Difference = 8.7932

### Comparison of Diffusion Coefficients—Actual vs. Predicted by Differential Version of Nernst-Hartley Equation

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Average Difference = 0.29753E-05
Average Percent Difference = 19.0653
Root Mean Square Difference = 0.35407E-05
Root Mean Square Percent Difference = 23.4170
COMPOUND HCL

THIS PROGRAM EVALUATES THE EFFECT OF CONCENTRATION ON THE ACTIVITY AND DIFFUSION COEFFICIENTS OF SALTS USING THE BROMLEY EQUATION AND THE Nernst-Hartley EQUATION FOR DIFFUSIVITY

\[ D(c) = D(0) \times (1 + c \times d(\ln(\gamma_\text{A})) / dc) \]

DILUTE SOLUTIONS

VALENCE OF COMP.A  1
VALENCE OF ANION  1
CATION A CONDUCTANCE-Mhos/equiv. -298 K, INFINITE DILUTION  349.8000
ANION CONDUCTANCE-Mhos/equiv. -298 K, INFINITE DILUTION  76.3500
BROMLEY EQ. SALT PARAMETER -CATION A, COMMON ANION  0.143300
Nernst LIMITING DIFFUSIVITY-CM²/SEC  0.33360E-04
DIFFUSIVITY CALCULATED FROM CONDUCTANCES-CM²/SEC  0.33371E-04
NUMBER OF DEPENDENT VARIABLES  3
NUMBER OF DATA POINTS  12

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<thead>
<tr>
<th>CONCENTRATION</th>
<th>ACTIVITY COEFF.</th>
<th>DIFFUSION COEFF.</th>
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AQUEOUS SOLUTION OF COMPOUND HCL

COMPARISON OF ACTIVITY COEFFICIENTS - ACTUAL VS. PREDICTED BY BROMLEY EQUATION

<table>
<thead>
<tr>
<th>CONCENTRATION</th>
<th>ACTUAL</th>
<th>PREDICTED</th>
<th>DIFFERENCE-%</th>
</tr>
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ROOT MEAN SQUARE-DIFFERENCE= 0.0056

AVERAGE PERCENT DIFFERENCE= 0.5802
ROOT MEAN SQUARE-PERCENT DIFFERENCE= 0.6603
### COMPARISON OF DIFFUSION COEFFICIENTS—ACTUAL VS. PREDICTED BY INTEGRAL VERSION OF Nernst-Hartley EQUATION

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<th>DIFFERENCE-%</th>
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**AVERAGE DIFFERENCE = 0.98083E-06**  
**AVERAGE PERCENT DIFFERENCE = 2.5466**

**ROOTMEAN SQUARE-DIFFERENCE = 0.10825E-05**  
**ROOTMEAN SQUARE-PERCENT DIFFERENCE = 2.7116**

### COMPARISON OF DIFFUSION COEFFICIENTS—ACTUAL VS. PREDICTED BY DIFFERENTIAL VERSION OF Nernst-Hartley EQUATION

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**AVERAGE DIFFERENCE = 0.80172E-05**  
**AVERAGE PERCENT DIFFERENCE = 18.5532**

**ROOTMEAN SQUARE-DIFFERENCE = 0.10324E-04**  
**ROOTMEAN SQUARE-PERCENT DIFFERENCE = 22.2623**
COMPOUND HBR

THIS PROGRAM EVALUATES THE EFFECT OF CONCENTRATION ON THE ACTIVITY AND DIFFUSION COEFFICIENTS
OF SALTS USING THE BROMLEY EQUATION AND THE Nernst-Hartley EQUATION FOR DIFFUSIVITY

\[ D(c) = D(0) \times (1 + c \times \frac{d(\text{GAMA})}{dc}) \]; DILUTE SOLUTIONS

| VALENCE OF COMP.A | 1 |
| VALENCE OF ANION | 1 |
| CATION A CONDUCTANCE-MHOS/EQIV.-298 K, INFINITE DILUTION | 349.8000 |
| ANION CONDUCTANCE-MHOS/EQIV.-298 K, INFINITE DILUTION | 78.1400 |
| BROMLEY EQ. SALT PARAMETER - CATION A, COMMON ANION | 0.175400 |
| Nernst LIMITING DIFFUSIVITY-CM**2/SEC | 0.34000E-04 |
| DIFFUSIVITY CALCULATED FROM CONDUCTANCES-CM**2/SEC | 0.34073E-04 |
| NUMBER OF DEPENDENT VARIABLES | 3 |
| NUMBER OF DATA POINTS | 6 |

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AQUEOUS SOLUTION OF COMPOUND HBR

COMPARISON OF ACTIVITY COEFFICIENTS-actual vs. predicted by bromley equation

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AVERAGE DIFFERENCE= 0.0068  AVERAGE PERCENT DIFFERENCE= 0.8469  
ROOT MEAN SQUARE-DIFFERENCE= 0.0072  ROOT MEAN SQUARE-PERCENT DIFFERENCE= 0.8957
### COMPARISON OF DIFFUSION COEFFICIENTS—ACTUAL VS. PREDICTED BY INTEGRAL VERSION OF Nernst-Hartley EQUATION

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**AVERAGE DIFFERENCE= 0.83365E-06**

**AVERAGE PERCENT DIFFERENCE= 2.3776**

**ROOTMEAN SQUARE-DIFFERENCE= 0.10735E-05**

**ROOTMEAN SQUARE-PERCENT DIFFERENCE= 2.9413**

### COMPARISON OF DIFFUSION COEFFICIENTS—ACTUAL VS. PREDICTED BY DIFFERENTIAL VERSION OF Nernst-Hartley EQUATION

<table>
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<th>CONCENTRATION</th>
<th>ACTUAL</th>
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<th>DIFFERENCE-%</th>
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**AVERAGE DIFFERENCE= 0.21465E-05**

**AVERAGE PERCENT DIFFERENCE= 6.1732**

**ROOTMEAN SQUARE-DIFFERENCE= 0.23158E-05**

**ROOTMEAN SQUARE-PERCENT DIFFERENCE= 6.5158**
COMPOUND CA\textsubscript{L2}

THIS PROGRAM EVALUATES THE EFFECT OF CONCENTRATION ON THE ACTIVITY AND DIFFUSION COEFFICIENTS OF SALTS USING THE BROMLEY EQUATION AND THE Nernst-Hartley EQUATION FOR DIFFUSIVITY

\[ D(c) = D(0) \times (1 + c \times \text{d}(\text{n(GAMA)}/\text{dc}) ; \text{DILUTE SOLUTIONS} \]

**VALENCE OF COMP.A**

**VALENCE OF ANION**

**CATION CONDUCTANCE-MHOS/EQIV.-298 K, INFINITE DILUTION**

59.5000

**ANION CONDUCTANCE-MHOS/EQIV.-298 K, INFINITE DILUTION**

76.3500

**BROMLEY EQ. SALT PARAMETER -CATION A, COMMON ANION**

0.094800

**Nernst LIMITING DIFFUSIVITY-CM**\(^2\)/SEC**

0.13350E-04

**DIFFUSIVITY CALCULATED FROM CONDUCTANCES-CM**\(^2\)/SEC**

0.13434E-04

**NUMBER OF DEPENDENT VARIABLES**

3

**NUMBER OF DATA POINTS**

11

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AQUEOUS SOLUTION OF COMPOUND CA\textsubscript{L2}

COMPARISON OF ACTIVITY COEFFICIENTS-ACTUAL VS. PREDICTED BY BROMLEY EQUATION

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AVERAGE PERCENT DIFFERENCE= 3.6333

ROOT MEAN SQUARE-DIFFERENCE= 0.1093

ROOT MEAN SQUARE-PERCENT DIFFERENCE= 5.9739
### Comparison of Diffusion Coefficients—Actual vs. Predicted by Integral Version of Nernst-Hartley Equation

<table>
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<th>Difference-%</th>
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AVERAGE DIFFERENCE = 0.39754E-05  
AVERAGE PERCENT DIFFERENCE = 32.0666

ROOTMEAN SQUARE-DIFFERENCE = 0.56651E-05  
ROOTMEAN SQUARE-PERCENT DIFFERENCE = 45.7862

### Comparison of Diffusion Coefficients—Actual vs. Predicted by Differential Version of Nernst-Hartley Equation

<table>
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AVERAGE DIFFERENCE = 0.97353E-05  
AVERAGE PERCENT DIFFERENCE = 78.3075

ROOTMEAN SQUARE-DIFFERENCE = 0.13503E-04  
ROOTMEAN SQUARE-PERCENT DIFFERENCE = 108.6670
COMPONENT BACl₂

This program evaluates the effect of concentration on the activity and diffusion coefficients of salts using the Bromley equation and the Nernst-Hartley equation for diffusivity:

\[ D(c) = D(0) \times (1 + c \times d(\ln(GAMA))/dc) \]; Dilute solutions

Valence of Comp. A = 2
Valence of Anion = 1
Cation A Conductance-Mhos/Equiv.-298 K, Infinite Dilution = 63.6500
Anion Conductance-Mhos/Equiv.-298 K, Infinite Dilution = 76.3500
Bromley Eq. Salt Parameter - Cation A, Common Anion = 0.063500
Nernst Limiting Diffusivity-CM²/Sec = 0.138500E-04
Diffusivity Calculated from Conductances-CM²/Sec = 0.139530E-04
Number of Dependent Variables = 3
Number of Data Points = 7

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<th>Activity Coeff.</th>
<th>Diffusion Coeff.</th>
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<tbody>
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<tr>
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<tr>
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<td>0.425000</td>
<td>0.115100E-04</td>
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<tr>
<td>0.500000</td>
<td>0.403000</td>
<td>0.116800E-04</td>
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<tr>
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<td>0.117900E-04</td>
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<tr>
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<tr>
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Aqueous Solution of Compound BACl₂

Comparison of Activity Coefficients-Actual vs. Predicted by Bromley Equation

<table>
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<tr>
<th>Concentration</th>
<th>Actual</th>
<th>Predicted</th>
<th>Difference-%</th>
</tr>
</thead>
<tbody>
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<td>0.425000</td>
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<td>1.249095</td>
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<td>0.396617</td>
<td>1.583762</td>
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<tr>
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<td>0.397000</td>
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Average Difference = 0.0060  Average Percent Difference = 1.3660  Root Mean Square Difference = 0.0063  Root Mean Square Percent Difference = 1.4216
### COMPARISON OF DIFFUSION COEFFICIENTS-ACTUAL VS. PREDICTED BY INTEGRAL VERSION OF Nernst-Hartley EQUATION

<table>
<thead>
<tr>
<th>CONCENTRATION</th>
<th>ACTUAL</th>
<th>PREDICTED</th>
<th>DIFFERENCE-%</th>
</tr>
</thead>
<tbody>
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AVERAGE DIFFERENCE= 0.74370E-06  AVERAGE PERCENT DIFFERENCE= 6.3315

ROOTMEAN SQUARE-DIFFERENCE= 0.11132E-05  ROOTMEAN SQUARE-PERCENT DIFFERENCE= 9.4443

### COMPARISON OF DIFFUSION COEFFICIENTS-ACTUAL VS. PREDICTED BY DIFFERENTIAL VERSION OF Nernst-Hartley EQUATION

<table>
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<tr>
<th>CONCENTRATION</th>
<th>ACTUAL</th>
<th>PREDICTED</th>
<th>DIFFERENCE-%</th>
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</thead>
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AVERAGE DIFFERENCE= 0.20799E-05  AVERAGE PERCENT DIFFERENCE= 17.7034

ROOTMEAN SQUARE-DIFFERENCE= 0.30841E-05  ROOTMEAN SQUARE-PERCENT DIFFERENCE= 26.1704
COMPOUND AM2SO4

THIS PROGRAM EVALUATES THE EFFECT OF CONCENTRATION ON THE ACTIVITY AND DIFFUSION COEFFICIENTS OF SALTS USING THE BROMLEY EQUATION AND THE Nernst-Hartley EQUATION FOR DIFFUSIVITY

\[ D(c) = D(0)*\left(1+c*d(\ln(GAMA))/dc\right) \]; DILUTE SOLUTIONS

VALENCE OF COMP.A 1
VALENCE OF ANION 2
CATION A CONDUCTANCE-MHOS/EQUIV.-298 K, INFINITE DILUTION 73.5000
ANION CONDUCTANCE-MHOS/EQUIV.-298 K, INFINITE DILUTION 80.0200
BROMLEY EQ. SALT PARAMETER - CATION A, COMMON ANION -0.028700
Nernst LIMITING DIFFUSIVITY-CM**2/SEC 0.15300E-04
DIFFUSIVITY CALCULATED FROM CONDUCTANCES-CM**2/SEC 0.15346E-04
NUMBER OF DEPENDENT VARIABLES 3
NUMBER OF DATA POINTS 12

<table>
<thead>
<tr>
<th>CONCENTRATION</th>
<th>ACTIVITY COEFF.</th>
<th>DIFFUSION COEFF.</th>
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<tbody>
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<td>0.200000</td>
<td>0.343000</td>
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<td>0.300000</td>
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<td>0.500000</td>
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AQUATEOUS SOLUTION OF COMPOUND AM2SO4

COMPARISON OF ACTIVITY COEFFICIENTS - ACTUAL VS. PREDICTED BY BROMLEY EQUATION

<table>
<thead>
<tr>
<th>CONCENTRATION</th>
<th>ACTUAL</th>
<th>PREDICTED</th>
<th>DIFFERENCE-%</th>
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AVERAGE DIFFERENCE = 0.0195
AVERAGE PERCENT DIFFERENCE = 12.0564
ROOT MEAN SQUARE - DIFFERENCE = 0.0221
ROOT MEAN SQUARE - PERCENT DIFFERENCE = 16.0001
### Comparison of Diffusion Coefficients—Actual vs. Predicted by Integral Version of Nernst-Hartley Equation

<table>
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<tr>
<th>Concentration</th>
<th>Actual</th>
<th>Predicted</th>
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<th>Difference-%</th>
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</table>

**Average Difference:** 0.28989E-05  
**Average Percent Difference:** 28.3884

**Root Mean Square Difference:** 0.33854E-05  
**Root Mean Square Percent Difference:** 32.5502

### Comparison of Diffusion Coefficients—Actual vs. Predicted by Differential Version of Nernst-Hartley Equation

<table>
<thead>
<tr>
<th>Concentration</th>
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<th>Predicted</th>
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</table>

**Average Difference:** 0.47087E-05  
**Average Percent Difference:** 44.3090

**Root Mean Square Difference:** 0.59686E-05  
**Root Mean Square Percent Difference:** 54.4470
Appendix E

SOLUTION ALGORITHM - 7 DIAGONAL ARRAY

The following matrix equation is to be solved for $x_i$:

$$A^7_i x_i = b^b 7_i$$  \hspace{1cm} (E-1)

where

$$A_i = \begin{bmatrix}
A^7_i & B^7_i & C^7_i & D^7_i & \mathbf{0} & \mathbf{0} & \mathbf{0} & \mathbf{0} & \mathbf{0} \\
E^7_i & A^7_2 & B^7_2 & C^7_2 & D^7_2 & \mathbf{0} & \mathbf{0} & \mathbf{0} & \mathbf{0} \\
F^7_i & E^7_3 & A^7_3 & B^7_3 & C^7_3 & D^7_3 & \mathbf{0} & \mathbf{0} & \mathbf{0} \\
G^7_i & F^7_4 & E^7_4 & A^7_4 & B^7_4 & C^7_4 & D^7_4 & \mathbf{0} & \mathbf{0} \\
\mathbf{0} & G^7_5 & F^7_5 & E^7_5 & A^7_5 & B^7_5 & C^7_5 & D^7_5 & \mathbf{0} \\
\mathbf{0} & \mathbf{0} & G^7_6 & F^7_6 & E^7_6 & A^7_6 & B^7_6 & C^7_6 & D^7_6 \\
\mathbf{0} & \mathbf{0} & \mathbf{0} & G^7_7 & F^7_7 & E^7_7 & A^7_7 & B^7_7 & C^7_7 \\
\mathbf{0} & \mathbf{0} & \mathbf{0} & \mathbf{0} & G^7_8 & F^7_8 & E^7_8 & A^7_8 & B^7_8 \\
\mathbf{0} & \mathbf{0} & \mathbf{0} & \mathbf{0} & \mathbf{0} & G^7_9 & F^7_9 & E^7_9 & A^7_9 \\
\mathbf{0} & \mathbf{0} & \mathbf{0} & \mathbf{0} & \mathbf{0} & \mathbf{0} & G^7_{10} & F^7_{10} & E^7_{10} \\
\end{bmatrix}$$

$$x_5 = \begin{bmatrix} x_1 \\ x_2 \\ x_3 \\ x_4 \\ x_5 \\ x_6 \\ x_7 \\ x_8 \\ x_9 \\ x_{10} \end{bmatrix}$$

\hspace{1cm} (E-2)
$A_1^7$ can be specified as:

$$A_1^7 = L_1^7 U_1^7$$  \hspace{1cm} (E-3)

where

$$L_1^7 = \begin{bmatrix}
\omega^7 & 0 & 0 & 0 & 0 & 0 & 0 & 0 & 0 \\
\beta^7 & \omega^7 & 0 & 0 & 0 & 0 & 0 & 0 & 0 \\
\alpha^7 & \beta^7 & \omega^7 & 0 & 0 & 0 & 0 & 0 & 0 \\
\gamma^7 & \alpha^7 & \beta^7 & \omega^7 & 0 & 0 & 0 & 0 & 0 \\
\beta^7 & \omega^7 & \alpha^7 & \beta^7 & \omega^7 & 0 & 0 & 0 & 0 \\
\alpha^7 & \beta^7 & \omega^7 & \alpha^7 & \beta^7 & \omega^7 & 0 & 0 & 0 \\
\gamma^7 & \alpha^7 & \beta^7 & \omega^7 & \alpha^7 & \beta^7 & \omega^7 & 0 & 0 \\
\beta^7 & \omega^7 & \alpha^7 & \beta^7 & \omega^7 & \alpha^7 & \beta^7 & \omega^7 & 0 \\
\gamma^7 & \alpha^7 & \beta^7 & \omega^7 & \alpha^7 & \beta^7 & \omega^7 & \alpha^7 & \beta^7 \\
\end{bmatrix}$$

$U_1^7 = \begin{bmatrix}
1 & \gamma^7 & \delta^7 & \phi^7 & 0 & 0 & 0 & 0 & 0 \\
0 & 1 & \gamma^7 & \delta^7 & \phi^7 & 0 & 0 & 0 & 0 \\
0 & 0 & 1 & \gamma^7 & \delta^7 & \phi^7 & 0 & 0 & 0 \\
0 & 0 & 0 & 1 & \gamma^7 & \delta^7 & \phi^7 & 0 & 0 \\
0 & 0 & 0 & 0 & 1 & \gamma^7 & \delta^7 & \phi^7 & 0 \\
0 & 0 & 0 & 0 & 0 & 1 & \gamma^7 & \delta^7 & \phi^7 \\
0 & 0 & 0 & 0 & 0 & 0 & 1 & \gamma^7 & \delta^7 \\
0 & 0 & 0 & 0 & 0 & 0 & 0 & 1 & \gamma^7 \\
0 & 0 & 0 & 0 & 0 & 0 & 0 & 0 & 1 \\
\end{bmatrix}$
matrix multiplication of (E-4) and E-5), setting the coefficients equal to those in (E-1), yields:

\[ e_i^7 = e_i \quad \text{for} \quad i = 4 \quad \text{---} \quad N \]  
\[ b_i^7 = b_i^7 / \omega_i^7 = \frac{D_i^7}{A_i^7} \]  
\[ c_i^7 = c_i^7 / \omega_i^7 = \frac{C_i^7}{A_i^7} \]  
\[ \delta_i^7 = \frac{c_i^7 - B_i^7 \delta_{i-1}}{\omega_i^7} \quad \text{for} \quad i = 2 \quad \text{---} \quad N-2 \]  
\[ \gamma_i^7 = B_i^7 / \omega_i^7 = \frac{B_i^7}{A_i^7} \]  
\[ \gamma_2^7 = \frac{b_2^7 - \alpha_2^7 \gamma_1^7}{\omega_2^7} \]  
\[ \gamma_i^7 = \frac{B_i^7 - \alpha_i^7 \gamma_{i-2}^7 - \delta_{i-1}^7}{\omega_i^7} \quad \text{for} \quad i = 3 \quad \text{---} \quad N-1 \]  
\[ \beta_2^7 = e_2^7 \]  
\[ \beta_3^7 = e_3^7 - \alpha_3^7 \gamma_1^7 \]  
\[ \beta_i^7 = e_i^7 - \alpha_i^7 \gamma_{i-3}^7 - \alpha_i^7 \gamma_{i-2}^7 \quad \text{for} \quad i = 4 \quad \text{---} \quad N \]  
\[ \gamma_3^7 = f_3^7 \]  
\[ \gamma_i^7 = f_i^7 \gamma_{i-3}^7 \quad \text{for} \quad i = 4 \quad \text{---} \quad N \]  
\[ \omega_1^7 = a_1^7 \]  
\[ \omega_2^7 = A_2^7 \gamma_1^7 \]  
\[ \omega_3^7 = A_3^7 \gamma_2^7 - \alpha_3^7 \gamma_1^7 \]  
\[ \omega_i^7 = A_i^7 \gamma_{i-1}^7 - \delta_{i-1}^7 - e_i^7 \gamma_{i-3}^7 \quad \text{for} \quad i = 4 \quad \text{---} \quad N \]
defining $FF_i^7 = U_i^7 X_i$

then $L_i^7 = FF_i^7$

where

$$L_i^7 = \begin{array}{c}
\omega_1 \\
\beta_2 \\
\gamma_3 \\
\alpha_4 \\
\beta_N \\
\omega_N
\end{array} \cdot \begin{array}{c}
X_i \\
\frac{FF_i^7}{bb_i}
\end{array}$$

solution to the sets of equations defined by (E-25)

$$FF_1^7 = \frac{bb_1^7}{\omega_1}$$

$$FF_2^7 = \frac{bb_2^7}{\omega_2}$$

$$FF_3^7 = \frac{bb_3^7}{\omega_3}$$

$$FF_4^7 = \frac{bb_4^7}{\omega_4}$$

$$FF_i^7 = \frac{bb_i^7}{\omega_i}$$

for $i = 4 \cdots N$ (E-30)

Returning to $U_i^7 X_i = FF_i^7$ (E-23)
solution to equations defined by (E-31) in terms of (E-30)

\[
X_N = F_i^7 = \frac{bb^7 - FF^7 - FF^7 N - 3^7 N - 2^7 N - 1^7 N}{N - 2^7 N - 1^7 N}
\]  
(E-32)

and

\[
F_i^7_{N-1} = \gamma_{N-1}^7 N - 1^7 X_N
\]  
(E-33)

\[
F_i^7_{N-2} = \gamma_{N-2}^7 N - 2^7 X_{N-1} + \delta_{N-2}^7 N
\]  
(E-34)

\[
F_i^7_{N-3} = \gamma_{N-3}^7 N - 3^7 N - 2^7 N - 1^7 X_{N-1} + \phi_{N-3}^7 N
\]  
(E-35)

\[
F_i^7_i = \gamma_i^7 X_{i+1} + \delta_i^7 X_{i+2} + \phi_i^7 X_{i+3} \quad \text{for } i = N-3 \ldots 1
\]  
(E-36)

so \(F_i^7_i = \gamma_i^7 X_2 + \delta_i^7 X_3 + \phi_i^7 X_4\)  
(E-37)

Rearranging in terms of \(X_i\)

\[
X_{N-1} = F_i^7_{N-1} - \gamma_{N-1}^7 X_N
\]  
(E-38)

\[
X_{N-2} = F_i^7_{N-2} - \gamma_{N-2}^7 X_{N-1} + \delta_{N-2}^7 X_N
\]  
(E-39)

\[
X_{N-3} = F_i^7_{N-3} - \gamma_{N-3}^7 X_{N-2} + \delta_{N-3}^7 X_{N-1} + \phi_{N-3}^7 X_N
\]  
(E-40)

\[
x_i = F_i^7_{i+1} - \gamma_{i+1}^7 X_{i+2} + \delta_{i+1}^7 X_{i+3} + \phi_{i+1}^7 X_1 + \delta_i^7 X_i + \phi_i^7 X_{i-1} + \phi_{i-1}^7 X_{i-2}
\]  
(E-41)

so \(x_i = F_i^7_{i+1} - \gamma_{i+1}^7 X_{i+2} + \delta_{i+1}^7 X_{i+3} + \phi_{i+1}^7 X_1\)  
(E-42)

from equation (E-30)

\[
F_i^7_{N-1} = \frac{bb^7 - FF^7 - FF^7 N - 3^7 N - 2^7 N - 1^7 N}{N - 2^7 N - 1^7 N}
\]  
(E-43)

for \(i = 1 \ldots N-4\)

with \(F_i^7_{1}, F_i^7_{2}, F_i^7_{3}\) given in equations (E-26) through (E-28).

The 7 diagonal algorithm then consists of separating the coefficient matrix \(A_i^7\) into an upper and lower diagonal matrix, which when multiplied and set equal to the terms in \(A_i^7\), allows values to be computed for the factors:

\[
\omega_i^7, \beta_i^7, \alpha_i^7, \text{ and } \delta_i^7
\]

By defining a vector \(F_i^7 = U_i^7 X_i\), advantage is taken of the upper diagonal \(U_i^7\) in which all diagonal elements are unity to solve for \(X_i\) in terms of vectors \(bb_i^7\) and vector \(F_i^7\), which is known in terms of the lower diagonal factors. \(X_i\) is then solved for starting with \(X_N\) and proceeding to \(X_1\). Thus in effect inverting the matrix \(A_i^7\) and solving for \(X_1\).
Appendix F

SOLUTION ALGORITHM - 5 DIAGONAL ARRAY

The following matrix equation is to be solved for $X_1$:

$$A_1^5 X_1 = bb5_1$$  \hspace{1cm} (F-1)

$$A_1^5 = 
\begin{bmatrix}
A_1^5 & B_1^5 & C_1^5 & 0 & 0 & 0 & 0 & 0 & 0 & 0 \\
A_2^5 & E_2^5 & B_2^5 & C_2^5 & 0 & 0 & 0 & 0 & 0 & 0 \\
A_3^5 & E_3^5 & A_3^5 & B_3^5 & C_3^5 & 0 & 0 & 0 & 0 & 0 \\
A_4^5 & F_4^5 & E_4^5 & A_4^5 & B_4^5 & C_4^5 & 0 & 0 & 0 & 0 \\
A_5^5 & 0 & 0 & F_5^5 & E_5^5 & A_5^5 & B_5^5 & C_5^5 & 0 & 0 \\
A_6^5 & 0 & 0 & 0 & F_6^5 & E_6^5 & A_6^5 & B_6^5 & C_6^5 & 0 \\
A_7^5 & 0 & 0 & 0 & 0 & F_7^5 & E_7^5 & A_7^5 & B_7^5 & C_7^5 \\
A_8^5 & 0 & 0 & 0 & 0 & 0 & F_8^5 & E_8^5 & A_8^5 & B_8^5 \\
A_9^5 & 0 & 0 & 0 & 0 & 0 & 0 & F_9^5 & E_9^5 & A_9^5 \\
A_{10}^5 & 0 & 0 & 0 & 0 & 0 & 0 & 0 & F_{10}^5 & E_{10}^5 \\
\end{bmatrix} \times 
\begin{bmatrix}
X_1 \\
X_2 \\
X_3 \\
X_4 \\
X_5 \\
X_6 \\
X_7 \\
X_8 \\
X_9 \\
X_{10} \\
\end{bmatrix} = 
\begin{bmatrix}
bb5_1 \\
bb5_2 \\
bb5_3 \\
bb5_4 \\
bb5_5 \\
bb5_6 \\
bb5_7 \\
bb5_8 \\
bb5_9 \\
bb5_{10} \\
\end{bmatrix}$$  \hspace{1cm} (F-2)
$A_1^5$ can be specified as:

$$A_1^5 = L_1^5 U_1^5$$  \hspace{1cm} \text{(F-3)}$$

where

$$L_1^5 = \begin{bmatrix}
\omega_1^5 & 0 & 0 & 0 & 0 & 0 & 0 & 0 \\
\beta_2^5 & \delta_2^5 & 0 & 0 & 0 & 0 & 0 & 0 \\
\alpha_3^5 & \beta_3^5 & \omega_3^5 & 0 & 0 & 0 & 0 & 0 \\
0 & \alpha_4^5 & \beta_4^5 & \omega_4^5 & 0 & 0 & 0 & 0 \\
0 & 0 & \alpha_5^5 & \beta_5^5 & \omega_5^5 & 0 & 0 & 0 \\
0 & 0 & 0 & \alpha_6^5 & \beta_6^5 & \omega_6^5 & 0 & 0 \\
0 & 0 & 0 & 0 & \alpha_7^5 & \beta_7^5 & \omega_7^5 & 0 \\
0 & 0 & 0 & 0 & 0 & \alpha_8^5 & \beta_8^5 & \omega_8^5 \\
0 & 0 & 0 & 0 & 0 & 0 & \alpha_9^5 & \beta_9^5 \omega_9^5 \\
0 & 0 & 0 & 0 & 0 & 0 & \alpha_{10}^5 & \beta_{10}^5 \omega_{10}^5
\end{bmatrix}$$

$$U_1^5 = \begin{bmatrix}
1 & \gamma_1^5 & \delta_1^5 & 0 & 0 & 0 & 0 & 0 & 0 \\
0 & 1 & \gamma_2^5 & \delta_2^5 & 0 & 0 & 0 & 0 & 0 \\
0 & 0 & 1 & \gamma_3^5 & \delta_3^5 & 0 & 0 & 0 & 0 \\
0 & 0 & 0 & 1 & \gamma_4^5 & \delta_4^5 & 0 & 0 & 0 \\
0 & 0 & 0 & 0 & 1 & \gamma_5^5 & \delta_5^5 & 0 & 0 \\
0 & 0 & 0 & 0 & 0 & 1 & \gamma_6^5 & \delta_6^5 & 0 \\
0 & 0 & 0 & 0 & 0 & 0 & 1 & \gamma_7^5 & \delta_7^5 \\
0 & 0 & 0 & 0 & 0 & 0 & 0 & 1 & \gamma_8^5 \\
0 & 0 & 0 & 0 & 0 & 0 & 0 & 0 & 1
\end{bmatrix}$$
matrix multiplication of (F-4) and (F-5), setting the coefficients equal to those in (F-1), yields:

\[ a_i^6 = F_i^5 \quad \text{for } i = 3 \quad \text{--- N} \]  
\[ a_i^5 = C_i^5/a_1^5 \quad \text{for } i = 1 \quad \text{--- N-2} \]  
\[ a_i^5 = C_i^5/a_1^5 = C_i^5/A_i^5 \]  
\[ \omega_i^5 = A_i^5 \]  
\[ \gamma_i^5 = B_i^5/a_1^5 = B_i^5/A_i^5 \]  
\[ \gamma_i^5 = B_i^5/a_1^5 = B_i^5/A_i^5 \quad \text{for } i = 2 \quad \text{--- N-1} \]  
\[ \beta_2^5 = E_2^5 \]  
\[ \beta_1^5 = E_1^{-\alpha_1} \gamma_1^{-2} \quad \text{for } i = 3 \quad \text{--- N} \]  
\[ \omega_2^5 = A_2^{-B_2^{-1}} \]  
\[ \omega_i^5 = A_i^{-B_i^{-1}} \gamma_{i-1}^{-2} \gamma_i^{-5} \quad \text{for } i = 3 \quad \text{--- N} \]  

defining \( F_i^5 = U_i^5 x_i^5 \)  
then \( L_i^5 F_i^5 = b_i^5 \)  

where

\[
 L_i^5 = \begin{bmatrix}
 a_1^5 & \omega_1^5 & \alpha_2^5 & \alpha_3^5 & \cdots & \alpha_N^5 \\
 b_1^5 & \beta_1^5 & \beta_2^5 & \beta_3^5 & \cdots & \beta_N^5 \\
 \omega_2^5 & \omega_3^5 & \omega_4^5 & \omega_5^5 & \cdots & \omega_N^5 \\
 \alpha_N^5 & \beta_N^5 & \omega_N^5 & \omega_N^5 & \cdots & \omega_N^5 \\
 \end{bmatrix} \]

solution to the sets of equations defined by (F-18)

\[ F_i^5 = b_i^5/a_1^5 \]  
\[ F_i^5 b_2^5 + F_i^5 \omega_2^5 = b_i^5 \]  
or \[ F_2^5 = b_2^5 F_2^5 b_2^5 \]

\[ F_3^5 = b_3^5 F_3^5 b_3^5 b_3^5 \]

\[ F_i^5 = b_i^5 F_i^5 a_i^5 F_i^5 a_i^5 \quad \text{for } i = 3 \quad \text{--- N} \]
returning to $V_i^5 X_i = F_i^5$

$$
\begin{array}{c|c|c|c}
1 & 5 & \delta_1^5 \\
\gamma_1 & 1 & 5 & \delta_2^5 \\
& \gamma_{N-2} & 1 & 5 & \delta_{N-2}^5 \\
& & \gamma_{N-1} & 1 & \delta_{N-1}^5 \\
& & & 1 & \delta_N^5 \\
\end{array}
$$

$X_i, X_{N-2}, X_{N-1}, X_N$

$$
X_i = F_i^5 = \frac{\beta_i^5 F_{N-i}^5}{F_{N-2}^5}$

$$
X_N = F_N^5 = \frac{b_{N-1}^5 F_{N-2}^5 - b_{N-2}^5 F_{N-3}^5}{\beta_N^5}
$$

and

$$
F_{N-1}^5 = X_{N-1} + \gamma_{N-1}^5 X_N
$$

$$
F_{N-2}^5 = X_{N-2} + \gamma_{N-2}^5 X_{N-1} + \delta_{N-2}^5 X_N
$$

$$
F_i^5 = X_i + \gamma_i^5 X_{i+1} + \delta_i^5 X_{i+2}
$$

so

$$
F_1^5 = X_1 + \gamma_1^5 X_2 + \delta_1^5 X_3
$$

rearranging in terms of $X_i$

$$
X_{N-1} = F_{N-1}^5 \gamma_{N-1}^5 X_N
$$

$$
X_{N-2} = F_{N-2}^5 \gamma_{N-2}^5 X_{N-1} + \delta_{N-2}^5 X_N
$$

$$
X_i = F_i^5 \gamma_i^5 X_{i+1} + \delta_i^5 X_{i+2} \quad \text{for } i = N-2 \ldots 1
$$

so

$$
X_1 = F_1^5 \gamma_1^5 X_2 + \delta_1^5 X_3
$$

from equation (F-23)

$$
F_i^5 = \frac{b_i^5 - F_{i-1}^5 F_{N-i}^5}{F_{N-2}^5} \quad \text{for } i = 1 \ldots N-3
$$

and $F_1^5$ and $F_2^5$ given in equations (F-20) and (F-21).

Use of the above algorithm inverts the $A_i^5$ matrix and solves for $X_i$. 

solution to equations defined by (F-24) in terms of (F-23)

$$
X_N = F_N^5 = \frac{b_{N-1}^5 F_{N-2}^5 - b_{N-2}^5 F_{N-3}^5}{\beta_N^5}
$$

$$
X_i = F_i^5 = \frac{\beta_i^5 F_{N-i}^5}{F_{N-2}^5}
$$

for $i = 1 \ldots N-3$
Appendix G

DERIVATION OF THE GENERAL EQUATION FOR DIFFUSION OF IONIC SPECIES IN THE RESIN PHASE — TERNARY SYSTEMS

A. Derivation using irreversible thermodynamics treatment

The Nernst-Planck equation for transport of ionic species in dilute electrolytes is given as:

\[ \overline{N}_k = -\overline{c}_k \cdot \sum_{j=1}^{4} \overline{L}_{kj} \cdot \nabla \overline{\mu}_j - \overline{c}_k \cdot F \cdot \nabla \overline{\phi} \cdot \sum_{j=1}^{4} \overline{L}_{kj} \cdot |z_j| \]  \hspace{1cm} (G-1)

for \( k = 1, 2, 3, 4 \)

where:

\( \overline{N}_k \) = molar flux of component \( k \)

\( \overline{c}_k \) = molar concentration of species \( k \) (moles/liter)

\( \overline{\mu}_j \) = chemical potential of species \( j \)

\( z_j \) = charge on species \( j \)

\( \overline{L}_{kj} \) = phenomenological coefficients

\( F \) = Faraday constant

\( \overline{\phi} \) = electric potential, volts

For electroneutrality in the resin, the following relationship holds:

\[ \sum_{i=1}^{4} |z_i| \cdot \overline{c}_i = \text{constant; charge of water, } |z_4| = 0 \]  \hspace{1cm} (G-2)

(1) constant = \( |z|_{\text{sites}} \cdot \overline{c}_{\text{sites}} \)

Then the gradient is given by:

\[ \sum_{i=1}^{4} |z_i| \cdot \nabla \overline{c}_i = 0 \]  \hspace{1cm} (G-3)
and since no electric current flows,

$$\sum_{i=1}^{4} |z_i| \cdot \vec{N}_i = 0. \tag{G-4}$$

Substituting (G-1) in (G-4),

$$\vec{V} = \left[ \sum_{i=1}^{4} \left( \frac{z_i \cdot c_i \cdot \left( \sum_{j=1}^{4} \vec{l}_{ij} \cdot \vec{V}_{ij} \right)}{F \cdot \sum_{i=1}^{4} \left( z_i \cdot c_i \right) \cdot \left( \sum_{j=1}^{4} \vec{l}_{ij} \cdot |z_j| \right)} \right) \right] \tag{G-5}$$

Putting \( \vec{V} \) from equation (G-5) into equation (G-1) yields:

$$\vec{N}_k = -c_k \cdot \sum_{j=1}^{4} \vec{l}_{kj} \cdot \vec{V}_{ij} + \left( \frac{c_k \cdot \sum_{i=1}^{4} \vec{l}_{ki} \cdot |z_i|}{4 \sum_{j=1}^{4} \vec{c}_j \cdot |z_j| \cdot \left( \sum_{i=1}^{4} \vec{l}_{ij} \cdot \vec{V}_{ij} \right) \cdot \sum_{i=1}^{4} \vec{c}_i \cdot z_i^2 \cdot \vec{l}_{ij}} \right) \tag{G-6}$$

for \( k = 1, 2, 3, 4 \)

From the chain rule,

$$\vec{V}_{ij} \cdot \frac{\partial \vec{V}_{ij}}{\partial c_j} \tag{G-7}$$
and substituting (G-7) in (G-6) gives,

\[ (G-8) \]

\[
\bar{N}_k = \left\{ \begin{array}{c}
-c_k \cdot \sum_{j=1}^{4} \bar{L}_{kj} \frac{\partial \mu_j}{\partial c_j} + \left\{ \begin{array}{c}
-c_k \cdot \sum_{i=1}^{4} \bar{L}_{ki} \cdot |z_i| \\
\left( \sum_{j=1}^{4} c_j \cdot |z_j| \cdot \sum_{i=1}^{4} \bar{L}_{ij} \frac{\partial \mu_j}{\partial c_j} \right) \\
\left( \sum_{i=1}^{4} c_i \cdot z_i^2 \cdot \bar{L}_{ij} \right)
\end{array} \right. \\
- \end{array} \right. 
\]

for \( k = 1, 2, 3, 4 \)

This can be expressed as:

\[ (G-9) \]

\[
\bar{N}_k = - \sum_{j=1}^{4} \bar{D}_{kj} \cdot \nabla \bar{c}_j 
\]

where:

\[ (G-10) \]

\[
\bar{D}_{kj} = \bar{c}_k \cdot \bar{L}_{kj} \frac{\partial \mu_j}{\partial c_j} - \left\{ \begin{array}{c}
4 \sum_{i=1}^{4} \bar{L}_{ki} \cdot |z_i| \\
\sum_{i=1}^{4} c_i \cdot |z_i| \cdot \bar{L}_{ij} \frac{\partial \mu_j}{\partial c_j} \\
\sum_{i=1}^{4} c_i \cdot z_i^2 \cdot \bar{L}_{ij}
\end{array} \right. \\
- \end{array} \right. 
\]

for \( k = 1, 2, 3, 4 \) and \( j = 1, 2, 3, 4 \)
B. Simplifying Assumptions

1.0 Elimination of Cross Phenomenological Coefficients

If we assume that cross phenomenological coefficients are zero,

\[ \bar{L}_{1k} = \bar{L}_{2k} = \bar{L}_{3k} = \bar{L}_{4k} = 0 \]  
(G-11)

and,

\[ \sum_{i=1}^{4} \bar{L}_{ki} \cdot |z_i| = \bar{L}_{kk} \cdot |z_k| \]  
(G-12)
resulting in:

\[ \sum_{i=1}^{4} \bar{c}_i \cdot z_i^2 \cdot \bar{L}_{ij} = \sum_{i=1}^{4} \bar{c}_i \cdot z_i^2 \cdot \bar{L}_{ii} \]  
(G-13)

The following diffusion coefficients result from substitution of (G-12) and (G-13) in equation (G-10):

\[ \bar{D}_{kj} = \bar{c}_k \left\{ \frac{\partial \bar{w}_c}{\partial \bar{c}_j} \right\} \left[ \bar{L}_{kk} \cdot (|z_k|) \cdot \bar{c}_j \cdot (|z_j|) \cdot \bar{L}_{jj} \cdot \frac{\partial \bar{w}_c}{\partial \bar{c}_j} \right] \left[ \frac{4}{\sum_{i=1}^{4} z_i^2 \cdot \bar{c}_i \cdot \bar{L}_{ii}} \right] \]

for \( k = 1, 2, 3, 4 \) and \( j = 1, 2, 3, 4 \)

and,

\[ \bar{D}_{kk} = \bar{c}_k \left\{ \bar{L}_{kk} - \frac{4}{\sum_{i=1}^{4} z_i^2 \cdot \bar{c}_i \cdot \bar{L}_{ii}} \right\} \frac{\partial \bar{w}_c}{\partial \bar{c}_k} \]

for \( k = 1, 2, 3, 4 \)

2.0 Pressure-Volume Effects

If pressure in the resin and the volume of the resin are
assumed to be constant, i.e.,

\[
\left( \frac{\partial (\pi u)}{\partial c_i} \right)_T = 0 \tag{G-16}
\]

then water diffusion is negligible, and,

\[
\frac{\overline{D}_{k4}}{4} \cdot \nabla \overline{c}_4 = 0 \tag{G-17}
\]

therefore,

\[
\overline{N}_k = - \sum_{j=1}^{3} \frac{\overline{D}_{kj}}{4} \cdot \nabla \overline{c}_j \tag{G-18}
\]

and

\[
\left( \sum_{j=1}^{3} |z_j| \cdot \nabla \overline{c}_j \right) = 0, \text{ since } z_4 = 0 \tag{G-19}
\]

C. Elimination of species 3 in sets of equations

By expanding equation (G-18), there results

\[
\overline{N}_k = - \frac{\overline{D}_{k1}}{4} \cdot \nabla \overline{c}_1 - \frac{\overline{D}_{k2}}{4} \cdot \nabla \overline{c}_2 - \frac{\overline{D}_{k3}}{4} \cdot \nabla \overline{c}_3 \tag{G-20}
\]

and in view of (G-19) which yields,

\[
-\nabla \overline{c}_3 = \left( \frac{|z_2|}{|z_3|} \right) \cdot \nabla \overline{c}_2 + \left( \frac{|z_1|}{|z_3|} \right) \cdot \nabla \overline{c}_1 \tag{G-21}
\]

so,

\[
\overline{N}_k = - \left( \frac{\overline{D}_{k1}}{4} - \left( \frac{|z_1|}{|z_3|} \right) \cdot \overline{D}_{k3} \right) \cdot \nabla \overline{c}_1 - \left( \frac{\overline{D}_{k2}}{4} - \left( \frac{|z_2|}{|z_3|} \right) \cdot \overline{D}_{k3} \right) \cdot \nabla \overline{c}_2 \tag{G-22}
\]

This can be shown as,

\[
\overline{N}_k = - \sum_{j=1}^{2} \left( \frac{\overline{D}_{kj}}{4} - \left( \frac{|z_j|}{|z_3|} \right) \cdot \overline{D}_{k3} \right) \cdot \nabla \overline{c}_j \tag{G-23}
\]

for \( k = 1, 2 \)

or,

\[
\overline{N}_k = - \sum_{j=1}^{2} \frac{\overline{D}_{kj}}{4} \cdot \nabla \overline{c}_j, \text{ for } k = 1, 2 \tag{G-24}
\]
D. Effective Diffusion Coefficients – chemical potential

By combining equations (G-15) and (G-16) to satisfy (G-23) and (G-24), there results,

\[
\bar{D}_{k,j} = \bar{c}_k \cdot |z_k \cdot z_j| \cdot \bar{L}_{k,k} \cdot \left( \begin{array}{c} \frac{\partial \bar{c}_3}{\partial \bar{c}_3} \cdot \bar{L}_{j,j} \cdot \bar{c}_j \cdot \frac{\partial \bar{\mu}_i}{\partial \bar{c}_i} \\ \frac{3}{2} \sum_{i=1} c_i z_i^2 \cdot \bar{L}_{i,i} \end{array} \right)
\]

for \( k = 1,2 \) and \( j = 1,2 \)

Assuming the "self" diffusion coefficient of species \( k \) is:

\[
\bar{D}_k = R \cdot T \cdot \bar{L}_{k,k}
\]  

(G-26)

The chemical potential is defined as:

\[
\bar{\mu}_k = R \cdot T \cdot \ln(\bar{a}_k)
\]  

(G-27)

then:

\[
\frac{\partial \bar{\mu}_k}{\partial \bar{c}_k} = R \cdot T \cdot \left( \frac{\partial \ln(\bar{a}_k)}{\partial \bar{c}_k} \right)
\]  

(G-28)

Substituting equations (G-26) and (G-27) in equation (G-24) yields,

\[
\bar{D}_{k,j} = \bar{c}_k \cdot |z_k \cdot z_j| \cdot \bar{D}_k \cdot \left( \begin{array}{c} \frac{\partial \ln(\bar{a}_3)}{\partial \bar{c}_3} \cdot \bar{L}_{j,j} \cdot \bar{c}_j \cdot \frac{\partial \ln(\bar{a}_i)}{\partial \bar{c}_i} \\ \frac{3}{2} \sum_{i=1} c_i z_i^2 \cdot \bar{L}_{i,i} \end{array} \right)
\]

for \( k = 1,2 \) and \( j = 1,2 \)
likewise

\[
\tilde{D}_{kk} = \tilde{c}_k \cdot \tilde{D}_k \cdot \left\{ \frac{\partial \ln(\tilde{a}_k)}{\partial \tilde{c}_k} - z_k^2 \cdot \frac{\partial \ln(\tilde{a}_3)}{\partial \tilde{c}_3} \right\} \quad (G-30)
\]

for \( k = 1, 2 \)

The activity of species \( k \) is given as,

\[
\tilde{a}_k = \tilde{\gamma}_k \cdot \tilde{c}_k \quad (G-31)
\]

where \( \tilde{\gamma}_k \) is the molar activity coefficient

and therefore,

\[
\frac{\partial \ln(\tilde{a}_k)}{\partial \tilde{c}_k} = \frac{1}{\tilde{c}_k} + \frac{\partial \ln(\tilde{\gamma}_k)}{\partial \tilde{c}_k} \quad (G-32)
\]

Substituting for \( \frac{\partial \ln(\tilde{a}_k)}{\partial \tilde{c}_k} \) in equations \((G-29)\) and \((G-30)\) yields,

\[
\tilde{D}'_{kk} = \tilde{c}_k \cdot \tilde{D}_k \cdot \left\{ \frac{1}{\tilde{c}_k} + \frac{\partial \ln(\tilde{\gamma}_k)}{\partial \tilde{c}_k} \right\}
\]

\[
- z_k^2 \cdot \left\{ \frac{3}{\sum \tilde{c}_i \cdot z_i^2 \cdot \tilde{D}_i} \right\}
\]

for \( k = 1, 2 \)
and, \[
\hat{b}_{k,j} = c_k \cdot (|z_k \cdot z_j|) \cdot \overline{b}_k \cdot \left( \frac{\partial \ln(\overline{\gamma}_3)}{\partial c_3} \right) - \overline{b}_j \cdot \left( 1 + c_j \cdot \frac{\partial \ln(\overline{\gamma}_j)}{\partial c_j} \right) \right] \left( \sum_{i=1}^{3} \frac{c_i \cdot z_i \cdot 2 \cdot \overline{b}_i}{c_i} \right)
\]

for \( k = 1, 2 \) and \( j = 1, 2 \)
Appendix H

**Quasilinearization of Non-linear Terms**

**Numerical Approximation of Resin Phase Continuity Equations**

- **Ternary System** -

Several terms appear in the resin phase continuity equations for the ternary system which are non linear, in addition to the equilibrium relationship. Multiplication of finite difference equations approximating those terms result in algebraic variables which cannot be separated, and the quasilinearization approach is used to resolve those terms. The following terms are treated accordingly.

\[
\frac{\partial^2 \bar{X}_i}{\partial \rho^2} \text{ at } j^{+1}\text{th time step}
\]

**quasilinearization, where** \(n+1\) **is the** \(n+1\) **th iteration**

leads to:

\[
\left(\frac{\partial \bar{X}_i}{\partial \rho}\right)_{n+1} = \left(\frac{\partial \bar{X}_i}{\partial \rho}\right)_n + \left(\frac{\partial \bar{X}_i}{\partial \rho}\right)_{n+1} \left(\frac{\partial \bar{X}_i}{\partial \rho}\right)_{n+1} - \left(\frac{\partial \bar{X}_i}{\partial \rho}\right)_n^{2}
\]

\[
= 2 \cdot \left(\frac{\partial \bar{X}_i}{\partial \rho}\right)_n \cdot \left(\frac{\partial \bar{X}_i}{\partial \rho}\right)_{n+1} - \left(\frac{\partial \bar{X}_i}{\partial \rho}\right)_n^{2}
\]

\[
\frac{\partial \bar{X}_i}{\partial \rho} \cdot \frac{\partial \bar{X}_i}{\partial \rho} \text{ at } j^{+1}\text{th time step}
\]

**quasilinearization leads to:**

\[
\left(\frac{\partial \bar{X}_i}{\partial \rho} \cdot \frac{\partial \bar{X}_i}{\partial \rho}\right)_{n+1} = \left(\frac{\partial \bar{X}_i}{\partial \rho} \cdot \frac{\partial \bar{X}_i}{\partial \rho}\right)_n + \left(\frac{\partial \bar{X}_i}{\partial \rho} \cdot \frac{\partial \bar{X}_i}{\partial \rho}\right)_{n+1} - \left(\frac{\partial \bar{X}_i}{\partial \rho} \cdot \frac{\partial \bar{X}_i}{\partial \rho}\right)_n \cdot \left(\frac{\partial \bar{X}_i}{\partial \rho}\right)_{n+1}
\]

continued
(H-2) continued
\[ + \left[ \frac{\partial \overline{x}_i}{\partial \rho} \right]_{n+1} - \left[ \frac{\partial \overline{x}_i}{\partial \rho} \right]_n \cdot \left[ \frac{\partial \overline{x}_i}{\partial \rho} \right]_n \]

since:
\[ a \left( \frac{\partial \overline{x}_i}{\partial \rho} \right) \cdot \left( \frac{\partial \overline{x}_i}{\partial \rho} \right) = \left( \frac{\partial \overline{x}_i}{\partial \rho} \right) \]

and:
\[ a \left( \frac{\partial \overline{x}_i}{\partial \rho} \right) \cdot \left( \frac{\partial \overline{x}_i}{\partial \rho} \right) = \left( \frac{\partial \overline{x}_i}{\partial \rho} \right) \]

clearing terms in (H-2) leads to:
\[ 0 = \left( \frac{\partial \overline{x}_i}{\partial \rho} \right)_{n+1} \cdot \left( \frac{\partial \overline{x}_i}{\partial \rho} \right)_{n+1} \cdot \left( \frac{\partial \overline{x}_i}{\partial \rho} \right)_n + \left( \frac{\partial \overline{x}_i}{\partial \rho} \right)_{n+1} \cdot \left( \frac{\partial \overline{x}_i}{\partial \rho} \right)_n \]

Since the concentration values from the \( n \)th iteration are known, they are treated as constants in the \( n+1 \)th iteration, and the iterations are repeated until convergence criteria are met at each time step. Satisfactory solution tolerances were met in 3 to 5 iterations in most cases studied.
Appendix I

COUPLING OF RESIN PHASE MATRIX EQUATIONS TO THE
FLUID PHASE MATRIX EQUATIONS - TERNARY FIXED BED MODEL

A. General

The seven diagonal array solution algorithm allows the solution of the coupled resin phase diffusion equations. The boundary conditions at the resin—electrolyte phase interface allow carrying the resin phase compositions into the solution of the coupled electrolyte phase equations, which equations are solved by the five diagonal array algorithm. This section deals with the coupling of the matrices and vectors describing the equations for both phases, through that boundary condition relationship.

B. Resin Phase Equations — Matrix Notation

In matrix notation, the coupled resin phase equations (5-81) and (5-82) are:

\[ A_i \cdot \bar{x}_1 = D_i + R_i \cdot \bar{x}_2 + E_1 \cdot x_1 (i,j+1)_{n+1} - E_2 \cdot x_2 (i,j+1)_{n+1} \]

and

\[ A_i \cdot \bar{x}_2 = \Pi_i + O_i \cdot \bar{x}_1 + \Omega_2 \cdot x_2 (i,j+1)_{n+1} - \Omega_1 \cdot x_1 (i,j+1)_{n+1} \]

C. Seven Diagonal Array Algorithm and Resolving of Resin Surface Concentration in terms of the Bulk Electrolyte Concentrations

The coefficients and vectors are loaded into a scratch 7 diagonal array (fig. 5-1) and (fig. E-2) in which the vectors \(bb_N\) and \(bb_{N-1}\) are the solution vectors for equations (I-2) and (I-1) respectively. These two vectors also contain the
boundary conditions:

\[ \Omega_2 \cdot x_2(bulk)_{(i,j+1)_{n+1}} \]

\[ \Omega_1 \cdot x_1(bulk)_{(i,j+1)_{n+1}} \]

and:

\[ E_1 \cdot x_1(bulk)_{(i,j+1)_{n+1}} \]

\[ E_2 \cdot x_2(bulk)_{(i,j+1)_{n+1}} \]

respectively.

Taking advantage of the solution algorithms given by equation (E-32),

\[ \bar{x}_2(M,i,j+1)_{n+1} = \frac{E}{F} \frac{\bar{x}_2(M,i,j+1)_{n+1}}{\omega_N} = \left( F_{-1}N \cdot \beta^7 \right) \frac{\bar{x}_2(M,i,j+1)_{n+1}}{\omega_N} \]

where \( b_{+}^7 = \pi_n(M,i,j) \)

but \( b_{+}^7 \) includes the term \( \left( \Omega_2 \cdot x_2(bulk)_{(i,j+1)_{n+1}} - \Omega_1 \cdot x_1(bulk)_{(i,j+1)_{n+1}} \right) \)

so

\[ \bar{x}_2(M,i,j+1)_{n+1} = \frac{E}{F} \frac{\bar{x}_2(M,i,j+1)_{n+1}}{\omega_N} \]

\[ + \left( \frac{\Omega_2 \cdot x_2(bulk)_{(i,j+1)_{n+1}} - \Omega_1 \cdot x_1(bulk)_{(i,j+1)_{n+1}}}{\omega_N} \right) \]

and therefore by equations (E-43) and (I-1):

\[ FF_{N-1}^7 = \left( \frac{b_{+}^7 N-1 FF_{N-4}^7 \cdot \theta^7_{N-4} FF_{N-3}^7 \cdot \alpha^7_{N-3} FF_{N-2}^7 \cdot \beta^7_{N-2}}{\omega_{N-1}} \right) \]

\[ + \left( \frac{E_1 \cdot x_1(bulk)_{(i,j+1)_{n+1}} - E_2 \cdot x_2(bulk)_{(i,j+1)_{n+1}}}{\omega_{N-1}} \right) \]

\[ = \frac{E}{F} + \left( \frac{E_1 \cdot x_1(bulk)_{(i,j+1)_{n+1}} - E_2 \cdot x_2(bulk)_{(i,j+1)_{n+1}}}{\omega_{N-1}} \right) \]
where:
\[ b_{N-1}^7 = d_n(M, i, j) \]  

so from equation (E-38) the following expression results:
\[ \bar{x}_1(M, i, j+1)_{n+1} = F_{N-1}^7 - \gamma_{N-1}^7 \cdot \bar{x}_2(M, i, j+1)_{n+1} \]  

Combining equation (I-10) with (I-12) yields:
\[ \bar{x}_1(M, i, j+1)_{n+1} = \epsilon \epsilon + \left[ \frac{E_1 n \cdot x_1(i, j+1)_{n+1} - E_2 n \cdot x_2(i, j+1)_{n+1}}{\omega_{N-1}^7} \right] \]
\[ - \gamma_{N-1}^7 \cdot \bar{x}_2(M, i, j+1)_{n+1} \]

and in terms of \( \bar{x}_2 \), using equations (I-10) and (I-9):
\[ \bar{x}_2(M, i, j+1)_{n+1} = \epsilon + \left[ \frac{\Omega_2 n \cdot x_2(i, j+1)_{n+1} - \Omega_1 n \cdot x_1(i, j+1)_{n+1}}{\omega_N^7} \right] \]
\[ - \left( \frac{\epsilon \epsilon \beta_{N}^7}{\omega_N^7} \right) \cdot \frac{E_1 n \cdot x_1(i, j+1)_{n+1} - E_2 n \cdot x_2(i, j+1)_{n+1}}{\omega_{N-1}^7 \cdot \omega_N^7} \cdot \beta_N^7 \]
In view of (I-13) and (I-14), the following can be shown:

\[ \bar{x}_1(M,i,j+1)_{n+1} = \varepsilon \varepsilon + \left( \frac{E_1n \cdot x_1(i,j+1)_{n+1} - E_2n \cdot x_2(i,j+1)_{n+1}}{\omega_{N-1}} \right) - \gamma_{N-1} \cdot \varepsilon + \left( \frac{\gamma_{N-1} \cdot \beta_{N-1} \cdot \psi_1 \cdot x_1(i,j+1)_{n+1}}{\omega_{N-1}^7} \right) + \gamma_{N-1} \cdot \psi_1 \cdot x_1(i,j+1)_{n+1} \]

\[ = \varepsilon \varepsilon \cdot \left( 1 + \left( \frac{\beta_{N-1} \cdot \gamma_{N-1}^7}{\omega_{N-1}^7} \right) \right) + \left( \frac{E_1n}{\omega_{N-1}} + \psi_1 \cdot \gamma_{N-1} \right) \cdot x_1(i,j+1)_{n+1} - \varepsilon \cdot \gamma_{N-1} \]

So,

\[ \bar{x}_1(M,i,j+1)_{n+1} = \tau_2 + \rho_1 \cdot x_1(i,j+1)_{n+1} - \rho_2 \cdot x_2(i,j+1)_{n+1} - \varepsilon \cdot \gamma_{N-1} \]

The resin phase concentrations at the boundary condition have now been described in terms of the bulk electrolyte phase concentrations at every column node, i = 1 to I.
D. Resolution of the Electrolyte Phase Concentrations — Five Diagonal Array

In matrix notation, the coupled electrolyte phase equations, (7-27) and (7-28), are:

\[
\begin{align*}
B_{1n}^{i} \cdot x_{1}^{i+1}(i,j+1)_{n+1} &= U_{1n}^{i} + V_{1n}^{i} \cdot \left( \frac{dx_{1}^{*}}{dx_{1}^{\text{int}}} \right)_{n} (i,j+1) \cdot \bar{x}_{1}(M,i,j+1)_{n+1} \\
&+ V_{1n}^{i} \cdot \left( \frac{dx_{1}^{*}}{dx_{2}^{\text{int}}} \right)_{n} (i,j+1) \cdot \bar{x}_{2}(M,i,j+1)_{n+1}
\end{align*}
\]

\[
\begin{align*}
B_{2n}^{i} \cdot x_{2}^{i+1}(i,j+1)_{n+1} &= U_{2n}^{i} + V_{2n}^{i} \cdot \left( \frac{dx_{2}^{*}}{dx_{2}^{\text{int}}} \right)_{n} (i,j+1) \cdot \bar{x}_{2}(M,i,j+1)_{n+1} \\
&+ V_{2n}^{i} \cdot \left( \frac{dx_{2}^{*}}{dx_{1}^{\text{int}}} \right)_{n} (i,j+1) \cdot \bar{x}_{1}(M,i,j+1)_{n+1}
\end{align*}
\]

Substituting (I-16) and (I-14) into (I-17) and (I-18) results in:

\[
\begin{align*}
B_{1n}^{i} \cdot x_{1}^{i+1}(i,j+1)_{n+1} &= U_{1n}^{i} + V_{1n}^{i} \cdot \left( \frac{dx_{1}^{*}}{dx_{1}^{\text{int}}} \right)_{n} (i,j+1) \cdot \left[ \tau_{2} + \rho_{1} \cdot x_{1}^{i+1}(i,j+1)_{n+1} \\
&- \rho_{2} \cdot x_{2}^{i+1}(i,j+1)_{n+1} - \varepsilon \cdot \gamma_{N-1}^{7} \right] + V_{1n}^{i} \cdot \left( \frac{dx_{1}^{*}}{dx_{2}^{\text{int}}} \right)_{n} (i,j+1) \\
&+ \varepsilon - \left( \frac{\varepsilon \cdot \beta_{N}^{7}}{\omega_{N}^{7}} \right) + \psi_{2} \cdot x_{2}^{i+1}(i,j+1)_{n+1} - \psi_{1} \cdot x_{1}^{i+1}(i,j+1)_{n+1}
\end{align*}
\]
and,

\[
B_2 \cdot x_{bulk}^{(i,j+1)}_{n+1} = U_2 \cdot \left\{ \begin{array}{l}
V_2 \cdot \left( \frac{dx_2}{dx_{int}} \right)_{n} \cdot \left( \begin{array}{c}
\varepsilon - \frac{\varepsilon \varepsilon \beta^7}{\omega_{N}} \\
+ \psi_2 \cdot x_{bulk}^{(i,j+1)}_{n+1} - \psi_1 \cdot x_{bulk}^{(i,j+1)}_{n+1}
\end{array} \right) \\
+ \left( \begin{array}{c}
V_2 \cdot \left( \frac{dx_2}{dx_{int}} \right)_{n} \cdot \left( \begin{array}{c}
\tau_2 + \rho_1 \cdot x_{bulk}^{(i,j+1)}_{n+1} - \rho_2 \cdot x_{bulk}^{(i,j+1)}_{n+1}
\end{array} \right)
\end{array} \right)
\end{array} \right)
\]

Collecting terms, the following expressions result:

(VI(i,j+1)_{n+1})

\[
B_1 \cdot x_{1}^{(i,j+1)}_{bulk} = U_1 \cdot \left\{ \begin{array}{l}
V_1 \cdot \left( \frac{dx_1}{dx_{int}} \right)_{n} \cdot \left( \begin{array}{c}
\varepsilon - \frac{\varepsilon \varepsilon \beta^7}{\omega_{N}} \\
+ \psi_1 \cdot x_{bulk}^{(i,j+1)}_{n+1}
\end{array} \right)
\end{array} \right)
\]

\[
\begin{array}{l}
\text{g1(i,j+1)_{n}} \\
+ \left( \begin{array}{l}
V_1 \cdot \left( \frac{dx_1}{dx_{int}} \right)_{n} \cdot \left( \begin{array}{c}
\rho_1 - V_1 \cdot \left( \frac{dx_1}{dx_{int}} \right)_{n} \cdot \psi_1 \\
\end{array} \right) \cdot x_{bulk}^{(i,j+1)}_{n+1}
\end{array} \right)
\end{array}
\]

\[
\begin{array}{l}
\text{g2(i,j+1)_{n}} \\
+ \left( \begin{array}{l}
V_1 \cdot \left( \frac{dx_1}{dx_{int}} \right)_{n} \cdot \left( \begin{array}{c}
\psi_2 - V_1 \cdot \left( \frac{dx_1}{dx_{int}} \right)_{n} \cdot \rho_2 \\
\end{array} \right) \cdot x_{bulk}^{(i,j+1)}_{n+1}
\end{array} \right)
\end{array}
\]
Final condensation of terms results in:

\[ B_{1n} \cdot x_{1} (i,j+1)_{n+1} = U_{1n} + VV_{1n} (i,j+1)_{n} + g_{1n} (i,j+1)_{n} \cdot x_{1, \text{bulk}} (i,j+1)_{n+1} + g_{2n} (i,j+1)_{n} \cdot x_{2, \text{bulk}} (i,j+1)_{n+1} \]

where from Figure (7-1),

\[ \alpha'_{1i} = \alpha_{1i} - g_{1n} (i,j+1)_{n} \]

Likewise,

\[ VV_{2n} (i,j+1)_{n} \]

\[ B_{2n} \cdot x_{2} (i,j+1)_{n+1} = U_{2n} + V_{2n} \cdot \left( \frac{d_{x_{2}}^{*}}{d_{x_{2}}^{\text{int}}} (i,j+1)_{n} \right) \cdot \left( \varepsilon - \frac{\varepsilon \cdot \beta_{N}^{7}}{\nu_{N}^{7}} \right) \]

\[ + V_{2n} \cdot \left( \frac{d_{x_{2}}^{*}}{d_{x_{1}}^{\text{int}}} (i,j+1)_{n} \right) \cdot \left( \tau_{2} - \varepsilon \cdot \gamma_{N-1}^{7} \right) \]

\[ h_{1n} + \frac{d_{x_{2}}^{*}}{d_{x_{2}}^{\text{int}}} (i,j+1)_{n} \cdot \frac{d_{x_{2}}^{*}}{d_{x_{1}}^{\text{int}}} (i,j+1)_{n} \cdot \psi_{2} - V_{2n} \cdot \frac{d_{x_{2}}^{*}}{d_{x_{2}}^{\text{int}}} (i,j+1)_{n} \cdot \rho_{2}^{*} \cdot x_{2, \text{bulk}} (i,j+1)_{n+1} \]

\[ h_{1n} + \frac{d_{x_{2}}^{*}}{d_{x_{2}}^{\text{int}}} (i,j+1)_{n} \cdot \rho_{1} - V_{2n} \cdot \frac{d_{x_{2}}^{*}}{d_{x_{2}}^{\text{int}}} (i,j+1)_{n} \cdot \psi_{1}^{*} \cdot x_{1, \text{bulk}} (i,j+1)_{n+1} \]

\[ B_{2n} \cdot x_{2} (i,j+1)_{n+1} = U_{2n} + VV_{2n} (i,j+1)_{n} + h_{2n} (i,j+1)_{n} \cdot x_{2, \text{bulk}} (i,j+1)_{n+1} + h_{1n} (i,j+1)_{n} \cdot x_{1, \text{bulk}} (i,j+1)_{n+1} \]
where from Figure (7-1),

\[
a_2' = a_2 - h_2(i,j+1)n
\]  \hspace{1cm} (I-26)

The five diagonal algorithm, called ZSOLVE, solves for values of \(x_1(i,j+1)n+1\) and \(x_2(i,j+1)n+1\) as illustrated in Fig. (7-1), and as outlined in Appendix F.

E. Resolution of Resin Phase Concentrations - Seven Diagonal Array

In order to compute the values for \(\bar{x}_1(M,i,j+1)n+1\) and \(\bar{x}_2(M,i,j+1)n+1\), we note that from Appendix E and Fig. (5-1),

\[
\bar{x}_2(M,i,j+1)n+1 = FF^7N \quad \text{as defined by equation (I-9), for which the } n+1 \text{th values of } x_1(i,j+1)n+1 \text{ are now known, and:}
\]

\[
\bar{x}_1(M,i,j+1)n+1 = FF^7N-1 - \gamma^7N-1 * \bar{x}_2(M,i,j+1)n+1
\]  \hspace{1cm} (I-12)

The \(FF^7N-1\) term can be computed from (I-10) since the \(x_1(i,j+1)n+1\) values are known, and \(\bar{x}_1(M,i,j+1)n+1\) values can be computed from (I-12) since \(\bar{x}_2(M,i,j+1)n+1\) values are known. Using the algorithm described in Appendix E for the remaining sequence of \(\bar{x}_1\) values, and keeping in mind that the \(\bar{x}_1\) and \(\bar{x}_2\) values alternate in the scratch array XSOLVE, the remaining resin phase concentrations are calculated.

The sequence is repeated, updating quasilinearized expressions at each iteration, until convergence is attained for the given time step.
NUMERICAL SOLUTION DEVELOPMENT
MODEL FOR ION EXCHANGE IN FIXED BEDS

BINARY SYSTEMS

A. Resin and Fluid Phase Matrix Equations

The resin phase equations in matrix notation are given in equation (5-71) as:

\[ A_{i} \cdot \bar{x}_{i} = D_{i} + E_{i} \cdot \bar{x}_{i} (i,j+1)_{n+1} \]  
\[ \bar{x}_{i+1} \]  
\[ \text{(J-1)} \]

The fluid phase equations in matrix notation are given as an adaptation of equation (7-27):

\[ B_{i} \cdot \bar{x}_{i} (i,j+1)_{n+1} = U_{i} + V_{i} \cdot \left[ \frac{dx_{i}}{dx_{1}} \right]_{(i,j+1)} \cdot \bar{x}_{i} (M,i,j+1)_{n+1} \]  
\[ \text{(J-2)} \]

B. Tridiagonal Treatment, Resin Phase

Recognizing that \( E_{i} \cdot \bar{x}_{i} (i,j+1)_{n+1} \) applies only at \( m = M \), the resin–film interface, then:

\[ A_{i} \text{ can be divided into two matrices, } L_{i} \text{ and } Q_{i}, \]

such that:

\[ A_{i} = L_{i} \cdot Q_{i} \]  
\[ \text{(J-3)} \]

where:

\[ L_{i} = \begin{bmatrix} w_{1,i} & 0 & \cdots & 0 & 0 \\ c_{2,i} & w_{2,i} & \cdots & 0 & 0 \\ \vdots & \vdots & \ddots & \vdots & \vdots \\ c_{m,i} & \cdots & w_{m,i} & \cdots & c_{M,i} \\ 0 & \cdots & \cdots & c_{M,i} & \cdots & WM,i \end{bmatrix} \]

\[ \text{lower diagonal matrix} \]
and,

\[
Q_{in} = \begin{bmatrix}
1 & q_{1,i} \\
& 1 & q_{2,i} \\
& & \ddots & \ddots \\
& & & 1 & q_{m,i}
\end{bmatrix}
\]

upper diagonal matrix

and since, from equation (5-72)

\[
A_{in} = \begin{bmatrix}
a_{1,i} & b_{1,i} & & \\
& c_{2,i} & a_{2,i} & b_{2,i} \\
& & \ddots & \ddots \\
& & & c_{M,i} & a_{M,i}
\end{bmatrix}
\]

If matrix \(L_n\) is multiplied by matrix \(Q_n\), and the result is set equal to matrix \(A_n\), there results the following algorithm:

\[
w_{1,i} = a_{1,i}
\]

\[
q_{1,i} = \frac{b_{1,i}}{w_{1,i}}
\]

\[
w_{m,i} = a_{m,i} - c_{m,i} \times q_{m-1,i}
\]

\[
q_{m,i} = \frac{b_{m,i}}{w_{m,i}}
\]

for \(m = 2\) to \(M-1\) and \(i = 1\) to \(I\), for the \(n\)th iteration

Define vector \(F_{in+1}^i \equiv Q_{in} \times x_{in+1}^i\)

where:

\[
F_{in+1}^i = \begin{bmatrix}
f_{1,i} \\
f_{m,i} \\
f_{M,i}
\end{bmatrix}
\]
and, 
\[ L_i \cdot F_{i,n}^{n+1} = D_i + E_i \cdot x_1^{(i,j+1)_{n+1}} \] 
then, 
\[ f_{i,n}^{n+1} = \begin{pmatrix} d_{i,i} \\ w_{i,i} \end{pmatrix}_n \] 
resulting in: 
\[ f_{m,i}^{n+1} = \begin{pmatrix} d_{m,i} - c_{m,i} \cdot f_{m-1,i} \\ w_{m,i} \end{pmatrix}_n \] 
for \( m = 2 \) to \( M-1 \)

At the resin surface, the relationship becomes:
\[ f_{M,i}^{n+1} = \begin{pmatrix} d_{M,i} - c_{M,i} \cdot f_{M-1,i} + e_i \cdot x_1^{(i,j+1)_{n+1}} \\ w_{M,i} \end{pmatrix}_n \] 

In addition, since \( Q_i \) is upper triangular with all diagonal elements equal to unity, then:
\[ \bar{x}_{1(M,i,j+1)_{n+1}} = f_{M,i}^{n+1} = \begin{pmatrix} d_{M,i} - c_{M,i} \cdot f_{M-1,i} + e_i \cdot x_1^{(i,j+1)_{n+1}} \\ w_{M,i} \end{pmatrix}_n \]

and \( \bar{x}_{1(M,i,j+1)_{n+1}} \) is now expressed as a function of \( x_1^{(i,j+1)_{n+1}} \).

C. Tridiagonal Treatment, Fluid Phase Resolution of Concentrations

Referring to the fluid phase coefficient matrix, \( B_{1i}^{n} \) (equation (7-29)), and vector \( V_{1i} \) (equation (7-31)), and using the revised notation for binary systems, i.e. \( B_{1i}^{n} = B_{i}^{n} \), and \( V_{1i} = V_{i} \), then:
but since $x_1(M,i,j+1)_{n+1}$ is now known in terms of $x_1(i,j+1)_{n+1}$, a revised coefficient matrix and solution vector can be derived for the fluid phase,

\[
B_i = \begin{bmatrix}
\alpha_1 & \beta_1 \\
\gamma_2 & \alpha_2 & \beta_2 \\
\gamma_i & \alpha_i & \beta_i \\
\gamma_I & \alpha_I \\
\end{bmatrix}_{n}
\]

and $V_i = \begin{bmatrix} v_1 \\ \vdots \\ v_i \\ \vdots \\ v_I \end{bmatrix}_{n}$

Also, substitution of (J-17) in equation (J-2) results in:

\[
\dot{v}_i = \alpha_i' = \alpha_i - \left( e_i \cdot v_i \cdot \frac{dx^*_{j+1}}{dx^*_{i,j+1}} \right)_{n}
\]

Also, substitution of (J-17) in equation (J-2) results in:

\[
V_i = \begin{bmatrix} \dot{v}_1 \\ \dot{v}_2 \\ \vdots \\ \dot{v}_i \\ \vdots \\ \dot{v}_I \end{bmatrix}_{n}
\]

where:

\[
\dot{v}_i = v_i \cdot \left( \frac{dM_i - cM_i f_{M-1,i}}{w_{M,i} n} \right)_{n}
\]

for $i = 1$ to $I$
The matrix equation then becomes:
\[
B'_{1n} \cdot x_{1,ij+1}^{n+1} = U_{1n} + V'_{1n}
\]  \hspace{1cm} (J-23)

\section*{D. Resin Phase – Resolution of Concentrations}

The values of \(x_{1,ij+1}^{n+1}\) are found with the tridiagonal bulk algorithm derived above for the fluid phase. These values are substituted in equation (J-17) for \(f_{M,i}^{n+1}\) to obtain the values for the resin phase compositions at the surface, \(x_{1,M,ij+1}^{n+1}\), at each \(i\) node. In view of the definition of \(F_{1n+1}\), equation (J-11):
\[
\overline{x}_{1,M,ij+1}^{n+1} = f_{m,i}^{n+1} - q_{m,i} \cdot \overline{x}_{1,m+1,ij+1}^{n+1}
\]  \hspace{1cm} (J-24)

for \(i = 1\) to \(I\) and \(m = 1\) to \(M-1\)

\section*{E. Convergence}

The remaining values of resin compositions at each node \(m\) can be calculated for each column node \(i\) at the \(n+1^{th}\) iteration during time step \(j+1\). The iterative process is continued at time step \(j+1\) until convergence criteria have been met for all values of \(\overline{x}_{1,m,ij+1}\) and \(x_{1,ij+1}\). The criterion for convergence normally was:
\[
\frac{(\text{value}_{n+1} - \text{value}_n)}{\text{value}_{n+1}} < 10^{-5}
\]  \hspace{1cm} (J-25)

\section*{F. Summary}

In summary, the defined resin solution vectors, \(f_{m,i}^{n+1}\), made up of the lower diagonal coefficients and solution vectors from the resin phase matrix equation are calculated from the resin center to the resin surface-fluid interface. The defined vectors, \(f_{M,i}^{n+1}\),
are then carried into the solution phase to obtain the column compositions through the equilibrium relationship. This relationship, as result of the quasilinearization procedure, expresses $x_{1}(i)_{\text{bulk int}}$ as a linear function of $x_{1}(M,i)_{\text{int}}$. The column matrix equations are then solved for $x_{1}(i)_{\text{bulk}}$.

The values of the vector $f_{M,i}$ are then calculated, and since $f(M,i) = x_{1}(M,i)$ the values of $x_{1}(M,i)$ are now known. Through use of the calculated resin phase matrix equation upper diagonal coefficients and the defined factors, $f_{m,i}$, the remaining compositions in the resin phase are calculated from the surface to the center dimension. Equilibrium estimates are refined at each iteration through the quasilinearization procedure and the process is repeated until convergence is attained. Normally, with proper "conditioning" of the column prior to the time step $j=1$, 5 iterations per time step were required to meet the convergence criteria at the beginning, dropping quickly to 3 and continuing to the end of the run. This was true of both the concentration and the activity models.
Appendix K

RFMAVD – FEATURES OF SIMULATION PROGRAM
FOR ION EXCHANGE IN FIXED BEDS
– BINARY SYSTEMS –

A. Main Program

1. Input and Display Options

RFMAVD, (Revised Final Mass Action (equilibrium) Variable (solid phase) Diffusivity), the binary model program, calls required input data, then displays the read in information plus key calculated parameters such as:

- diffusion coefficients of ions in electrolyte phase at infinite dilution
- diffusion coefficients of salts in electrolyte phase at infinite dilution
- diffusion coefficients of salts in electrolyte phase corrected to feed normality
- activity coefficients of salts at feed normality
- Reynolds number
- Peclet number
- initial column concentration values, $x_1(m,i)$, $x_1(i)$, $x_*(i)$, $x_{\text{bulk}}^*(i)$, $x_{\text{int}}^*(i)$
- initial values of film coefficients, $k_{\text{eff}}(i)$
- initial average Biot number
- percent bed capacity available after initialization procedure
plug flow time to completely utilize available
capacity in solution and resin
d debug options selected

2. **Run Options** available include:
   a. Operation in a "batch" mode.
   b. Operation with resin phase equilibrium parameters
correlated in equivalent fraction or mole fraction
for the resin phase.
   c. Operation with activity driving force vs.
concentration driving force.
   d. Operation with Kataoka et al (50) electric field
model for equal valence exchange.
   e. Selection of a particle form factor, $\alpha = 0,1,2,$
for slab, cylinder, or sphere.
   f. Fractional adjustment of the constants in the
Carberry (11) or Koloini et al (54) film
coefficient correlation to observe effect on fit
of ECH curves.
   g. Selection of a bypass option in which a fraction of
the feed is assumed to bypass directly from the
feed inlet to mix with the effluent stream.
   h. Option of no printout until $x_{1\text{bulk}}$ (column outlet)
reaches a preselected value, i.e., first appearance
of "S" curve at column exit. Key column variables
are written out to a separate file during that "blind
period" for every 10 dimensionless time steps, with
every third file being replaced by the current data update. This allows examination and early "killing" of runs which appear to be of no interest to the user.

i. Since on some runs where, because of use of small grid spacing, column length spacing, and many time steps, CPU time can be up to 1-2 hours -- all required data is read out into a file every 50 minutes, which can be used to restart the program at that point if the computer is inadvertently shut down, or if the system has a batch queue time limit. In the latter case, an option is provided to shut the program down prior to the time limit for resubmission to the queue.

3. Output Options

o At selected times and intervals:
   - Profiles at 0.1 column lengths for species 1 for:
     - fluid phase composition
     - interface (equilibrium) composition of fluid
     - interface (equilibrium) composition of resin surface
     - film coefficient
   o and at a second selected interval
     - either
       - all profiles for j and j+1 time steps for
         \[ \bar{x}_1(m,i), x_1(i), x^*_1(i), D_1(m,i), \frac{\partial x^*_1(m,i)}{\partial x_1(M,i)} \]

and
\[ \frac{\partial x^*_1(i)}{\partial x_1(M,i)} \]
Complete flux balance for species 1 in both phases, showing convection flux, axial dispersion flux, holdup flux, film diffusion flux, and resin accumulation flux. In addition, an approximation of the B.C. flux is computed by a backward difference approximation for comparison to the resin accumulation flux. Species 1 film diffusion flux is also calculated by the net difference resulting from the fluid phase fluxes for comparison to the resin accumulation flux. All printed at column intervals preselected.

B. RFMAVD Subroutines

1. **Input**

   Reads in data, performs computations to obtain key program control parameters and constants, and writes out prerun information described above.

2. **INISHL**

   Computes the initial distribution for column nodes 2 to IMAX, of \( x_1(MMAX,i) \), \( x_1(i) \), and \( x_1^*(i) \), by a procedure described in Chapter VII, Section C, paragraph 2. This procedure is based on a flux balance in which diffusion flux into the top resin layer is set equal to the flux across the film. The equilibrium values, \( x_1^*(i) \), calculated by subroutine ISOTEM from the \( x_1(MMAX,i) \) values, and the film coefficients \( k_{eff}(i) \) calculated by subroutine KONST, are used to obtain
the $x_1(i)$ values which permit the flux balance to close at each column node. This subroutine is used in conjunction with an initializing procedure in the Main Program in which the flux at column node 1 is balanced, based on the Boundary Condition described by equation (7-21) in Chapter VII, Section B. If the first column increment is treated as a CSFTR, then

$$x_1(1) = \frac{x_1(2) + x_{\text{in}} \cdot \text{Pe} \cdot \Delta \text{ column}}{1 + \text{Pe} \cdot \Delta \text{ column}}$$

The procedure in the Main Program solves the monotonic non-linear relationship with a false positioning procedure (secant method) embodied in subroutine AKM33.

3. KONST

Computes $k_l(i)$ and the Biot numbers $(i)$, at eff 

every $j+1$ time step using time step $j$ values of

(1) average film concentrations if the pseudo electric field method is utilized, or (2) calculated values of

film concentrations $x_1^*(i)$ and $x_1(i)$ and $k_l(i)$ if the Kataoka et al (50) electric field model is selected (applies to homovalent exchange only).

Additionally calculates the $j^{+1}$th and $n^{th}$ iteration values of both the resin phase and the electrolyte phase matrix coefficients and solution vectors.

4. ISOTEM

Computes the values of electrolyte phase concentrations $x_1^*(i)$ in equilibrium with the resin
surface compositions $\bar{x}_1(\text{MMAX},i)$. This subroutine utilizes Lee's quasilinearization procedure to solve the non-linear equilibrium relationship. Since this procedure is a variant of the Newton-Raphson method, if a guess value close to the solution is utilized, rapid convergence is obtained. Since the concentrations do not change rapidly between time steps, use of the value of $x_1^*(i)$ from time step $j$ as a first guess makes this procedure work rapidly.

5. **LEECAL**

Calculates the derivatives required for the quasilinearization procedure which allows expression of $x_1^*(i)$ as a linear function of $\bar{x}_1(\text{MMAX},i)$. This procedure is:

$$
x_1^*(i) = x_1^*(i) + \left\{ \frac{d \left[ f[x_1^*(i), \bar{x}_1(\text{MMAX},i)] \right]}{d \bar{x}_1(\text{MMAX},i)} \right\}_n \int_n^{x_1^*(i)} \left[ \bar{x}_1(\text{MMAX},i)_{n+1} - \bar{x}_1(\text{MMAX},i)_n \right]
$$

$$
+ \left\{ \frac{d \left[ f[x_1^*(i), \bar{x}_1(\text{MMAX},i)] \right]}{d x_1^*(i)} \right\}_n \int_n^{x_1^*(i)} \left[ x_1^*(i) - x_1^*(i) \right]
$$
Therefore the derivatives given in the expression must be updated at each iteration, \( n + 1 \), with \( n \)th values of the concentrations. The subroutine computes the derivatives by numerical perturbation, and calls the subroutines GAMCAL and GAMRCL to obtain the activity coefficients.

6. **GAMCAL**

Computes solution ionic strength, \( I \), and the electrolyte phase ionic species molal activity coefficients, \( \gamma_1 \) and \( \gamma_2 \), which are derived from the Bromley equation. In addition, the derivatives, \( d\gamma_i/dI \) are calculated for use in the quasilinearization procedure used to compute the equilibrium concentrations in ISOTEM.

7. **GAMRCL**

Computes resin phase ion species activity coefficients based on the Redlich-Kister equation for binary systems, with parameters correlated in either mole or equivalent fraction concentration units, based on concentration arguments from the various calling subroutines.

8. **LEEKON**

Performs the solution to the coupled resin and electrolyte matrices at each time step, \( j+1 \), and iteration, \( n \), as described in Appendix J. Portions of the solution vector and coefficient matrices involving the boundary condition at the resin-film interface
are updated at the beginning of this subroutine using \( n \)th iteration values of compositions. The \( n+1 \)th values of \( x_1(m,i) \) and \( \bar{x}_1(i) \) are computed, and ISOTEM is called to calculate values of \( x_1^{*}(i) \) from the latest values of \( \bar{x}_1(m,i) \).

9. DFDNNU

If the activity driving force model is used then in arriving at the term \( \frac{\partial \bar{D}_1(m,i)}{\partial \bar{x}_1(m,i)} \), which is needed in the resin phase continuity equation, the following expression must be evaluated and added to the simpler partial derived for the concentration driving force model:

\[
\frac{\partial \bar{D}_1 \cdot \bar{D}_2 \cdot \bar{x}_1 \cdot \bar{x}_2 \cdot \bar{y}_1}{\partial \bar{x}_1} \left[ \frac{\partial n}{\partial \bar{y}_1} \right]
\]

The expression is separated into parts \( u \) and \( v \), and the relationship \( d(uv) = u \cdot dv + v \cdot du \) is used in the solution. The derivative of part \( u \) is taken analytically, and \( dv \) is solved by numerical perturbation.

10. AKM33

This subroutine, developed by Dr. A.K.S. Murthy, involves the secant method false positioning technique to solve non-linear relationships which are monotonic with respect to the variable whose value is required.
C. Main Program Function Details

Performs the following functions:

- calls Input

- performs the initialization procedure involving calling the subroutine INISHL. Prints out initial concentrations and other parameters resulting from the initialization procedure

- performs the calculation procedure for dimensionless time, \( j = 1 \) to \( j_{max} \), by performing the following functions:
  - at iteration \( n = 1 \), sets all concentrations to values from the last time step \( j \) as a first approximation
  - at iterations \( n = 2 \) to \( n(\text{convergence}) \), resets all \( j+1^{th} \) concentration approximations calculated at the \( n-1^{th} \) iteration to those calculated at the \( n^{th} \) iteration. These values are used to compute the derivatives required in the quasilinearization procedure.
  - computes the effective diffusion coefficients, \( \bar{D}_{eff}(m,i) \), based on the \( j^{th} \) concentration values. The effective diffusion coefficients can be (1) based on the harmonic mean of the individual species diffusion coefficients (Fick's Law), (2) based on the concentration model of Nernst-Planck, or (3) include activity effects so that the activity driving force is utilized instead of concentration.
o computes the partial derivatives of $\overline{D}_{\text{eff}}(m,i)$ with respect to $\overline{x}_i(m,i)$ when using the N-P model. When the activity driving force model is used, calls subroutine DFDVNU to obtain the second derivative terms.

o calls LEEKON for each iteration, which in turn calls KONST and LEECAL to obtain the $n^{th}$ values of $\overline{x}_i(m,i)$ and $x_1(i)$ at time step $j+1$. Also, LEEKON calls ISOTEM to obtain the $n^{th}$ values of $x_1^*(i)$.

o checks the indicator, ICOUNT, which represents the number of variables not meeting convergence criteria, and if >0, recycles through another iteration. The user inputs the maximum number of iterations desired before moving on, from $j$ to $j+1$. Since this override represents moving on without total convergence, a warning message is written out giving the count of variables not meeting convergence criteria. Iterations normally ranged between 3-5 for convergence. If the system was ill conditioned due to poor initialization, or if one species reached a very low value (binary), increasing the maximum number of iterations above, say, 20 did not provide convergent solutions. Simulations containing non-convergent sections were discarded as suspect.

o controls all output options.
TERNEX — FEATURES OF SIMULATION PROGRAM
FOR ION EXCHANGE IN FIXED BEDS
— TERNARY SYSTEMS —

A. Main Program

1. Input and Display Options

TERNEX, (TERNary Ion EXchange), the ternary model program, calls required input data, then displays the read in information plus key calculated parameters such as:

- diffusion coefficients of ions in electrolyte phase at infinite dilution
- diffusion coefficients of salts in electrolyte phase at infinite dilution
- diffusion coefficients of salts in electrolyte phase corrected to feed normality
- activity coefficients of salts at feed normality
- Reynolds number
- Peclet number

and displays the initial values of \( x_j(i) \), \( x_*^j(i) \), \( j_{\text{max}} \), \( j_{\text{bulk}} \), \( j_{\text{int}} \), and \( k_\text{eff} \) for the three species.

Additionally, the column average concentration for each species is shown for each phase after initialization to give an indication of column "preloading". Debugging options are shown as selected by the user, if required. Unlike the binary model, RFMAVD, only a few run options are available.
2. **Run Options:**

a. Option of no printout until $x_{1 \text{bulk}}$ reaches a preselected value, i.e., first appearance of "S" curve at column exit. Key column variables are written out to a separate file during that "blind period", for every 10 dimensionless time steps, with every third file being replaced by the current data update. This allows examination and early "killing" of runs which appear to be of no interest to the user.

b. Since on some runs where, because of use of small grid spacing, column length spacing, and many time steps, CPU time can be up to 1-2 hours, all required data is read out into a file every 50 minutes which can be used to restart the program at that point if the computer is inadvertently shut down, or if the system has a batch queue time limit. In the latter case, an option is provided to shut the program down prior to the time limit for resubmission to the queue.

3. **Output Options:**

   o At selected times
      - Column average concentration for each species, both phases.
      - Profiles @ 0.1 column lengths
        - fluid phase - each component
        - film coefficients - each component.

   o At other selected times
Either

Complete flux balance for each species in both phases, showing convection flux, dispersion flux, holdup flux, film diffusion flux, matching resin boundary condition flux, resin boundary condition flux, and resin accumulation flux. In addition, an approximation of the B.C. flux is computed by a backward difference approximation, and species 3 fluxes are computed and also arrived at by difference from species 1 and 2 fluxes for comparison. All printed at column intervals preselected.

Or

Complete composition printout of \( \bar{x}_i \), \( x_{ibulk}^i \), and \( x_{int}^i \) for \( i = 1,2,3 \) at both \( j \) and \( j+1 \) time steps at all column and resin nodes.

B. TERNEX Subroutines

1. INPUT

Reads in data, performs computations to obtain key program control parameters and constants, and writes out prerun information described above.

2. ENISHL

Supported by subroutines INSHD and FLXCAL, performs the flux balance solution to obtain initial compositions of \( \bar{x}_i(M,1) \), \( x_i(j) \), and \( x_{int}^i(i) \), using the Newton-Raphson method, for solution of non-linear equations.
3. KFCALC
Calculates $k_{\text{eff}}(i)$ and $B_i(k)$ (Biot no.), for $k = 1,2,3$ at every $j+1$ calculation using $j$th values of fluid phase compositions. Also computes these values as called by ENISHL for initialization.

4. DRVCAL
Calculates the derivatives:
\[
\begin{align*}
\frac{d\{x_k^\ast(i,j+1)\}}{d\{x_k^\ast(M,i,j+1)\}} \\
\frac{d\{f(x_k^\ast(i,j+1))\}}{d\{x_k^\ast(i,j+1)\}}
\end{align*}
\]
for $k = 1,2,3$ and $i = 1$ to $I$
This is performed at each time step, $j+1$, and at each iteration, $n$. The method (70) for differentiation of implicit functions is utilized (uses Cramer's Rule).

5. GAMRCL
Computes resin phase ionic species activity coefficients based on the Redlich-Kister equation for ternary systems, with parameters correlated in equivalent fractions, based on concentration arguments from the various calling subroutines.

6. GAMCAL
Computes electrolyte phase ionic species molal activity coefficients, $\gamma_k$ (where $k = 1,2,3$), based on the Bromley equation plus the derivatives, $d\gamma_k/dI$ (where $I$ = ionic strength), based on concentration arguments from the various calling subroutines.
7. **EQUILB**

Computes, with its support subroutine, EQICAL, the values of \( x_k^{*}(i,j+1)_n \) from input values of \( x_k(M,i,j+1)_n \) for \( k = 1,2,3 \) at \( i=1 \) to \( I \) based on the Newton-Raphson method for solution of non-linear equations. EQICAL computes the numerical derivatives of the functions with respect to \( x_1^{*}(i,j+1)_n \) and \( x_2^{*}(i,j+1)_n \) for use in the Newton-Raphson method for solving non-linear equations.

8. **DFTYCL**

Computes:

\[
\frac{\delta D^{*}(m,i,j)}{\delta x_k(m,i,j)}, \quad \frac{\delta D^{*}(m,i,j)}{\delta x_k(m,i,j)}, \quad \frac{\delta D^{*}(m,i,j)}{\delta x_k(m,i,j)}
\]

for \( k = 1,2,3 \) and \( \ell = 1,2,3 \) at \( m = 1 \) to \( M \) and \( i = 1 \) to \( I \) utilizing concentrations calculated at time step \( j \) versus \( j+1 \).

9. **CONCAL**

Computes the \( j+1^{th} \) and \( n^{th} \) iteration values of resin phase matrix coefficients and solution vectors.

10. **ZCONCL**

Computes the \( j+1^{th} \) and \( n^{th} \) iteration values of electrolyte phase matrix coefficients and solution vectors.

11. **ZSOLVE**

Scratch 5-diagonal array algorithm which solves for electrolyte phase concentrations \( x_k(i,j+1)_n \) \( (k = 1,2) \) \( _{\text{bulk}} \) for \( i = 1 \) to \( I \) at each time step, \( j+1 \), and each iteration, \( n \),
based on coefficients and vectors computed by ZCONCL.

12. XISOLVE and X2SOLVE

Scratch 7-diagonal array algorithm which solves for resin phase concentrations \( x_k(m,i,j+1)n(k = 1,2) \) for \( m = 1 \) to \( M \) and \( i = 1 \) to \( I \), at each iteration \( n \), and time step, \( j+1 \), with coefficients and vectors computed by CONCAL. Calls ZCONCL and ZSOLVE to get latest values of \( x_k(i,j+1)n \), since these are part of the solution bulk vector for \( x_k(M,i,j+1)n(k = 1,2) \) for \( i = 1 \) to \( I \). As discussed in Chapter V, Section D, the algorithms give better precision if the larger of \( x_1(1,i,j+1)n+1 \) or \( x_2(1,i,j+1)n+1 \) is in the first position in the scratch array, the decision being based on their respective values at the \( n \)th iteration.

C. Main Program Function Details

Performs following functions:

- calls INPUT
- calls ENISHL
- performs the calculation procedure from dimensionless time, \( j = 1 \) to \( j_{\text{max}} - 1 \), by performing the following functions:
  - calls EQUILB, KFCALC, and DFTYCL to get \( j \)th values of variables
  - at \( n = 1 \), sets all concentrations to values at last time step \( j \) as a first approximation, where \( n \) is iteration number
at $n_2$ to $n_{\text{convergence}}$, resets all $j+1^{th}$
concentration approximations calculated at the
$n-1^{th}$ iteration to those calculated at the $n^{th}$
iteration. These values are used to compute the
derivatives required in the quasilinearization
procedure.

- calls X1SOLVE or X2SOLVE

- checks the indicator, ICOUNT, which represents the
  number of variables not meeting convergence criteria,
  and if>0, recycles through another iteration. The
  user inputs the maximum number of iterations desired
  before moving on, from $j$ to $j+1$. Since this
  override represents moving on without total convergence,
  a warning message is written out giving the count of
  variables not meeting convergence criteria.
  Iterations normally ranged between 3-5 for convergence.
  If the system was ill conditioned due to poor
  initialization, or if one species reached a very low
  value (binary), increasing the maximum number of
  iterations above say 20 did not provide convergent
  solutions. Simulations containing non-convergent
  sections were discarded as suspect.

- controls all output options.
LISTING OF PROGRAMS DEVELOPED IN THIS RESEARCH

The following listing gives a brief description of programs developed in the course of this research project. All programs are written in Fortran and are compatible with the Fortran 77 standards as compilable on the DEC VAX 11/780 SYSTEM. Since the eight basic programs comprise over 12,000 lines of source code, the source code listings are not included in this dissertation. The source code listings are given in a supplement which is available in the New Jersey Institute of Technology library. In addition, source codes may be obtained on magnetic tapes for a nominal fee either from the NJIT Chemical Engineering and Chemistry Computational Center or from the author at the following address: D. W. H. Roth, Jr., Allied Corporation, P.O. Box 1021R, Morristown, NJ 07960. The following is a list of programs in the chapter order in which they were utilized.

1. **Chapter 2 — RIONFT**
   This program is used to regress non-linear ion exchange equilibrium data to obtain equilibrium constants and resin phase activity coefficient parameters for binary and ternary systems. The main features are covered in Appendix C. Including diagnostics and commentary statements, the program has about 2600 lines of coding.

2. **Chapter 3 — DIFACT**
   This program utilizes the Bromley equation to compute activity coefficients of salts in aqueous solution. In addition, the Nernst-Hartley equation is used to predict the values of
diffusion coefficients for salts at concentrations up to 6 molal. Both an integral and a differential form of the equation are used for comparison. The program has about 160 lines of coding.

3. **Chapter 5 - BMAVDM**

This program is based on the batch binary ion exchange dynamic model, i.e., the electrolyte concentration does not change during the exchange. It can handle equilibrium correlated in either mole or equivalent fraction concentration units, and the driving force can be defined either as a concentration or an activity difference. About 880 lines of coding.

4. **Chapter 5 - DFUSFT**

This program is designed to fit ion exchange "self" diffusion coefficients by modeling experimental batch ion exchange curves. It consists of BMAVDM appended to A. K. S. Murthy's non-linear regression program discussed in Appendix B. It can fit parameters using either the concentration or activity driving force models, plus a two parameter diffusion model involving an ion pair interaction factor. There are about 1420 lines of coding.

5. **Chapter 5 - TRYBCH**

This program is based on the batch ternary ion exchange dynamic model, i.e., the electrolyte concentration does not change during the exchange. It runs with either concentration or activity driving force. It is used to test the ability of "self" diffusion coefficients correlated with DFUSFT to predict ternary system behavior. About 2700 lines of coding.
6. Chapter 6 - DISPRS

This program models dispersive diffusion experiments to correlate axial dispersion coefficients in packed columns based on step change in inlet concentration. The resulting effluent concentration histories (ECH) or "S" curves are matched by adjusting the axial dispersion coefficient and are discussed in Chapter 6. About 400 lines of coding.

7. Chapter 7 - RFMAVD

This is the dynamic ion exchange column modeling program for binary systems as described in Appendix K. There are about 1920 lines.

8. Chapter 7 - TERNEX

This is the dynamic ion exchange column modeling program for ternary systems as described in Appendix L. About 2380 lines.
BIBLIOGRAPHY


